

Dictra 1D multiphase moving phase boundary simulations under local equilibrium conditions

TMS 2015

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- Software package for simulation of diffusion controlled reactions in multi-component alloys.
- Simulation on geometries, which may be reduced to one spatial coordinate (planar, cylindrical or spherical)
- Linked to Thermo-Calc, which provides all necessary thermodynamic properties.
- The result of more than 20 years and 60 man-years R&D at:
 - Royal Institute of Technology in Stockholm, Sweden
 - Max-Planck Institute für Eisenforschung in Düsseldorf, Germany
 - Thermo-Calc Software



Thermo-Calc Software

Outline

- Multiphase simulations
- Moving phase boundary simulations
- Combining the above



Thermo-Calc Software

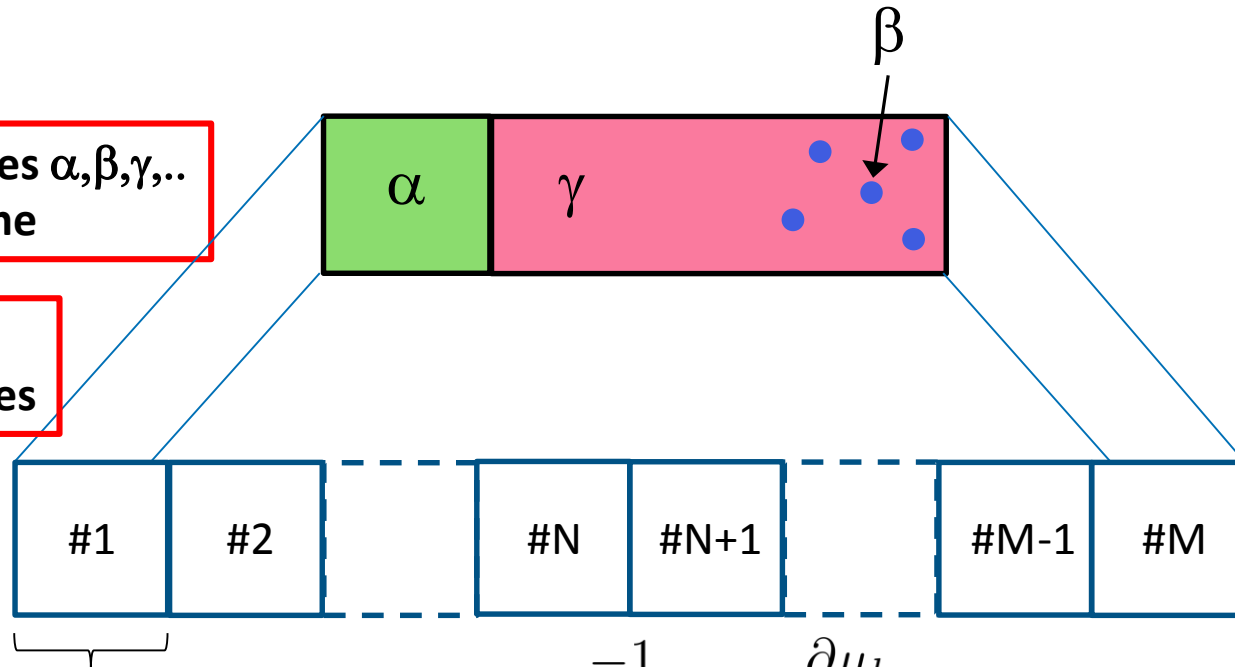
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Multiphase simulations

Allow a set of phases $\alpha, \beta, \gamma, \dots$
in each finite volume

No front-tracking
No explicit interfaces



Equilibrium calculation

Phase fractions
Phase compositions
Chemical potentials
Mobilities

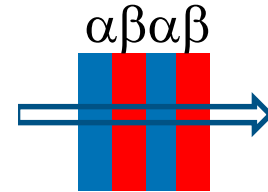
$$J_k = \frac{-1}{V_m} M_k x_k \frac{\partial \mu_k}{\partial z}$$

$$\simeq \frac{-1}{V_m} \sqrt{[M_k x_k]_{\#N}^{\text{eff}} [M_k x_k]_{\#N+1}^{\text{eff}}} \frac{\Delta \mu_k}{\Delta z}$$

Estimates of $[M_k x_k]^{\text{eff}} = \Gamma_k^*$

What are the kinetics of a
Multiphase mixture?

$$\Gamma_k^* = \left[\sum_{\phi} \frac{f^{\phi}}{\Gamma_k^{\phi}} \right]^{-1}$$



Wiener bounds,
rule-of-mixtures

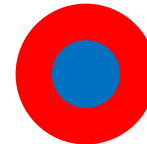
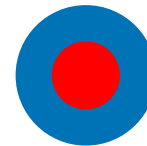
$$\Gamma_k^* = \sum_{\phi} f^{\phi} \Gamma_k^{\phi}$$



$$\Gamma_k^* = \Gamma_k^{\alpha} + \frac{A_k^{\alpha}}{1 - \frac{A_k^{\alpha}}{3\Gamma_k^{\alpha}}}$$

$$A_k^{\alpha} = \sum_{\phi \neq \alpha} \frac{f^{\phi}}{\frac{1}{\Gamma_k^{\phi} - \Gamma_k^{\alpha}} + \frac{1}{3\Gamma_k^{\alpha}}}$$

$$\begin{cases} \Gamma_k^{\alpha} = \min [\Gamma_k^{\beta}, \Gamma_k^{\gamma}, \Gamma_k^{\delta}, \dots] \\ \Gamma_k^{\alpha} = \max [\Gamma_k^{\beta}, \Gamma_k^{\gamma}, \Gamma_k^{\delta}, \dots] \end{cases}$$



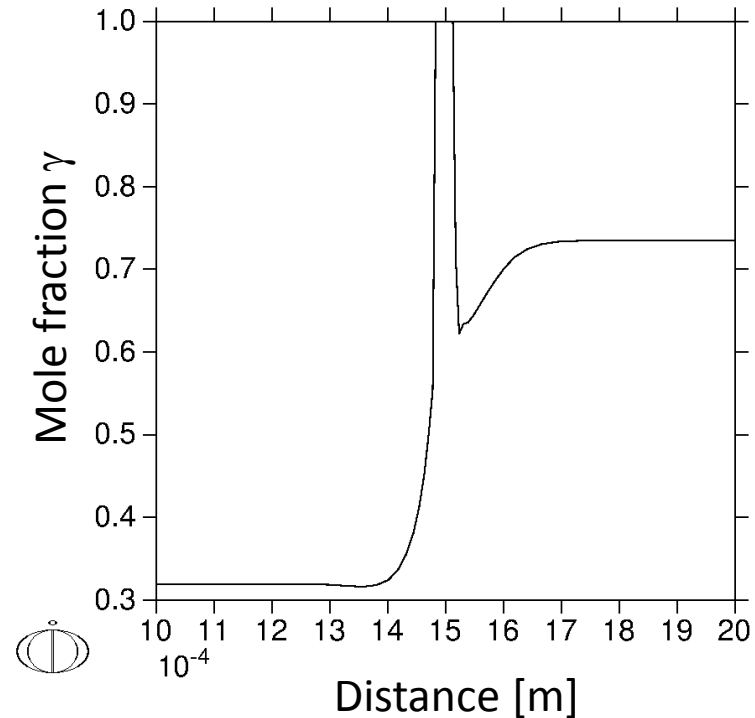
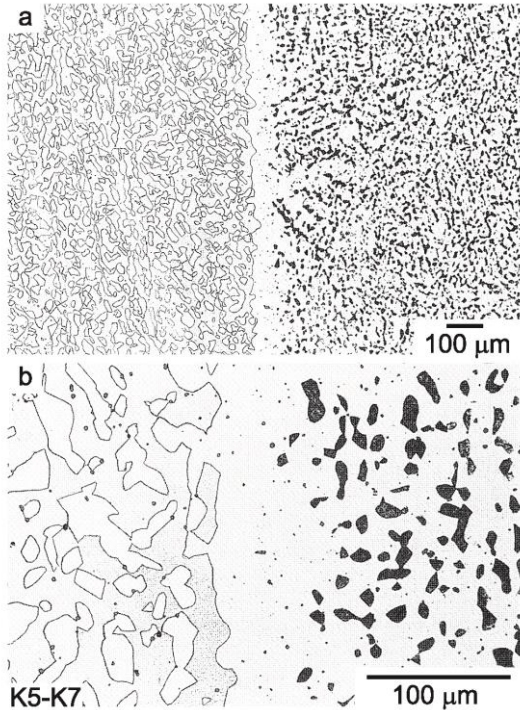
Hashin-Shtrikman
bounds

Wiener, Thesis, Königl Sächs Gesell Wissen, 1912
Rayleigh, Phil Mag 34(1892)481
Hashin, Shtrikman, J Appl Phys 33(1962)3125

Multiphase simulations

Example: Fe-Cr-Ni α + γ diffusion couple

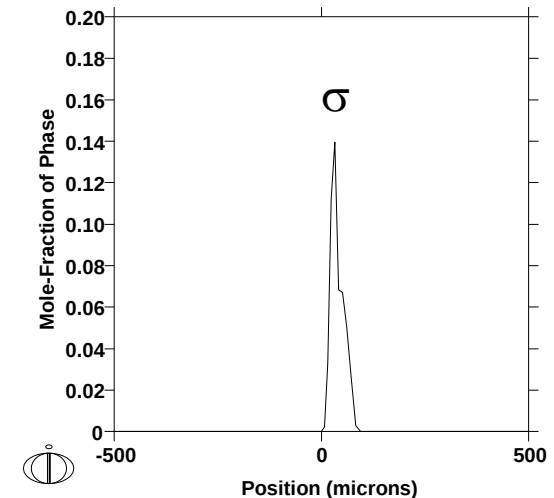
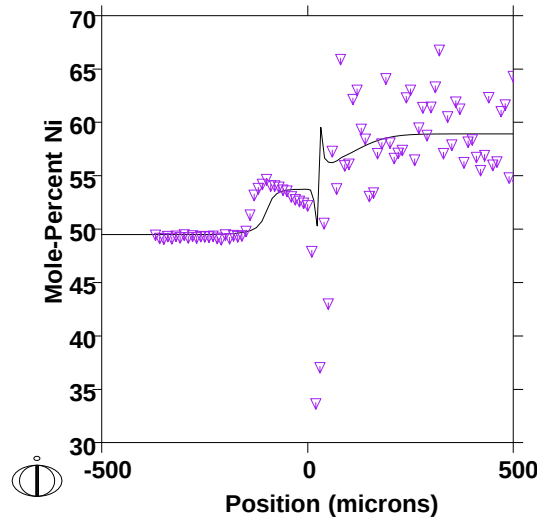
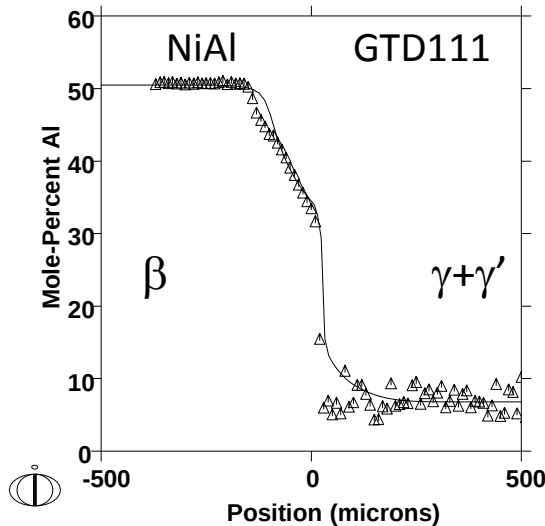
- 25.7 Cr 6.5 Ni - 42.3 Cr 27.6 Ni (at-%)
- 100 h, 1100°C
- Experiment by Engström



Multiphase simulations

Example: NiAl-coating / Ni superalloy diffusion couple

- 50.5 Al - 6.9 Al 9.5 Co 16.6 Cr 6.2 Ti (at-%, bal. Ni)
- 96 h, 1050°C
- Experiment by Perez, Patterson, Sohn



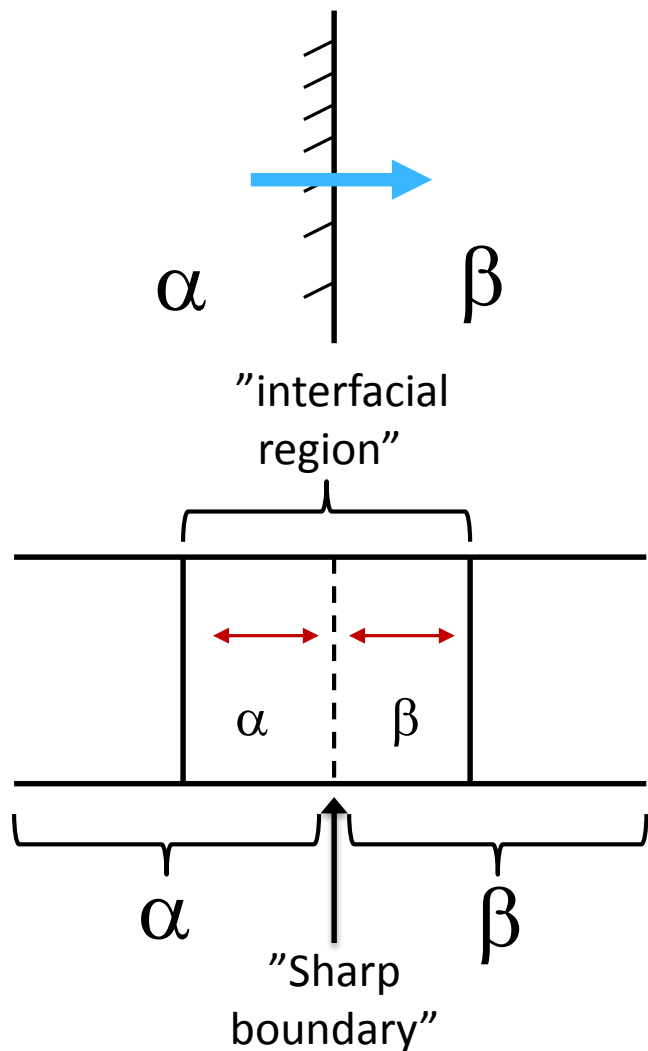


Thermo-Calc Software

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Local equilibrium – finite width interface



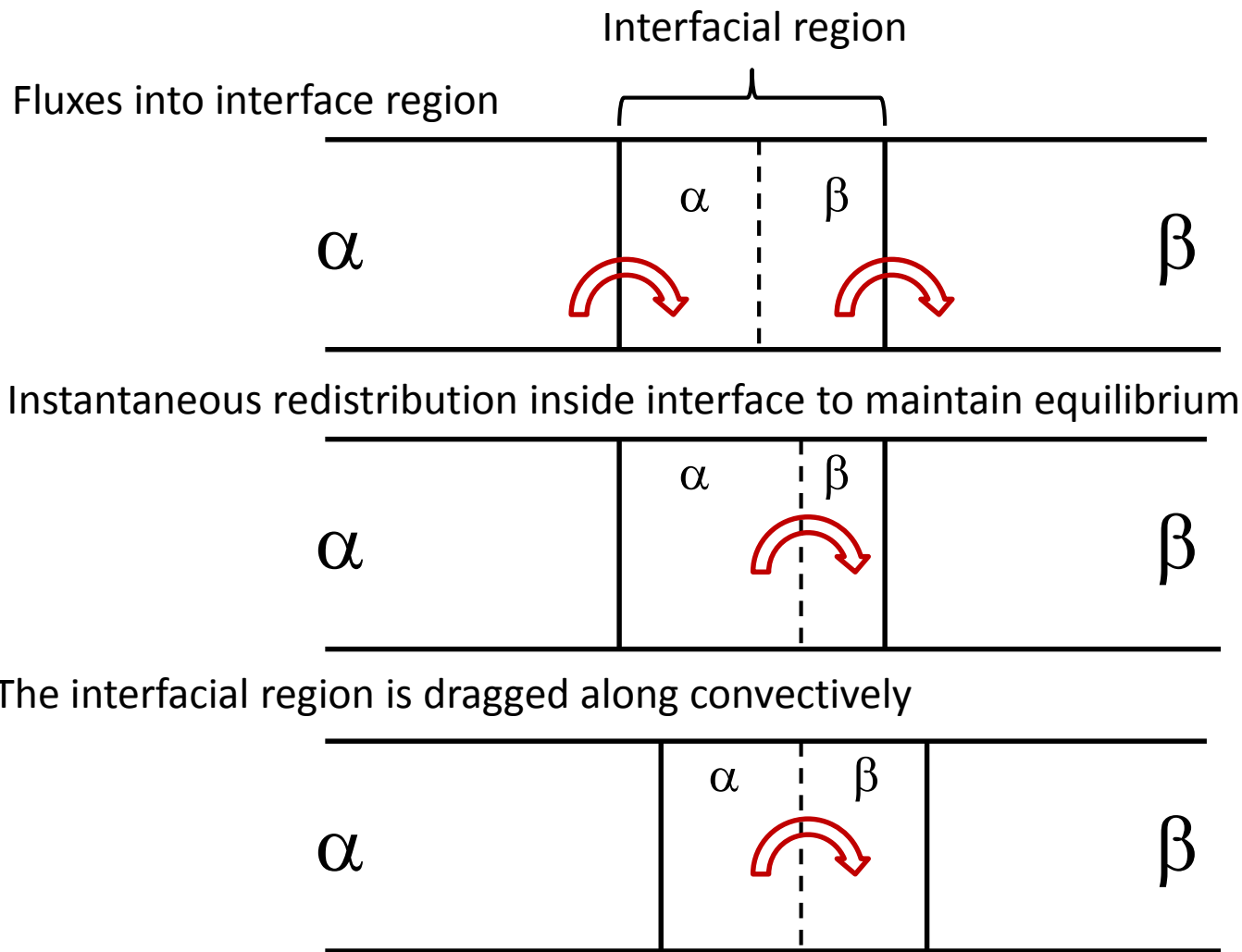
A recently added model in DICTRA in addition to the originally implemented Sharp interface model

Assume

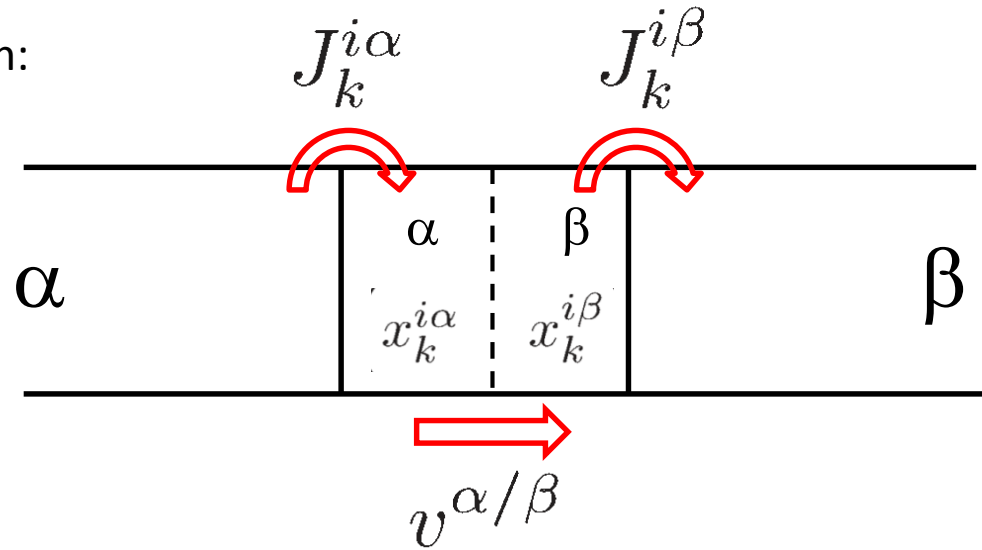
- An interfacial region of constant, finite width
- Local equilibrium holds inside the interfacial region
- Kinetics is infinitely fast in the interfacial region
- A sharp boundary in the middle of the interfacial region separate the phases
- Interface migration is caused by mass transfer over the sharp boundary

$$v^{\alpha/\beta} = -\vec{J}V_m$$

Conceived sequence of events leading to interface migration



Fluxes into interfacial region:

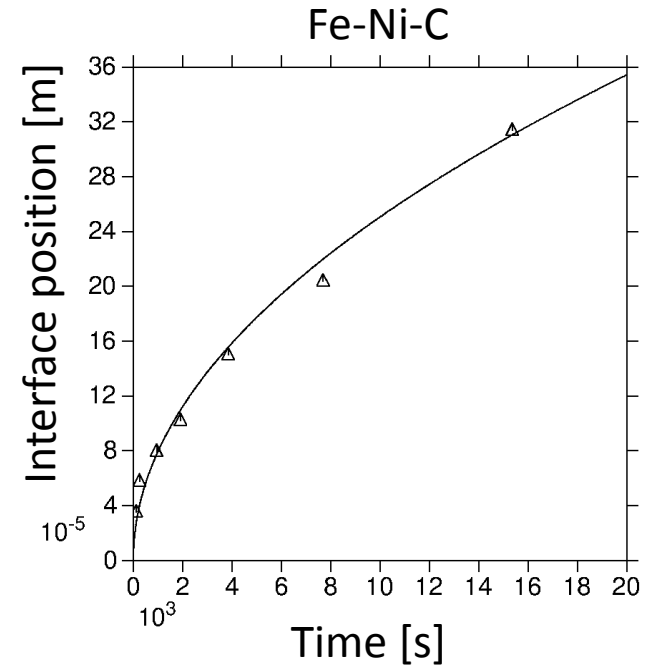
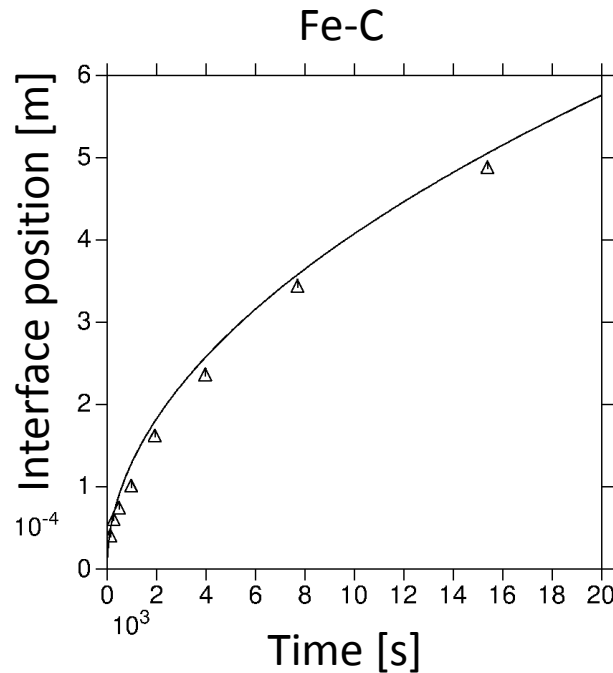
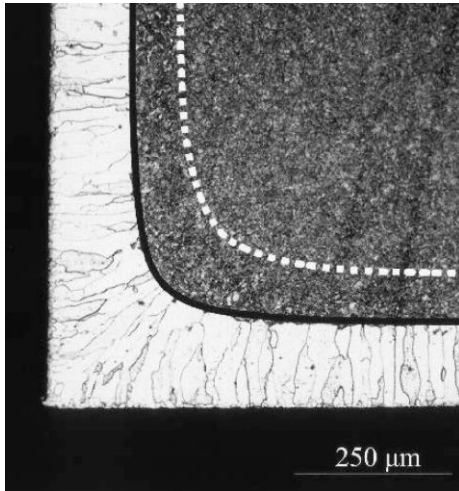


Interface velocity given by:

$$v^{\alpha/\beta} = V_m \frac{\sum \frac{\partial f^\alpha}{\partial N_k} \left(J_k^{i\alpha} - J_k^{i\beta} \right)}{\sum \frac{\partial f^\alpha}{\partial N_k} \left(x_k^{i\alpha} - x_k^{i\beta} \right)}$$

Example: Ferrite growth during decarburization

- $\gamma \rightarrow \alpha$
- Experiments by Phillion et al.
- Fe-Ni-C: Ferrite growth under NPLE conditions (no long-range Ni diffusion)



Local equilibrium, sharp vs finite width interface

The original moving boundary model in Dictra

First implemented in Dictra v27

Local equilibrium – sharp interface

Interface velocity found by solving
Set of flux balance equations

$$v (c_k^\alpha - c_k^\gamma) = J_k^\alpha - J_k^\gamma$$
$$k = 1, \dots, n - 1$$

Fast and efficient

Local equilibrium – finite interface

Explicit expression for interface velocity

$$v^{\alpha/\beta} = V_m \frac{\sum \frac{\partial f^\alpha}{\partial N_k} (J_k^{i\alpha} - J_k^{i\beta})}{\sum \frac{\partial f^\alpha}{\partial N_k} (x_k^{i\alpha} - x_k^{i\beta})}$$

Robust

Both methods are available in the Dictra software



Thermo-Calc Software

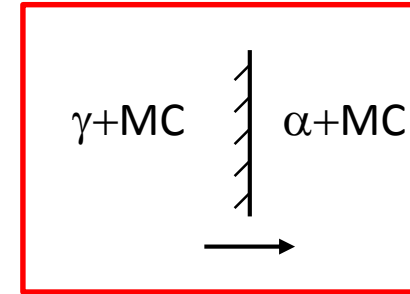
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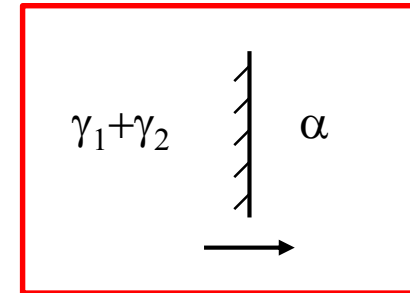
Combining a moving phase boundary model with the multiphase approach

Why?

Phase(s) present on both sides of an interface



Reaction product inside spinodal



Combined multiphase – finite width-local eq.

Three sets of phases:

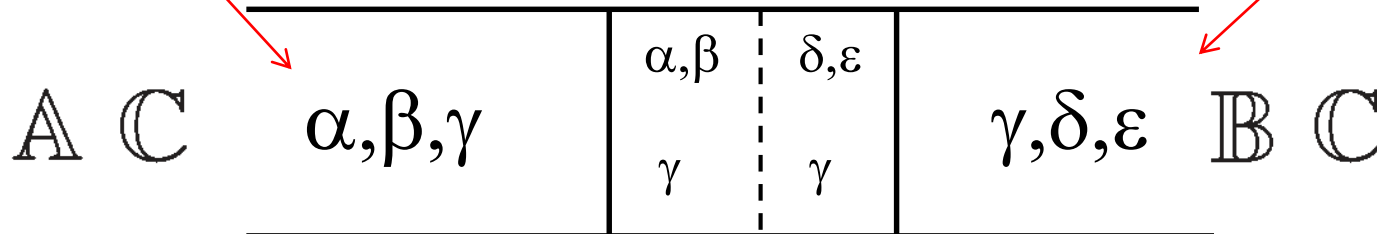
$\mathbb{A}$: α, β Allowed only on the left side of the interface

$\mathbb{B}$: δ, ε Allowed only on the right side of the interface

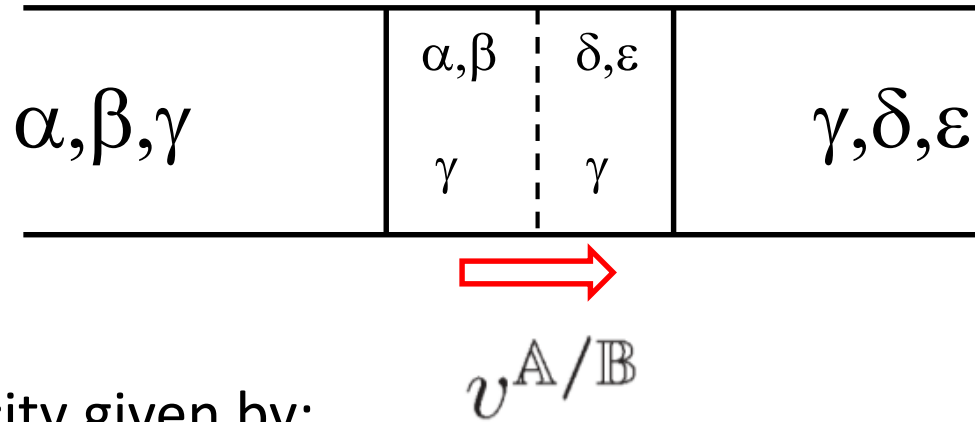
$\mathbb{C}$: γ Allowed on both sides of the interface

Multiphase mixture
 α, β, γ allowed

Multiphase mixture
 $\gamma, \delta, \varepsilon$ allowed



Combined multiphase – finite width-local eq.

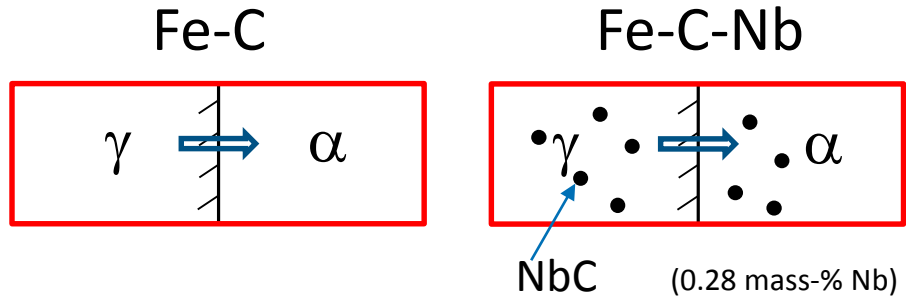
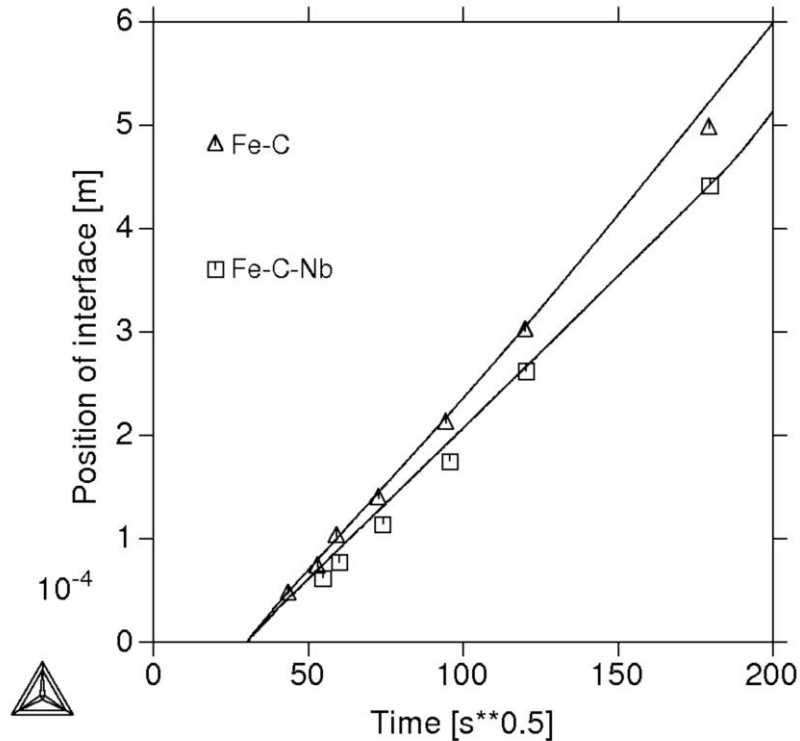


Interface velocity given by:

$$v^{\mathbb{A}/\mathbb{B}} = V_m \frac{\sum_{\alpha \in \mathbb{A}} \sum_k \frac{\partial f^\alpha}{\partial N_k} (J_k^{i\mathbb{A}} - J_k^{i\mathbb{B}}) - \sum_{\delta \in \mathbb{B}} \sum_k \frac{\partial f^\delta}{\partial N_k} (J_k^{i\mathbb{A}} - J_k^{i\mathbb{B}})}{\sum_{\alpha \in \mathbb{A}} \sum_k \frac{\partial f^\alpha}{\partial N_k} (x_k^{i\mathbb{A}} - x_k^{i\mathbb{B}}) - \sum_{\delta \in \mathbb{B}} \sum_k \frac{\partial f^\delta}{\partial N_k} (x_k^{i\mathbb{A}} - x_k^{i\mathbb{B}})}$$

Combined multiphase – finite width-local eq.

Example: Carburization of Fe-C and Fe-C-Nb



Exp. by Togashi and Nishizawa
Carburization at 1073 K

Carbon mobilities fitted to binary case

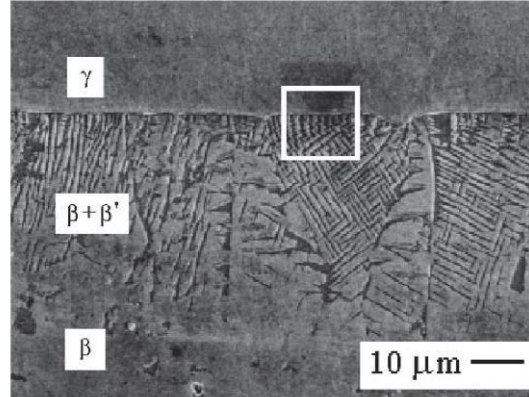
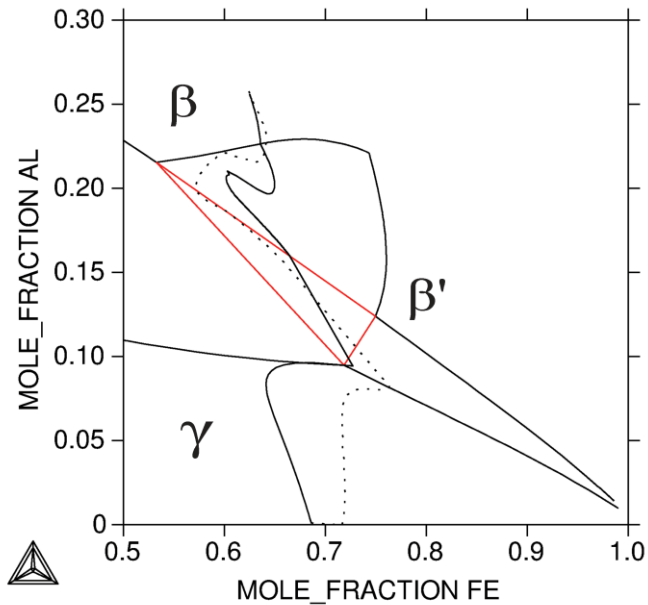
Combined multiphase – finite width-local eq.

Example: Al-Fe-Ni Diffusion couple

Diffusion path

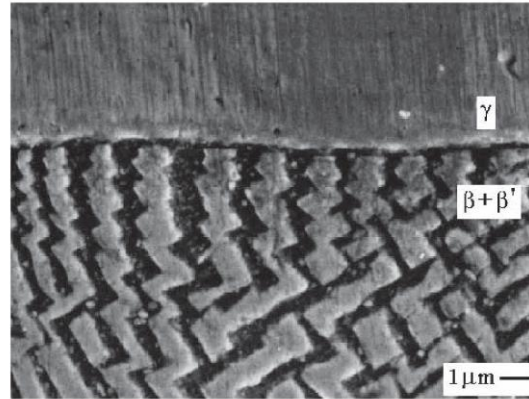
Dotted: Experiment

Solid: Simulation



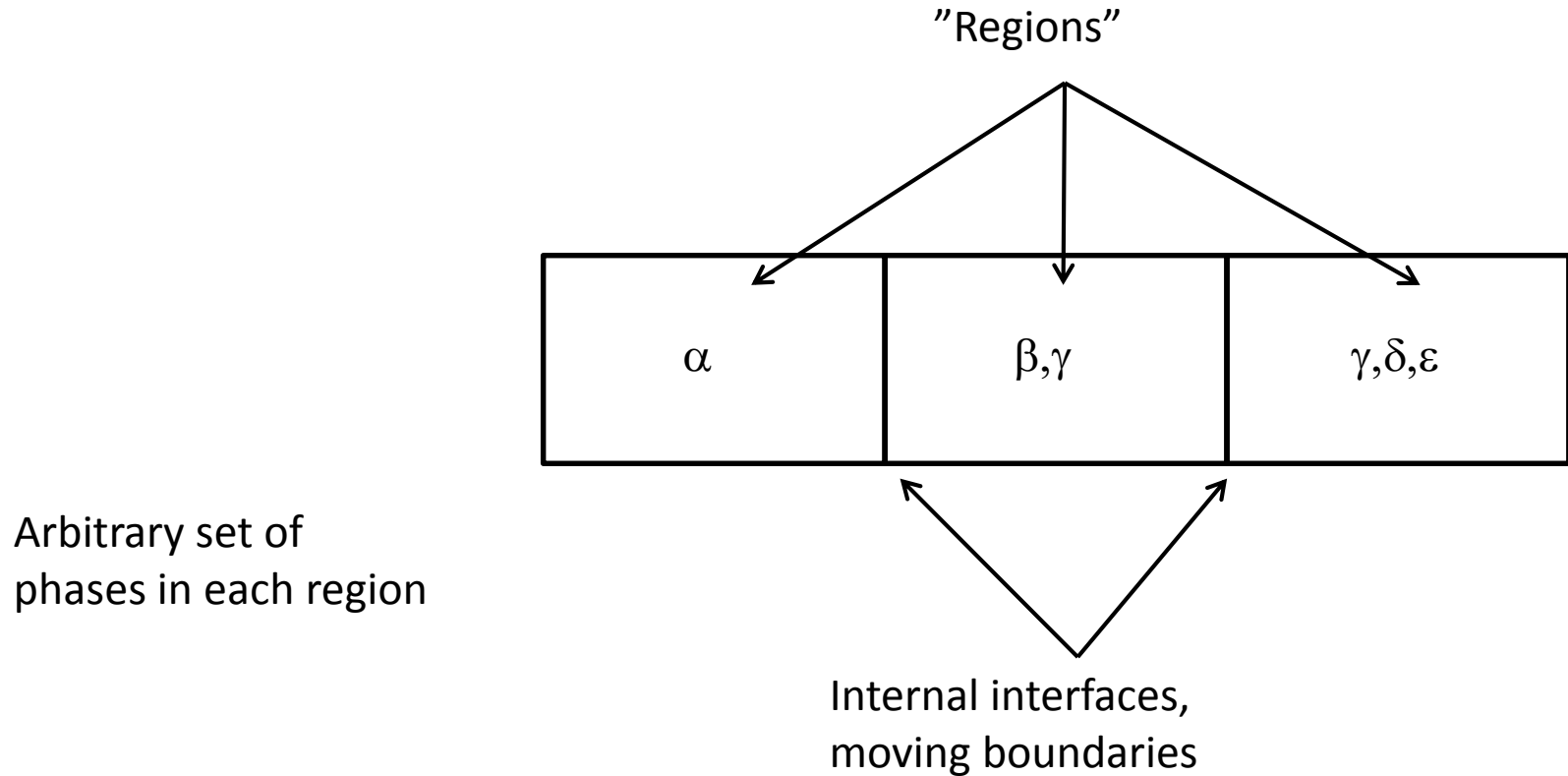
Experiment by Sohn,
Puccio, Dayananda

1273 K
48 h



Al mobilities of database
Increased by a factor two

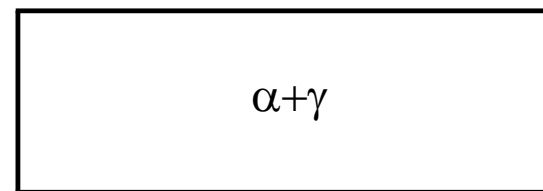
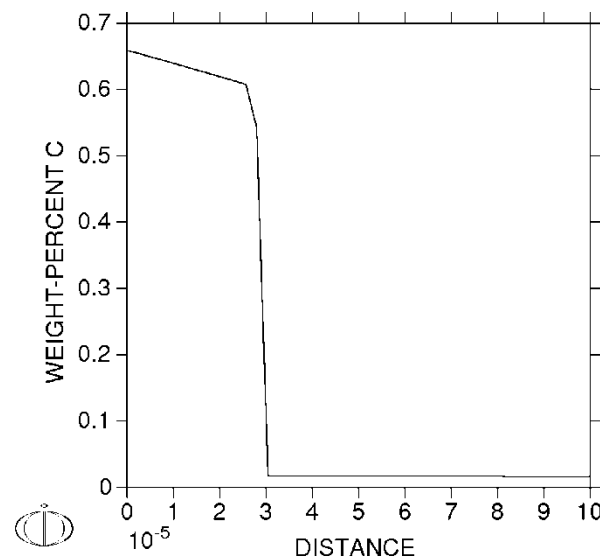
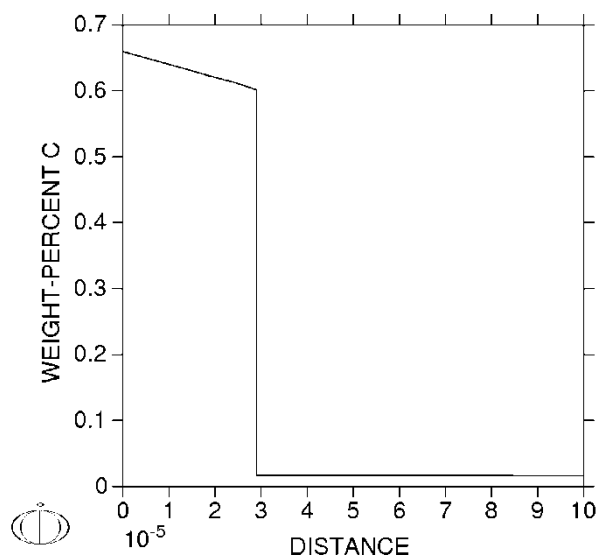
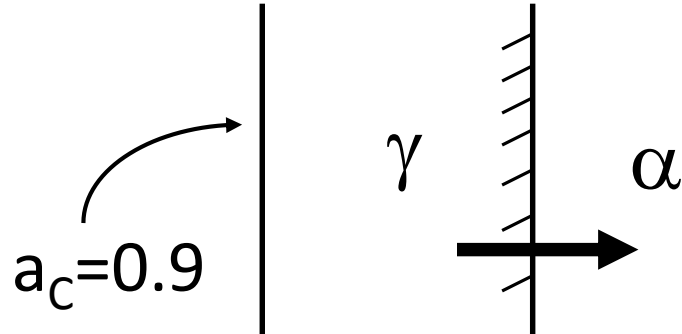
Dictra set-up using new model



Remaining requirement: one unique phase in each region

Different set-ups, same results

Carburizing plain carbon steel
at 750°C



Summary

- Two alternative models for moving phase boundary simulations in Dictra
- Dictra diffusion simulations can now be set up in a very general manner
- More important than ever to carefully consider all assumptions when setting up a simulation

Thank you!