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# Application of Computational Thermodynamics to Steel Processing and Alloy Design

André Costa e Silva  
EEIMVR-UFF  
Volta Redonda Brazil

# Outline

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  - Empiricism, theories, models, diagrams, data...
- **Where are we (what can we do)?**
  - CALPHAD and “related techniques”
- **Where do we want to go?**
  - Using CALPHAD to solve practical applications and gain insight on complex problems.

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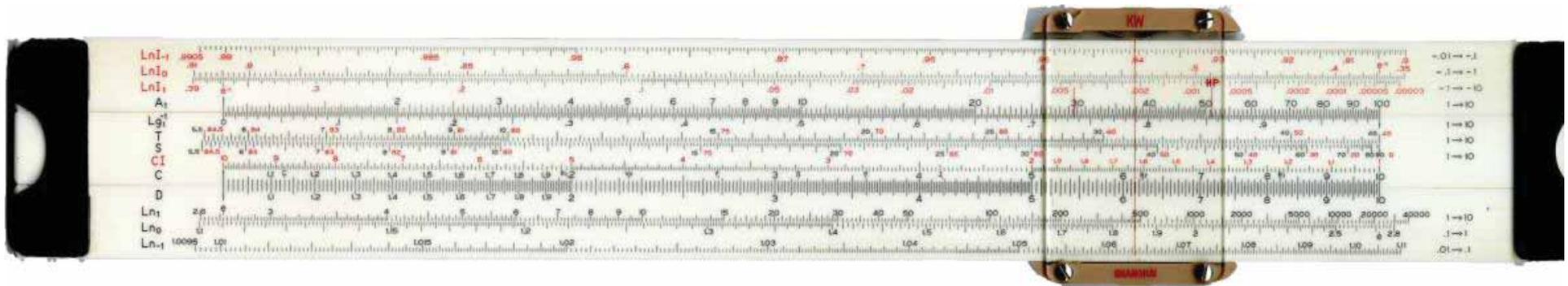
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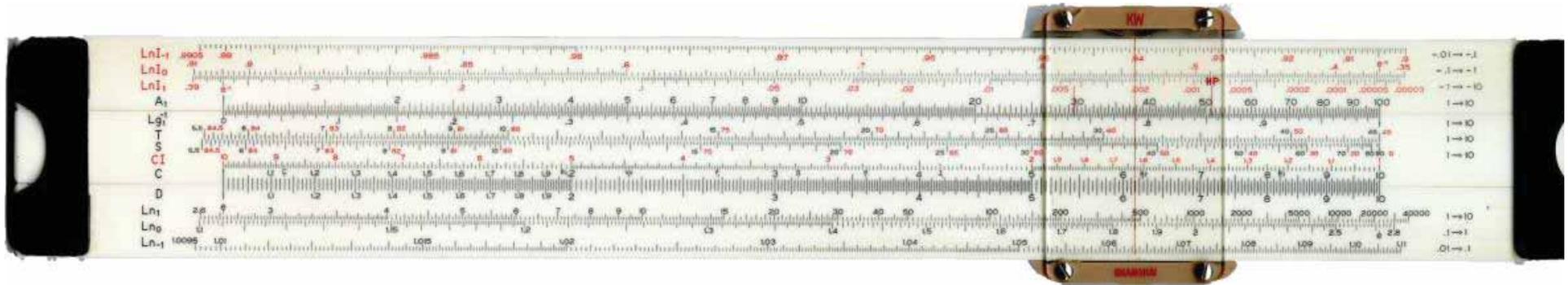
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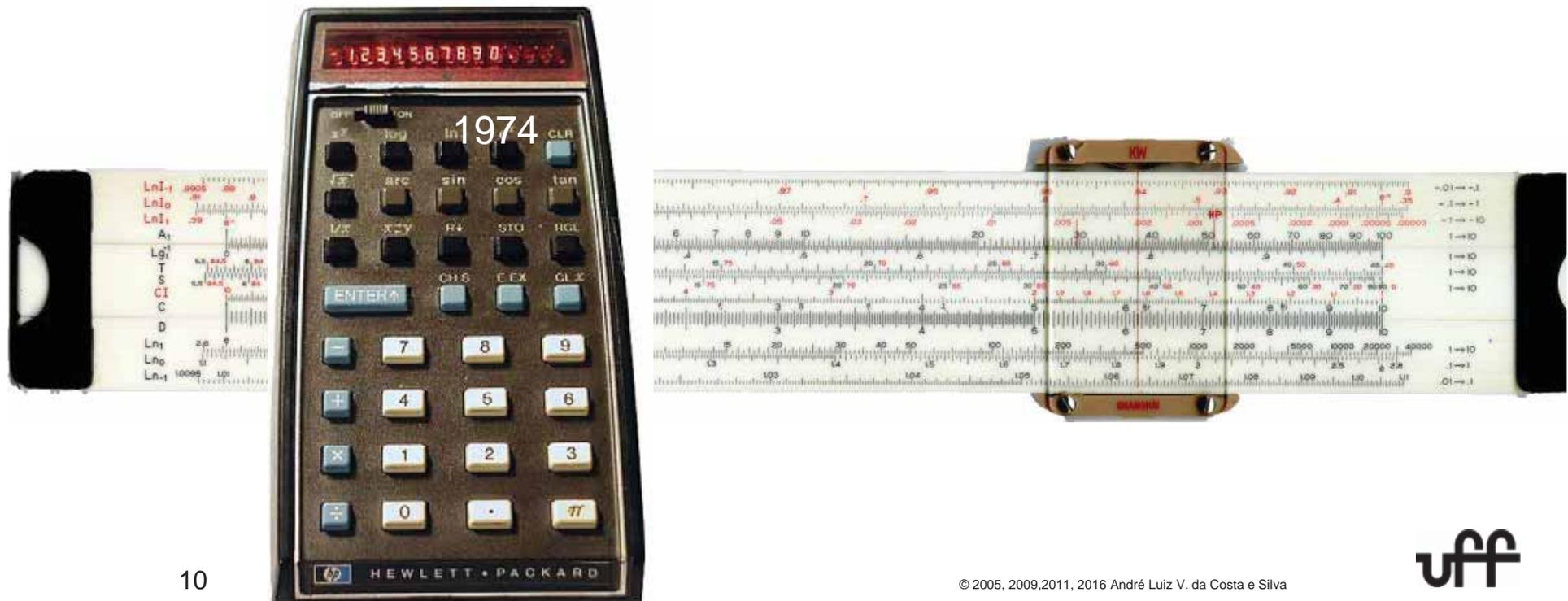


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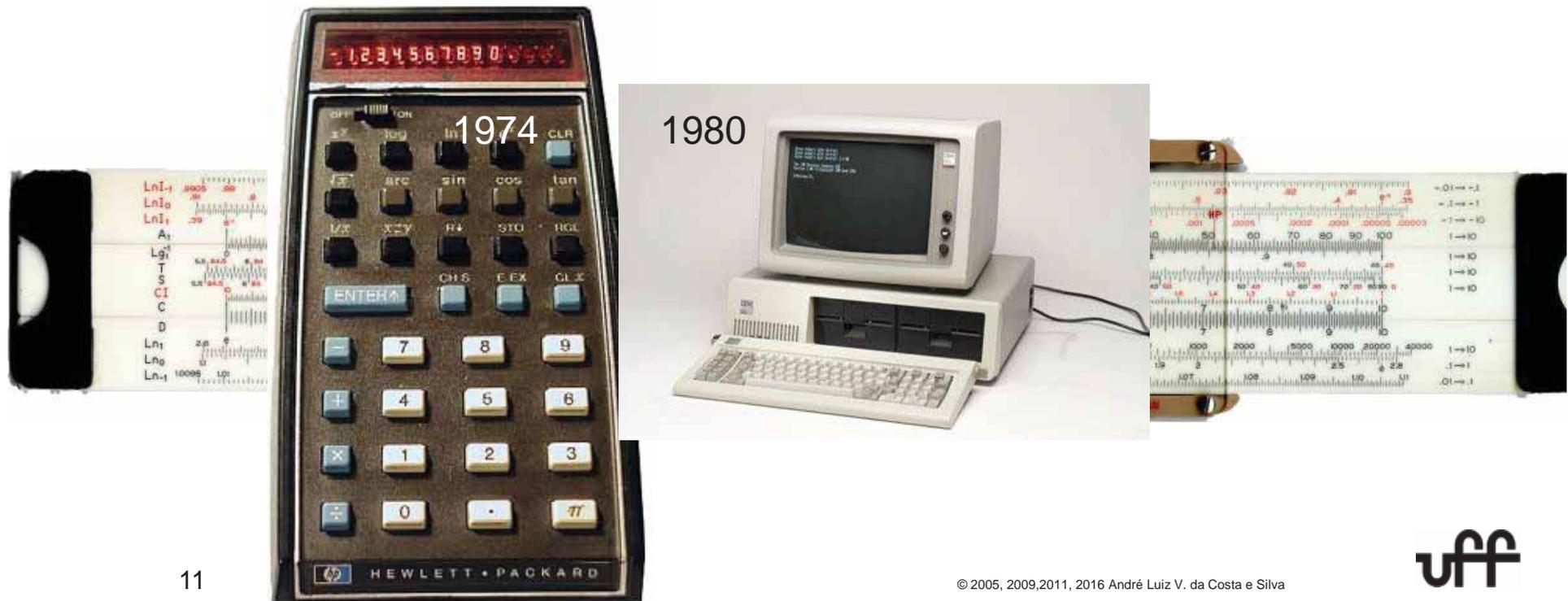


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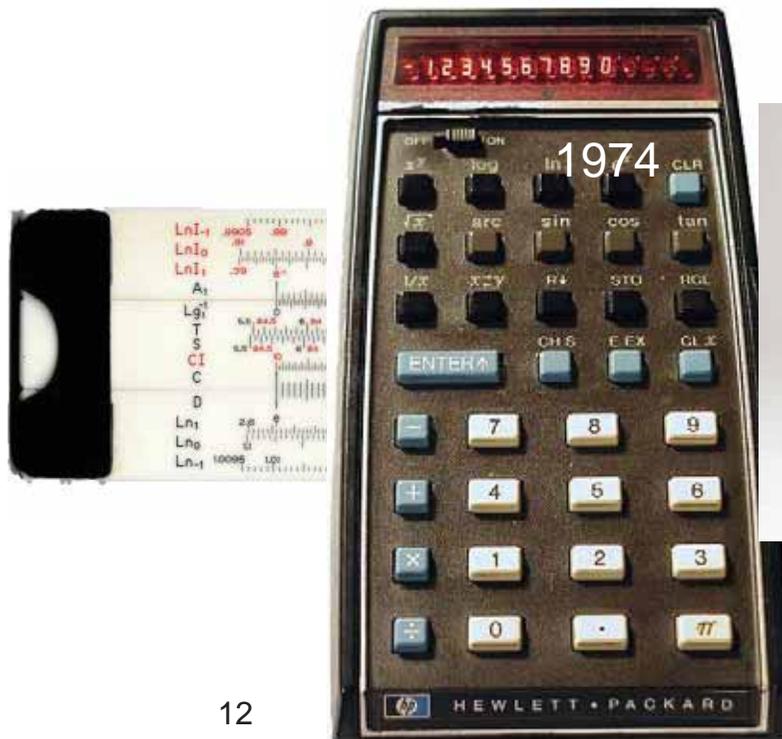


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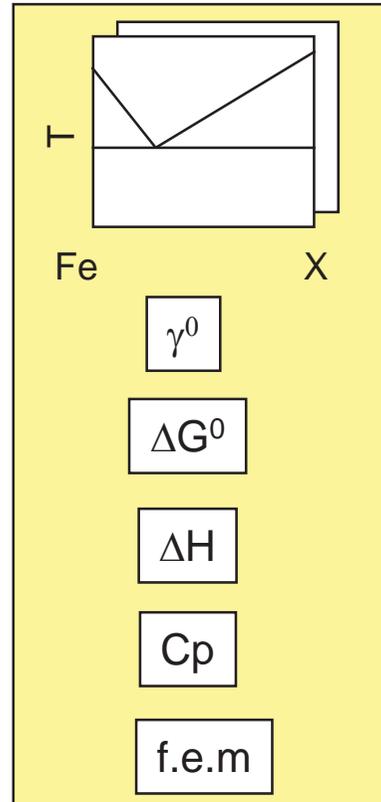
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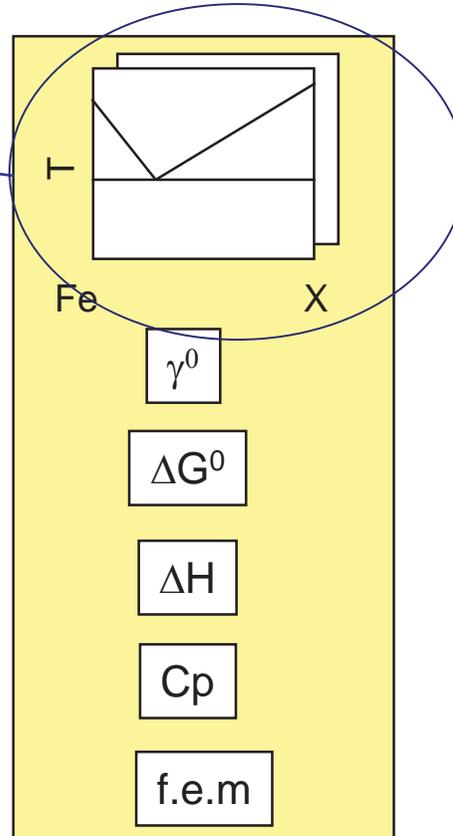
## Experimental measurements



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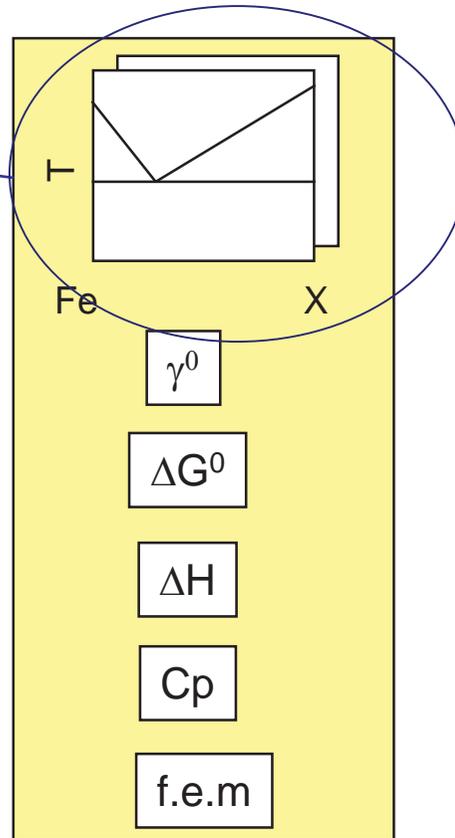
Direct reading of phase diagrams:  
Binaries, ternaries, ...  
Isoactivities,  
Isotherms  
*Liquidus* surfaces



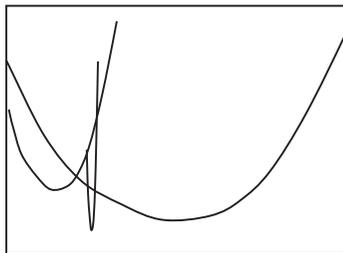
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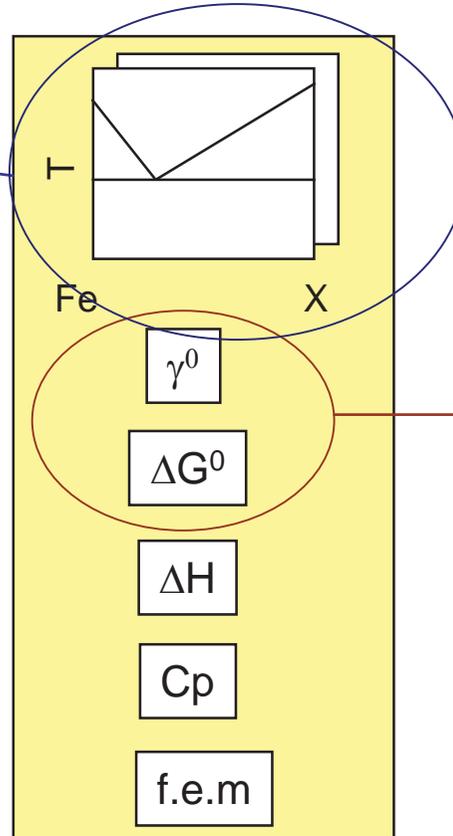
Attempts at “rationalization”  
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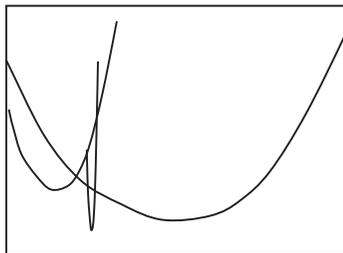
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Calculations for dilute solutions:  
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Solubility products,  
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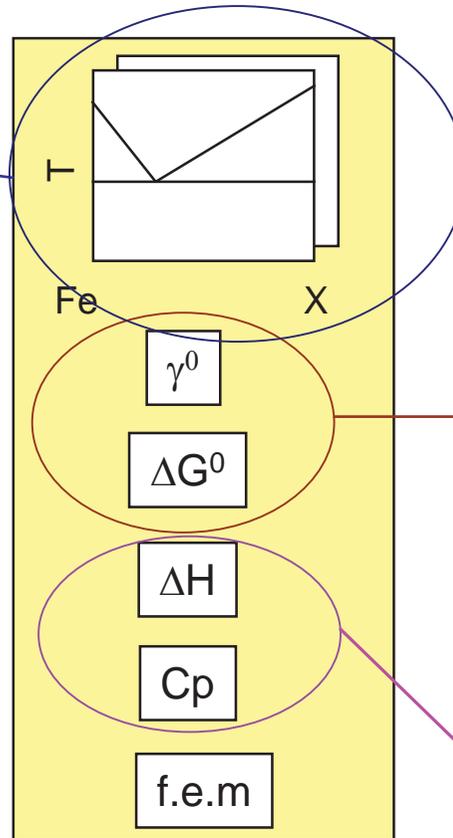
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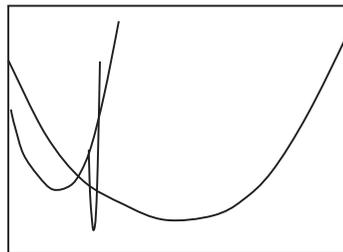
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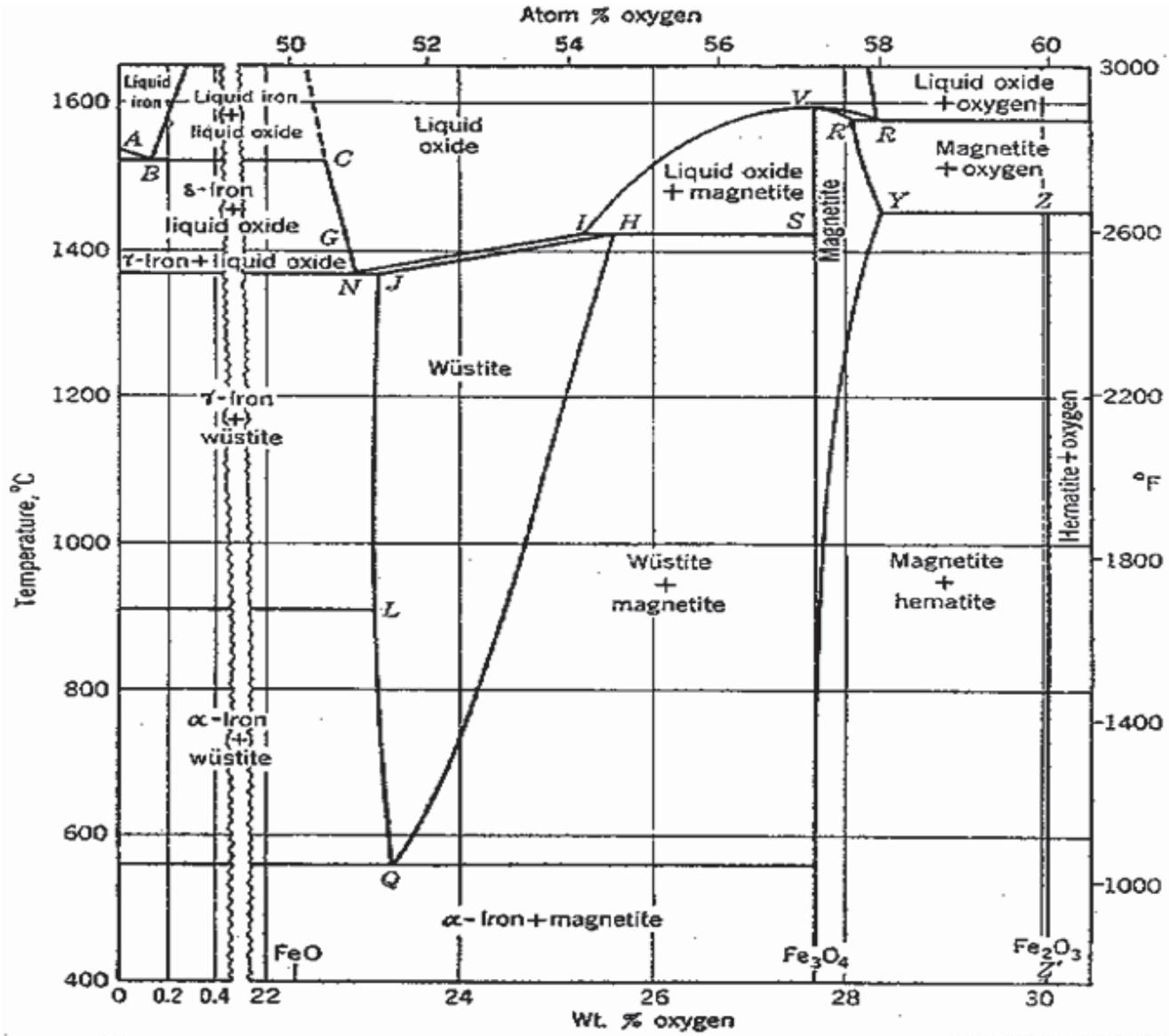
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Heat balances using  
 EXTERNAL  
 information about  
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 system.

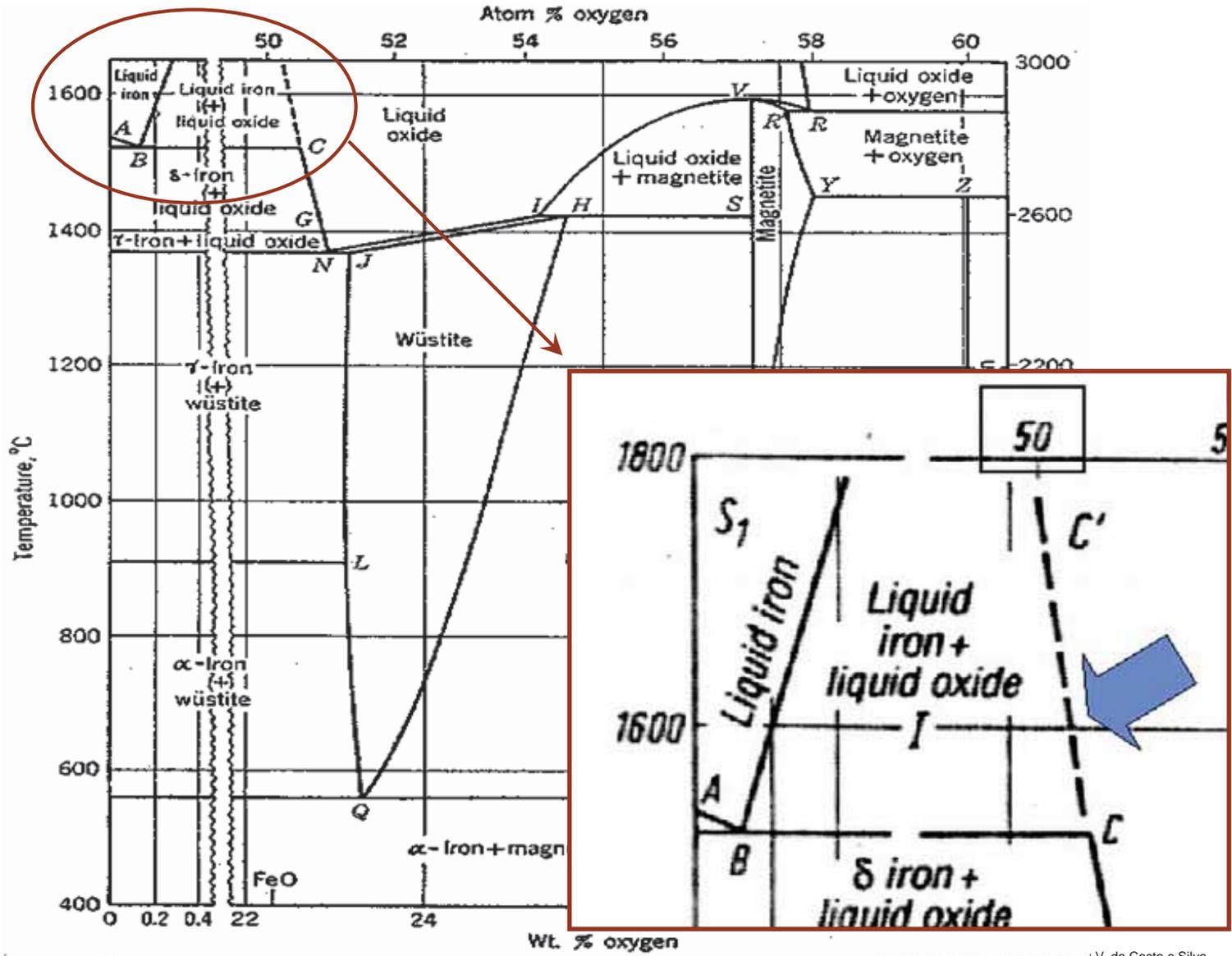
# Example: O solubilty in Liquid Fe



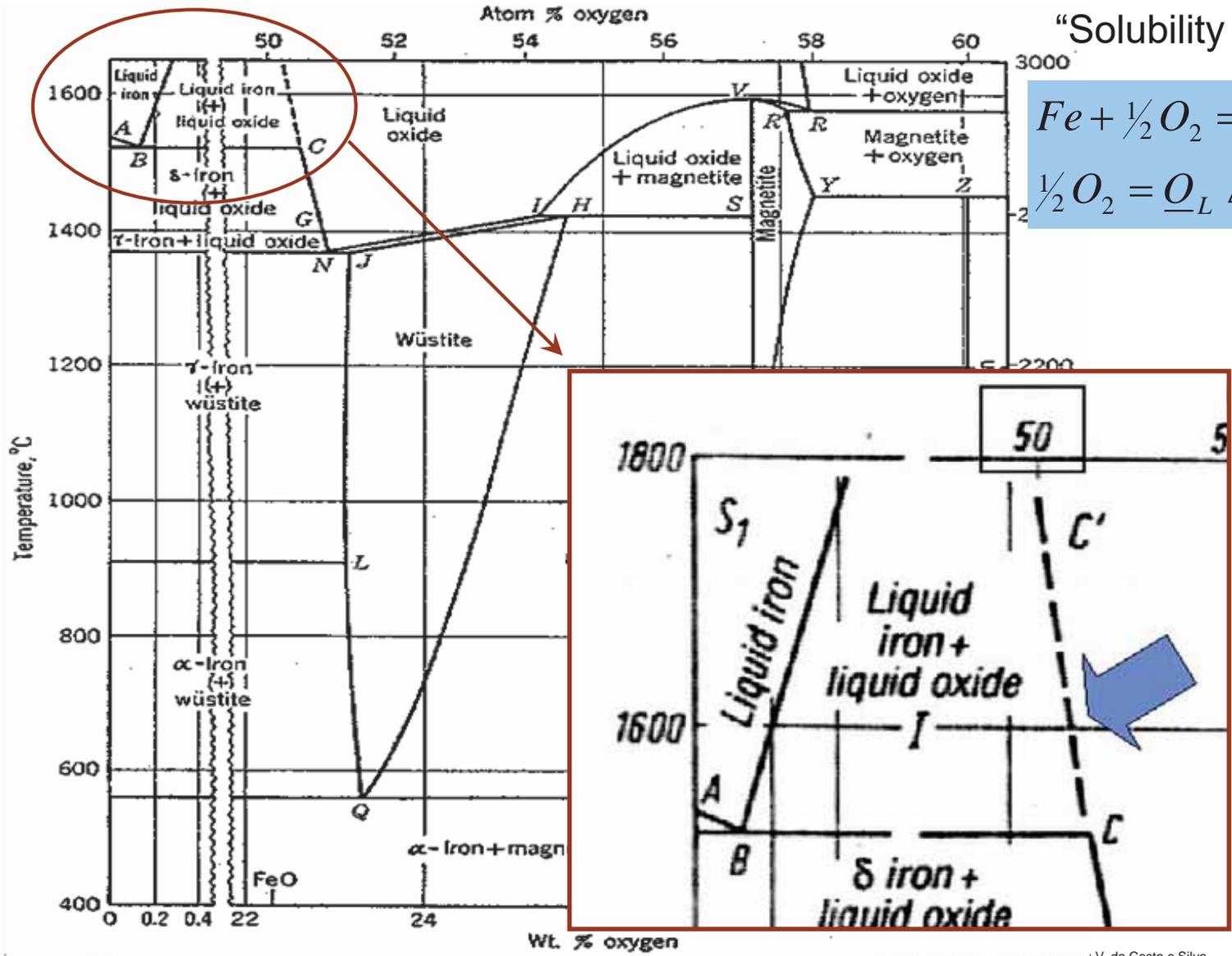
© 2000, 2005, 2011, 2010 André Luiz V. da Costa e Silva



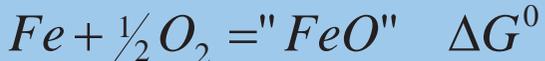
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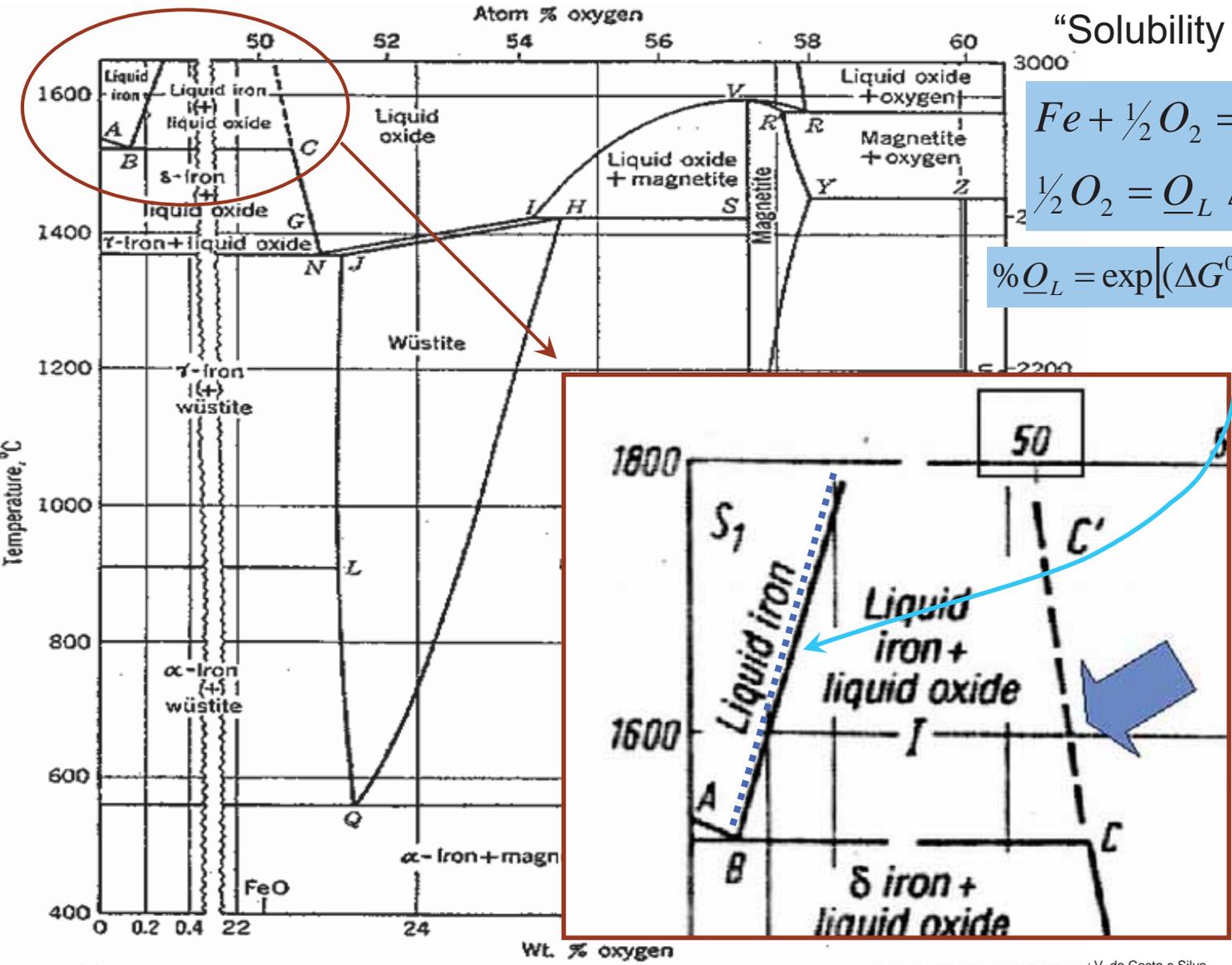
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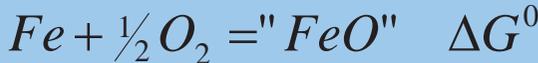
“Solubility limit”



# Example: O solubility in Liquid Fe



“Solubility limit”



$$\% \underline{O}_L = \exp [(\Delta G^0 - \Delta G_{O,L}^{M,1\%}) / RT]$$



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- **“Brave” applications**
  - Alloy design
  - Process design
  - Forecasting

# Some current examples of “brave” applications

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- **Steelmaking Alloy and Processing design**
  - 6 elements at least (besides Fe), some in ppm quantities, high temperatures, need to be “very, very” close to reality or lose your job!

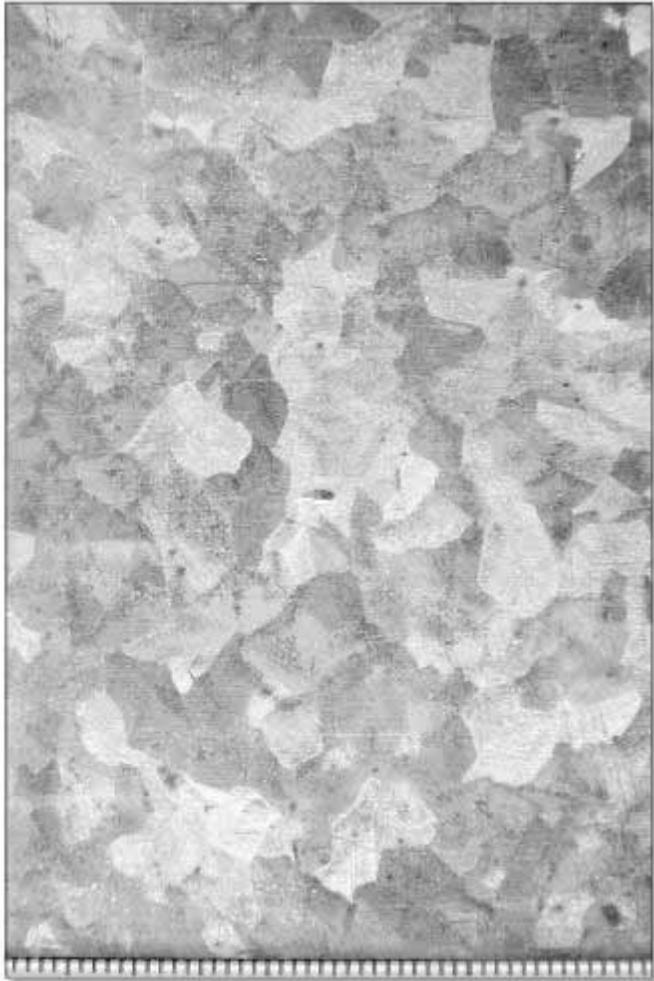
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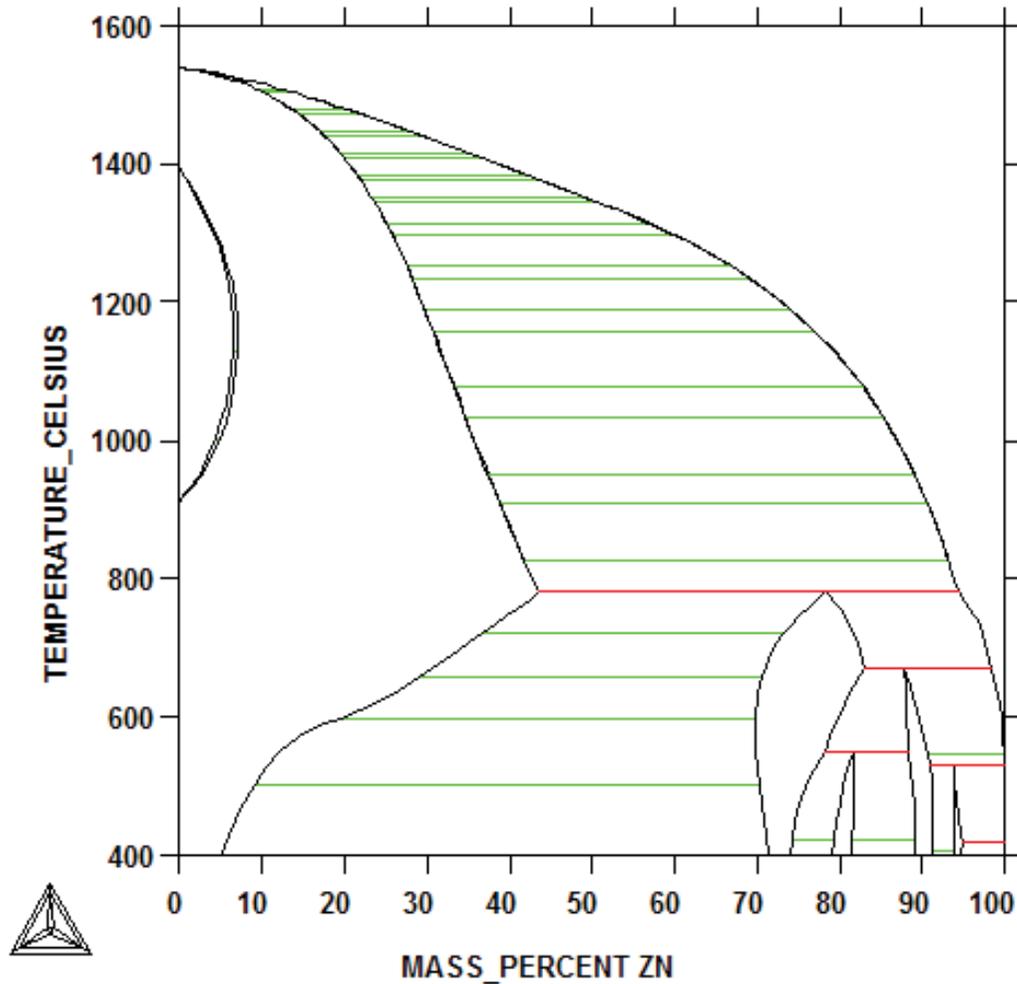
- **Steelmaking Alloy and Processing design**
  - 6 elements at least (besides Fe), some in ppm quantities, high temperatures, need to be “very, very” close to reality or lose your job!
- **Forecasting Alloy behavior**
  - Component ageing (inaccessible experiments...)
  - Understanding oxidation...

# Understanding galvanizing- small additions to Zn bath have large effects...

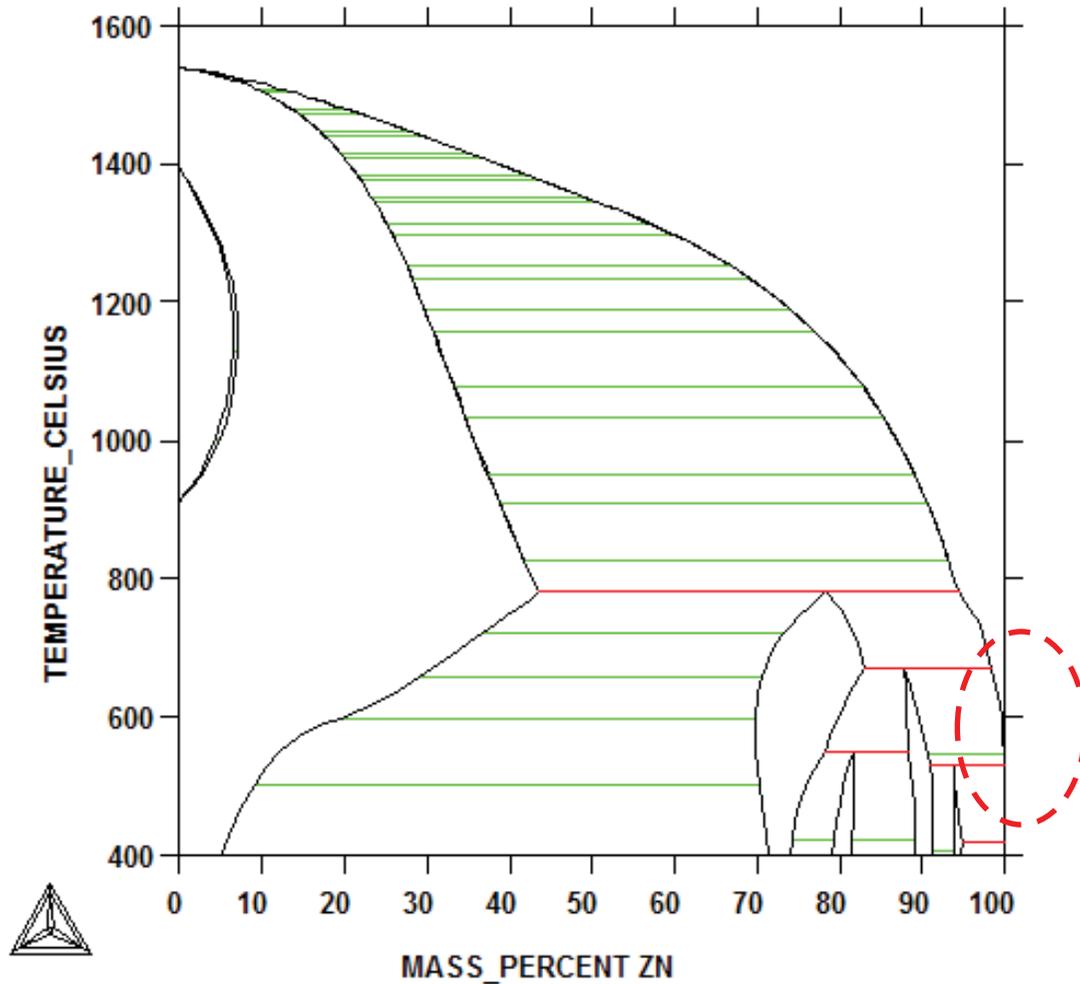
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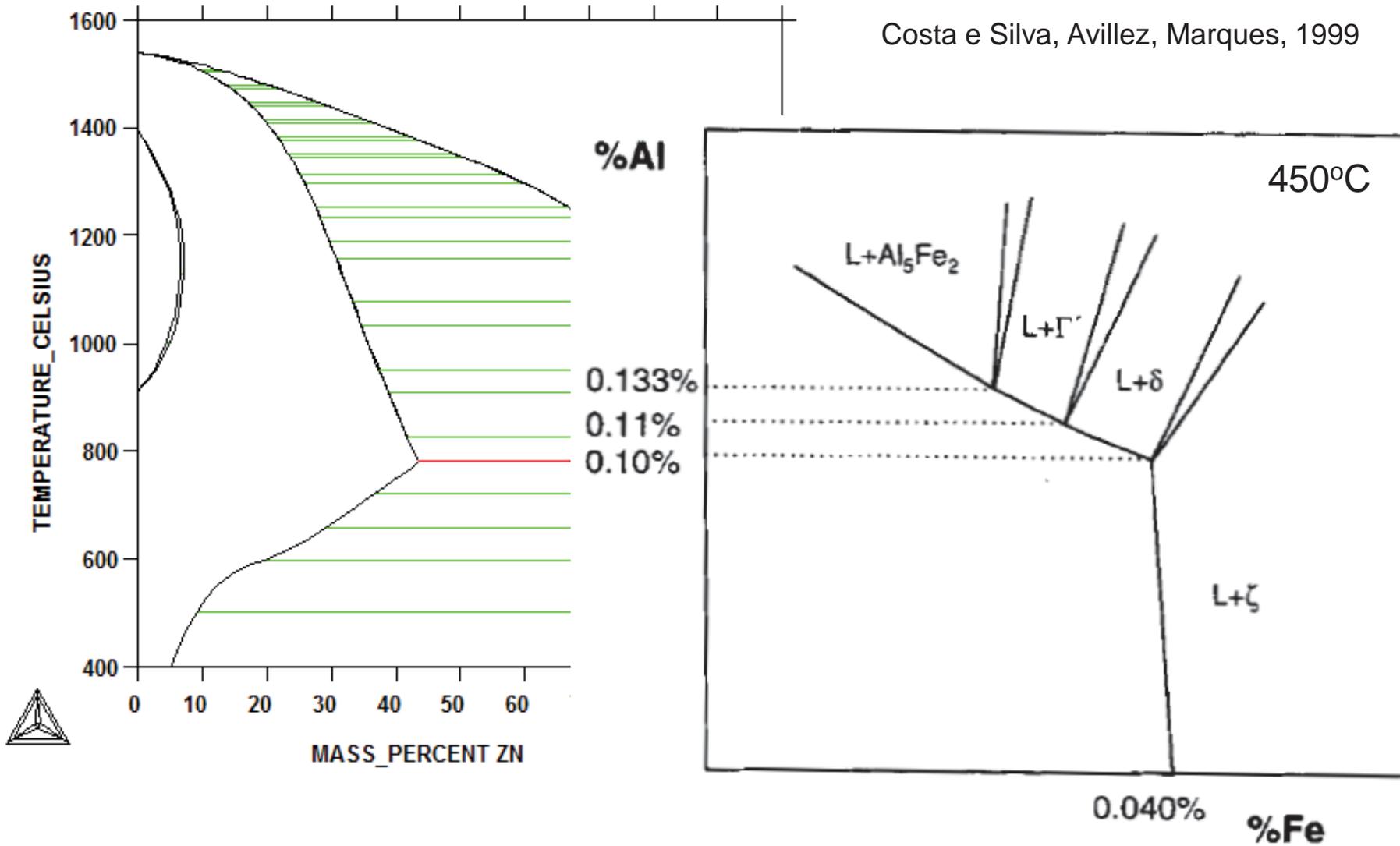
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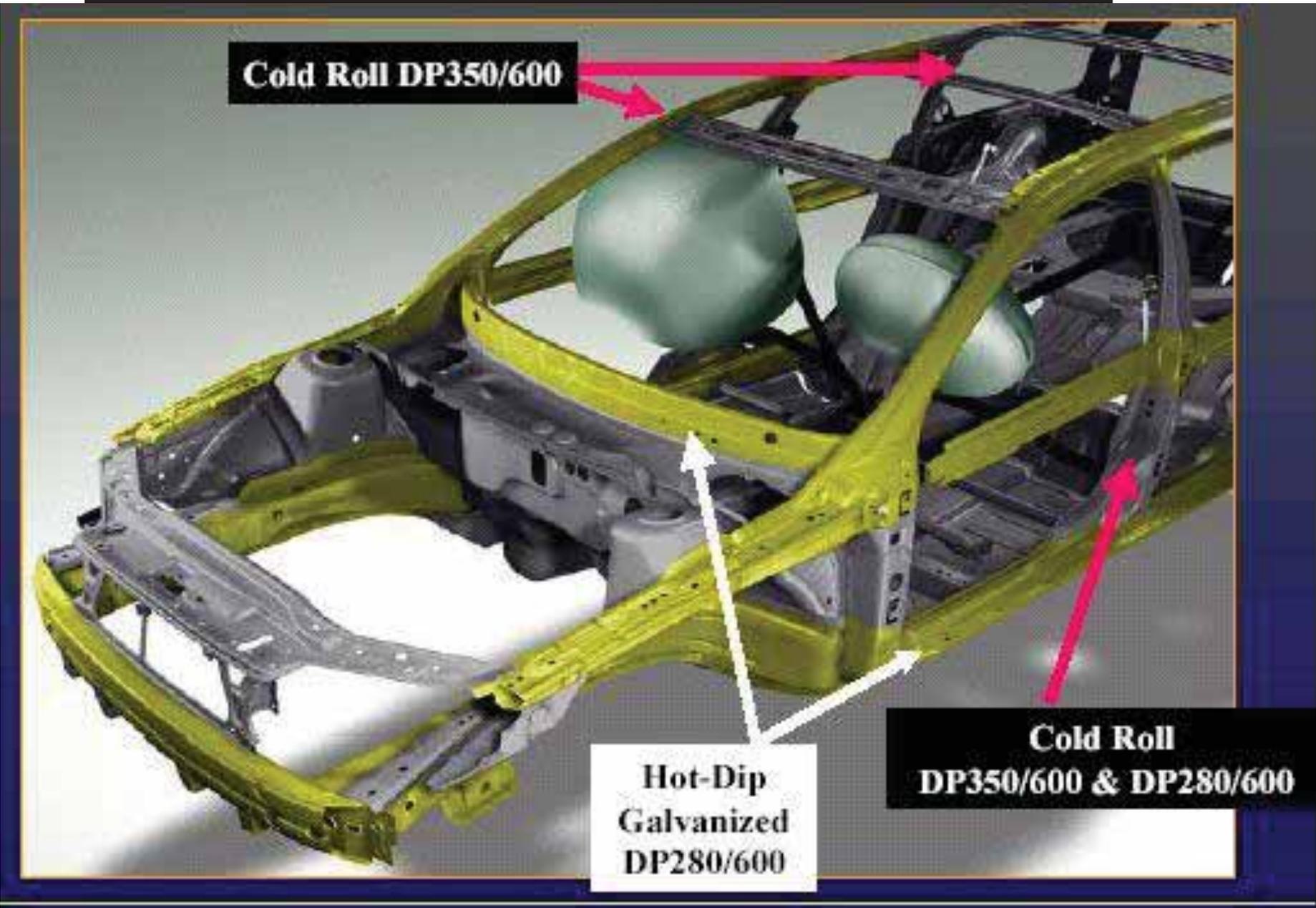
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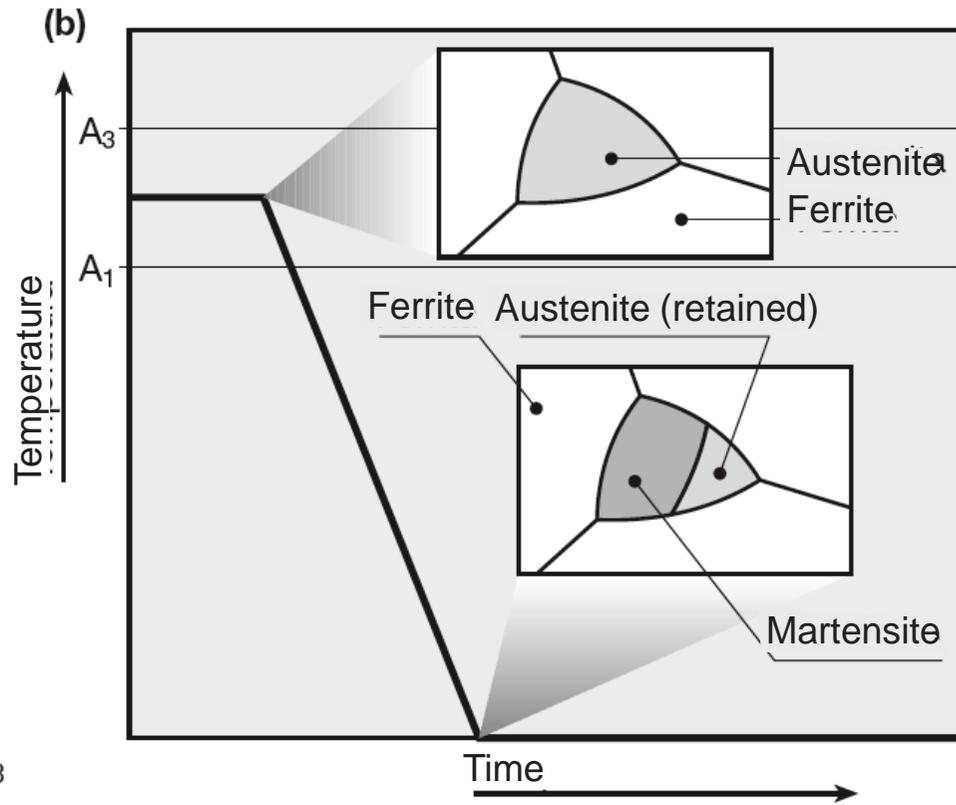
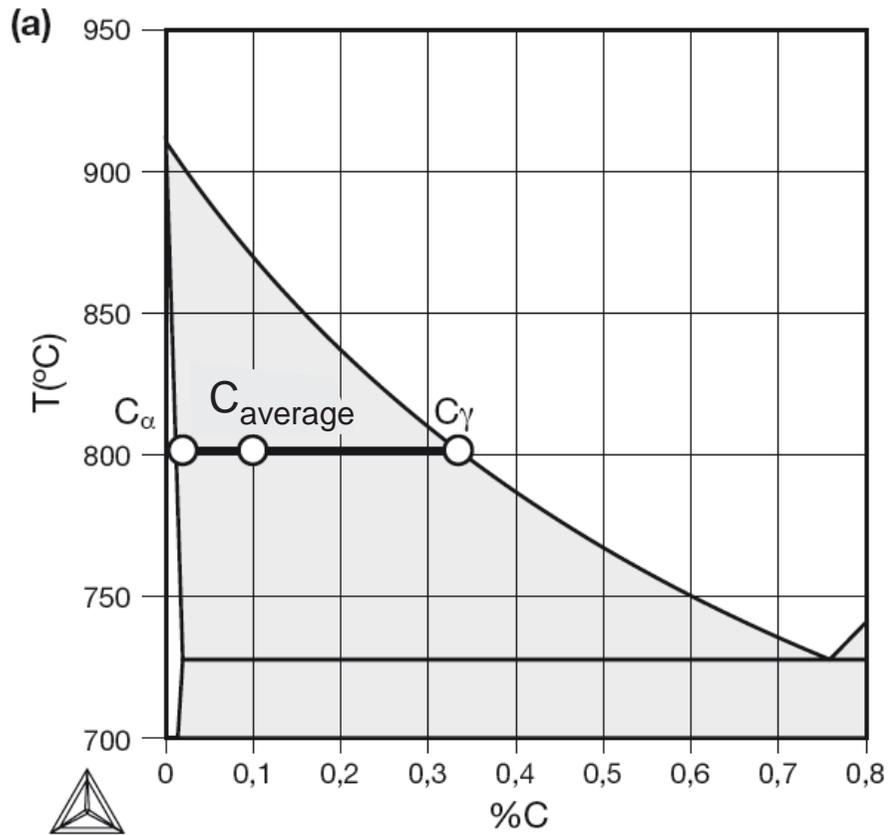
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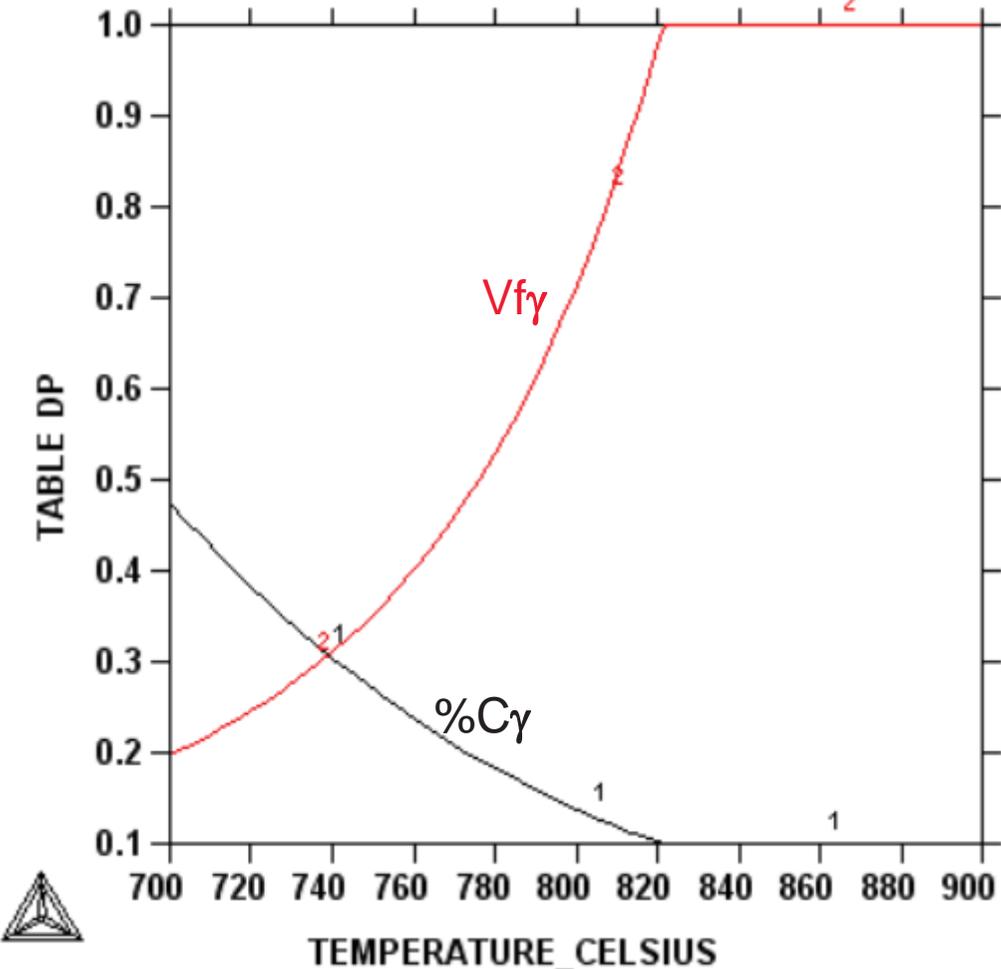
# Dual phase steels



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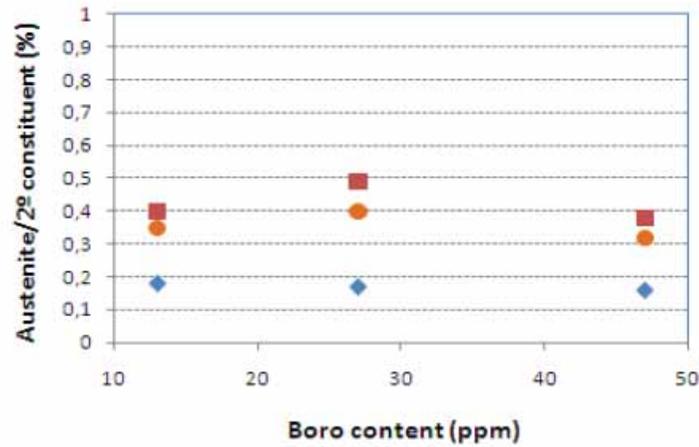


# Dual Phase constitution- Effect of B addition

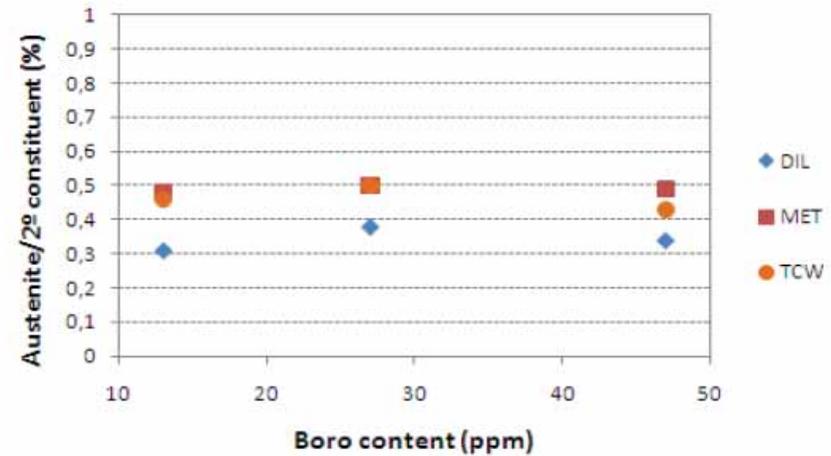


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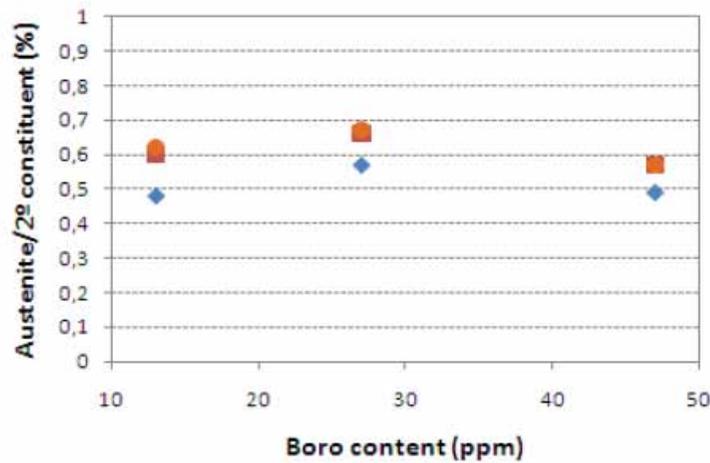
Murari, Costa e Silva, Avillez, 2015



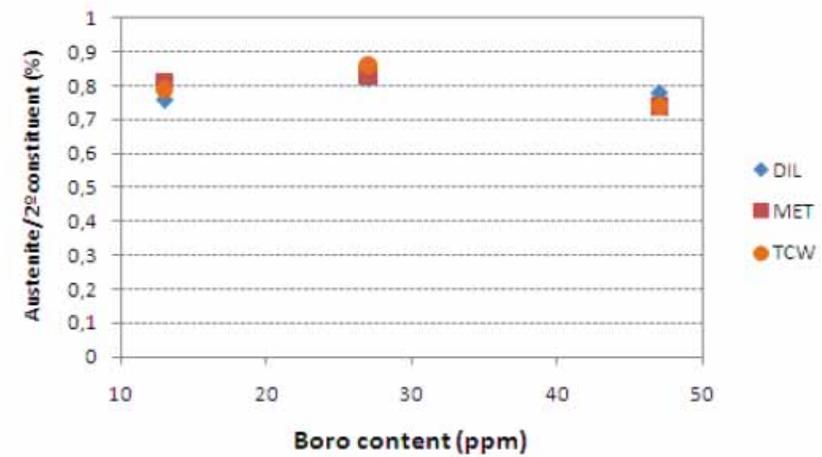
(a) 760°C



(b) 780°C



(c) 800°C



(d) 820°C



# Example: Spring steels, tire cord, bearings

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1955-80



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1955-80



41

1980-20...

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42

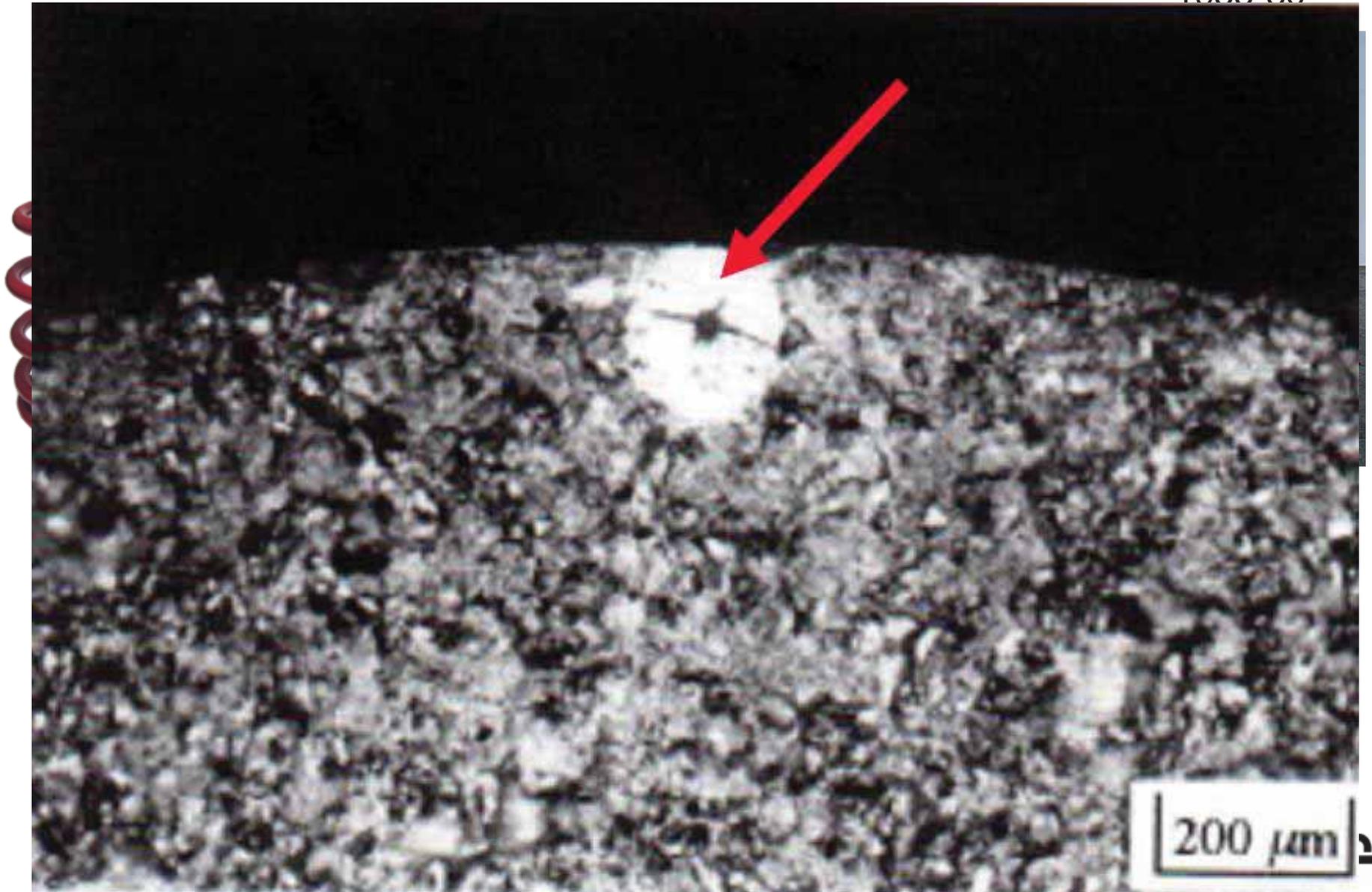
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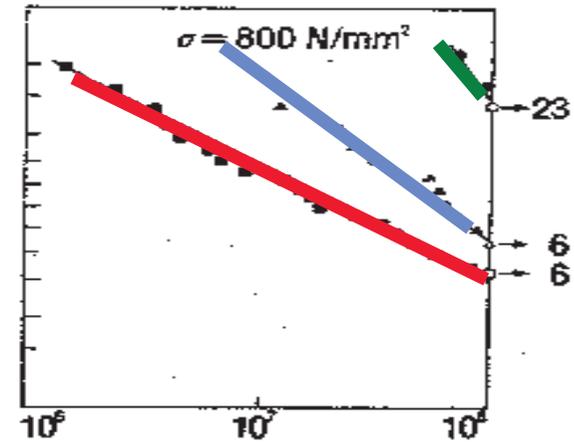
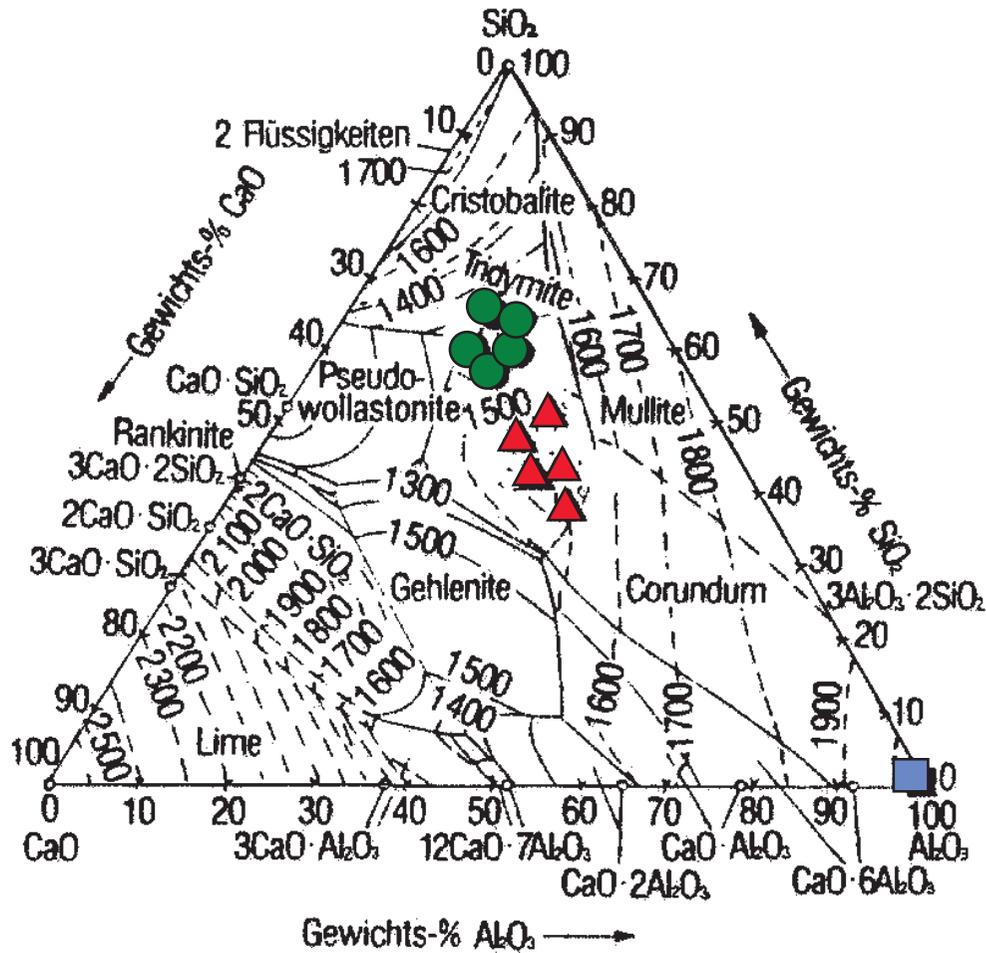
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# Inclusions vs. Fatigue

- SAE 9254 - Oshiro et al. 1989

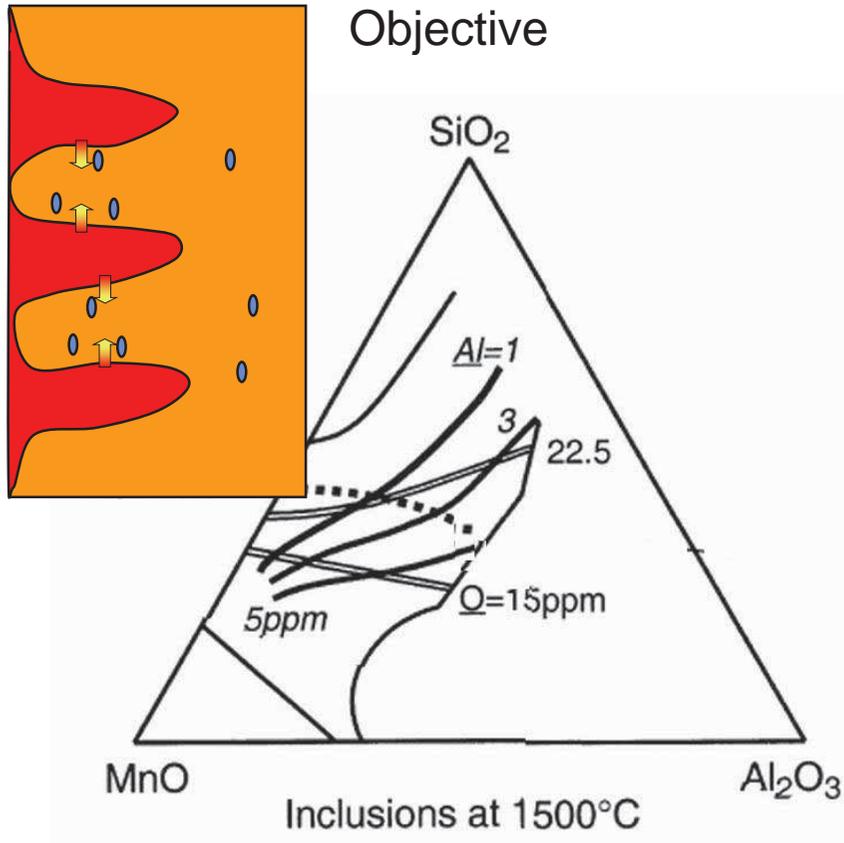


Number of cycles

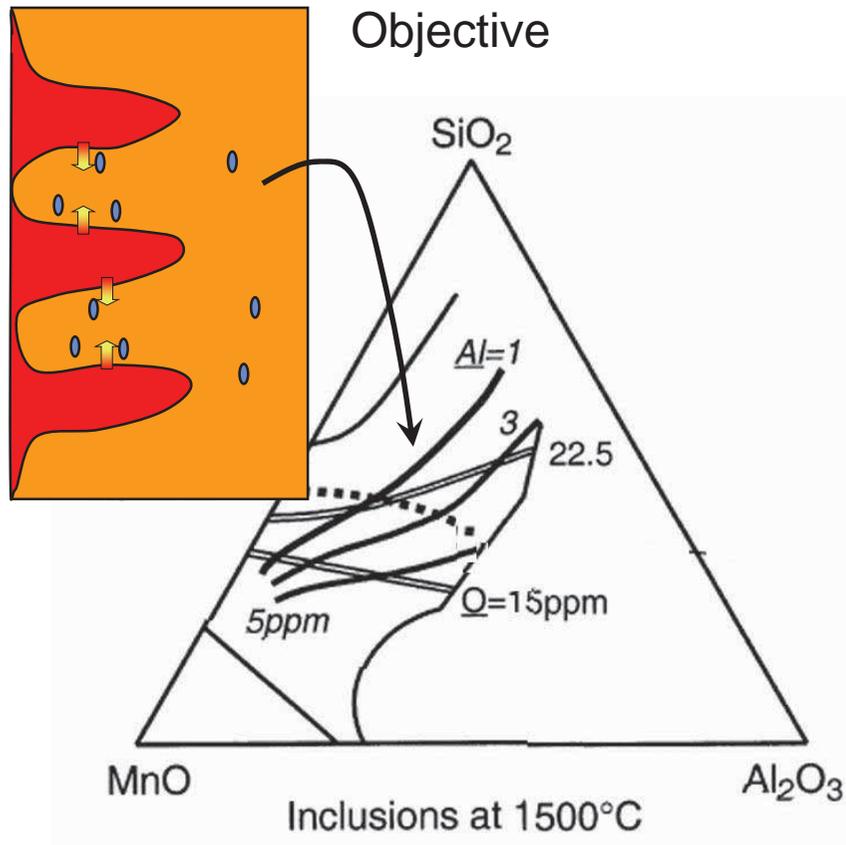
# The strategy in slag-metal equilibrium processing

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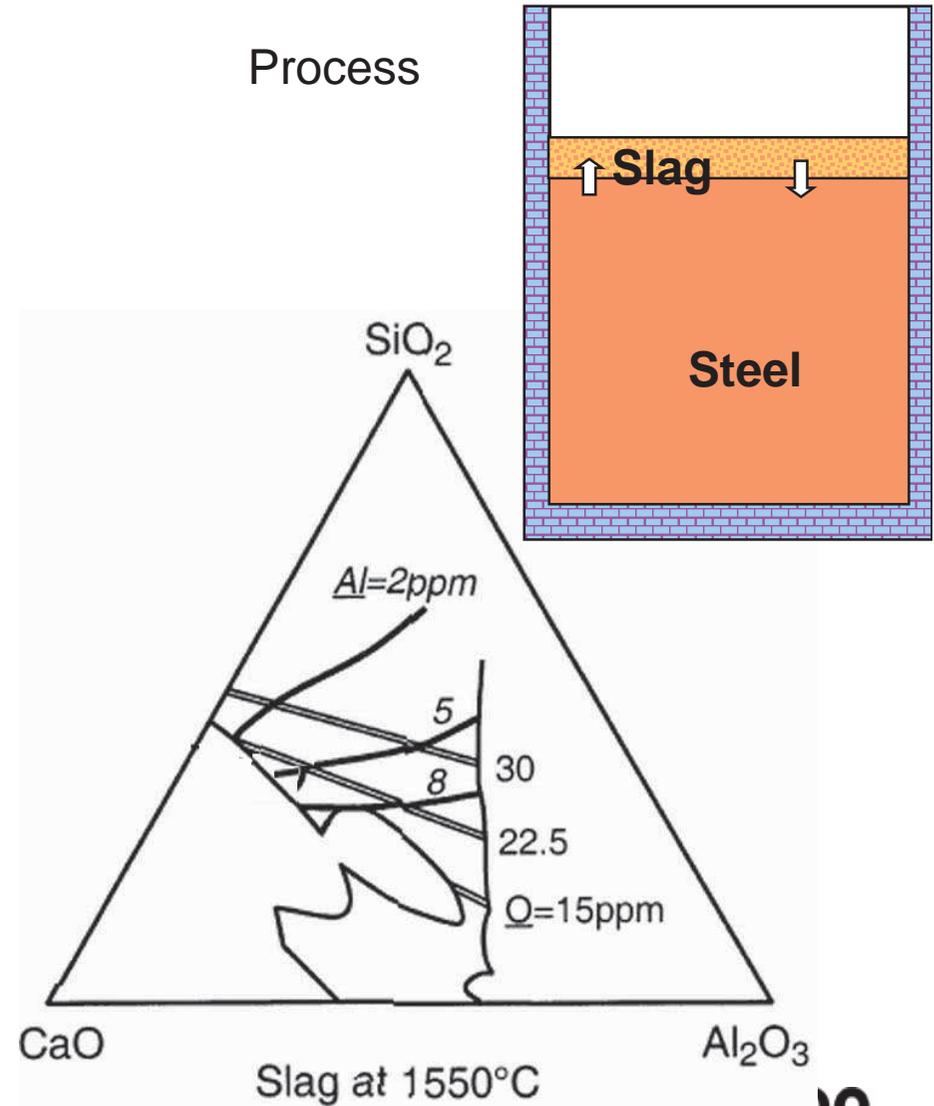
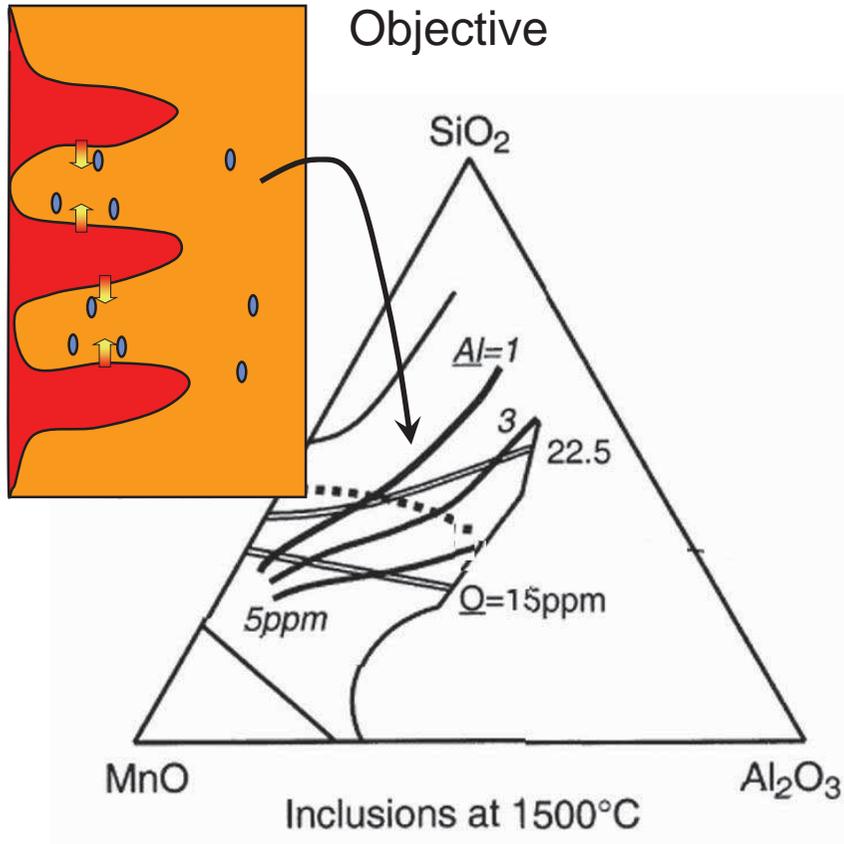
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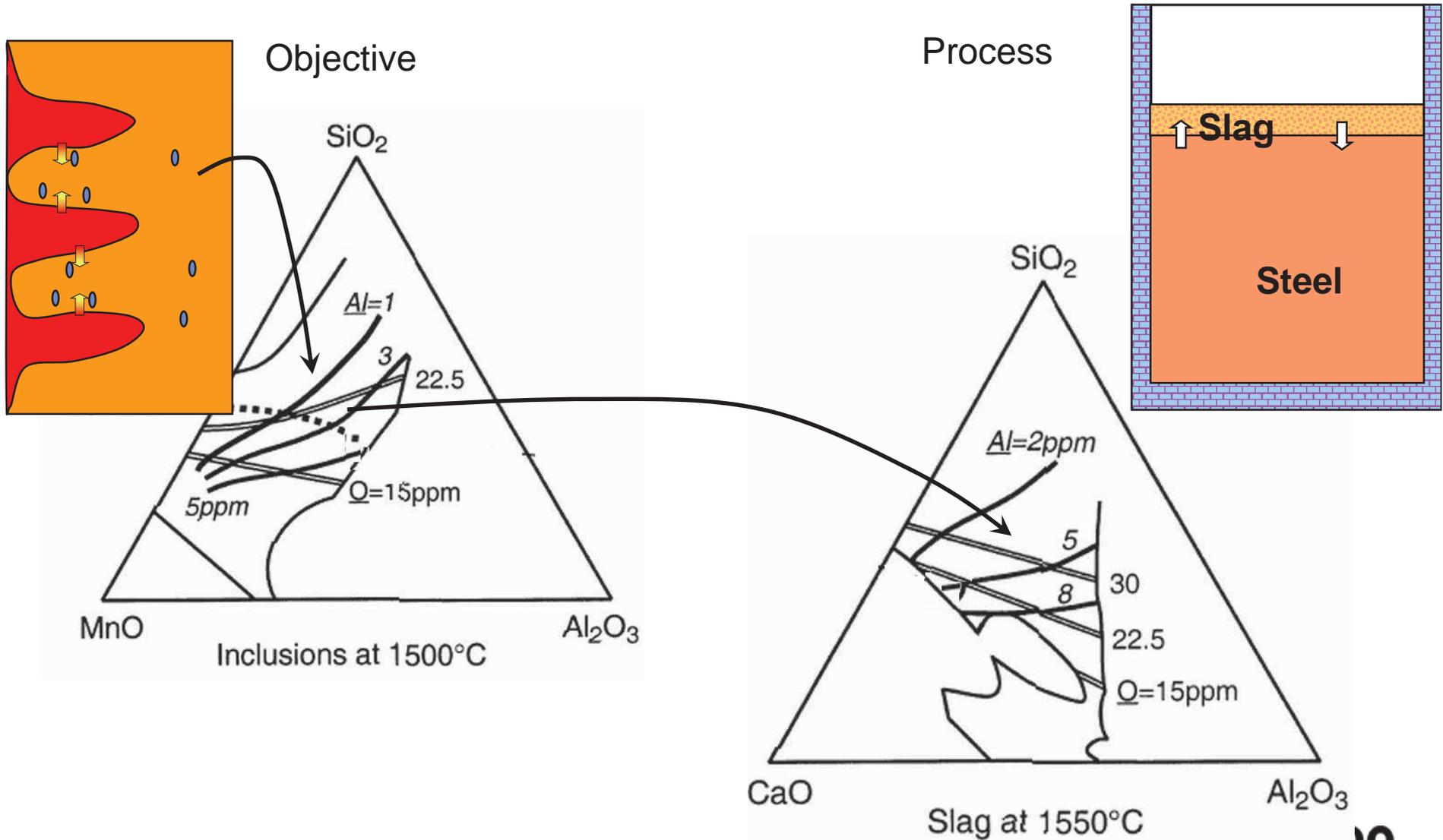
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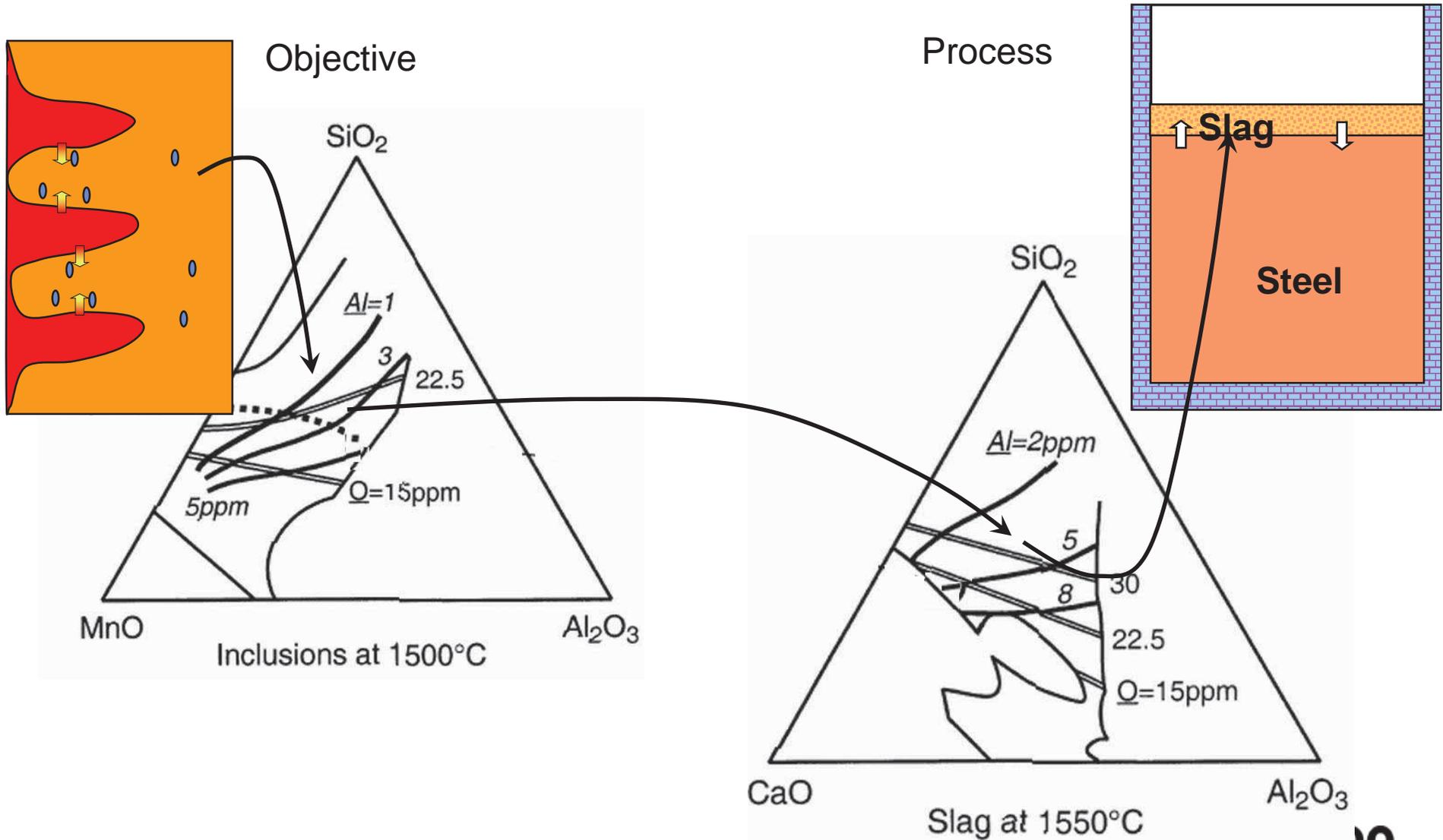
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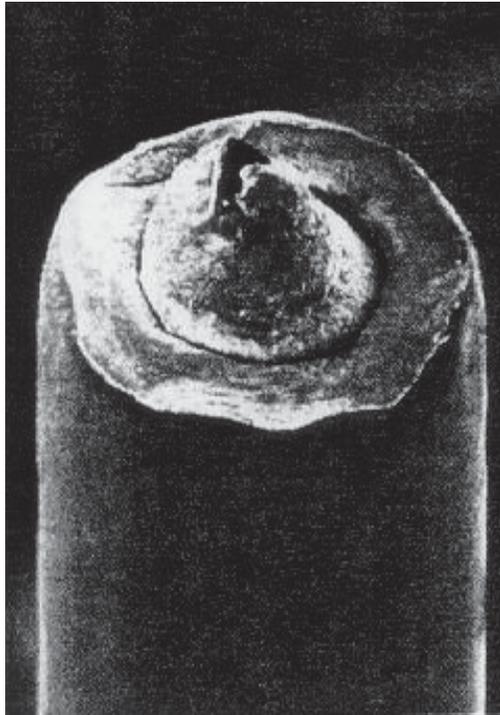


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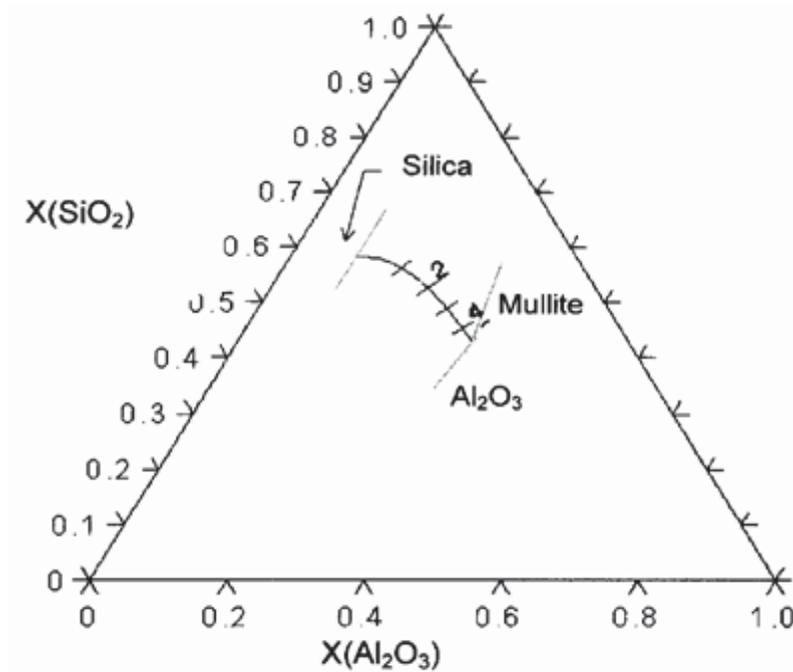


↔  
50  $\mu\text{m}$

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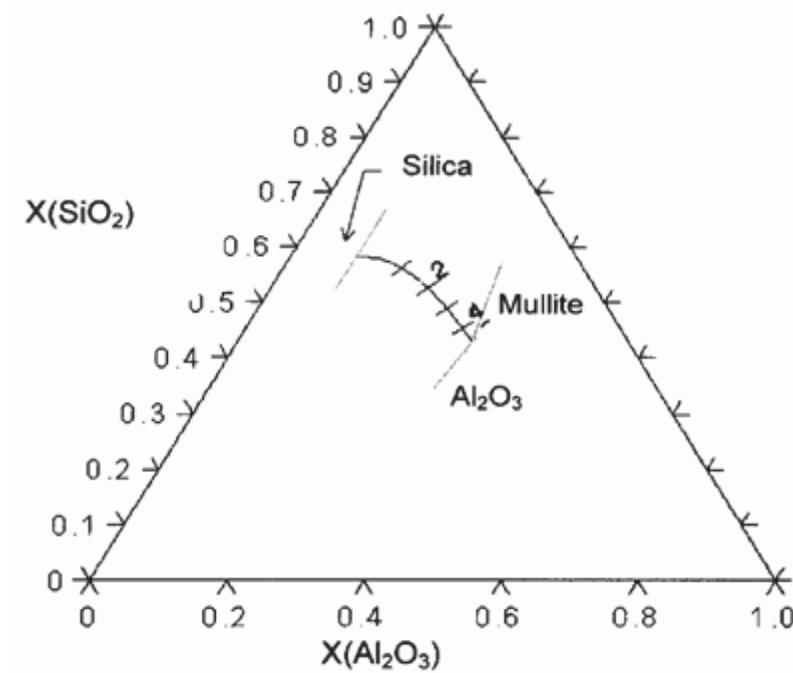
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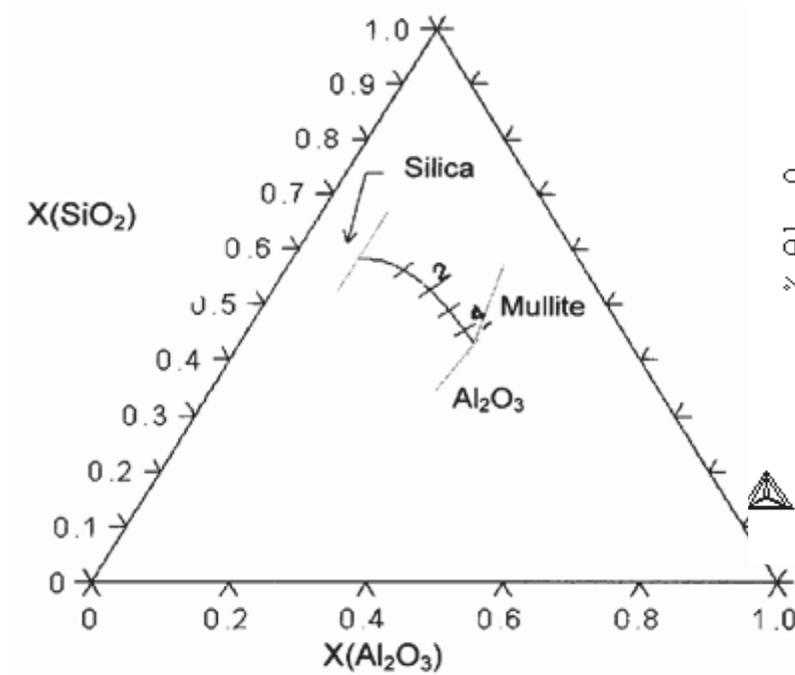
Steel C=0,8%, Mn=0.6%, Si=0,3%

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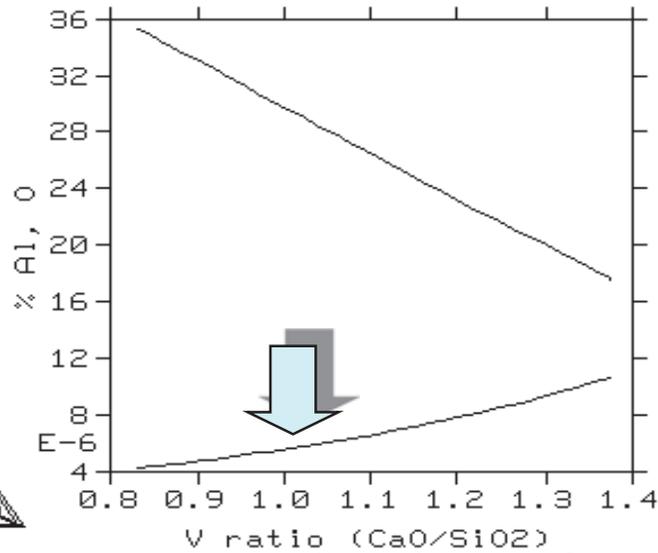
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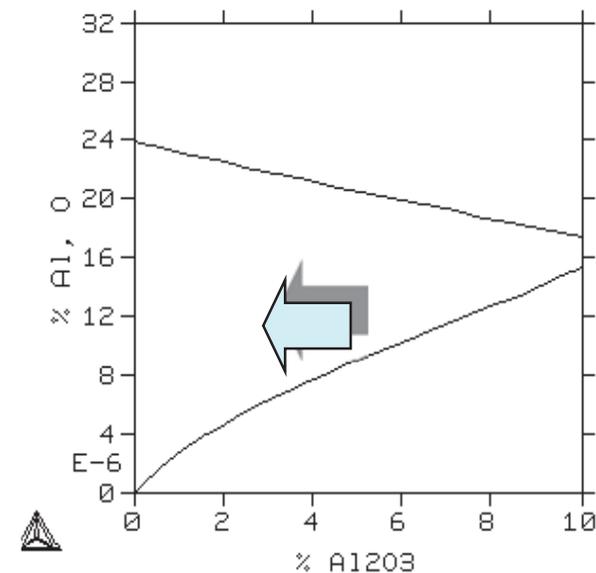


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Effect of slag basicity  $\% \text{Al}_2\text{O}_3=5$

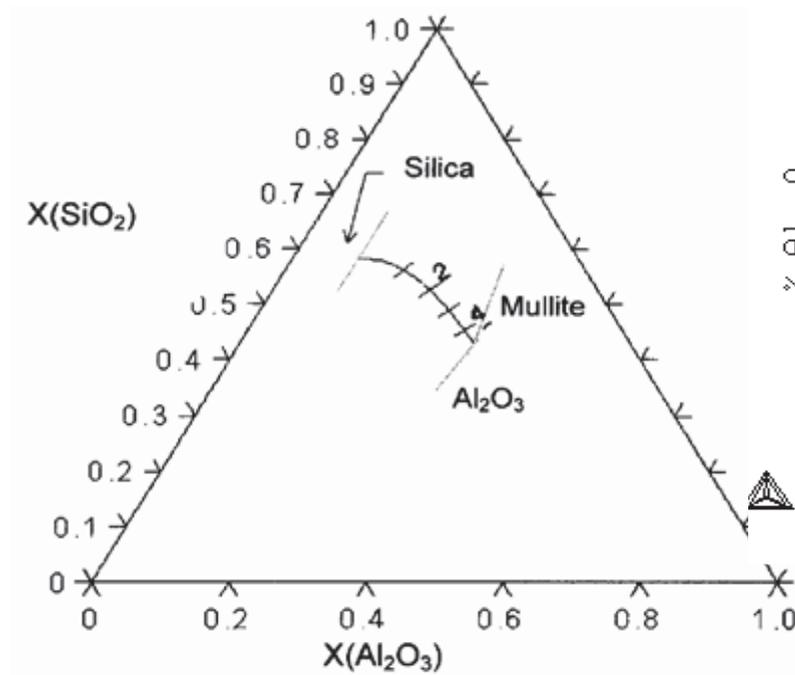


Effect of slag  $\text{Al}_2\text{O}_3$  content V=1.2



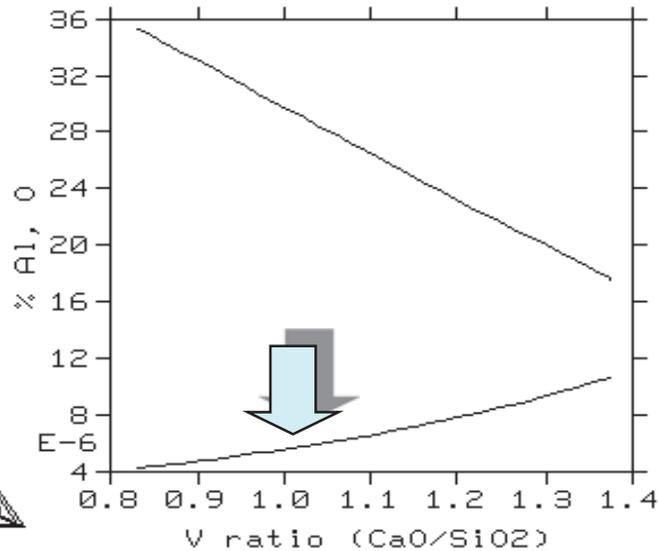
Oertel, Costa e Silva, 1999, Costa e Silva, 1999

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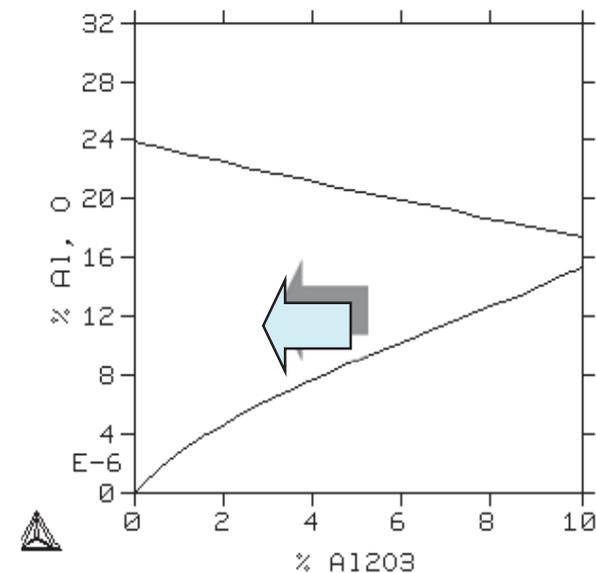


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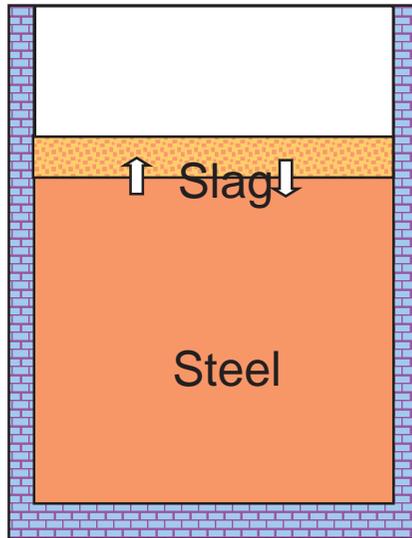


# Minimizing Al and O in bearing steels (52100 or 100Cr6)

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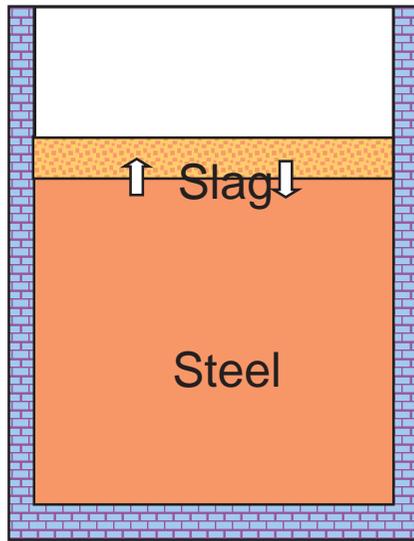
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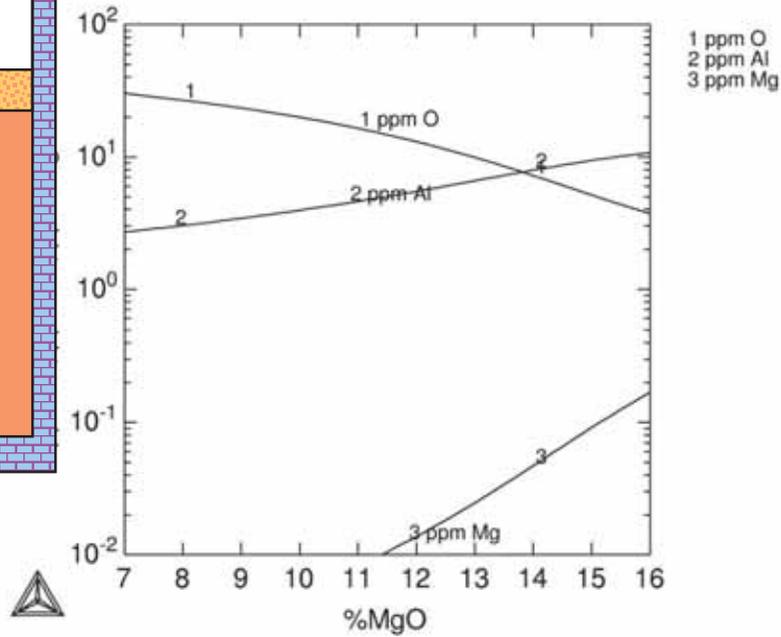


Slag metal equilibrium- Slag is  $\text{CaO}$ ,  $\text{MgO}$ ,  $\text{SiO}_2$ ,  $\text{Al}_2\text{O}_3$ , mostly

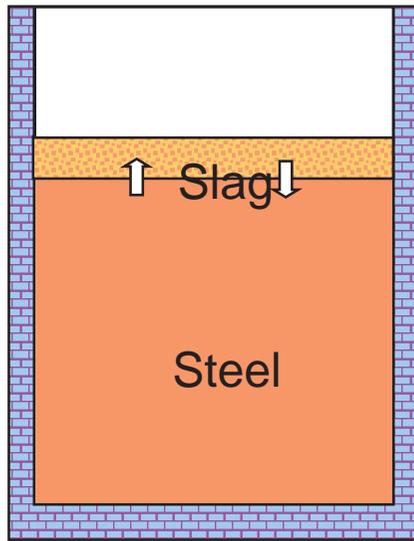
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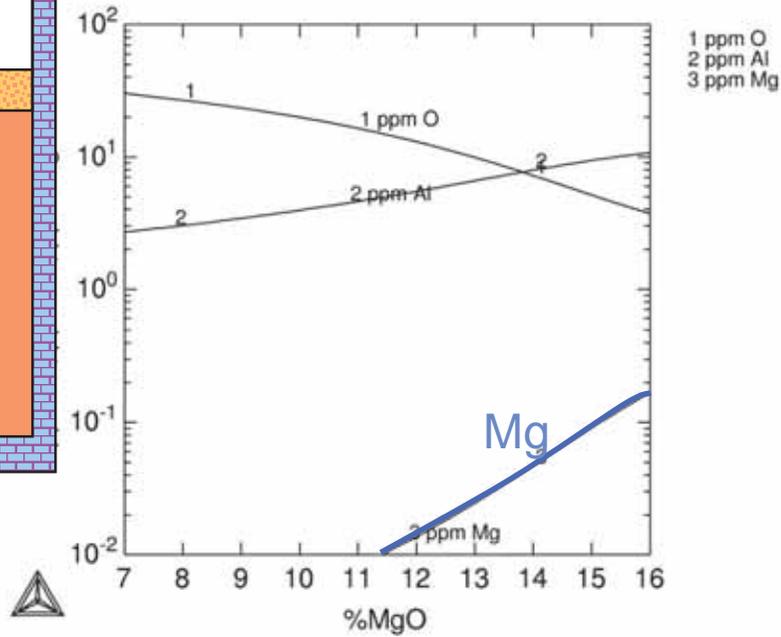
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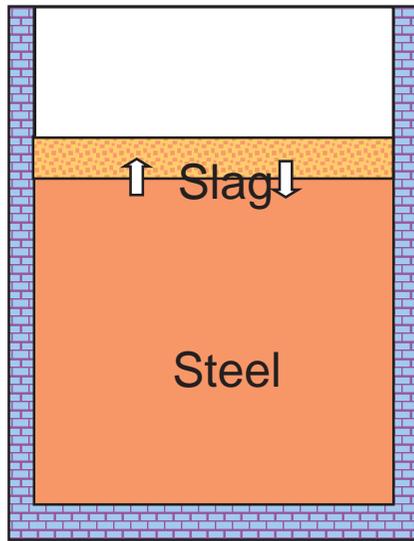
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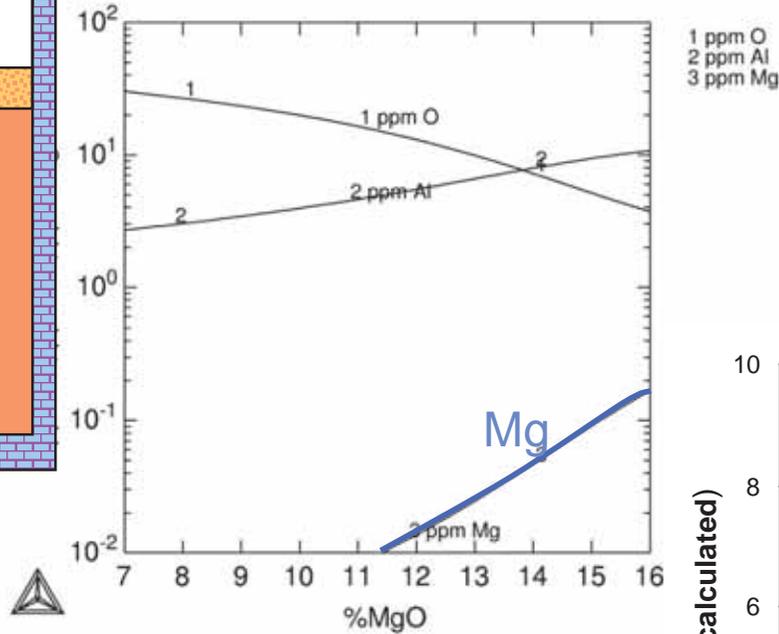
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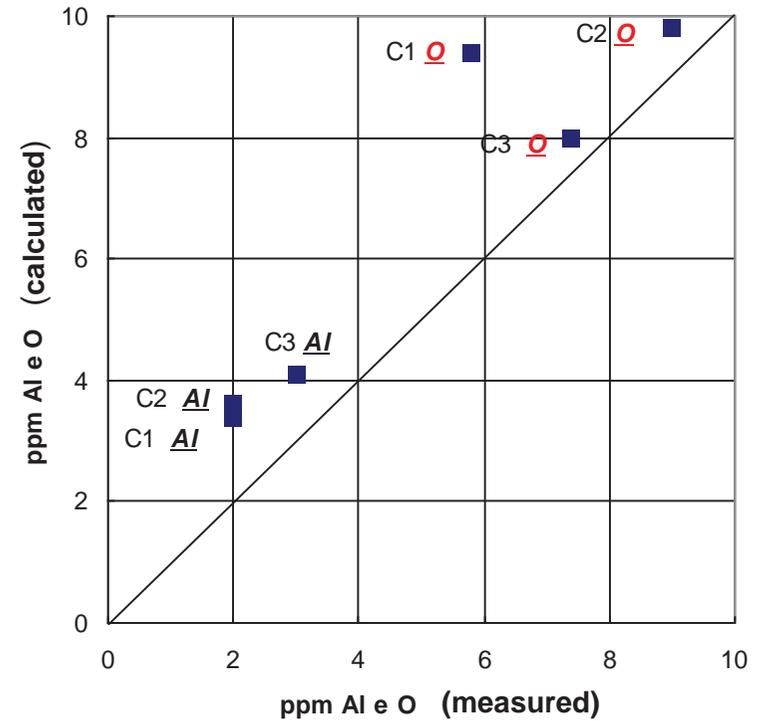
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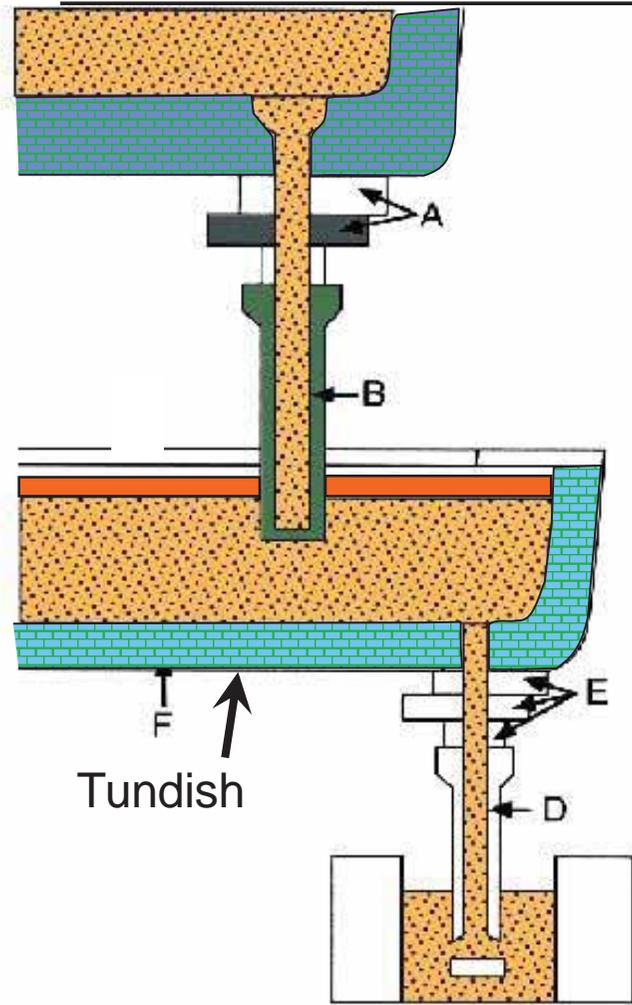
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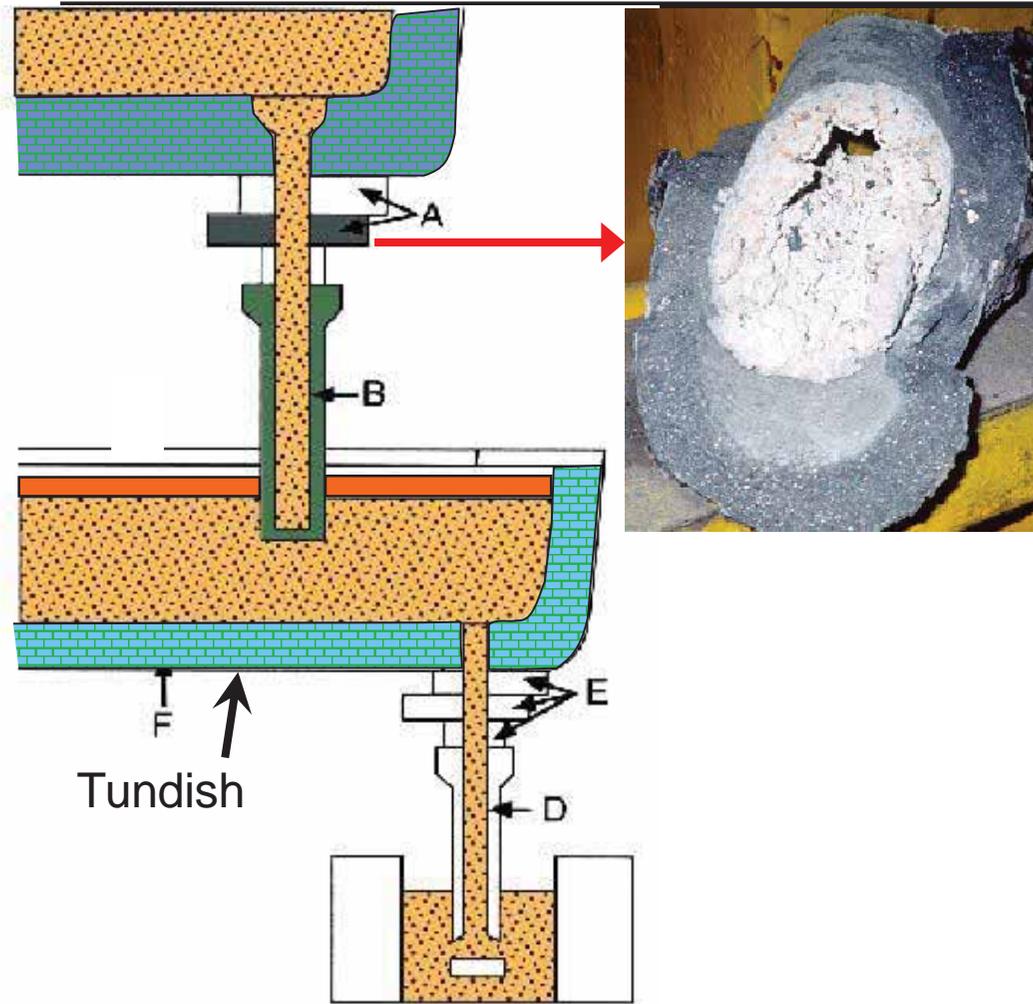
Costa e Silva, 2006, Costa e Silva, 2013



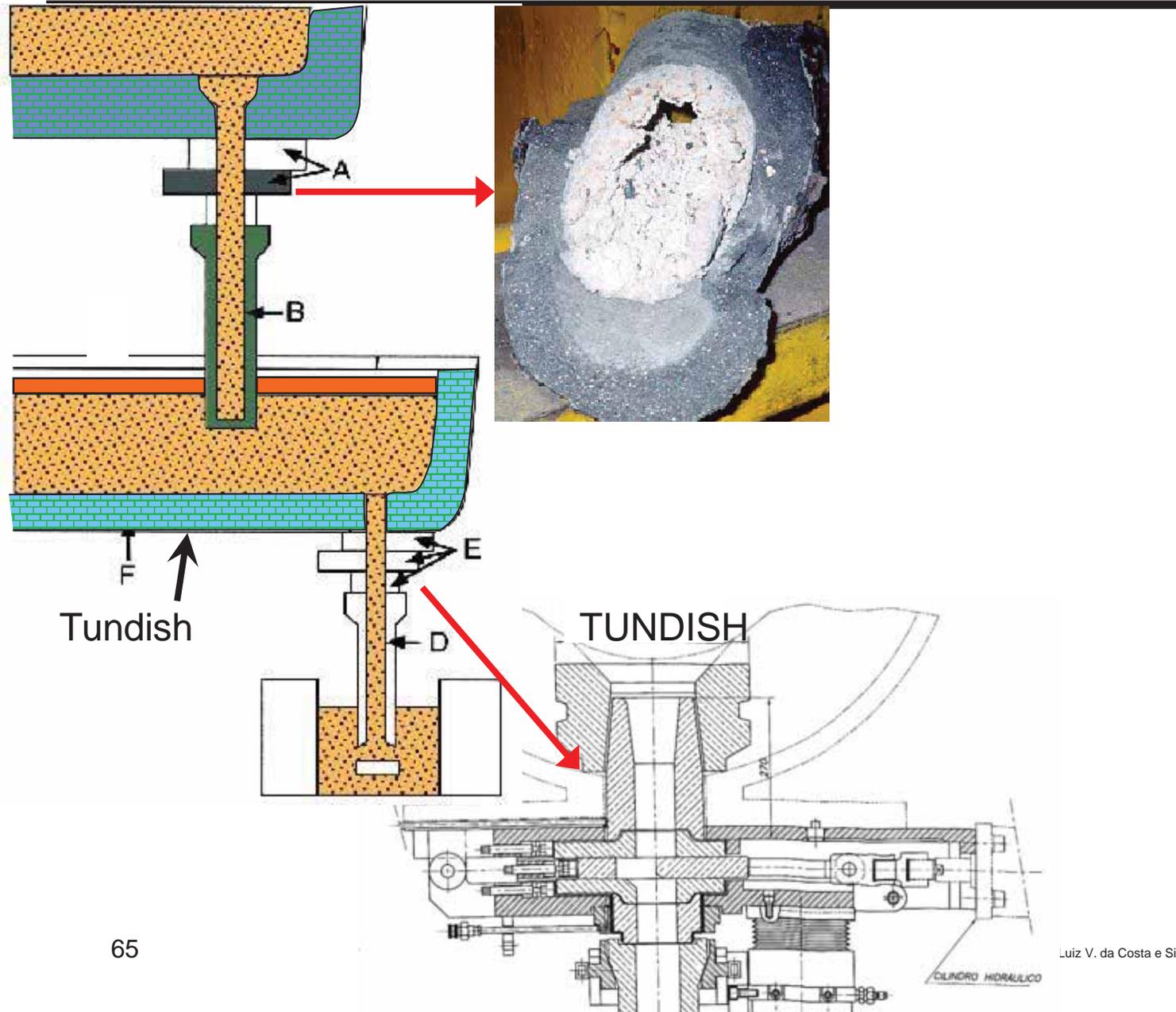
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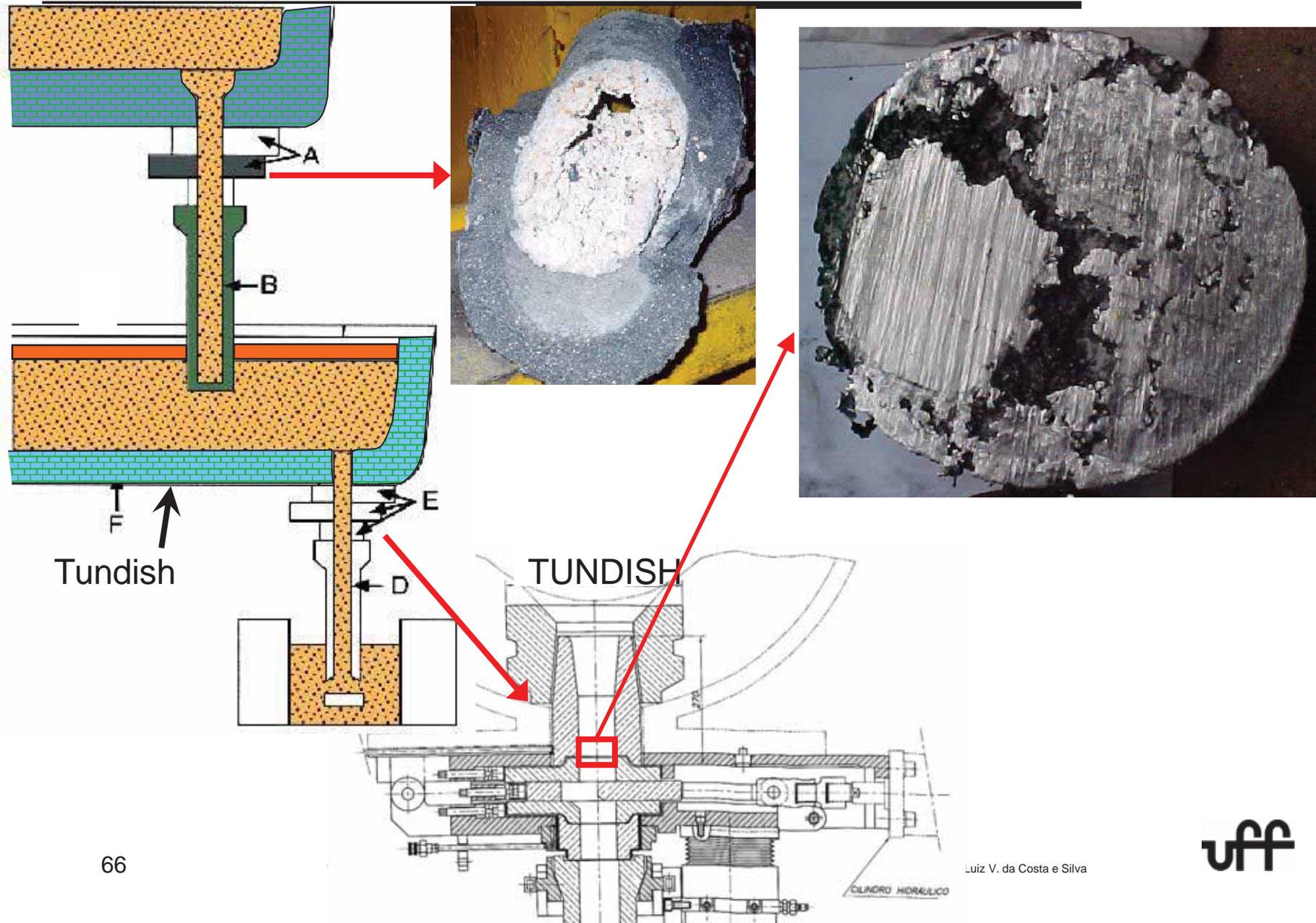
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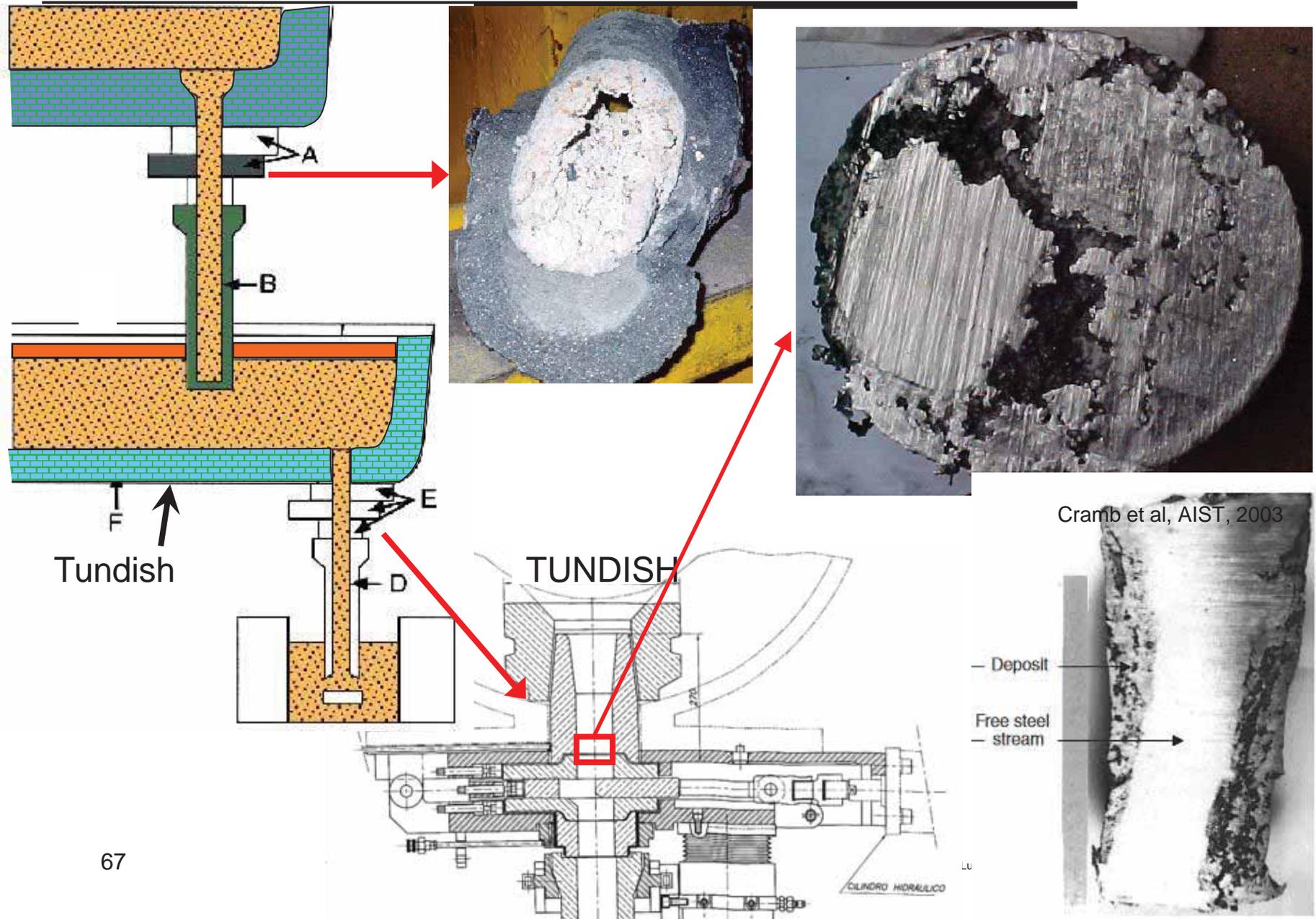
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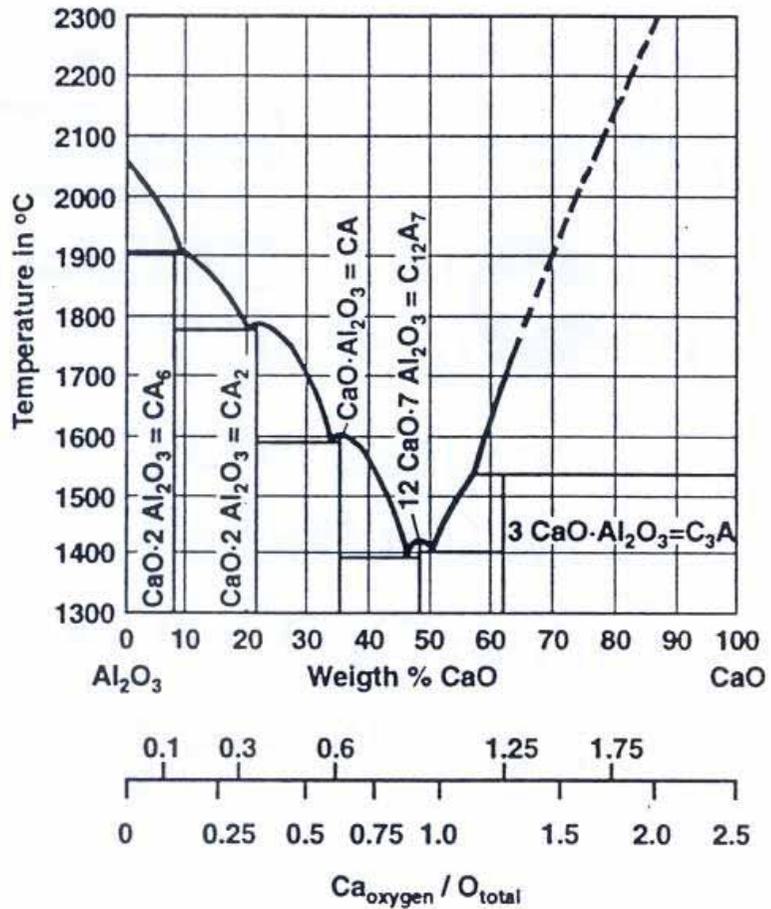


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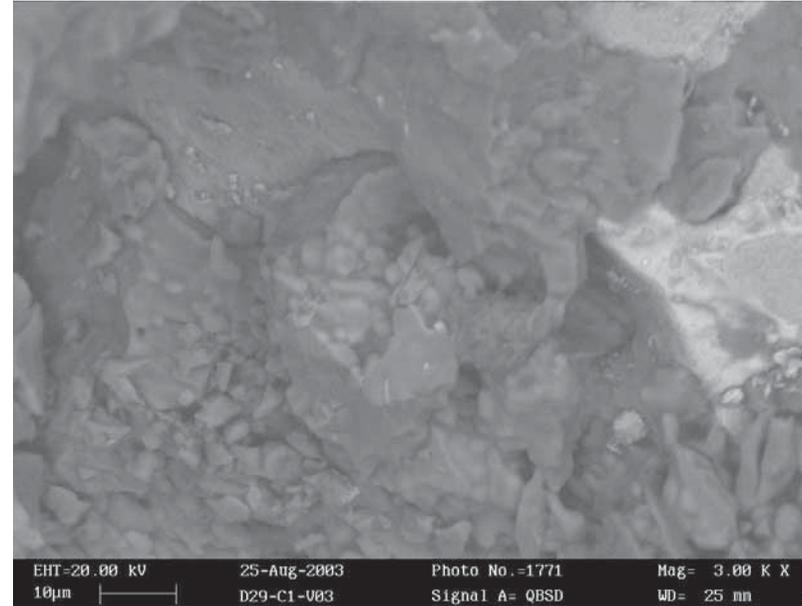
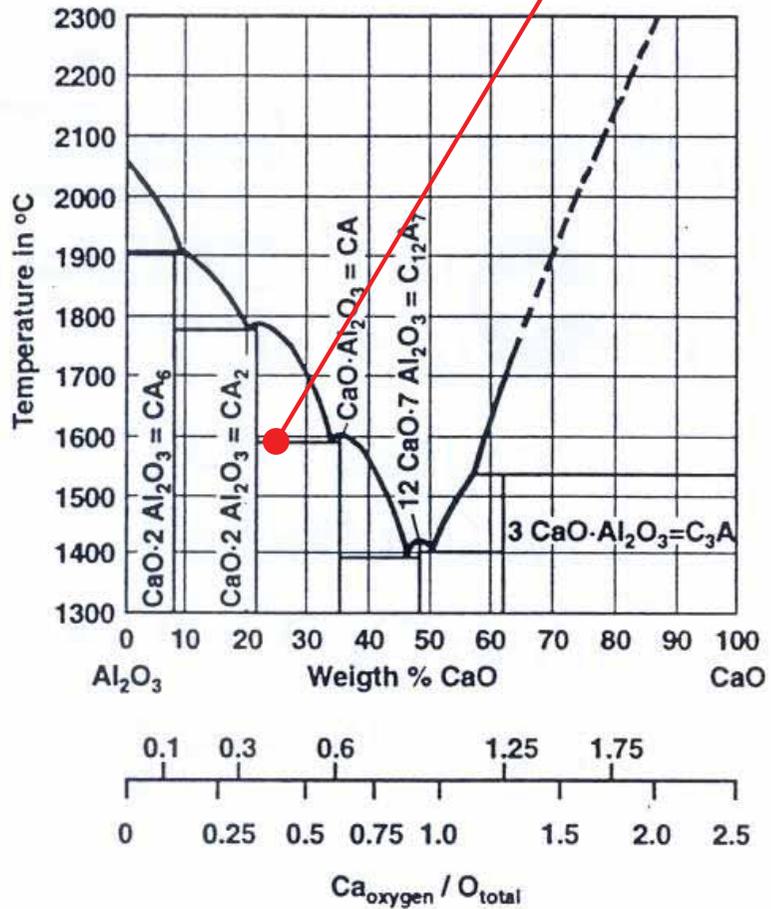
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## Semi-empirical- Increasing Ca additions



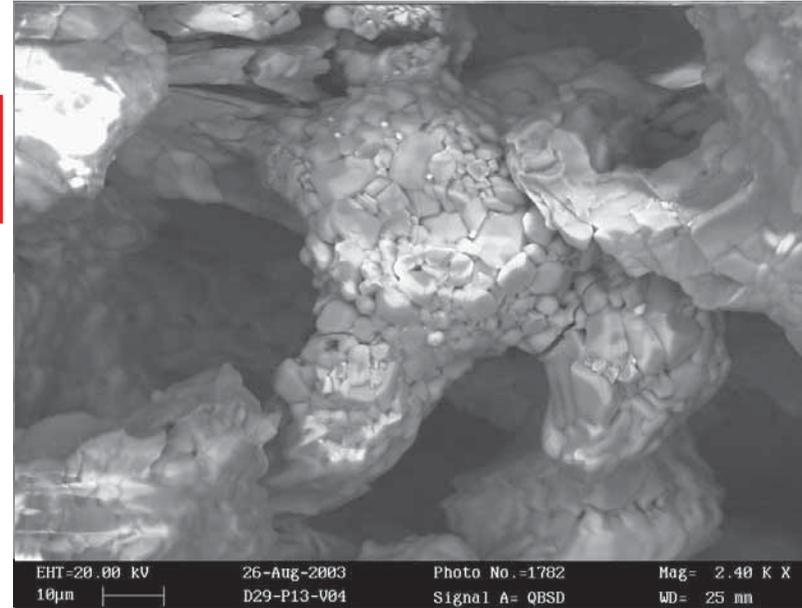
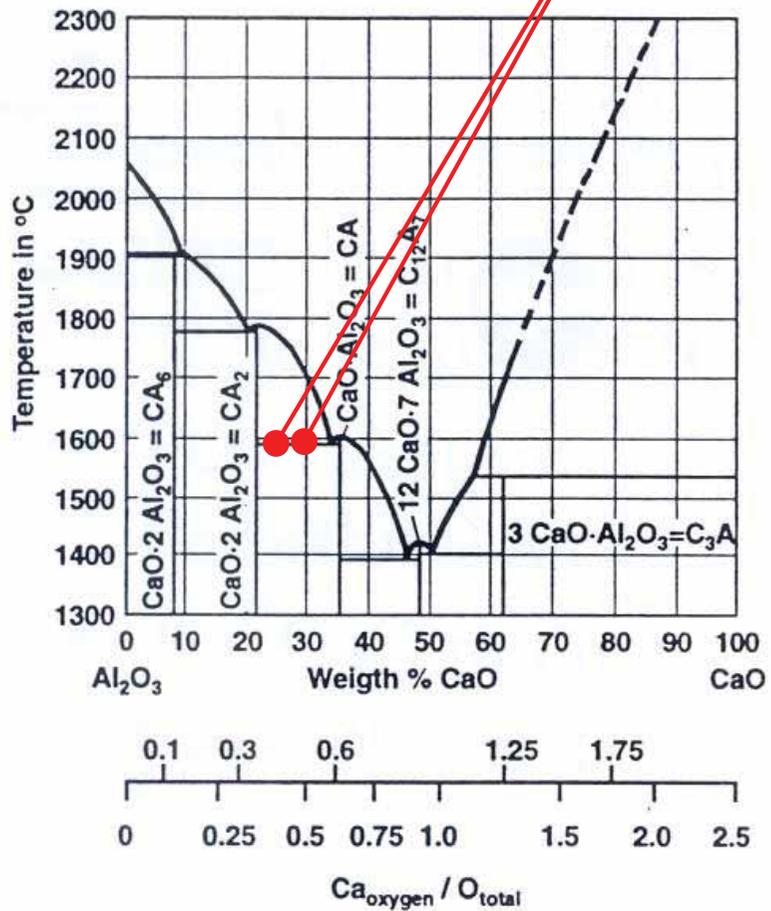
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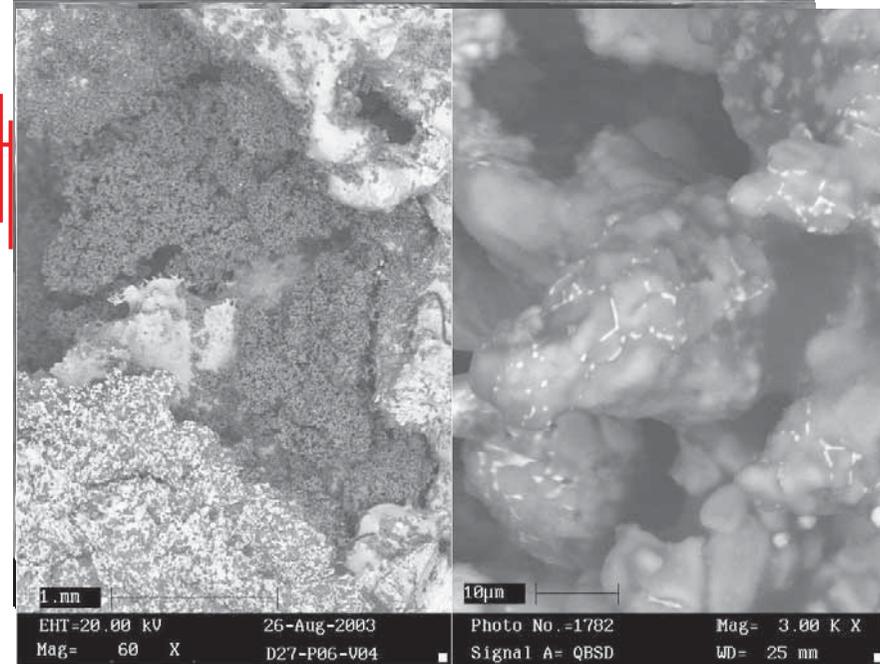
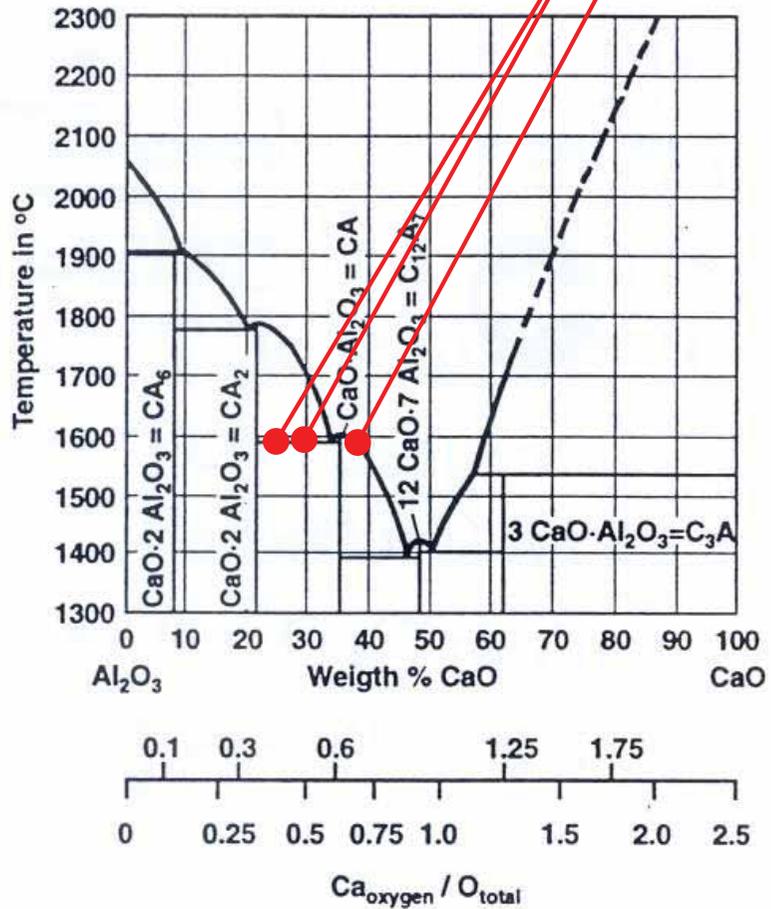
# “Classical Al<sub>2</sub>O<sub>3</sub> clogging”

## Semi-empirical- Increasing Ca additions



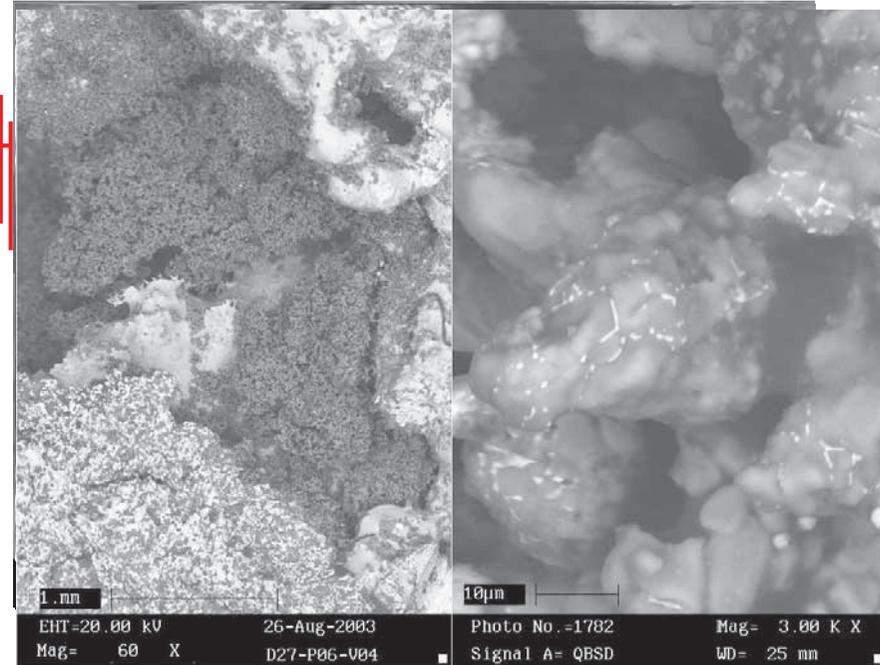
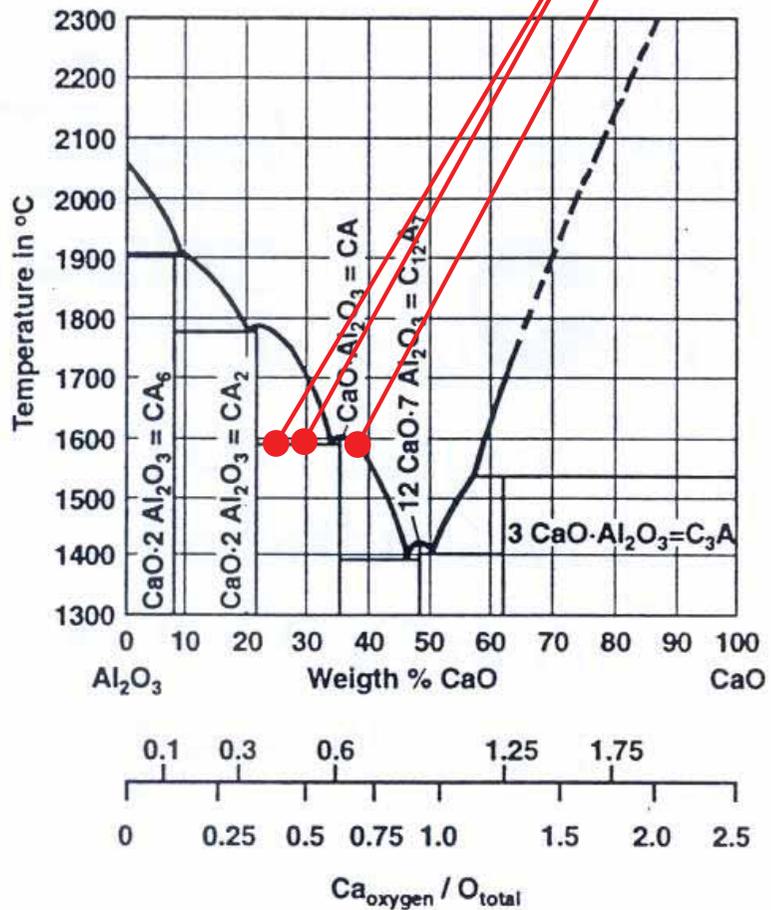
# “Classical Al<sub>2</sub>O<sub>3</sub> clogging”

## Semi-empirical- Increasing Ca additions



# “Classical Al<sub>2</sub>O<sub>3</sub> clogging”

## Semi-empirical- Increasing Ca additions

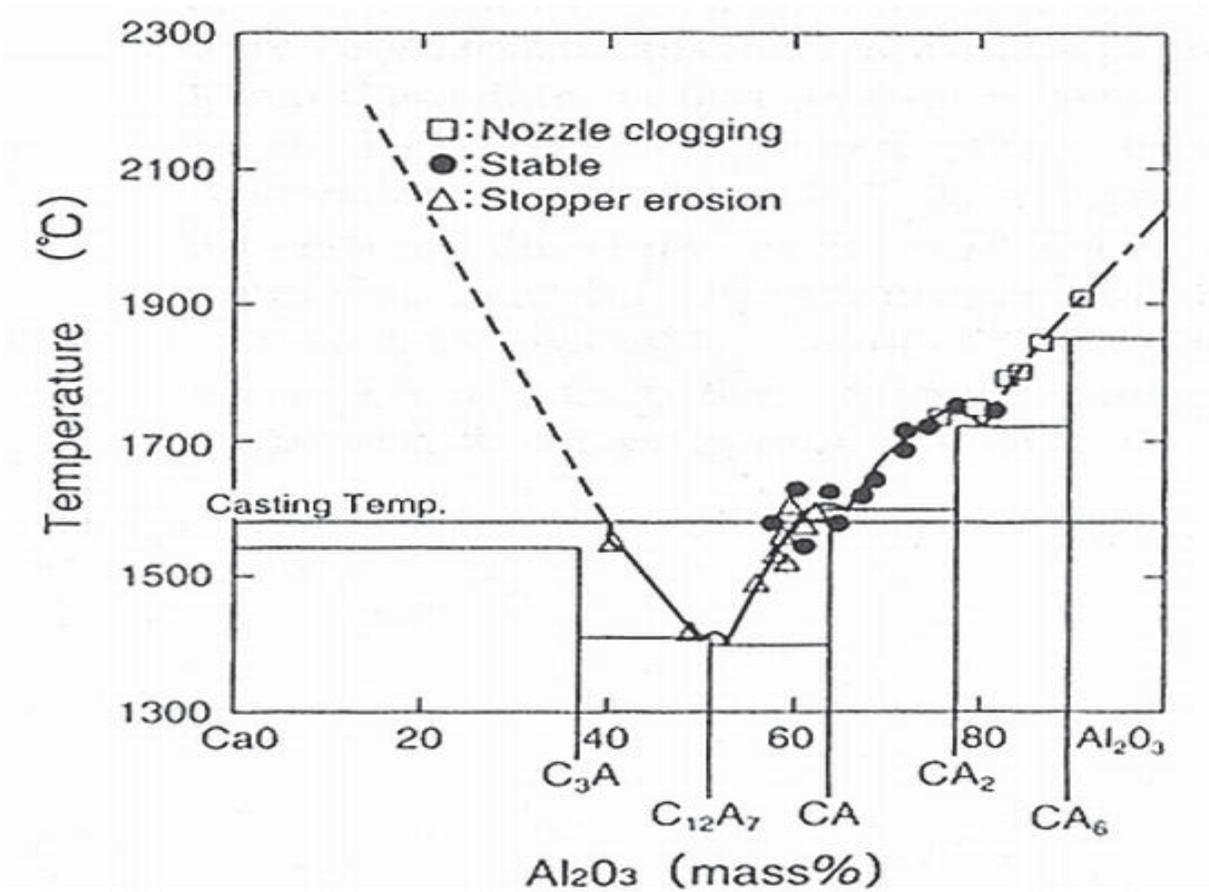


Courtesy AM Tubarão, 2006

Total O, S and T must be constant

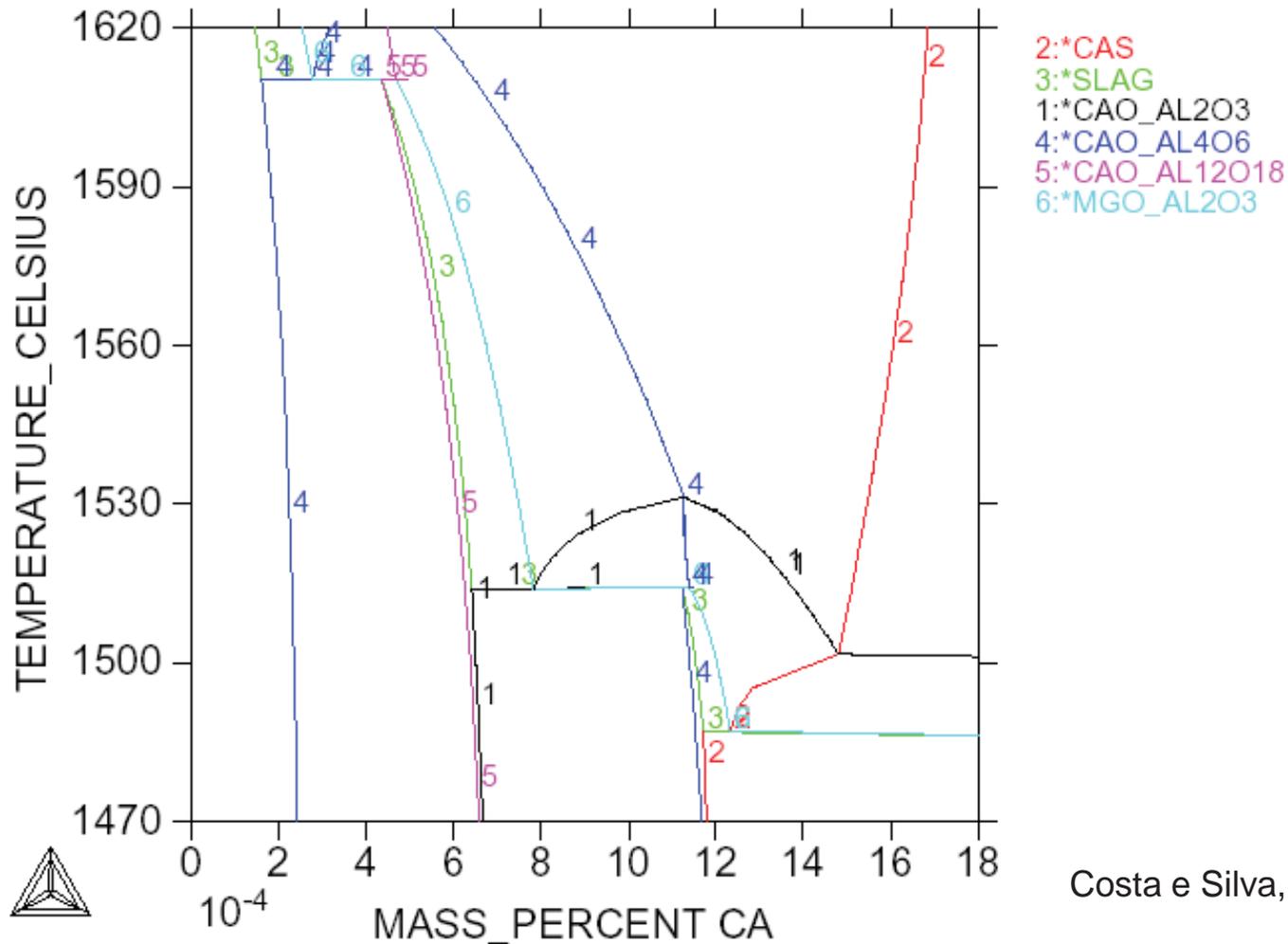
# Tailoring the Steel composition for Liquid Inclusions (“castability window”)

CaO-Al<sub>2</sub>O<sub>3</sub> pseudo-binary



# Tailoring the Steel composition for Liquid Inclusions (“castability window”)

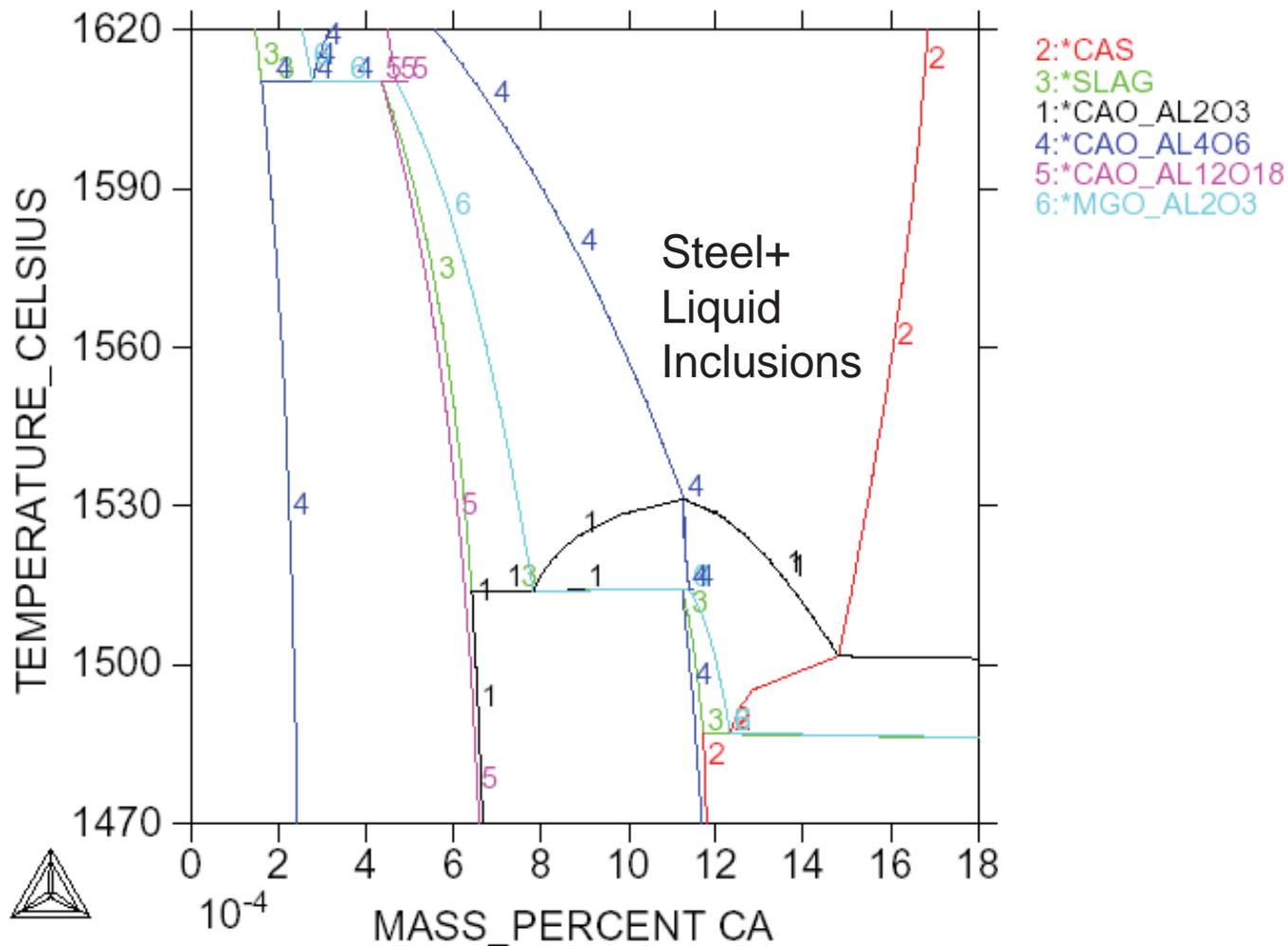
DATABASE:SLAG2  
 P=1E5, N=1, W(AL)=2E-4,W(MG)=2.5E-7, W(O)=2E-5, W(S)=2E-4, W(SI)=2E-3;



Costa e Silva, 2006, 2007

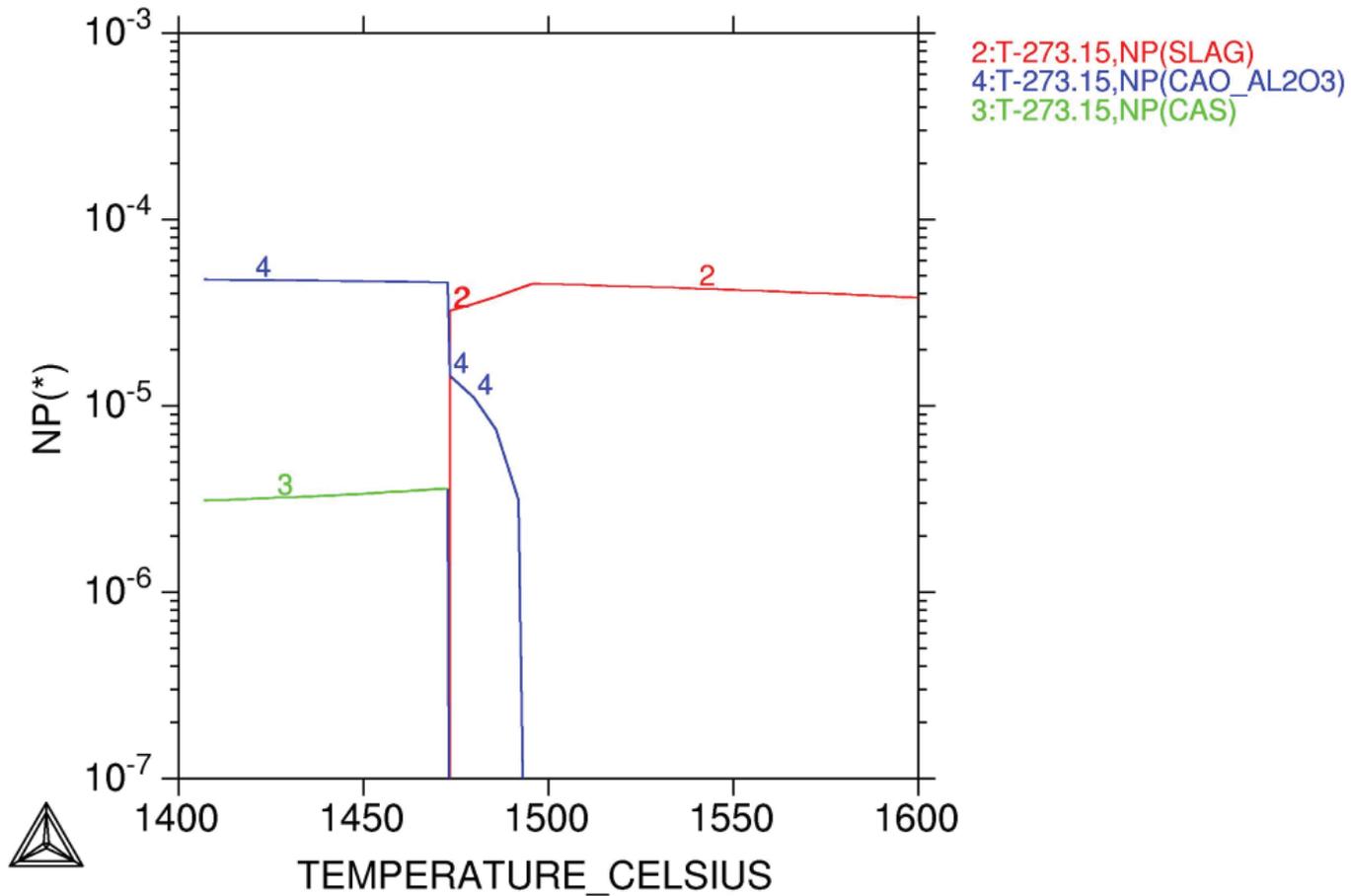
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DATABASE:SLAG2  
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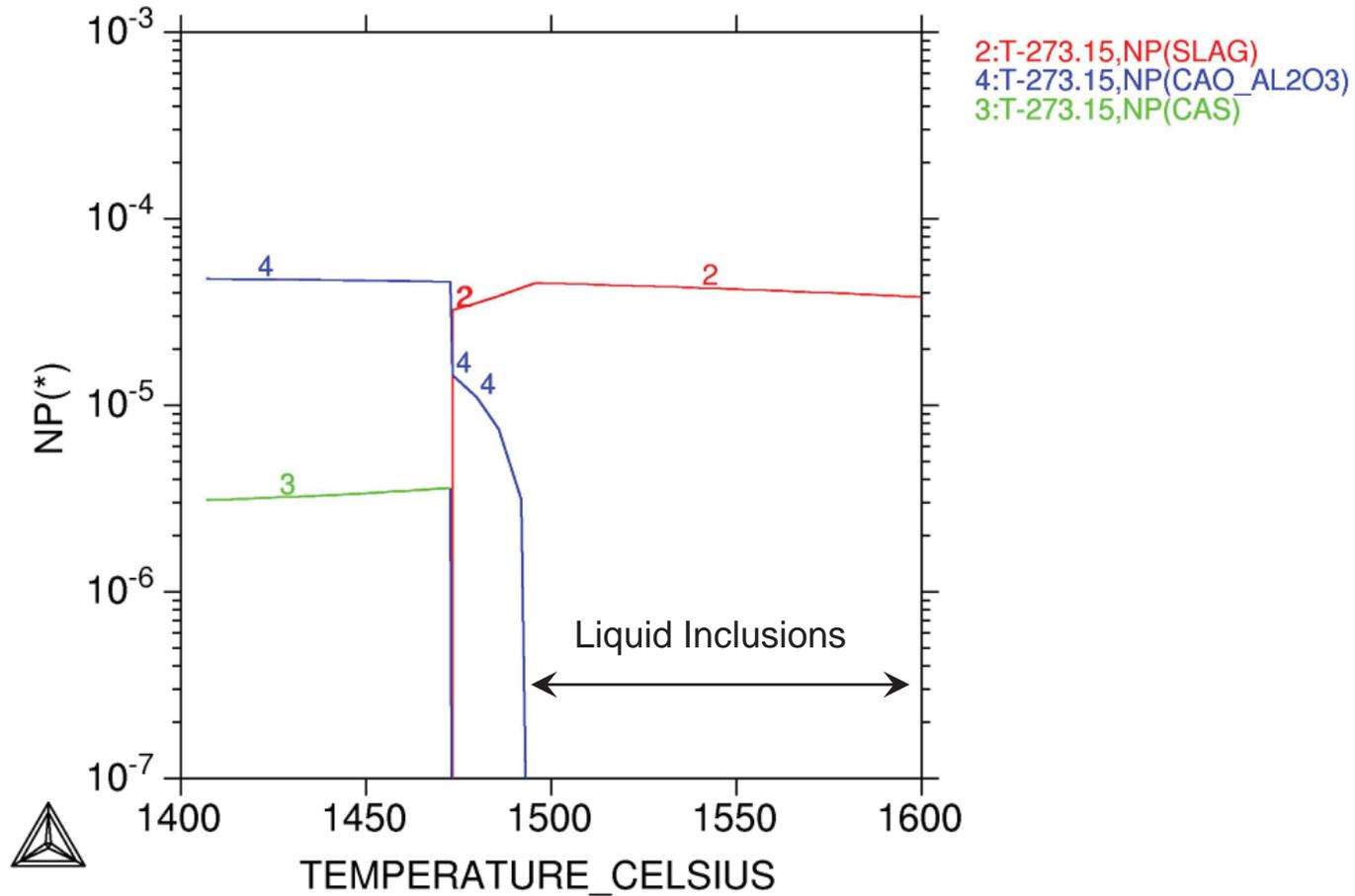
# Simple and direct way:

**Example %Al=0,02, Ca=6ppm, O=8ppm, %S=0,007,%Si=0,1**



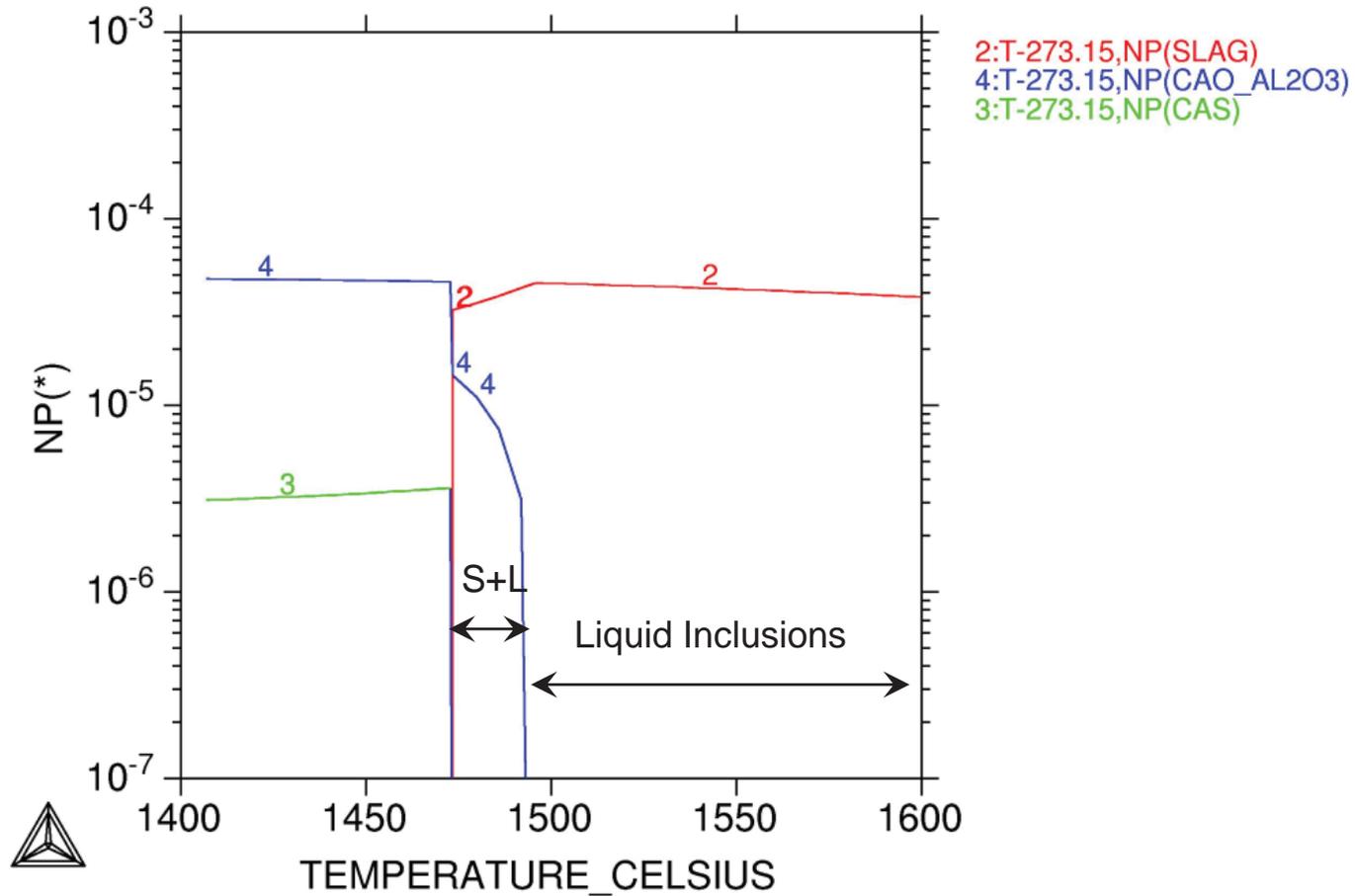
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**Example %Al=0,02, Ca=6ppm, O=8ppm, %S=0,007,%Si=0,1**



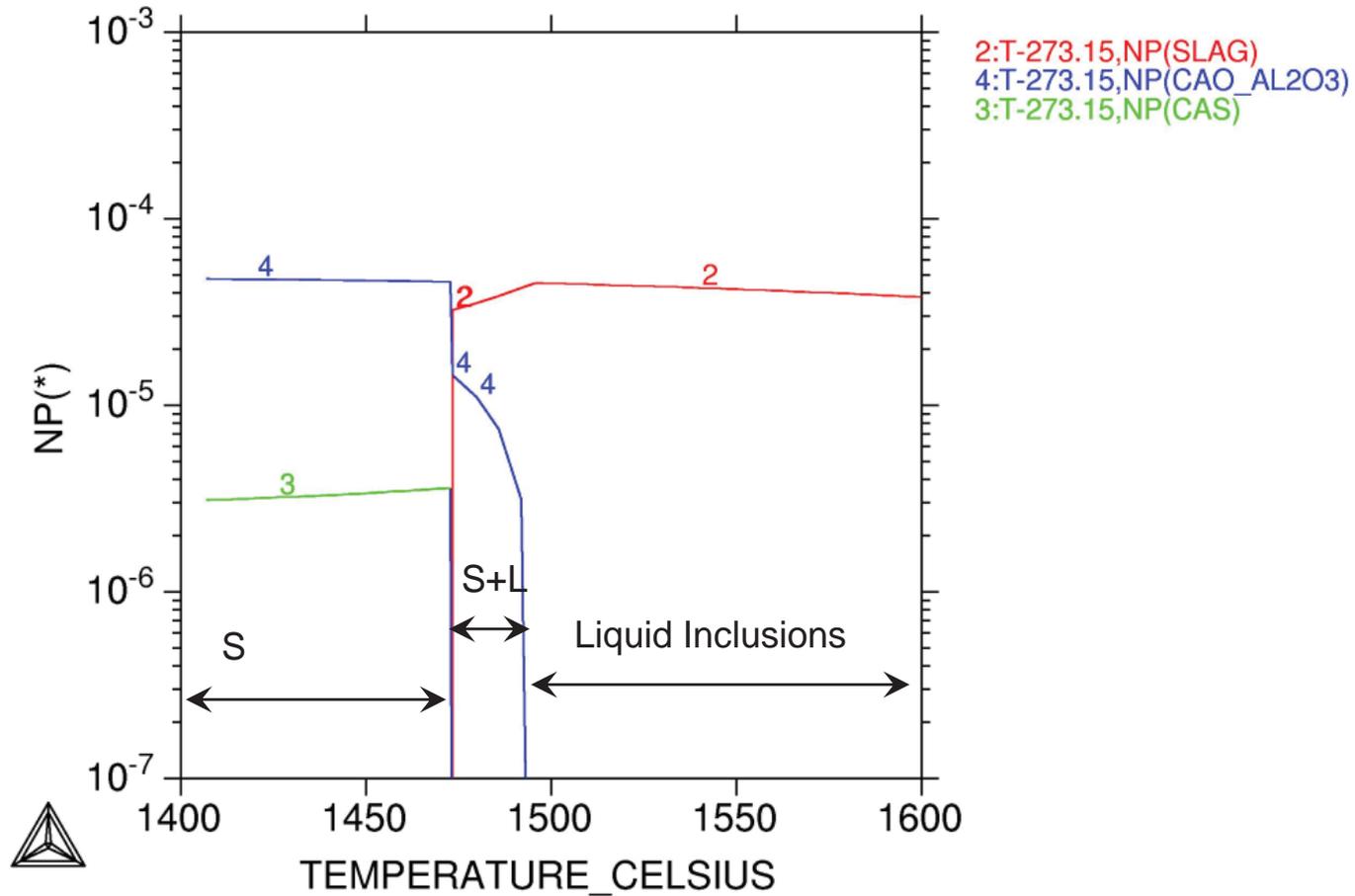
# Simple and direct way:

**Example %Al=0,02, Ca=6ppm, O=8ppm, %S=0,007,%Si=0,1**

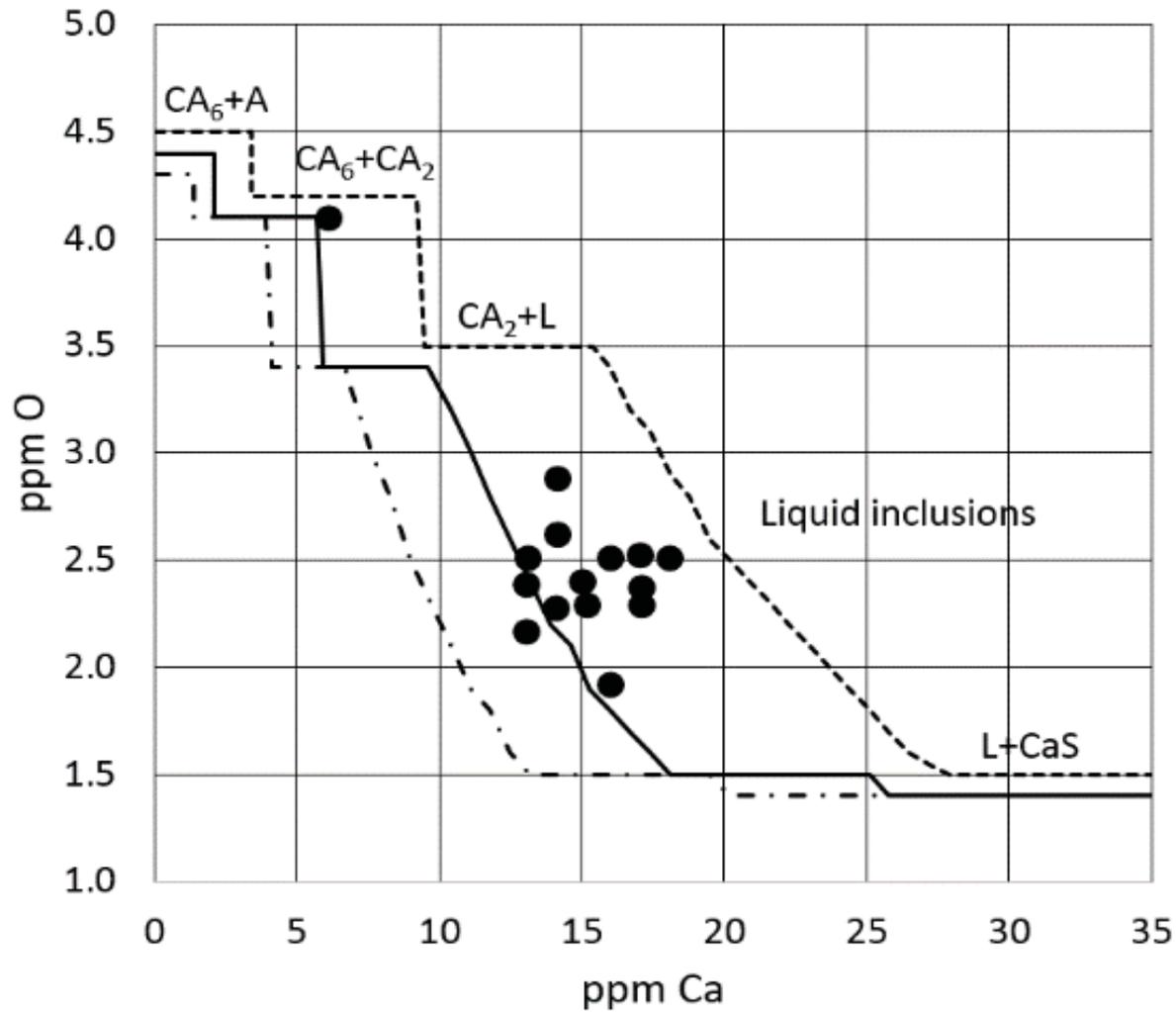


# Simple and direct way:

**Example %Al=0,02, Ca=6ppm, O=8ppm, %S=0,007,%Si=0,1**



# Monitoring Ca injection via soluble oxygen (CELOX®)



Mercier, 2014, Kattner et al, to be published, 2016

# Spinel clogging in Si-Mn long products - Ladle metallurgy

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# Spinel clogging in Si-Mn long products - Ladle metallurgy

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1. Low S- High B<sub>2</sub>  
Good de-ox (low  
*μ O steel ) O steel eel O steel O steel*  
)

# Spinel clogging in Si-Mn long products - Ladle metallurgy

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## 1. Protect refractory- **MgO**

# Spinel clogging in Si-Mn long products - Ladle metallurgy

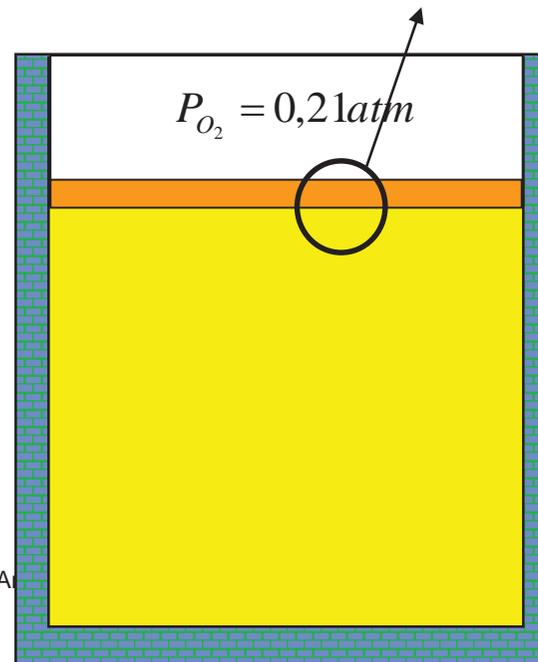
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1. Protect refractory- **MgO**
2. Fluid slag and slag from EAF - **significant %Al<sub>2</sub>O<sub>3</sub>**

# Spinel clogging in Si-Mn long products - Ladle metallurgy

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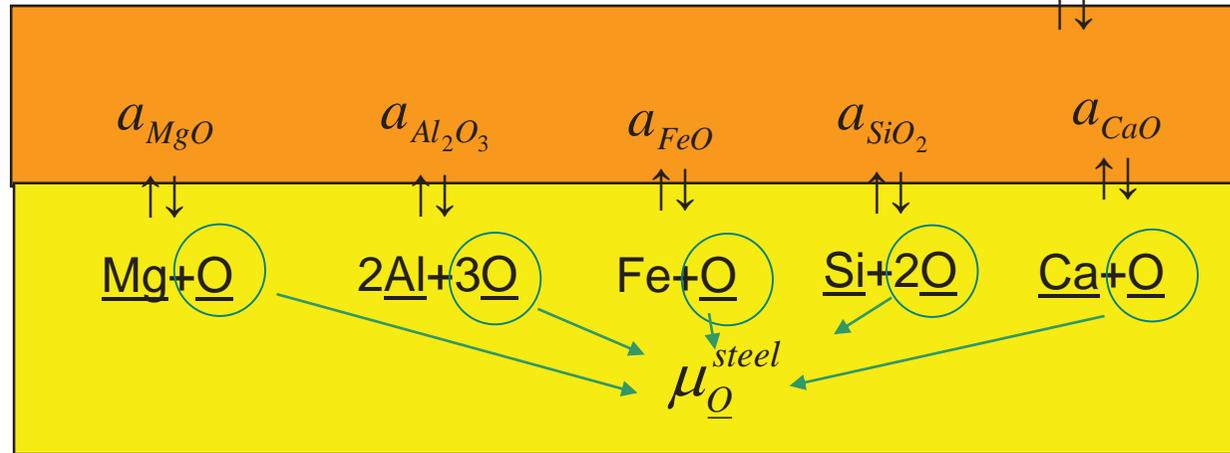
1. Protect refractory- **MgO**
2. Fluid slag and slag from EAF - **significant %Al<sub>2</sub>O<sub>3</sub>**



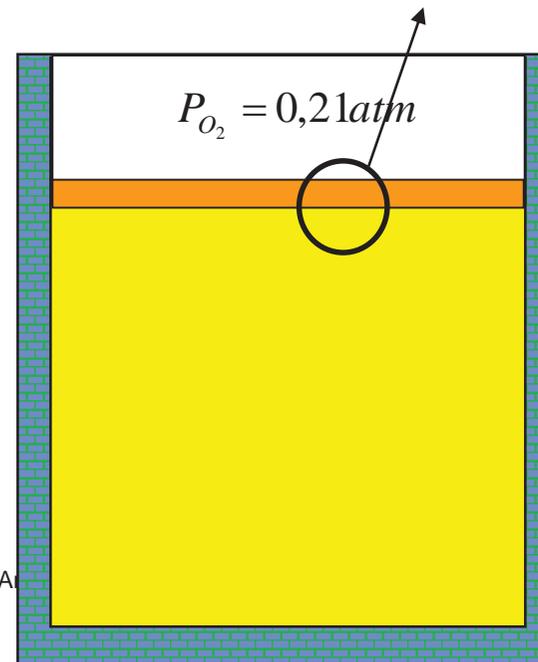
# Spinel clogging in Si-Mn long products - Ladle metallurgy

$$P_{O_2} = 0,21atm$$

1. Protect refractory- **MgO**
2. Fluid slag and slag from EAF - **significant %Al<sub>2</sub>O<sub>3</sub>**



Botelho et al 2016

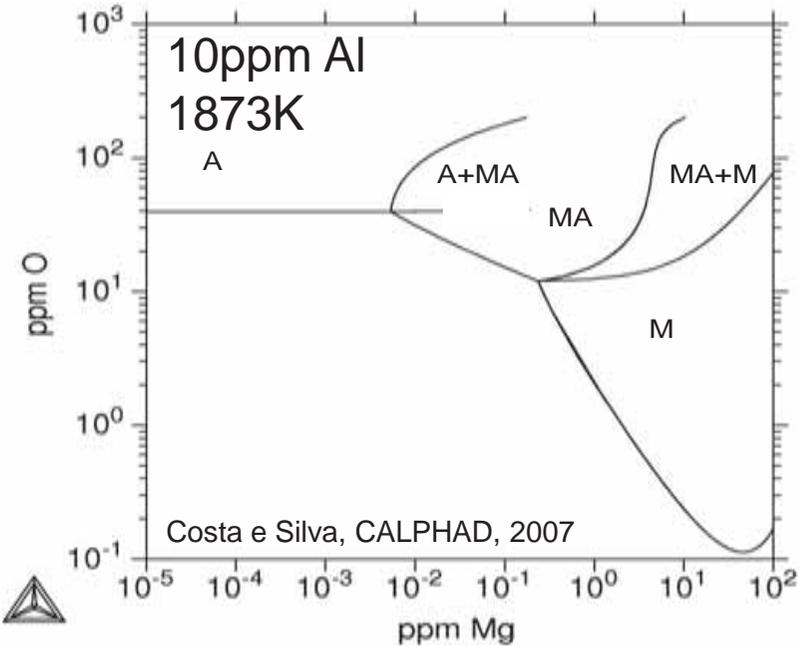
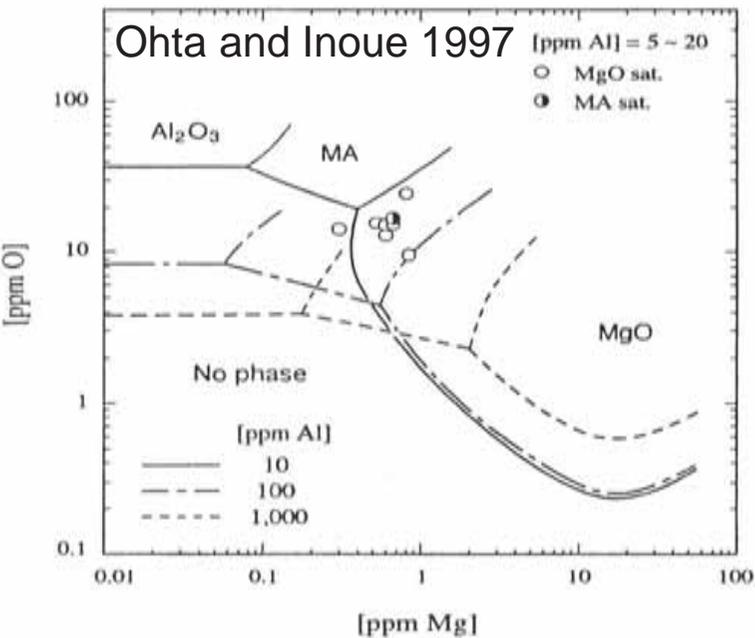


A Costa e Silva, CALPHAD 2016

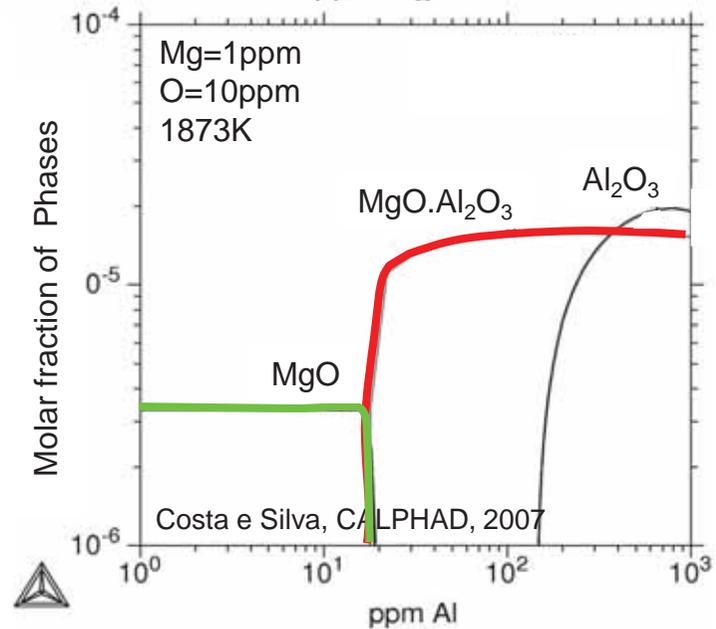
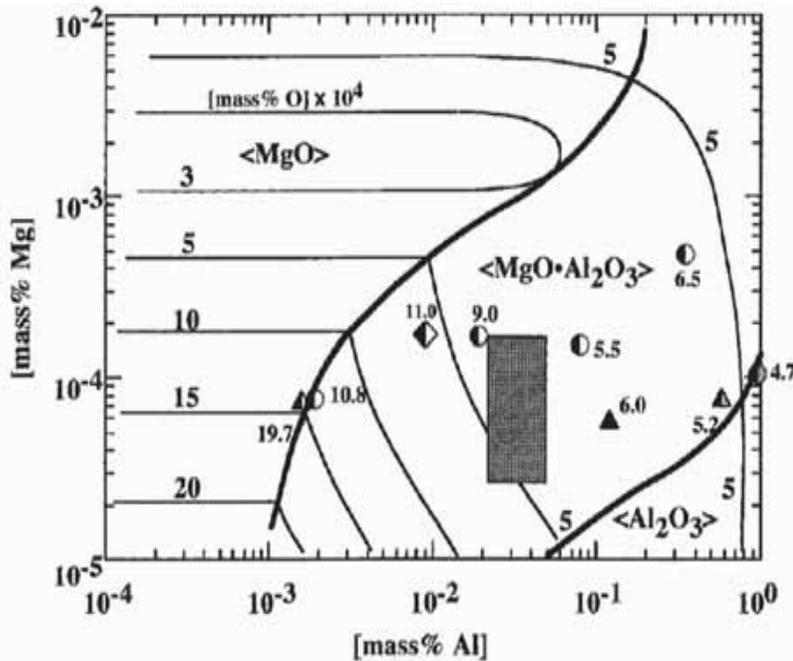
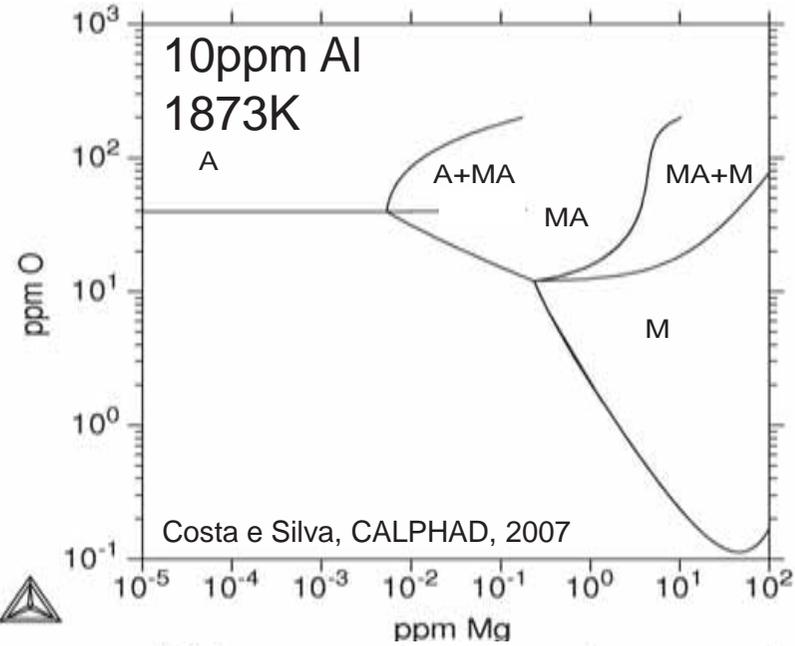
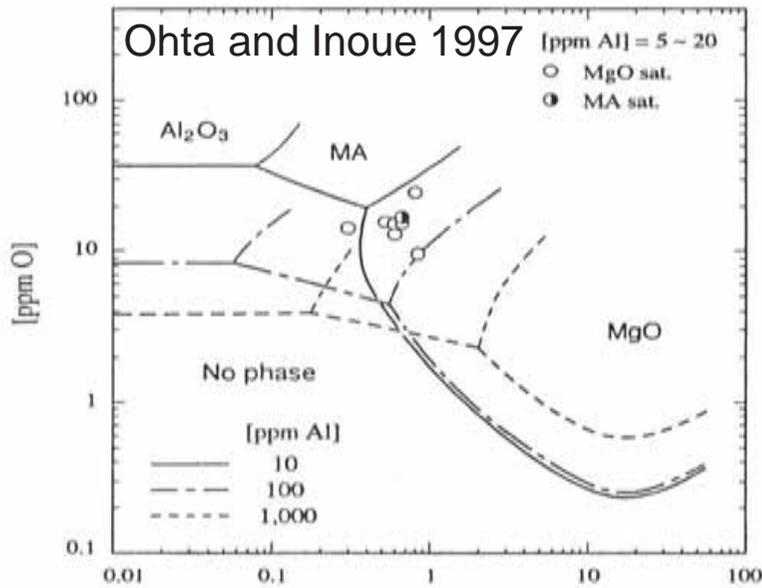
# Spinel studies in high-end steel (Solubility of oxides in Al-Fe-Mg-O)

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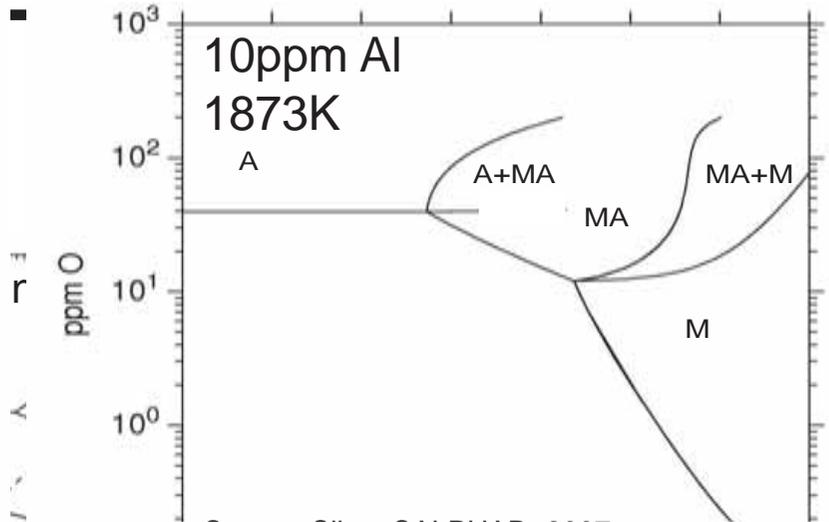
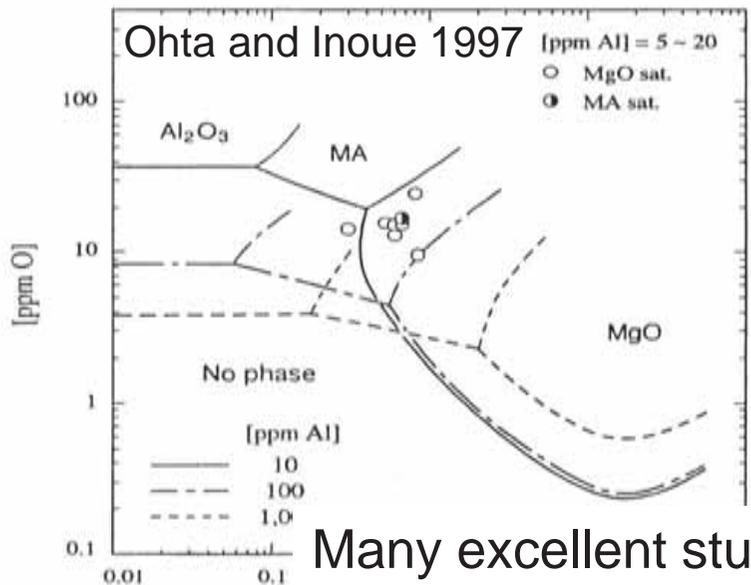
# Spinel studies in high-end steel (Solubility of oxides in Al-Fe-Mg-O)



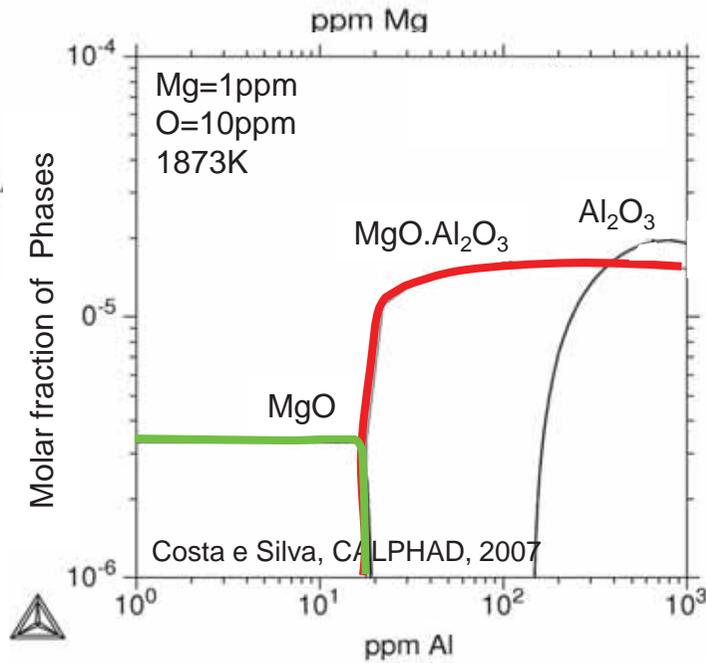
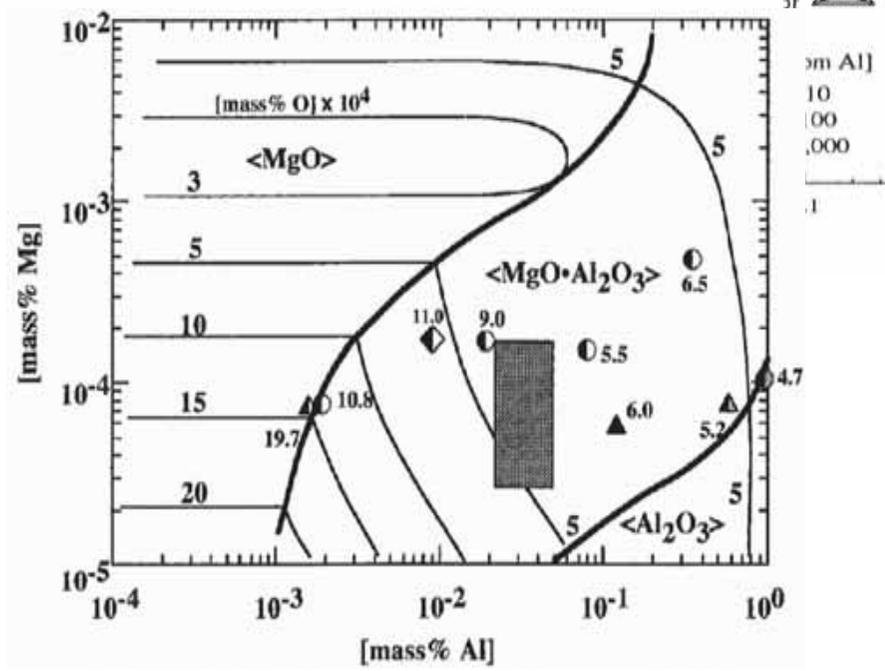
# Spinel studies in high-end steel (Solubility of oxides in Al-Fe-Mg-O)



# Spinel studies in high-end steel (Solubility of oxides in Al-Fe-Mg-O)



Many excellent studies in high alloy and in high value steels!



# Evaluation of slag-metal equilibrium for Si-Mn deoxidized long products (1585°C, in Ladle Furnace (LF))

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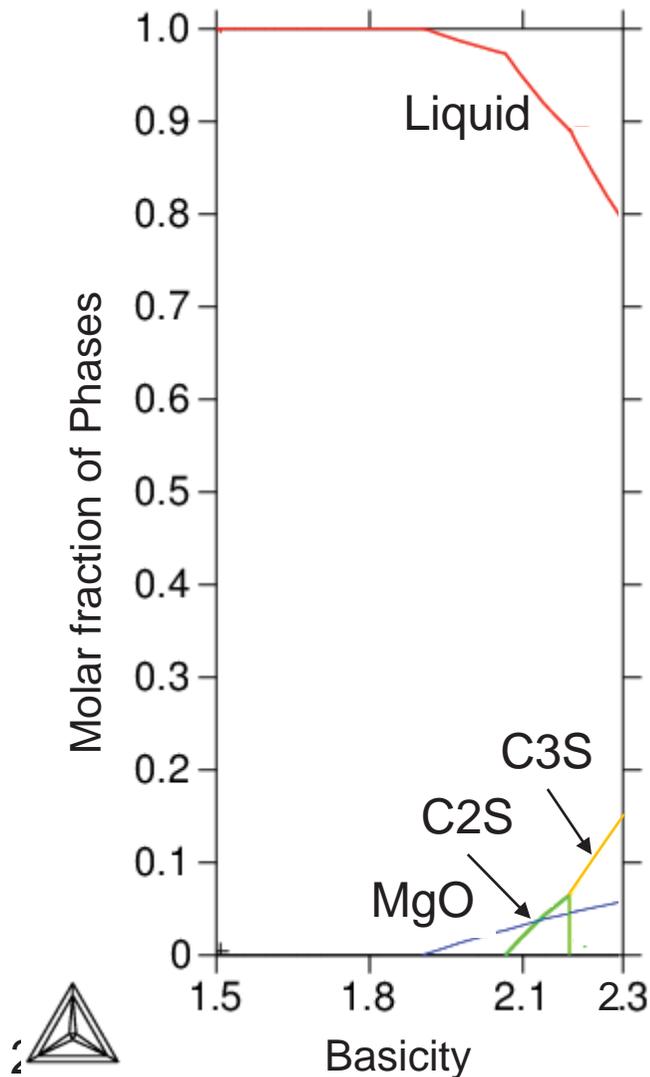
CaO	MgO	Al <sub>2</sub> O <sub>3</sub>	CaF <sub>2</sub>
52,27	10,26	3,06	9,00

## Evaluation of slag-metal equilibrium for Si-Mn deoxidized long products (1585°C, in Ladle Furnace (LF))

CaO	MgO	Al <sub>2</sub> O <sub>3</sub>	CaF <sub>2</sub>	Steel 1.35%Mn 0.3%Si
52,27	10,26	3,06	9,00	

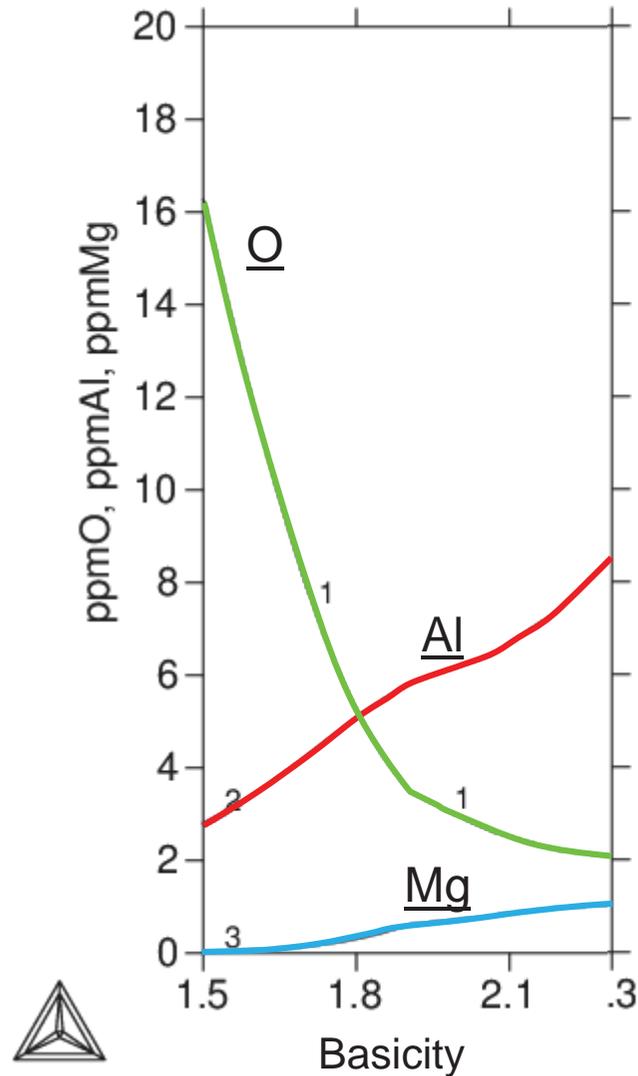
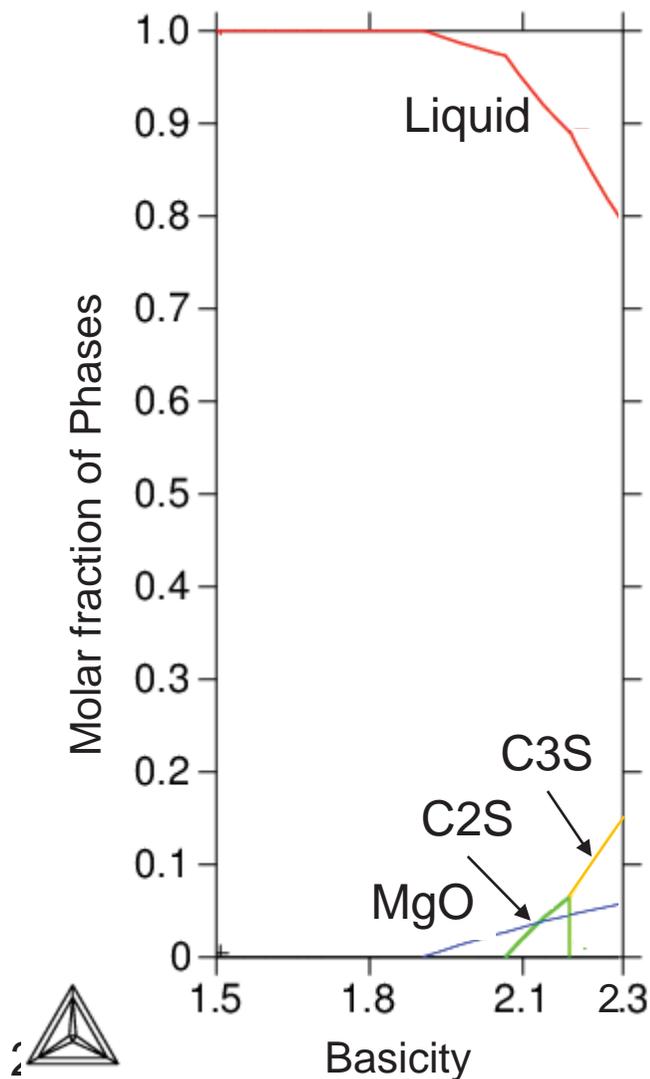
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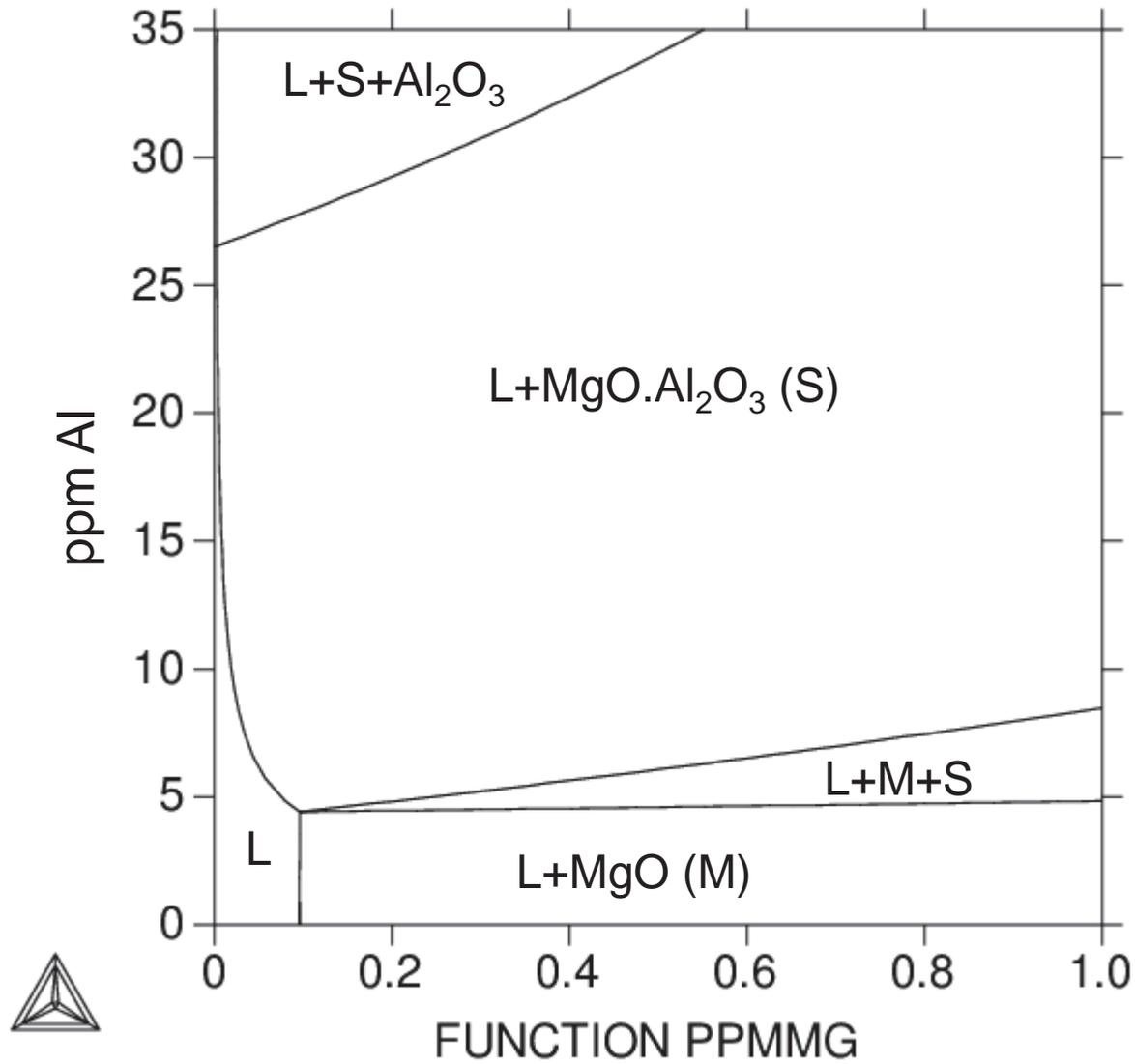


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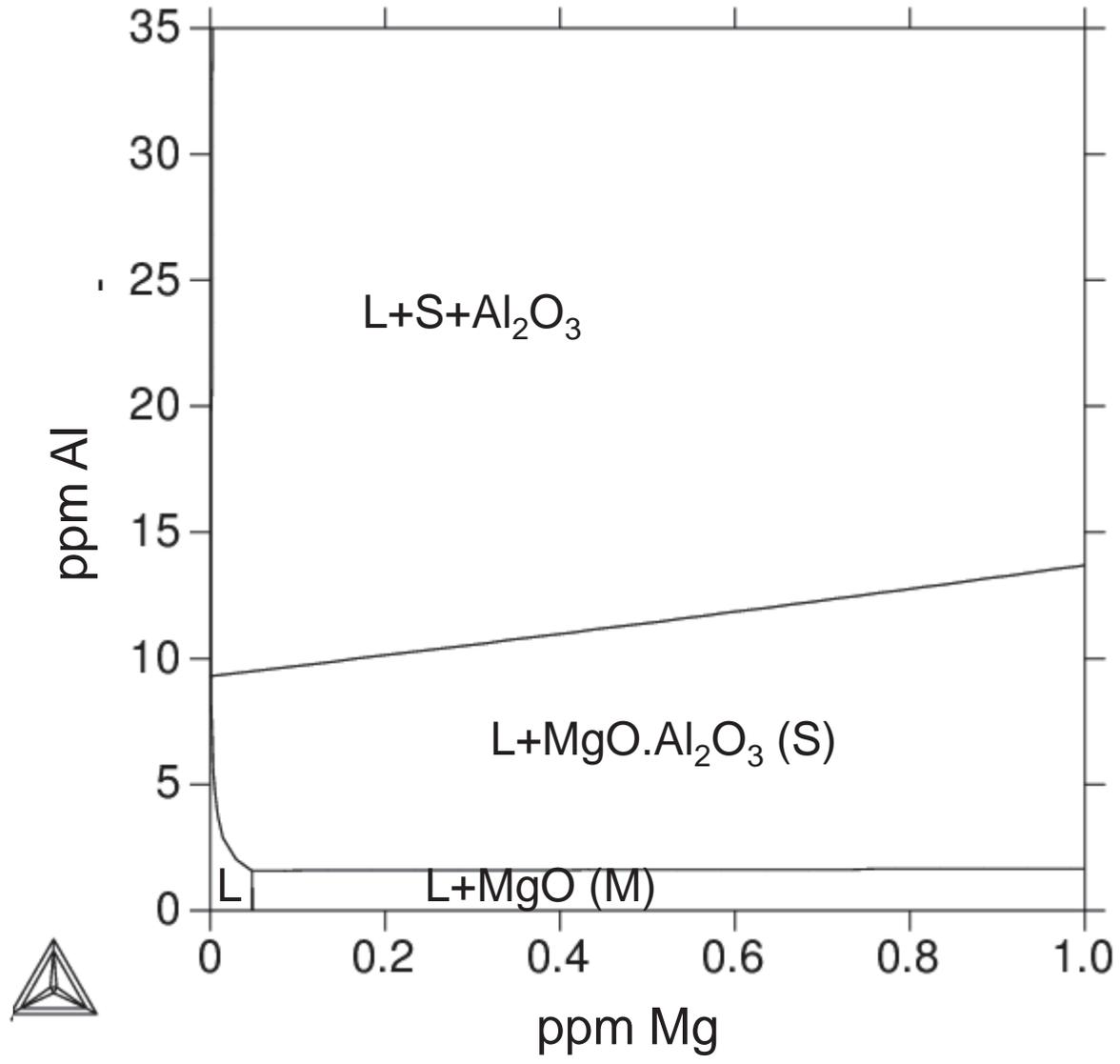
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52,27	10,26	3,06	9,00	



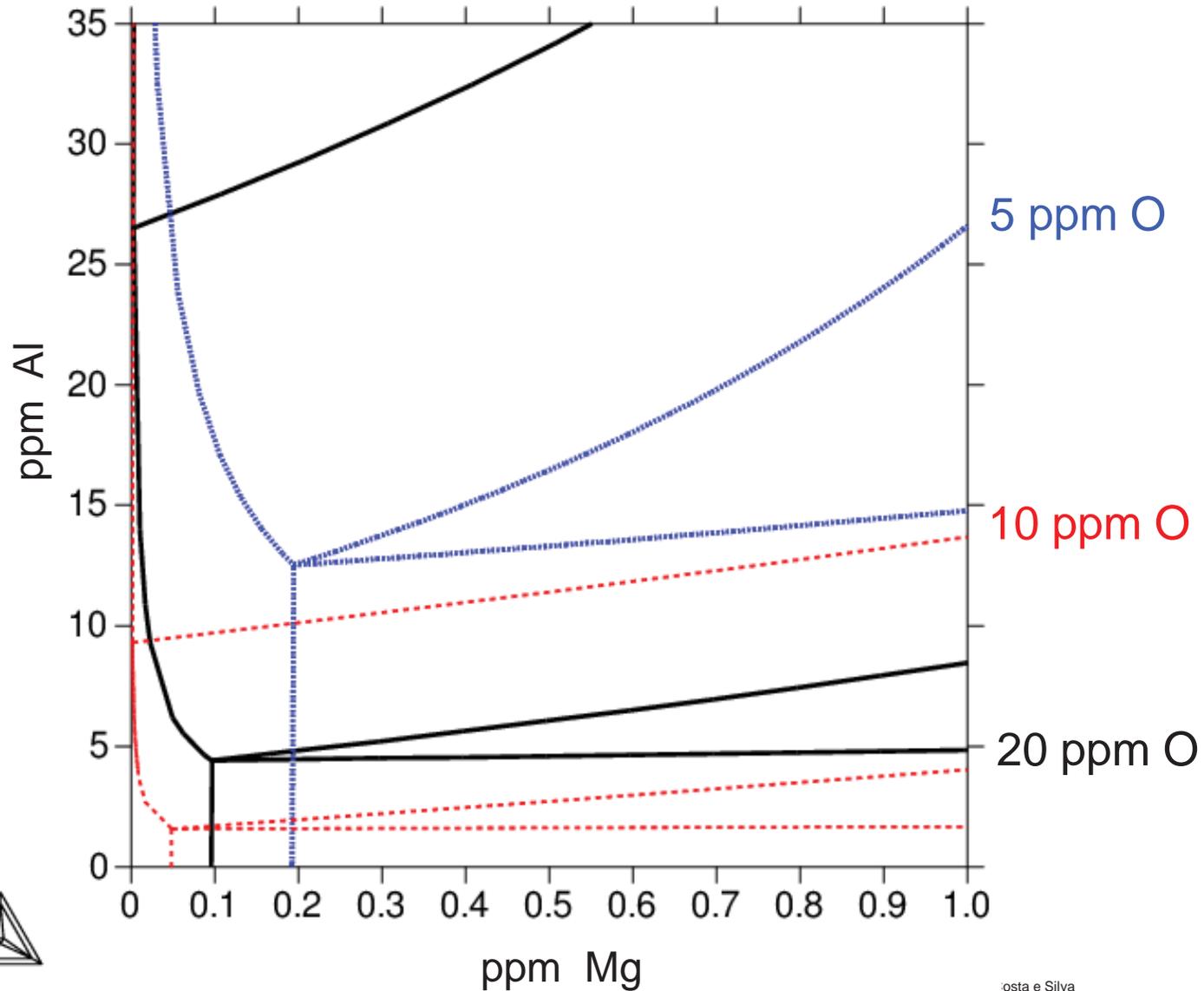
# What happens during casting: 1550°C, 10 ppm O



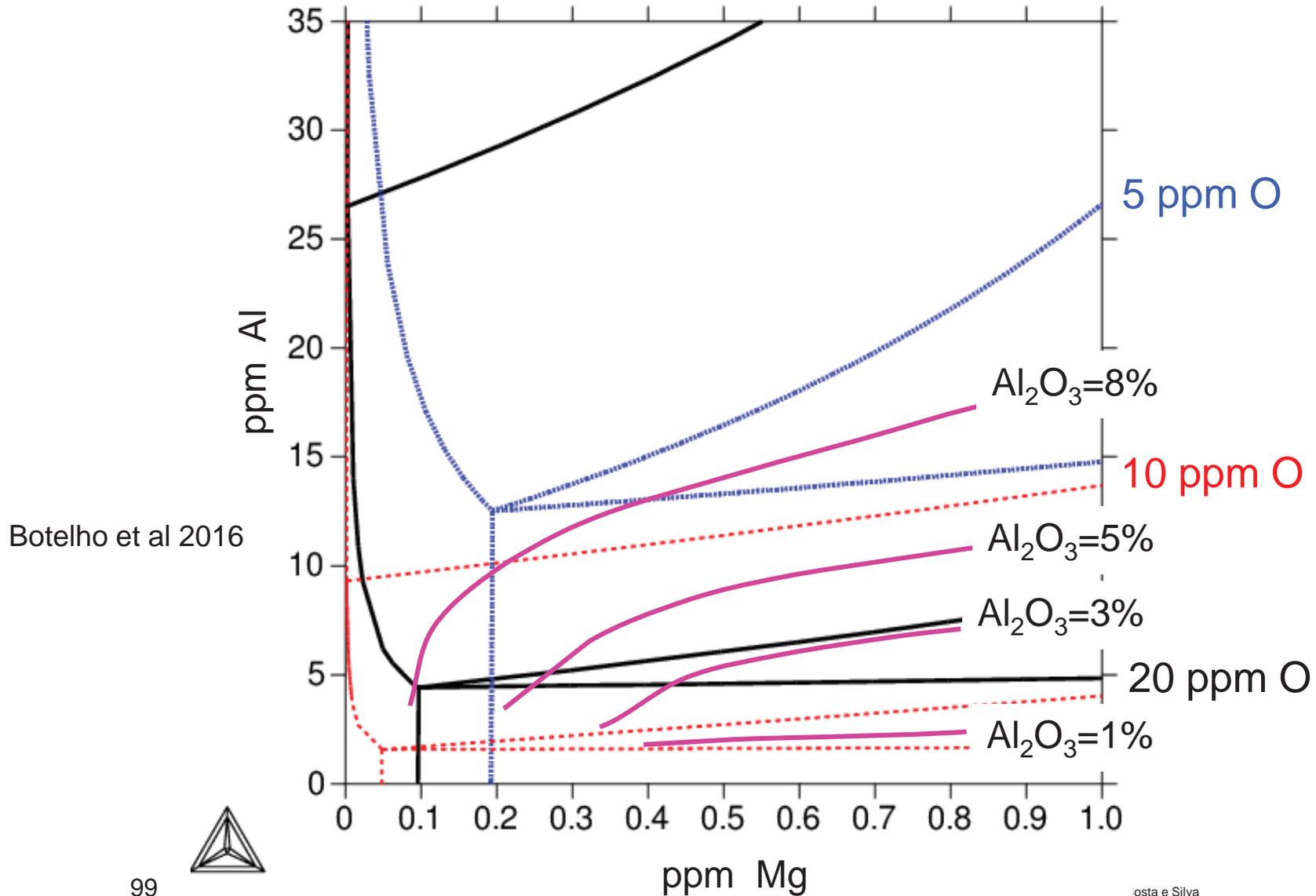
# What happens during casting: 1550°C, 20 ppm O



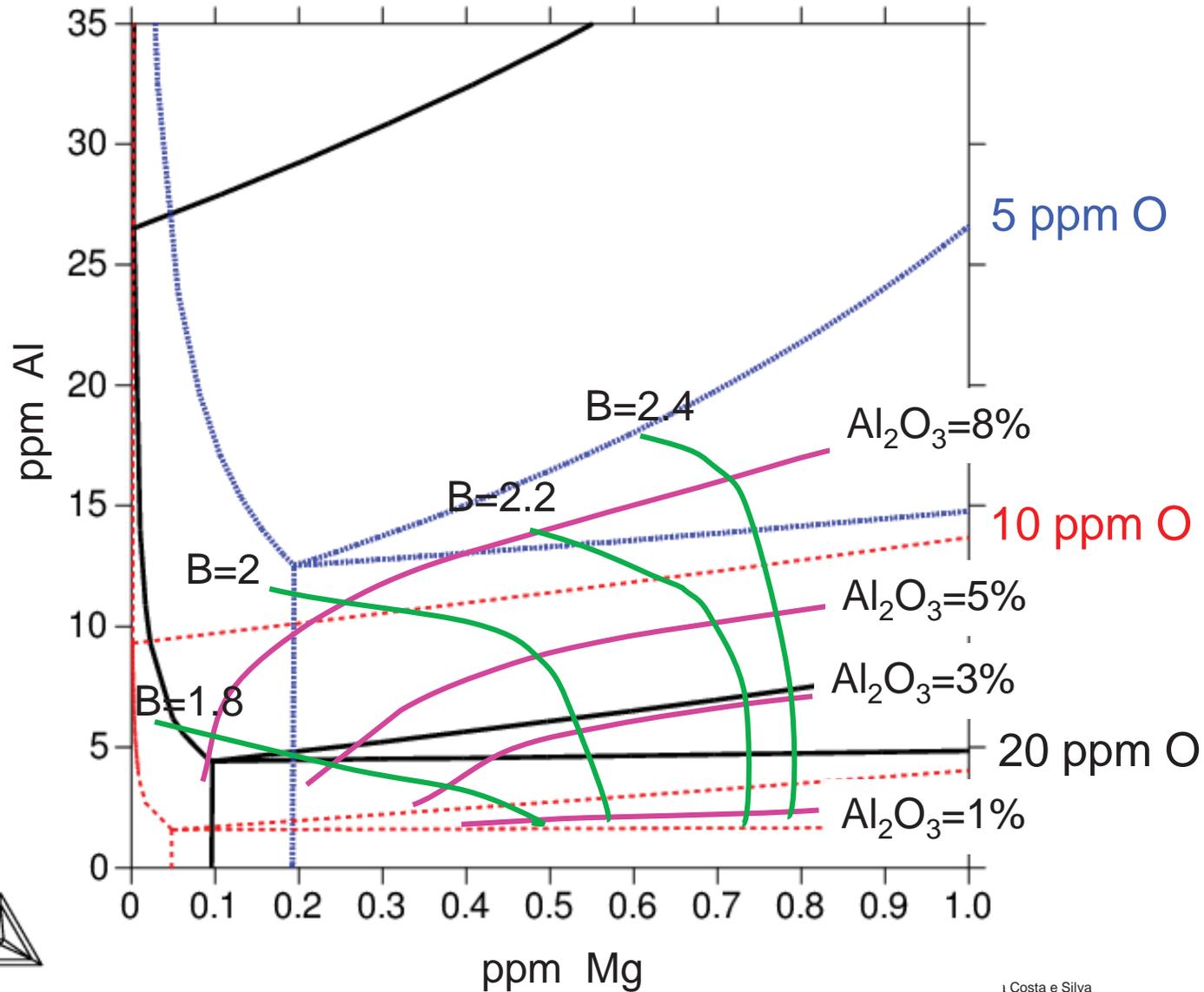
# Explore processing conditions: Different slags in LF and possible reoxidation conditions



# Explore processing conditions: Different slags in LF and possible reoxidation conditions



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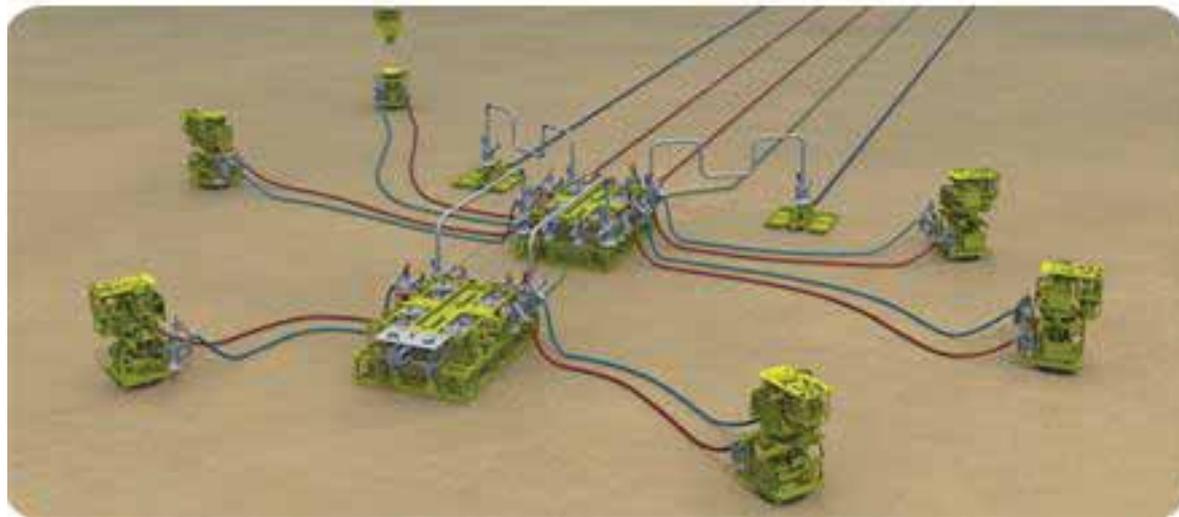


# Sub-sea manifolds – Extensive use of duplex SS

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# Sub-sea manifolds – Extensive use of duplex SS

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# Sigma phase cracking – large Duplex Stainless steel casting

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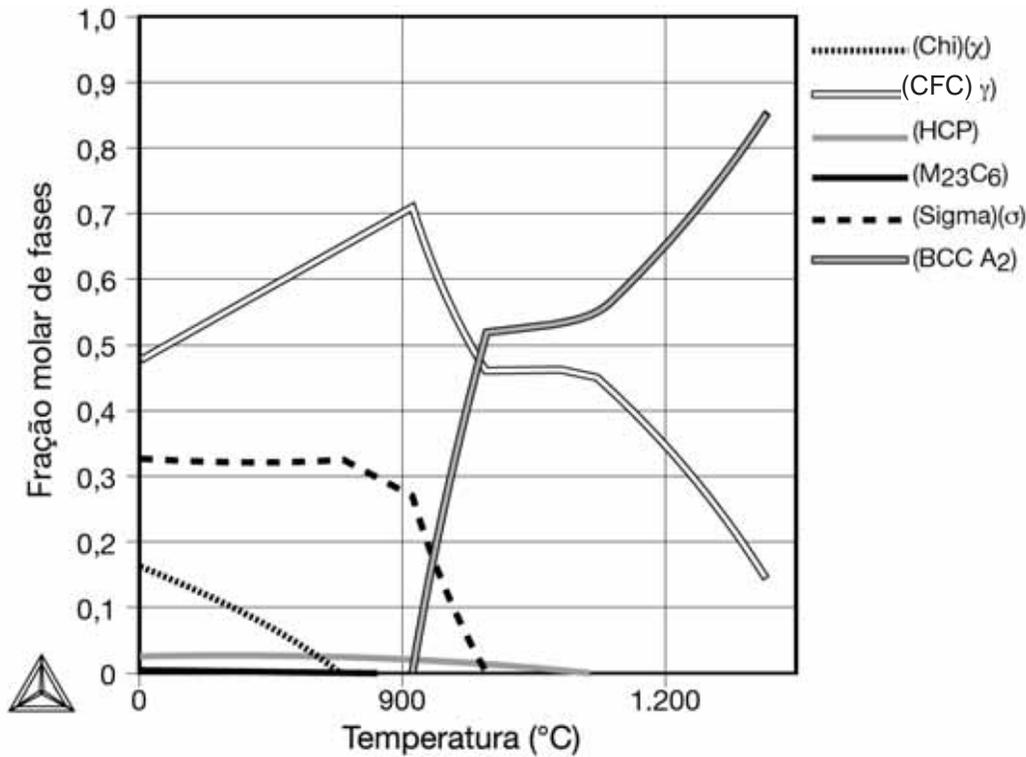


M Martins, 2008

# Sigma in ASTM A890 Gr 6A

Costa e Silva, Avillez, 2009

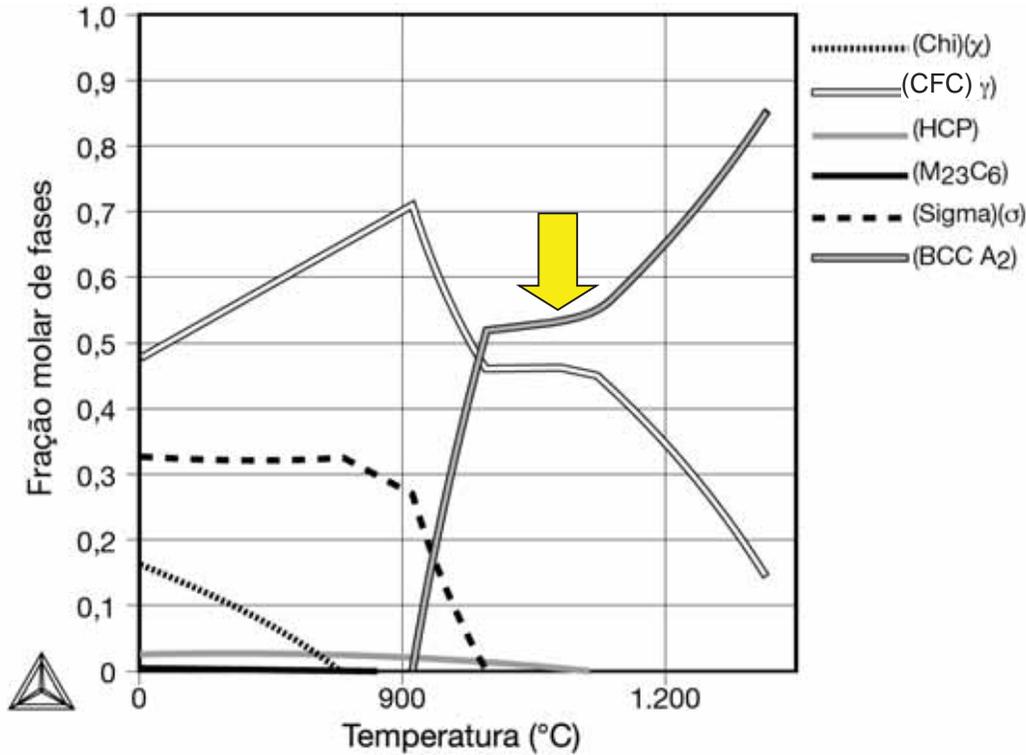
$N=1$ ,  $P=1.01325E5$ ,  $W(C)=2E-4$ ,  $W(CR)=0.2584$ ,  $W(NI)=5.94E-2$ ,  $W(MN)=8.8E-3$ ,  
 $W(SI)=9.2E-3$ ,  $W(MO)=3.79E-2$ ,  $W(N)=2.5E-3$ ,  $W(CU)=1.5E-2$ ;



# Sigma in ASTM A890 Gr 6A

Costa e Silva, Avillez, 2009

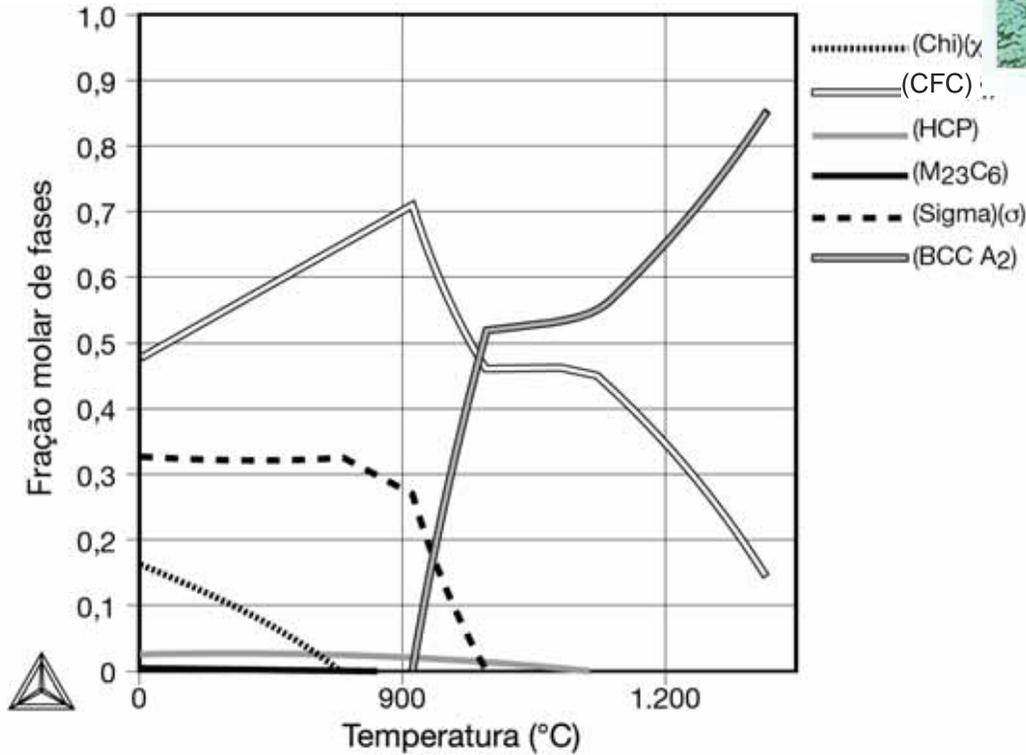
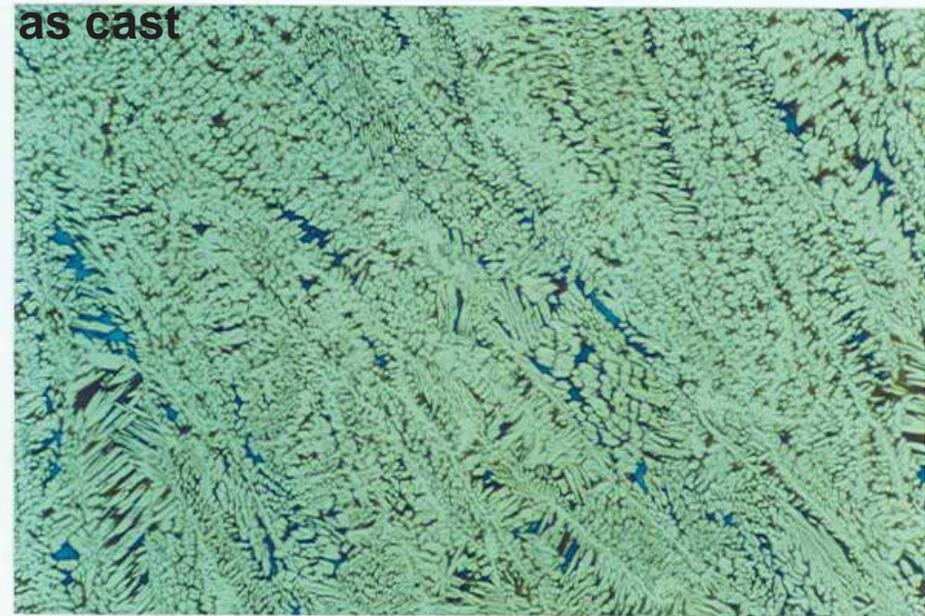
$N=1$ ,  $P=1.01325E5$ ,  $W(C)=2E-4$ ,  $W(CR)=0.2584$ ,  $W(NI)=5.94E-2$ ,  $W(MN)=8.8E-3$ ,  
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# Sigma in ASTM A890 Gr 6A

Costa e Silva, Avillez, 2009

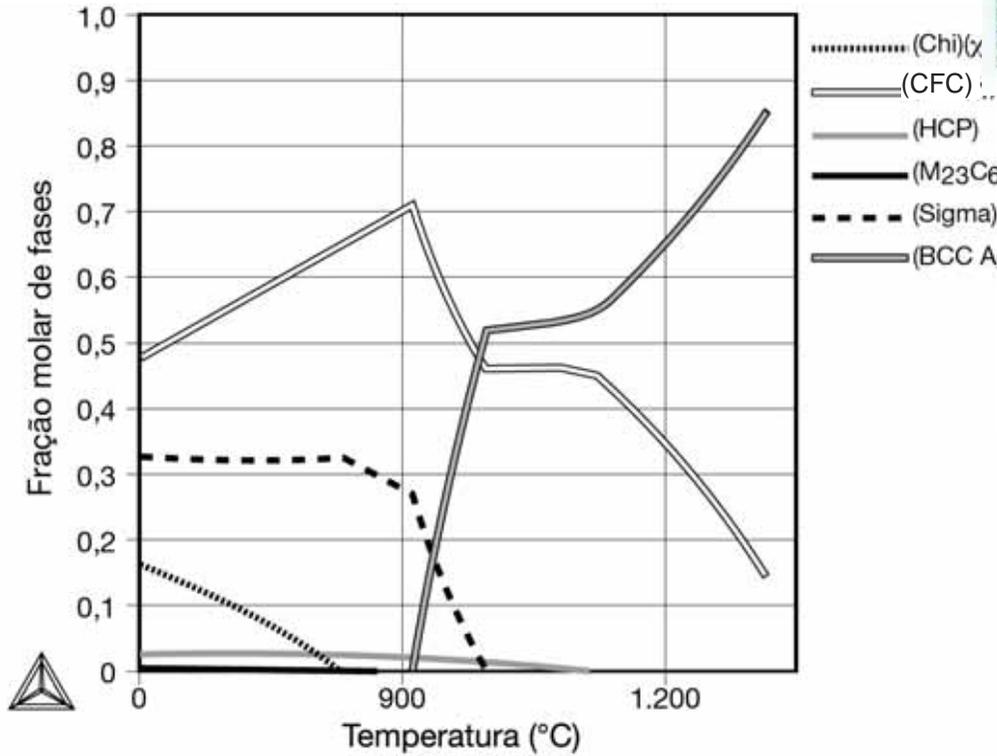
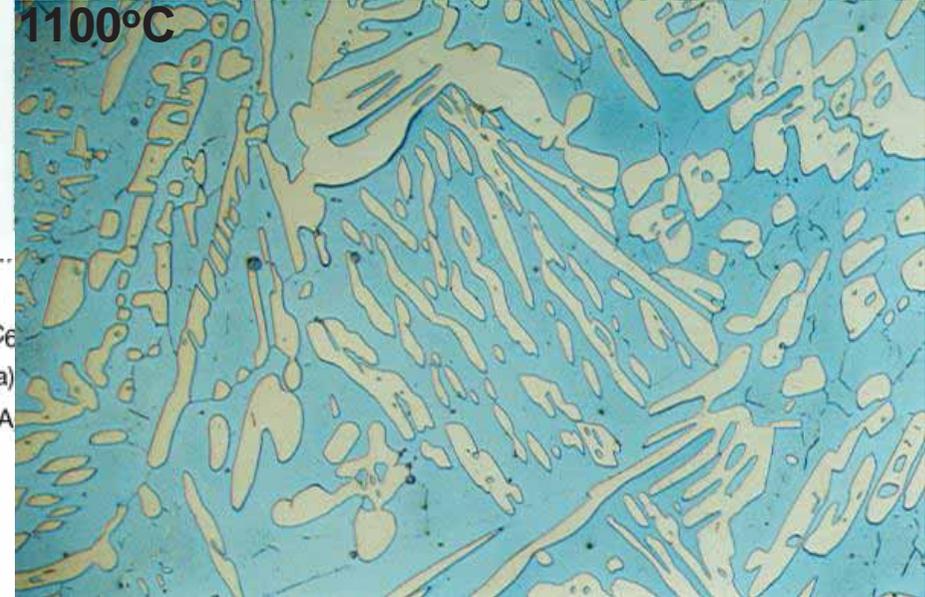
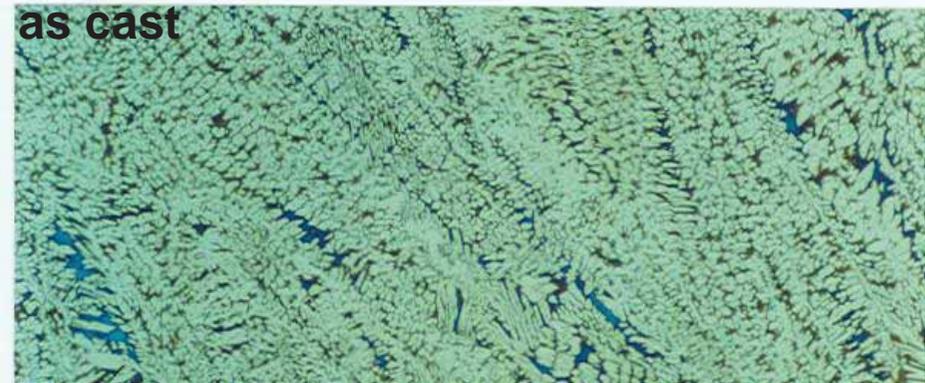
N=1, P=1.01325E5, W(C)=2E-4, W(CR)=0.2584, W(NI)=5.  
W(SI)=9.2E-3, W(MO)=3.79E-2, W(N)=2.5E-3, W(CU)=1.5



# Sigma in ASTM A890 Gr 6A

Costa e Silva, Avillez, 2009

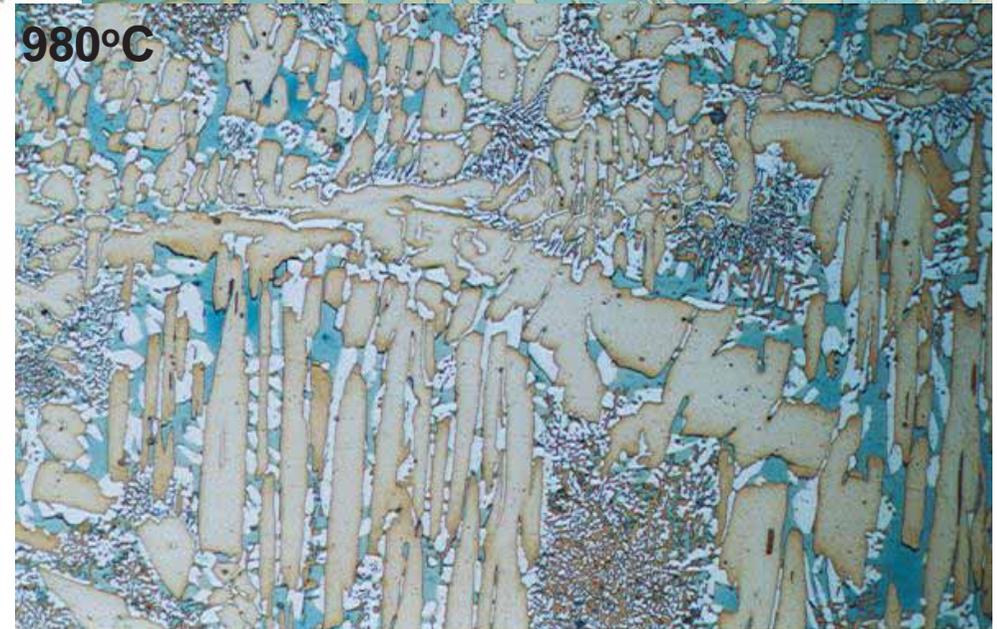
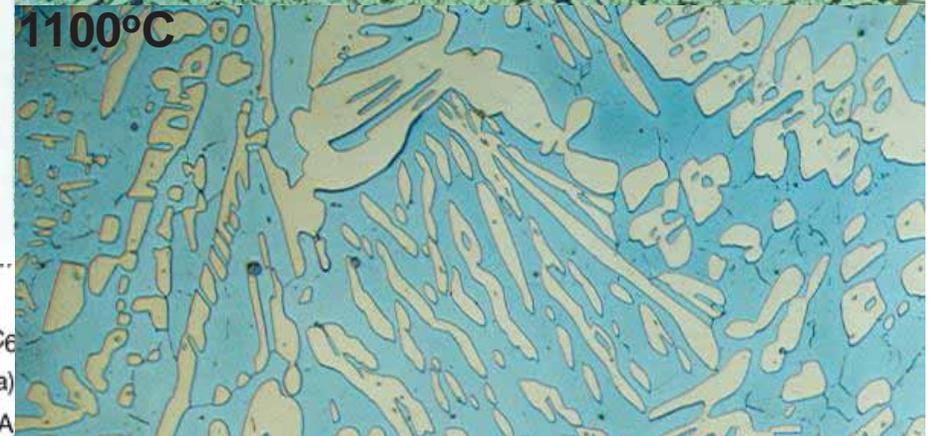
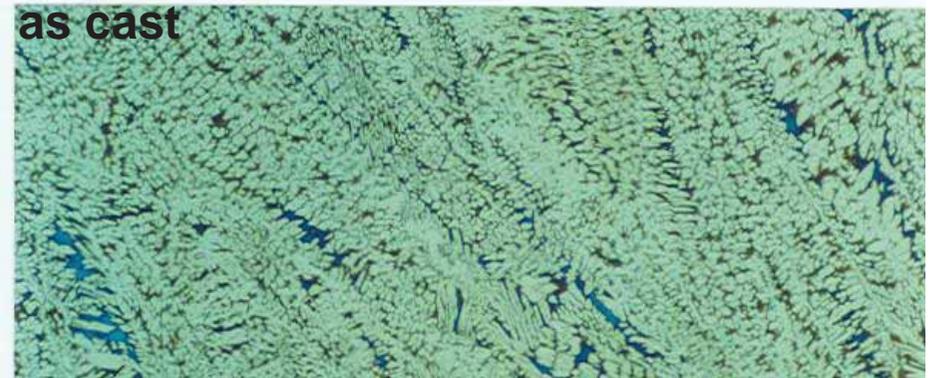
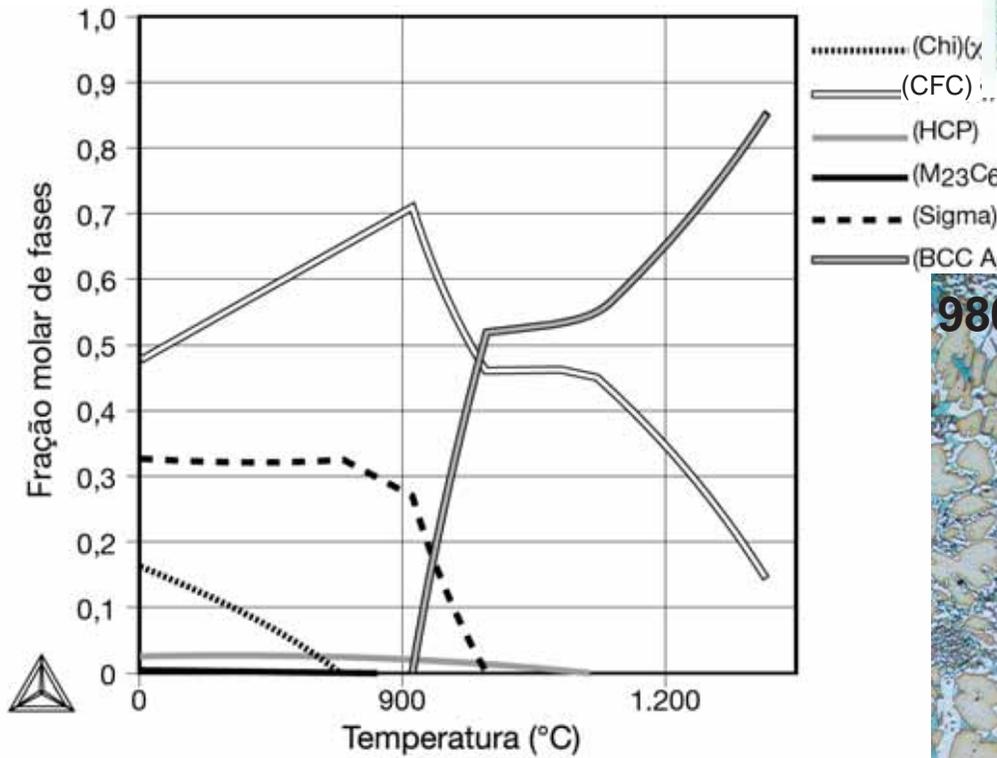
N=1, P=1.01325E5, W(C)=2E-4, W(CR)=0.2584, W(NI)=5.  
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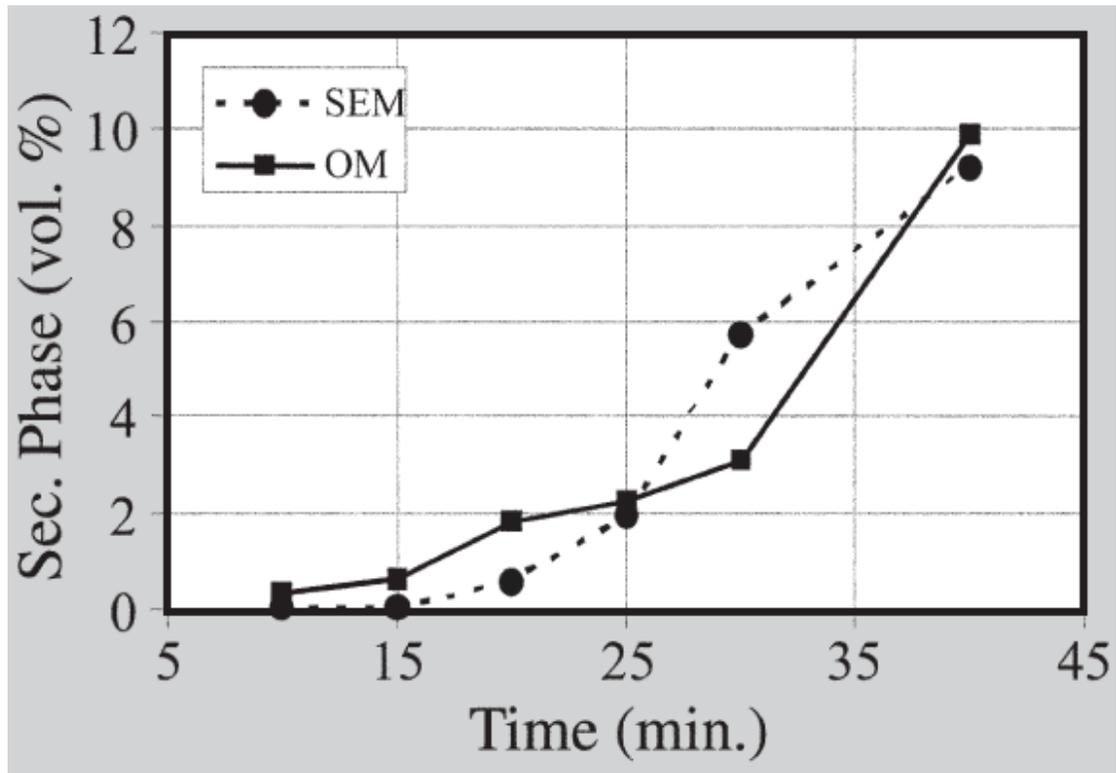
# Sigma in ASTM A890 Gr 6A

Costa e Silva, Avillez, 2009

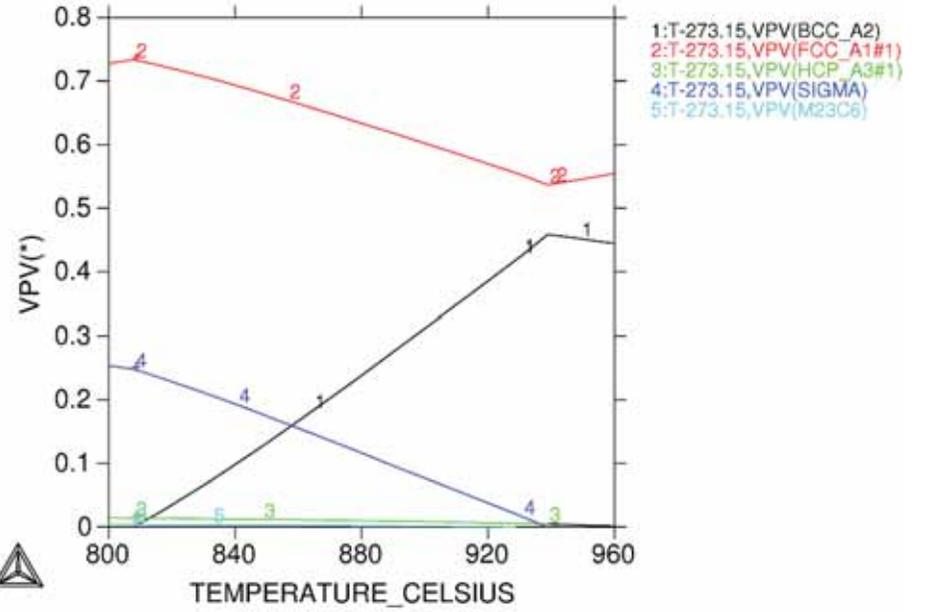
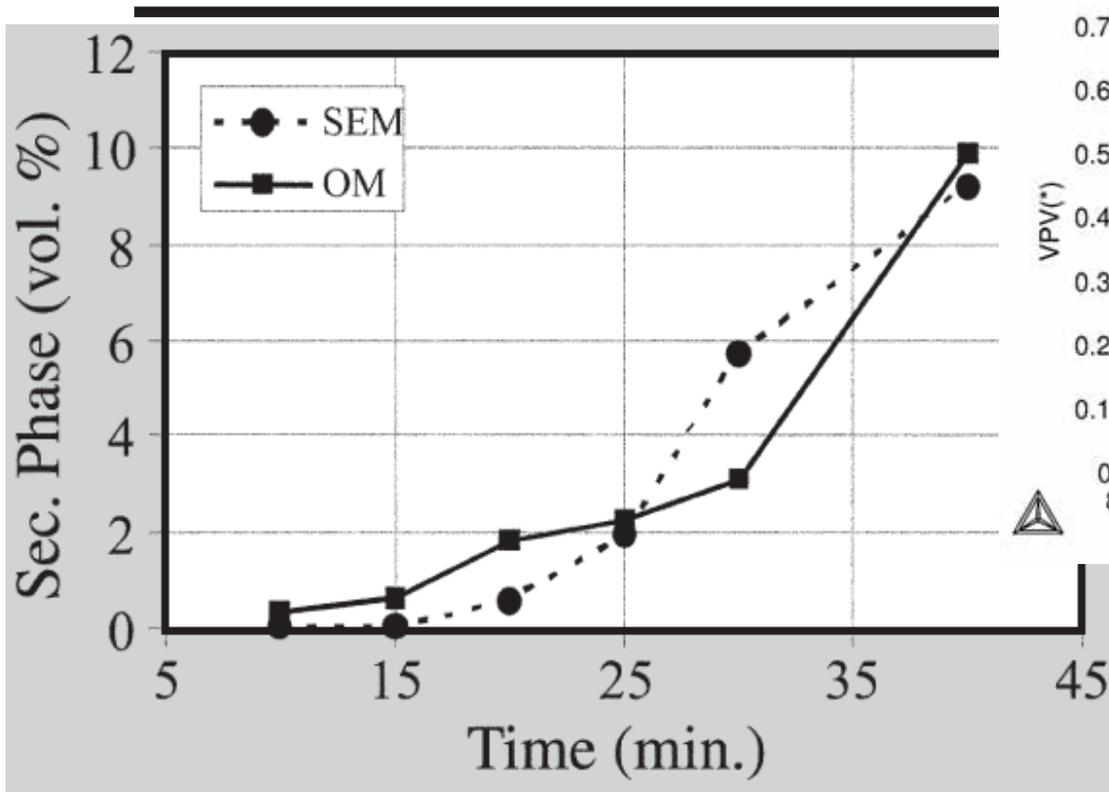
N=1, P=1.01325E5, W(C)=2E-4, W(CR)=0.2584, W(NI)=5.  
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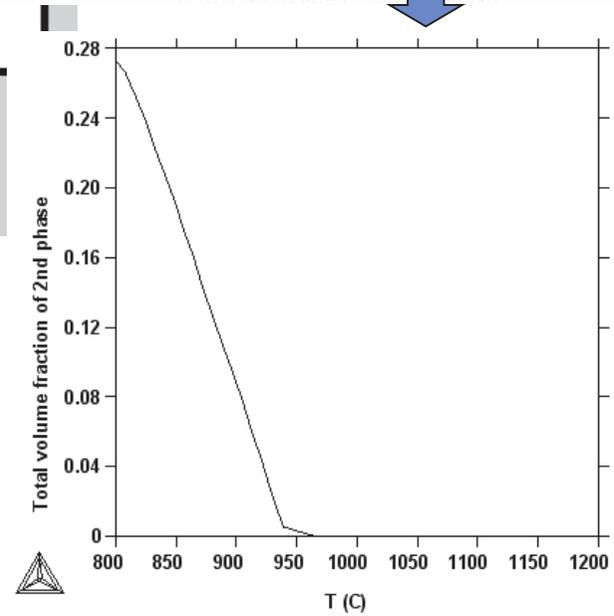
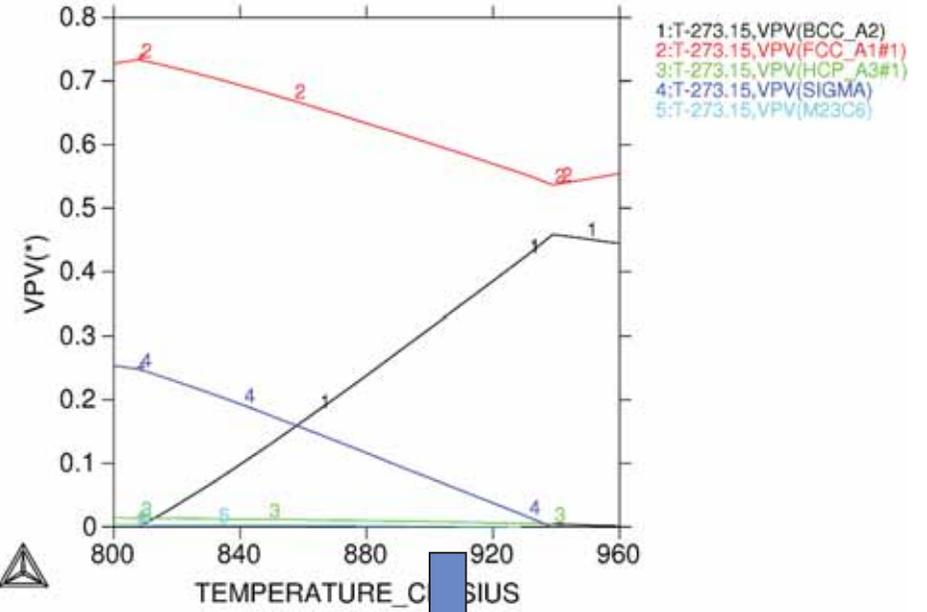
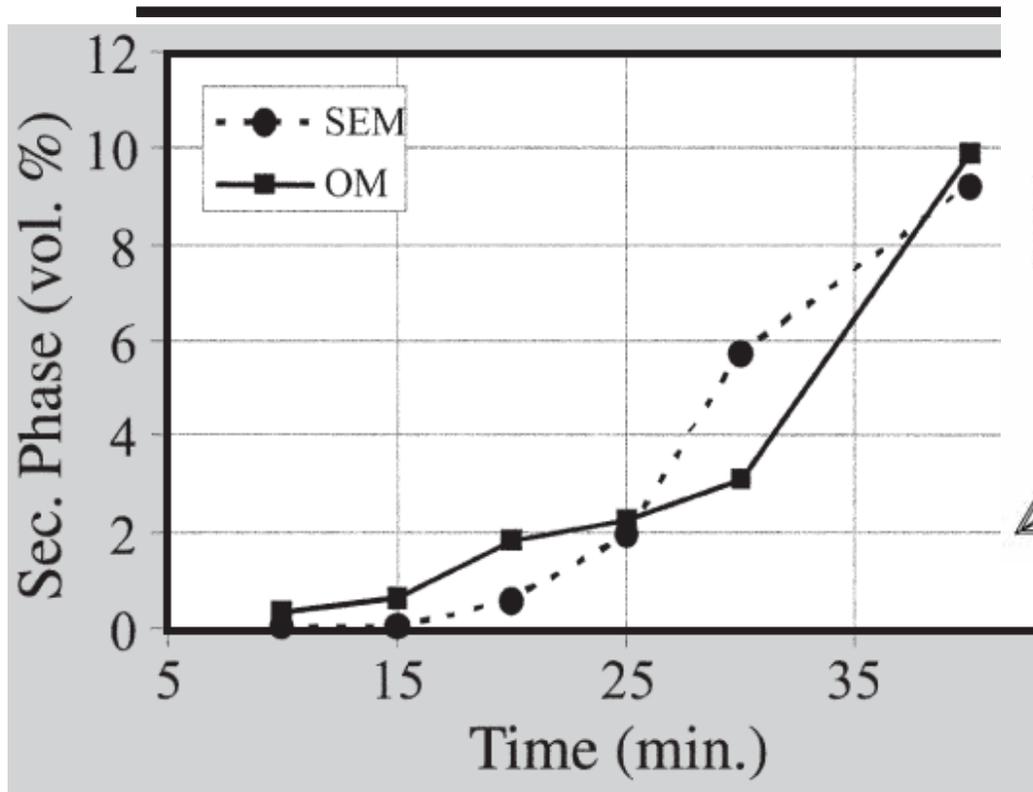
# Second phase precipitation in 2205 at 900°C, I. Calliari, JOM 2009



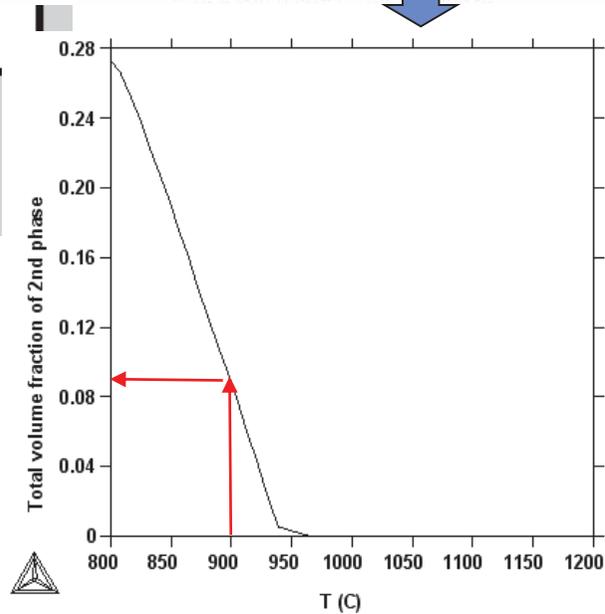
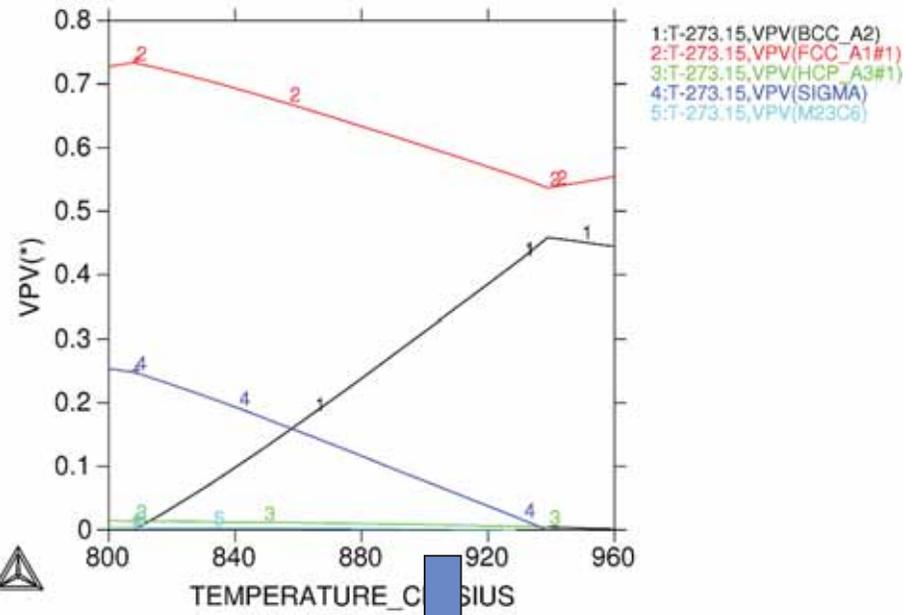
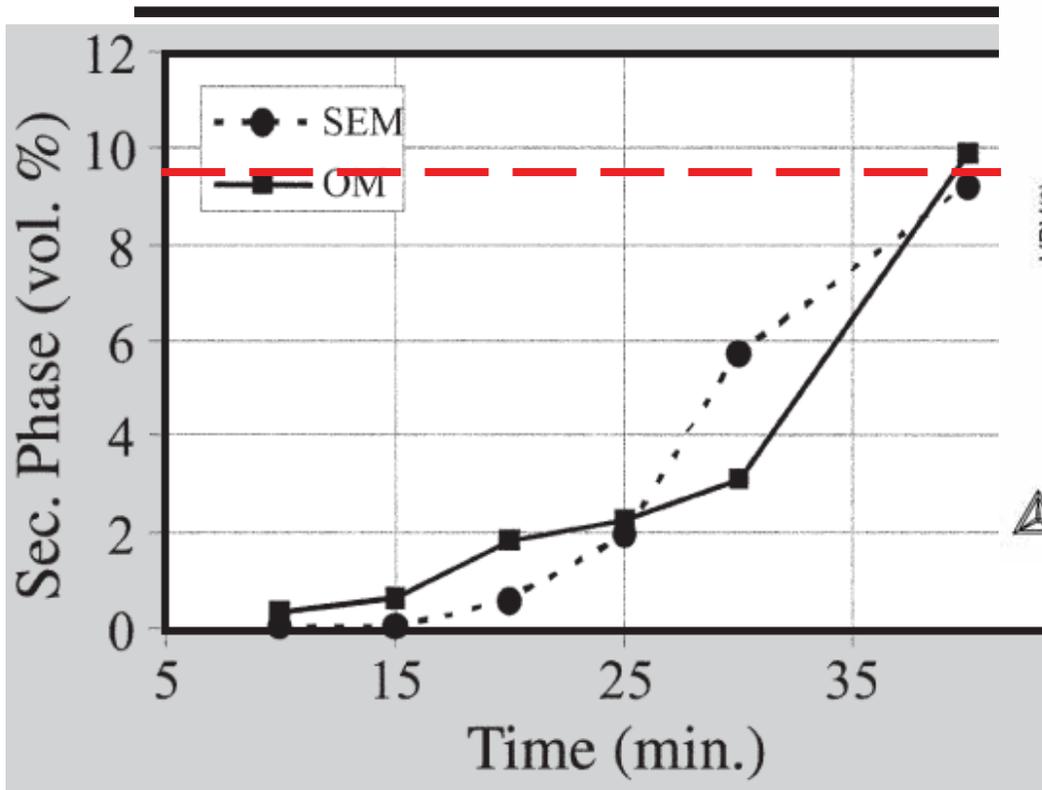
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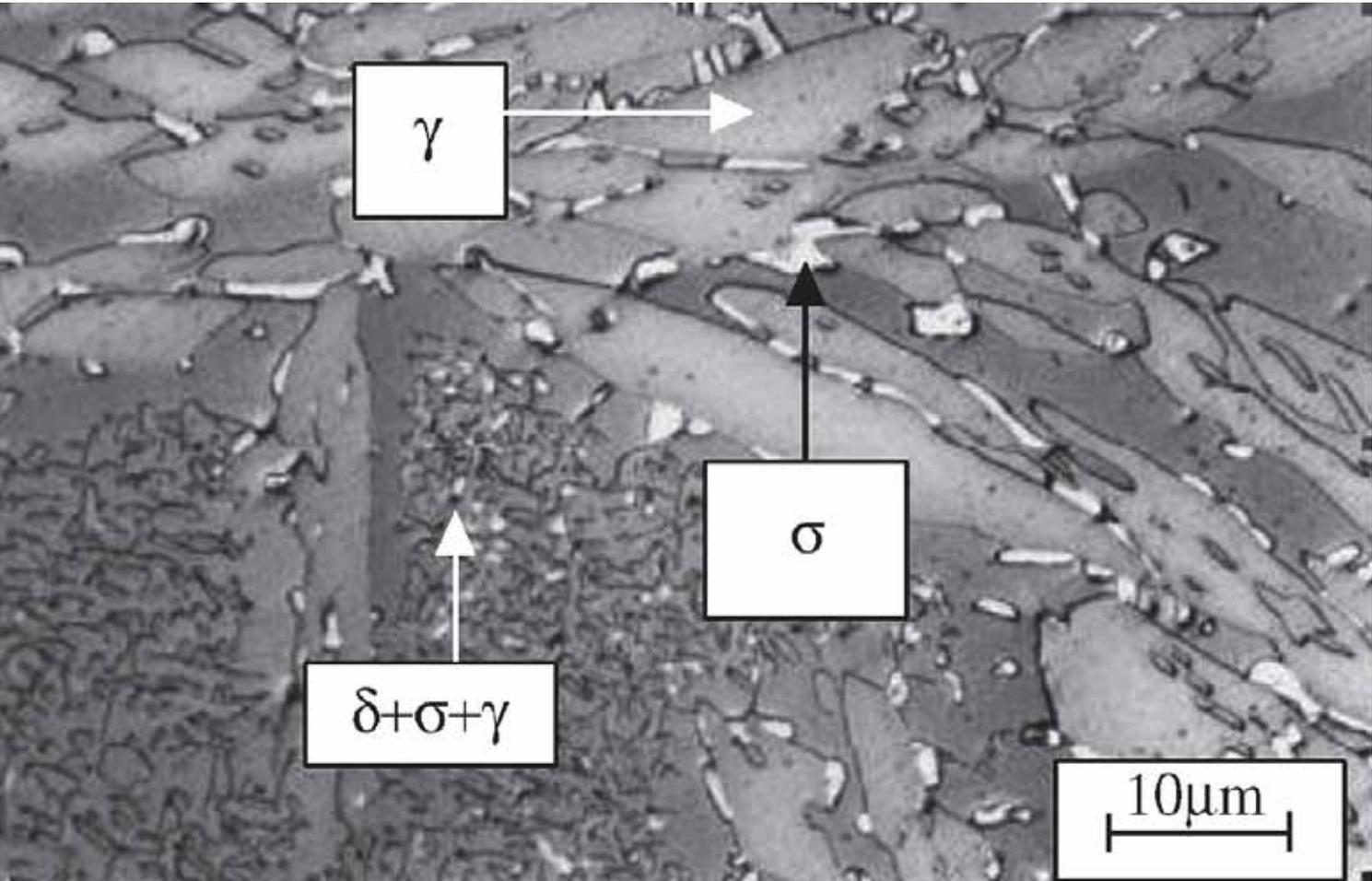


# Second phase precipitation in 2205 at 900°C. I. Calliari. JOM 2009



# ASTM A890 Gr 6A

## Isothermal 1040 °C $\sigma$ location

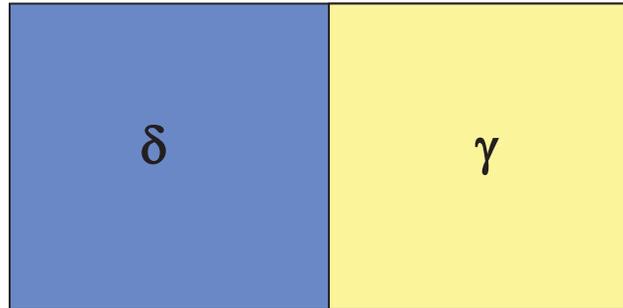


Martins 2005

# DICTRA Modeling (CALPHAD 2009)

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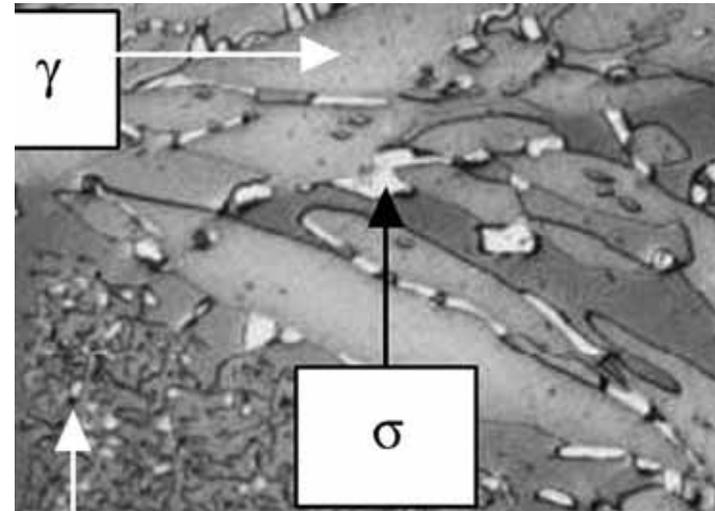
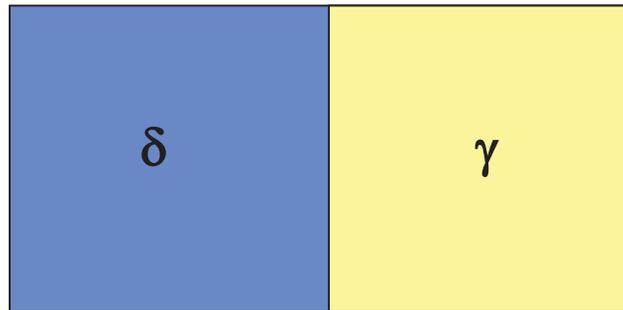
Modeling the formation of sigma at  $\gamma$ - $\delta$  interface



# DICTRA Modeling (CALPHAD 2009)

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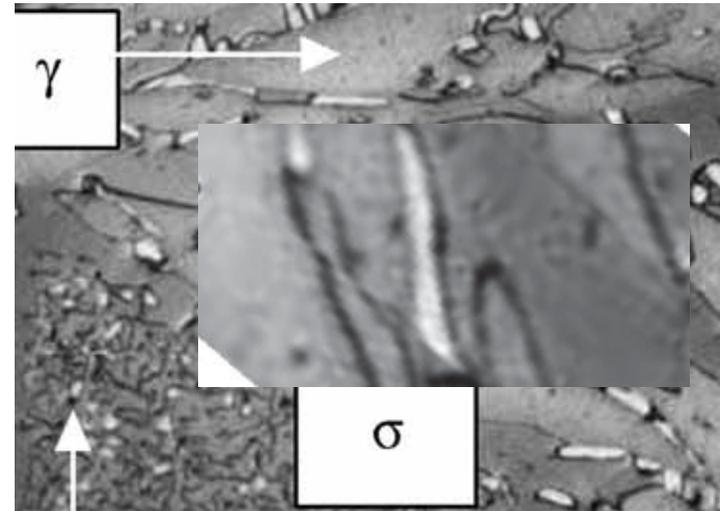
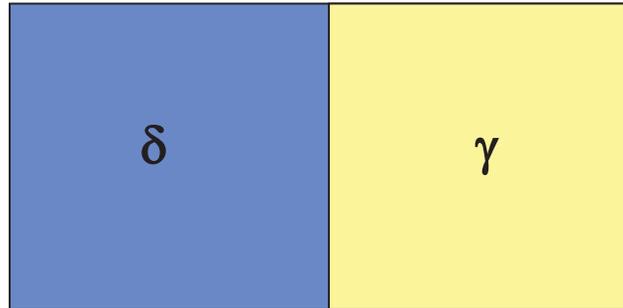
Modeling the formation of sigma at  $\gamma$ - $\delta$  interface



# DICTRA Modeling (CALPHAD 2009)

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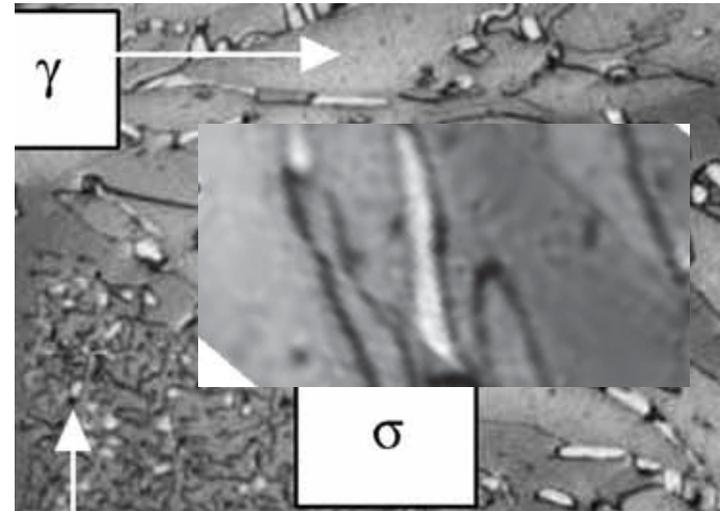
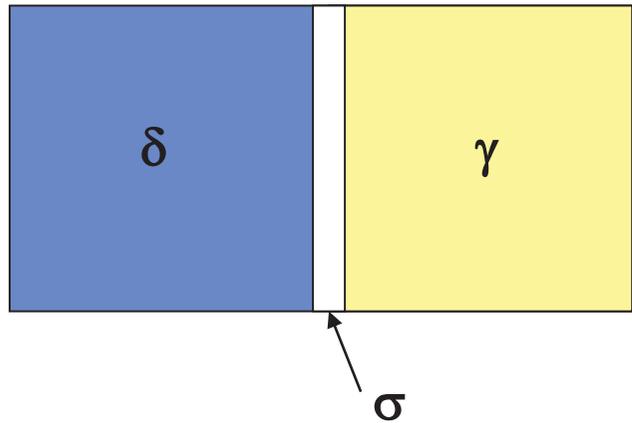
Modeling the formation of sigma at  $\gamma$ - $\delta$  interface



# DICTRA Modeling (CALPHAD 2009)

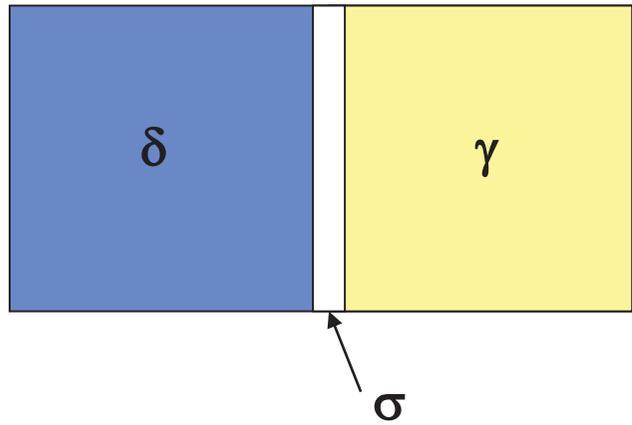
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Modeling the formation of sigma at  $\gamma$ - $\delta$  interface

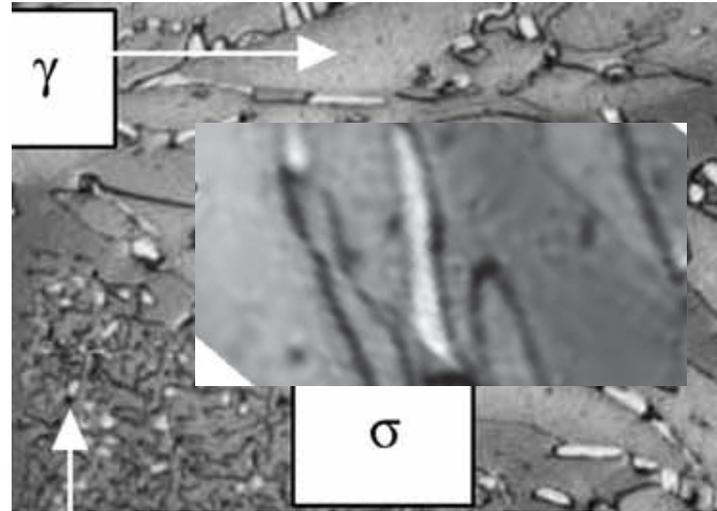
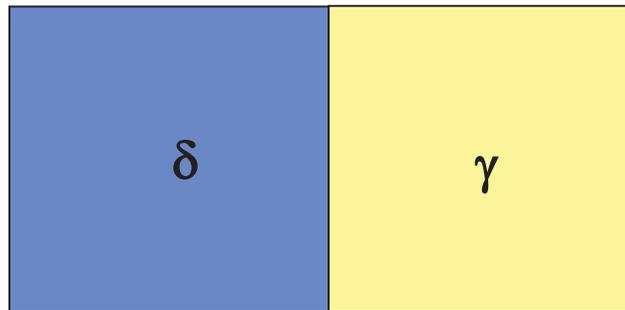


# DICTRA Modeling (CALPHAD 2009)

Modeling the formation of sigma at  $\gamma$ - $\delta$  interface

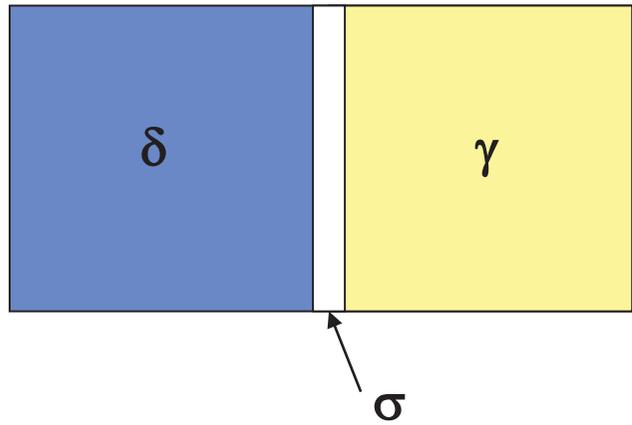


Cooperative growth - Modeling the growth

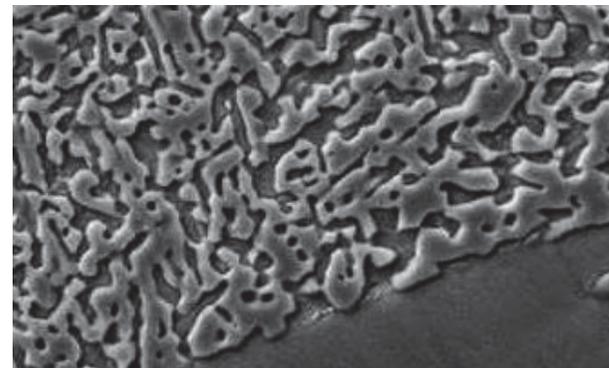
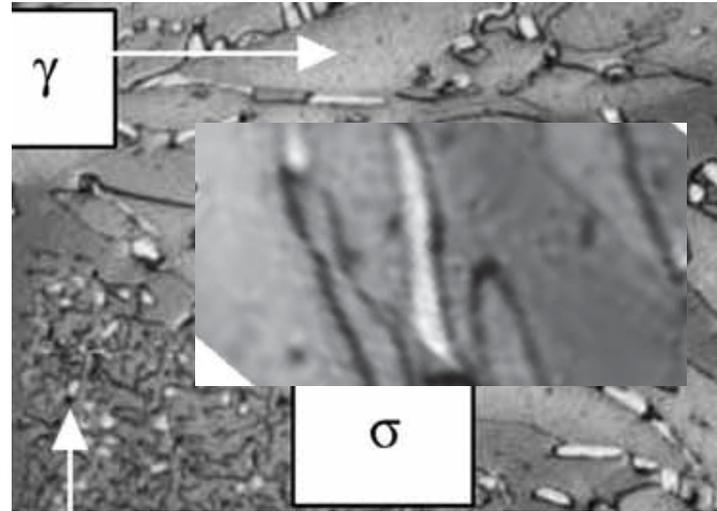
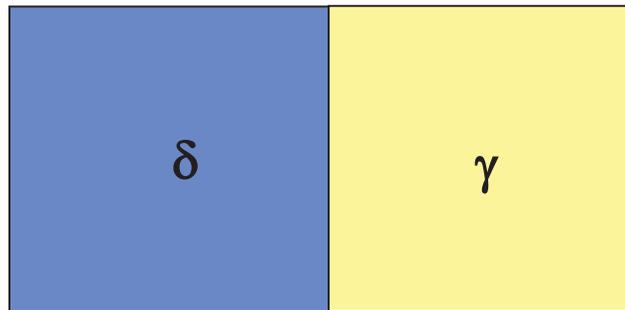


# DICTRA Modeling (CALPHAD 2009)

Modeling the formation of sigma at  $\gamma$ - $\delta$  interface

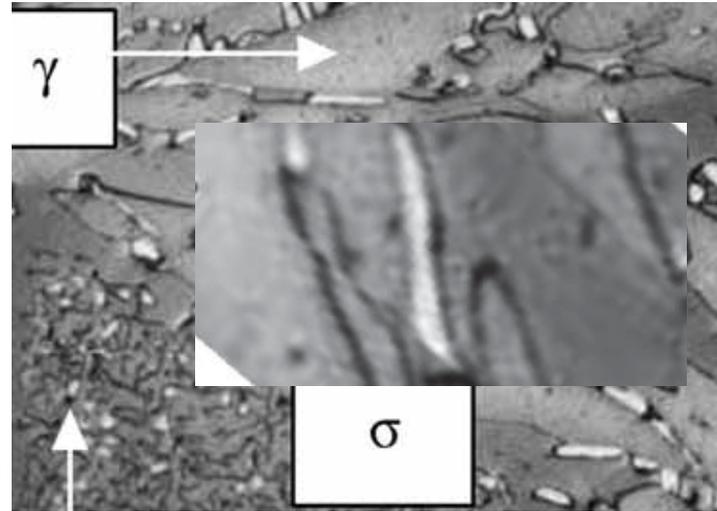
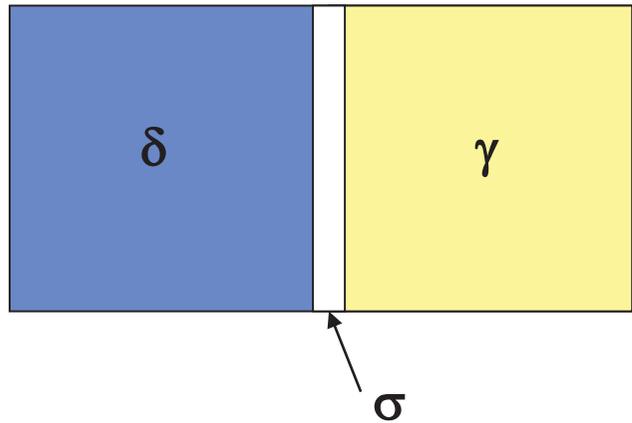


Cooperative growth - Modeling the growth

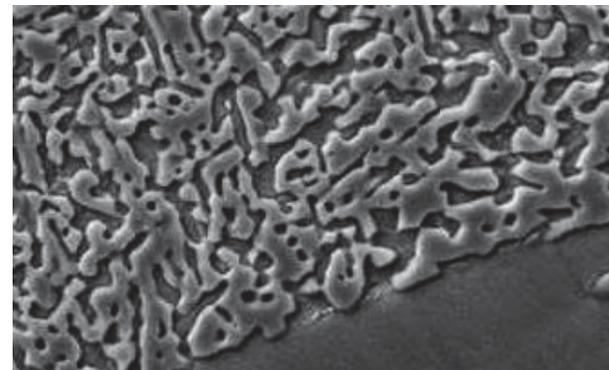
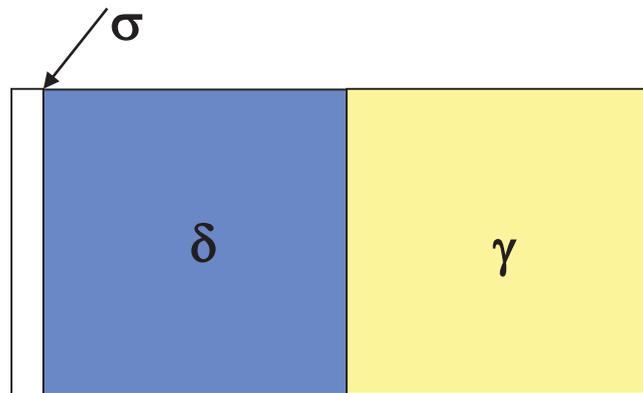


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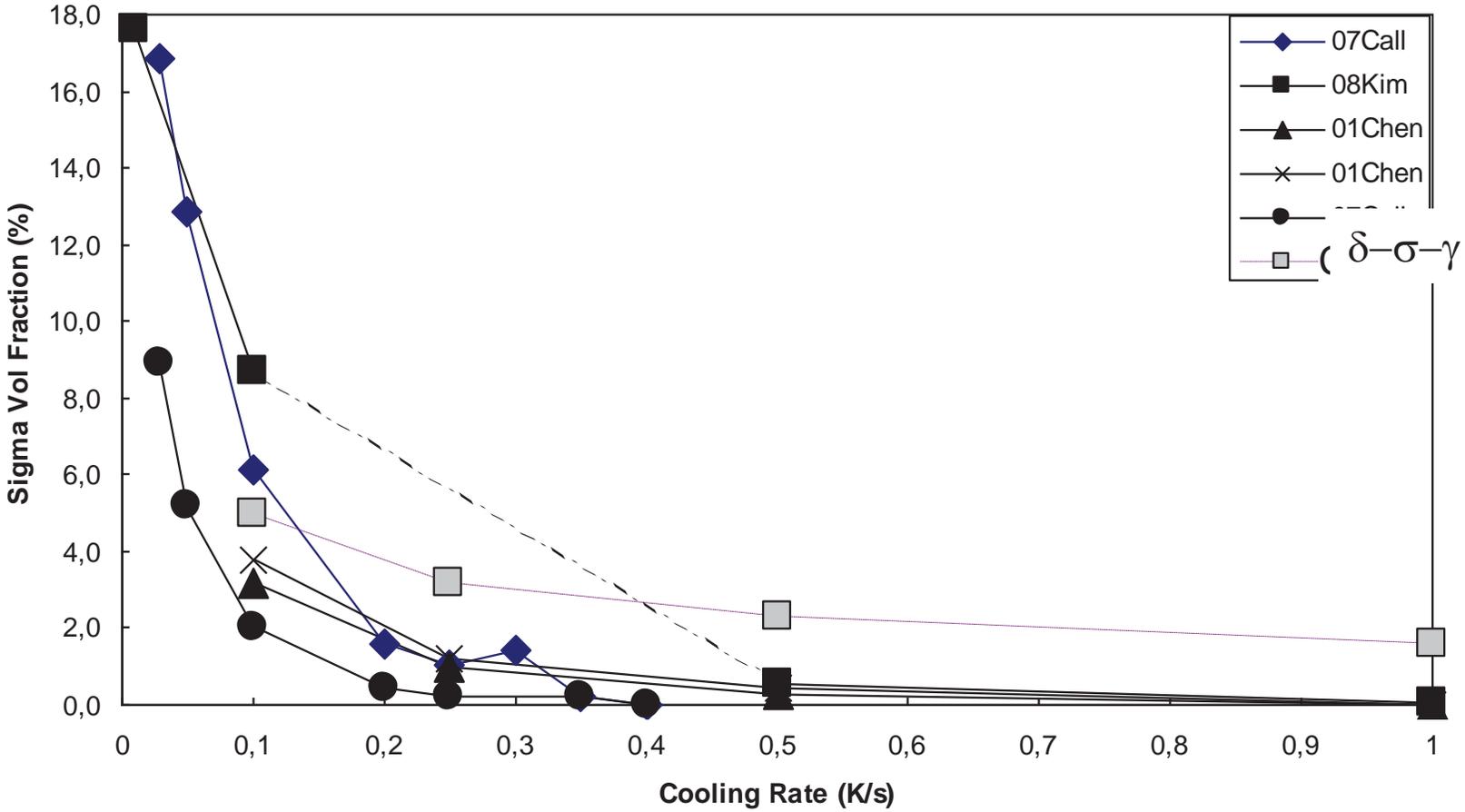
Modeling the formation of sigma at  $\gamma$ - $\delta$  interface



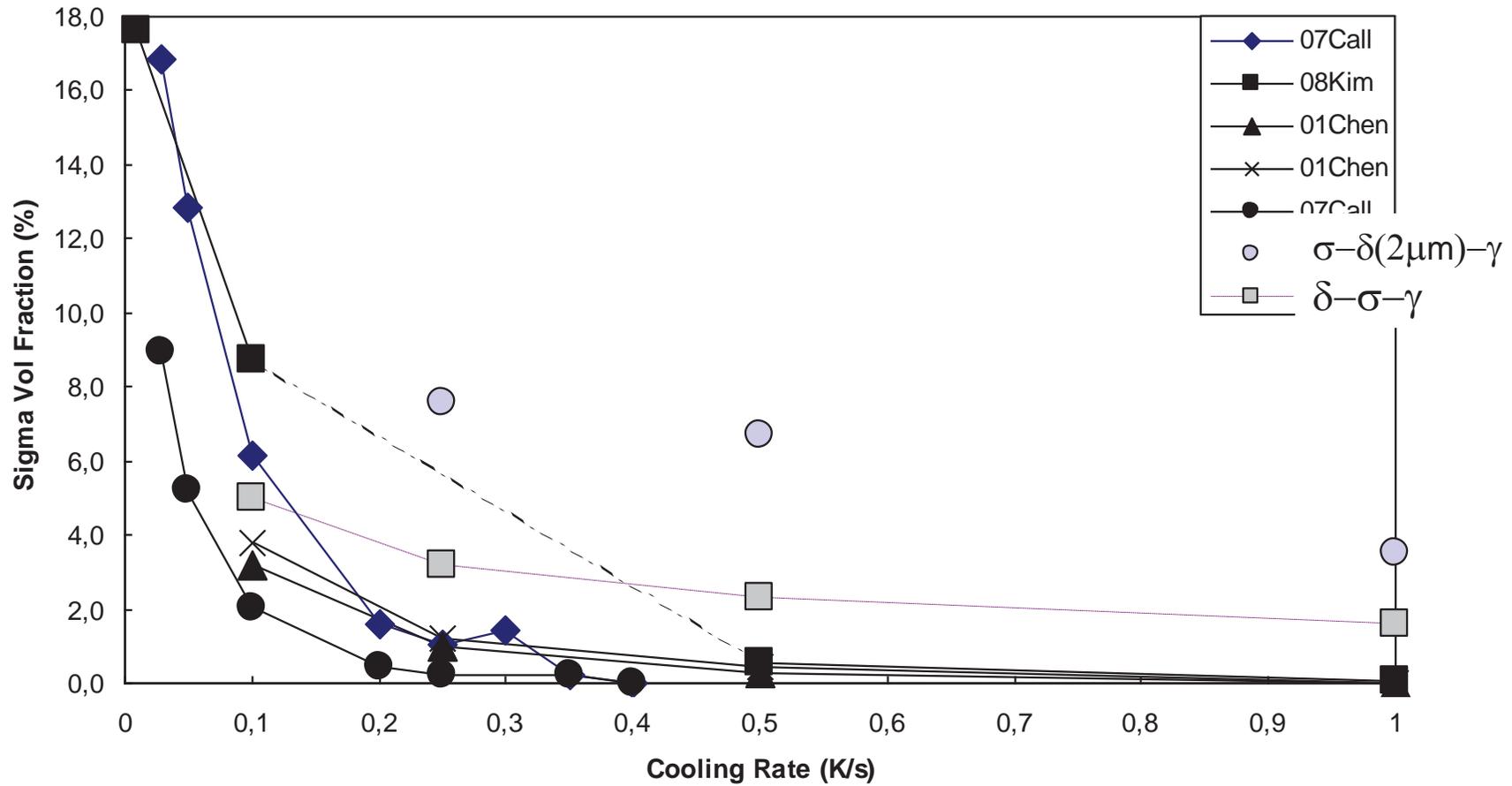
Cooperative growth - Modeling the growth



# DICTRA results of $\sigma$ and $\gamma$ growth into $\delta$

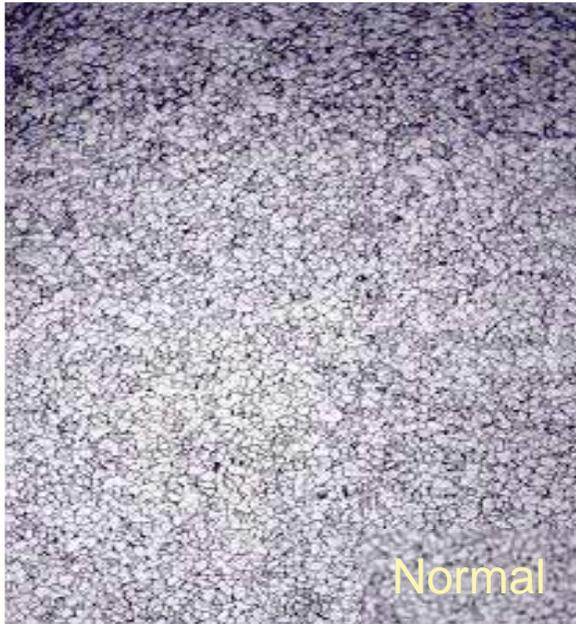


# DICTRA results of $\sigma$ and $\gamma$ growth into $\delta$



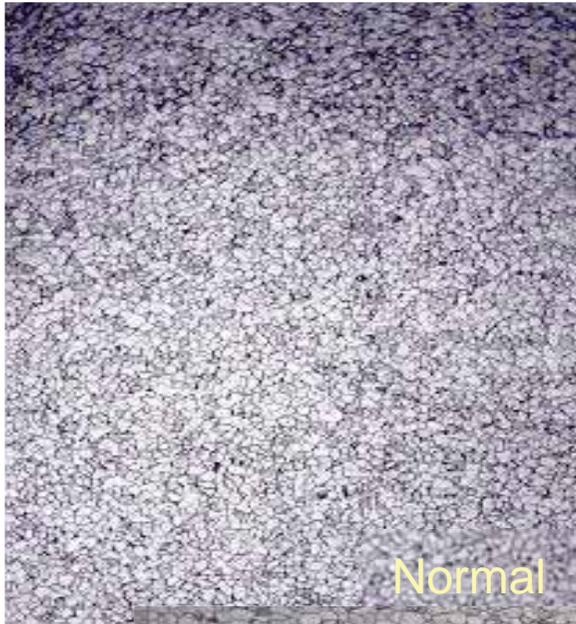
# N control in steel for cold forming

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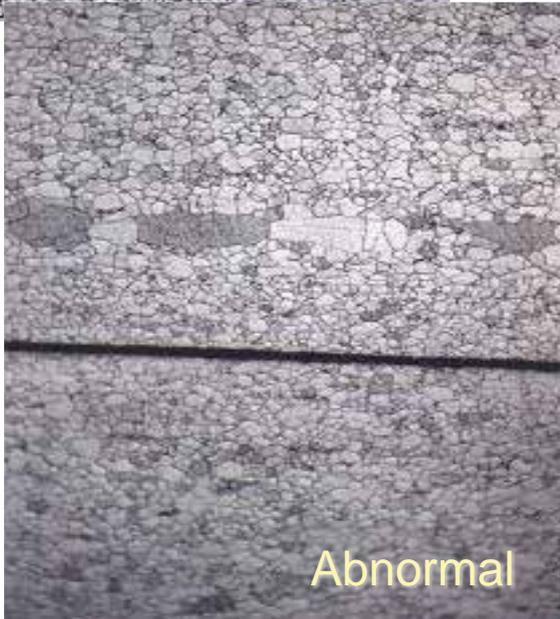


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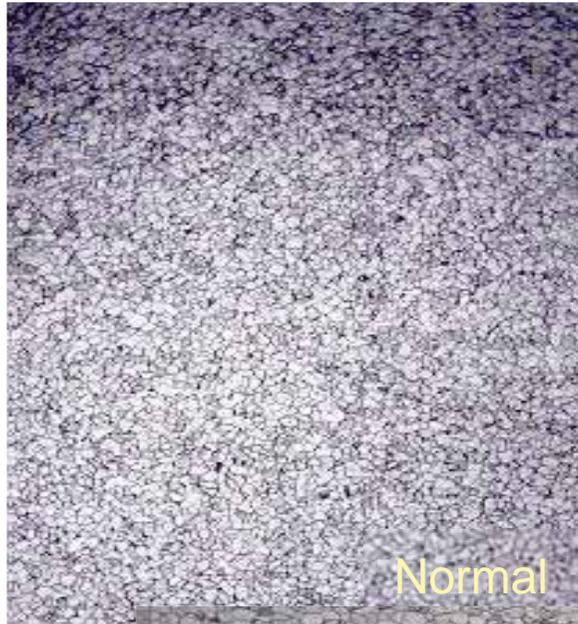


Normal

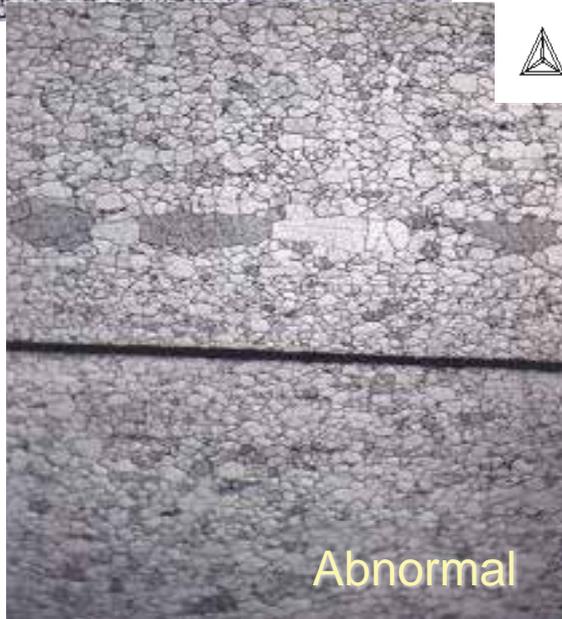


Abnormal

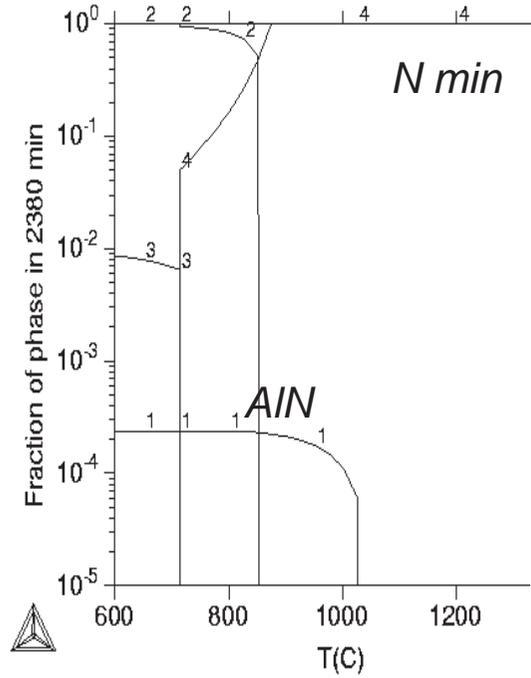
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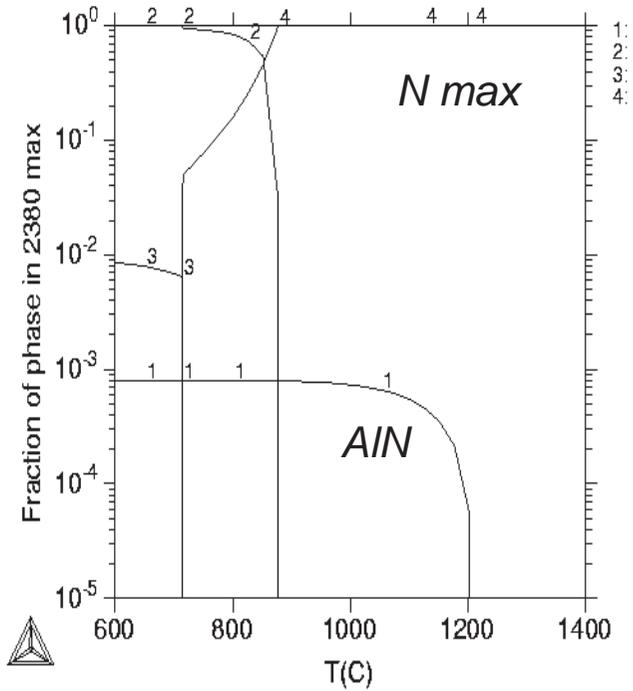
Normal



Abnormal



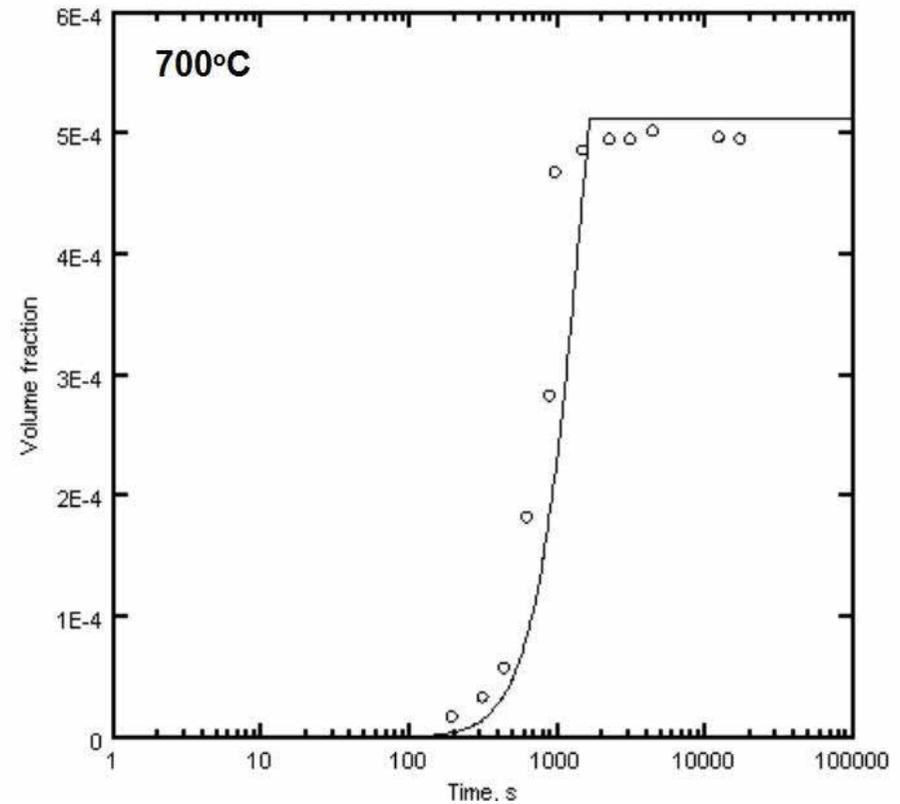
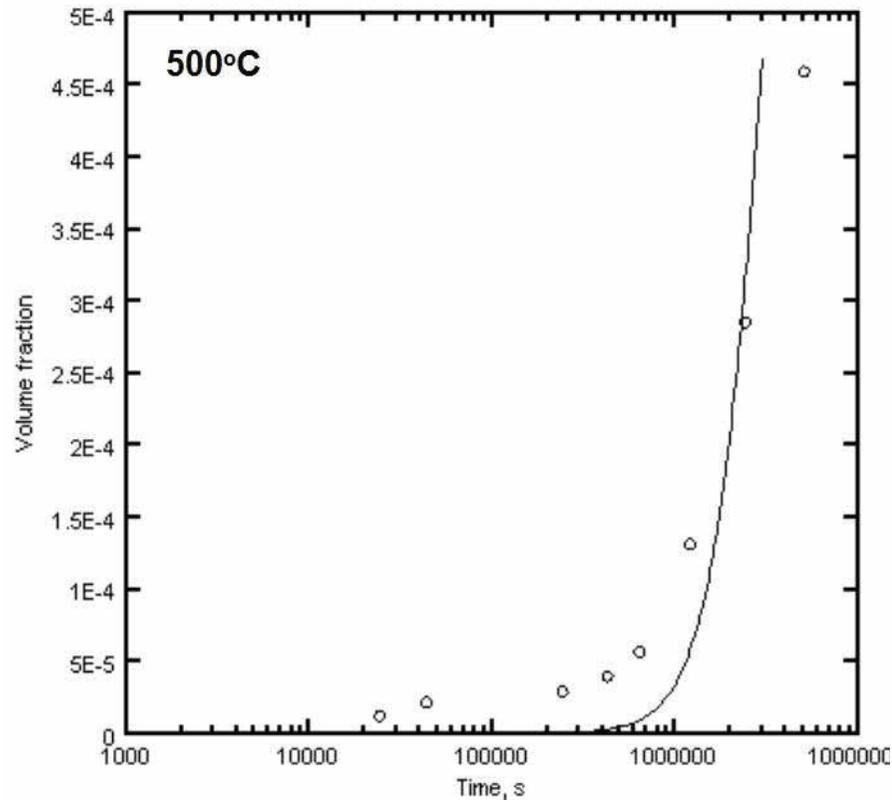
Rennó et al 1997



# Kinetics of Precipitation of AlN with CALPHAD data (PRISMA)

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# Final deposit (Yucca?) precipitation in Alloy 22 (NiCrMoW) (10.000 yrs)

## L. Kaufman, P. Turchi, DICTRA calculations

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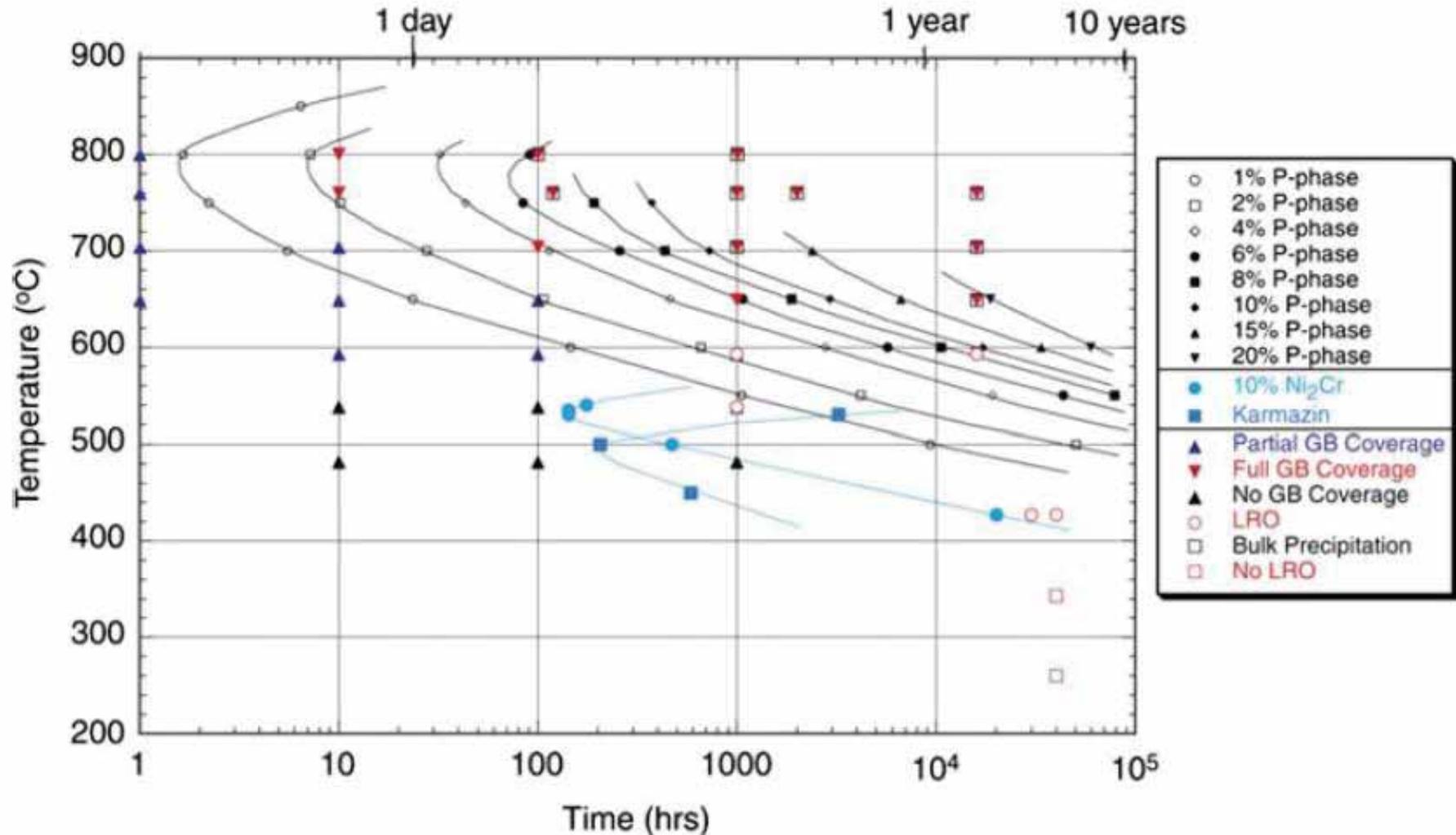
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For the “first” 10,000 years, the EPA would retain the dose limit of 15 millirem per year.

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# Final deposit (Yucca?) precipitation in Alloy 22 (NiCrMoW) (10.000 yrs) L. Kaufman, P. Turchi, DICTRA calculations

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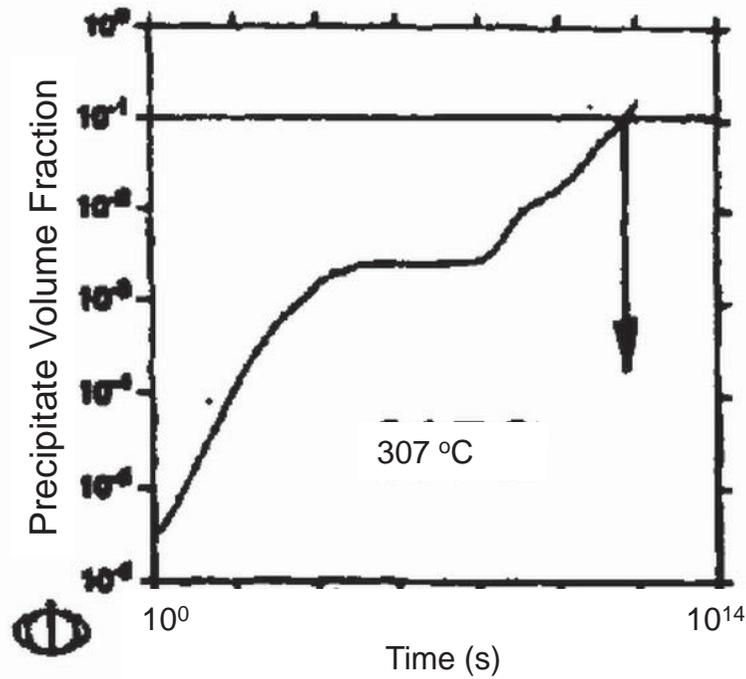
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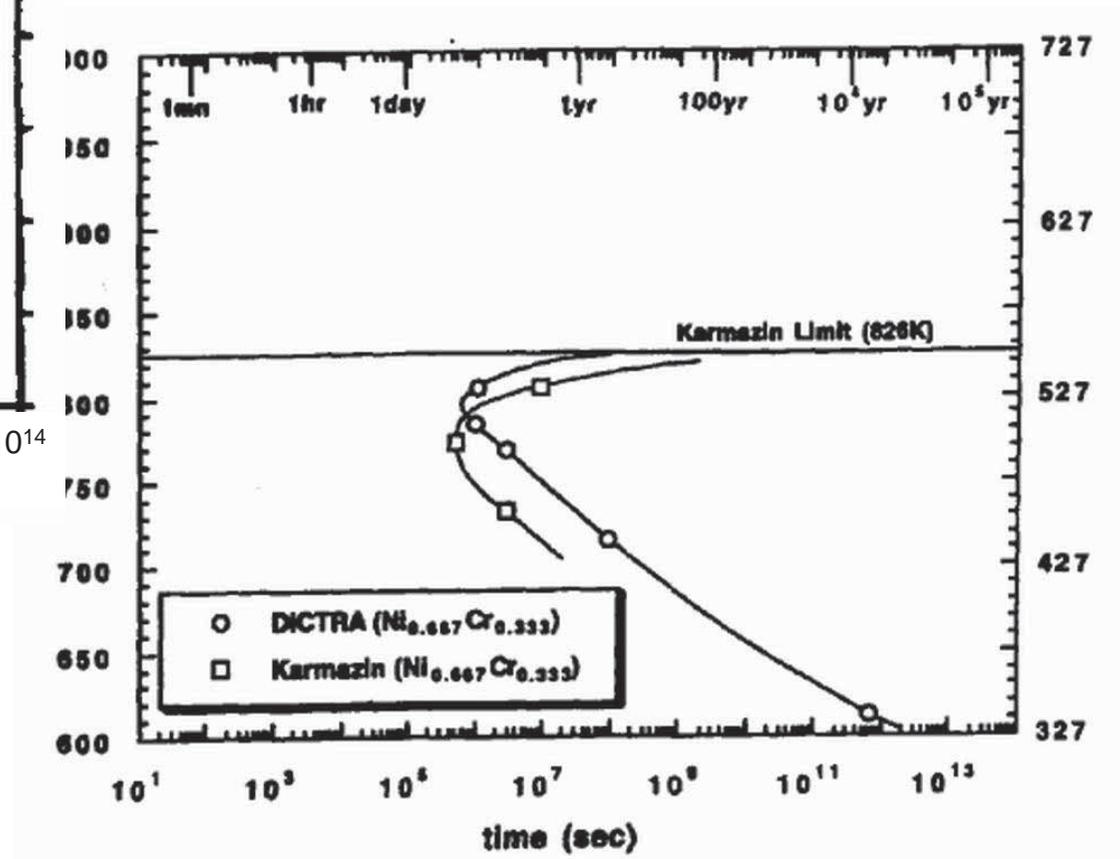
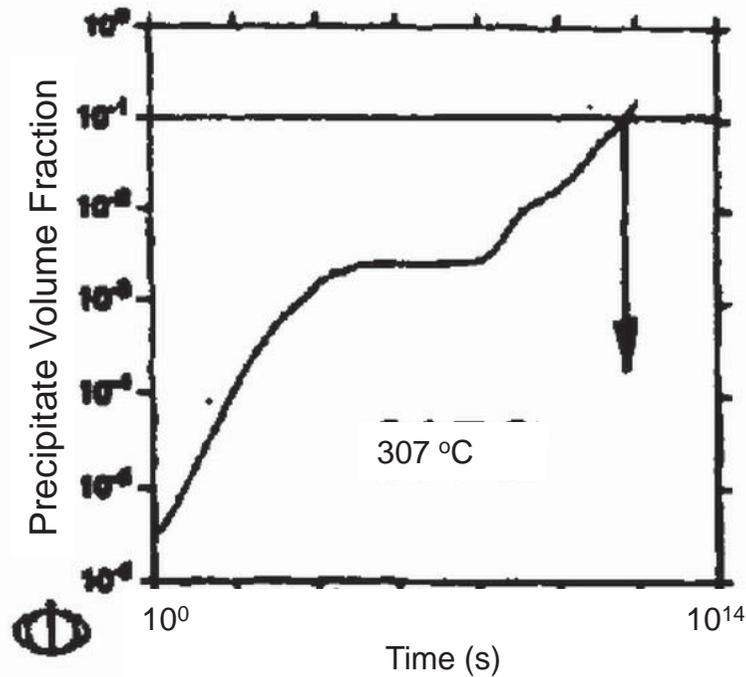
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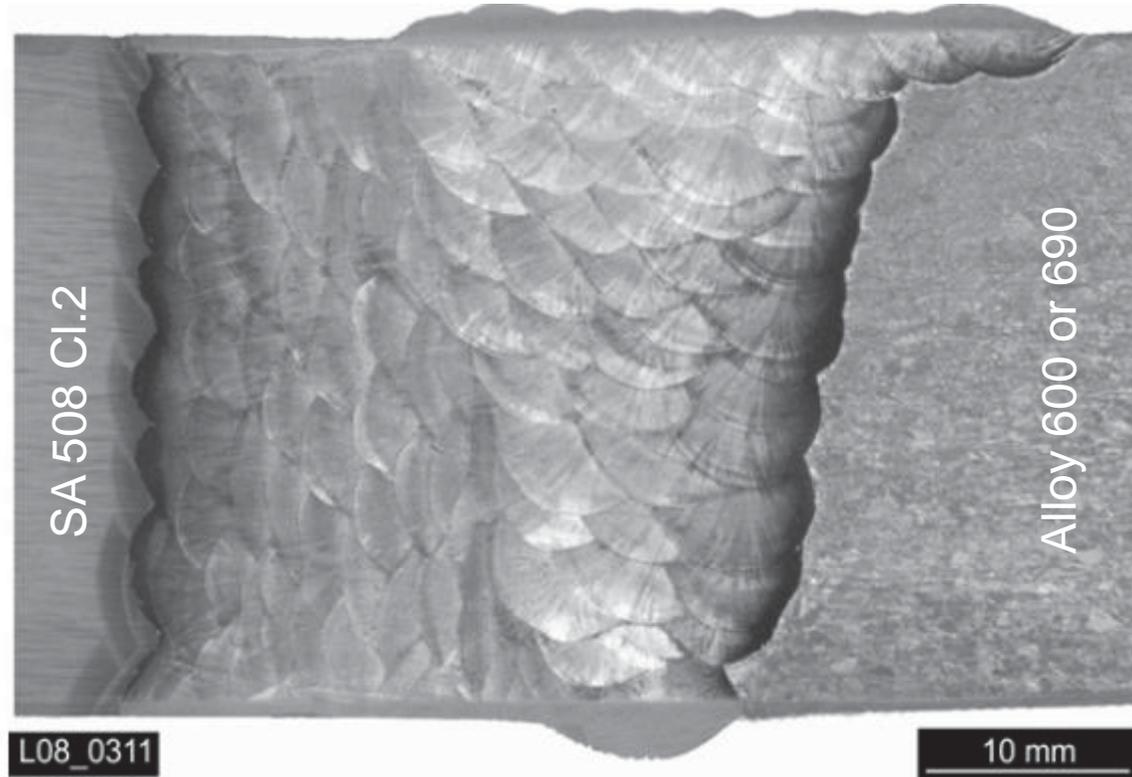
## L. Kaufman, P. Turchi, DICTRA calculations

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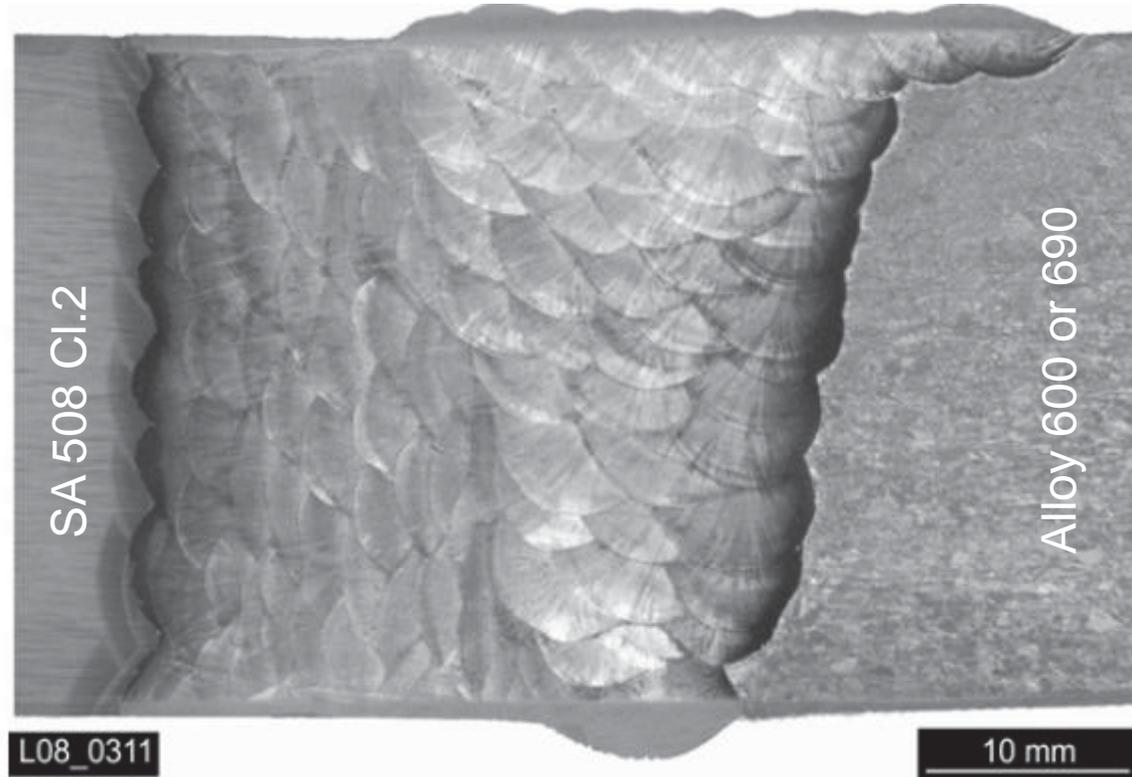
# Ageing of “safe ends” in PWR’s

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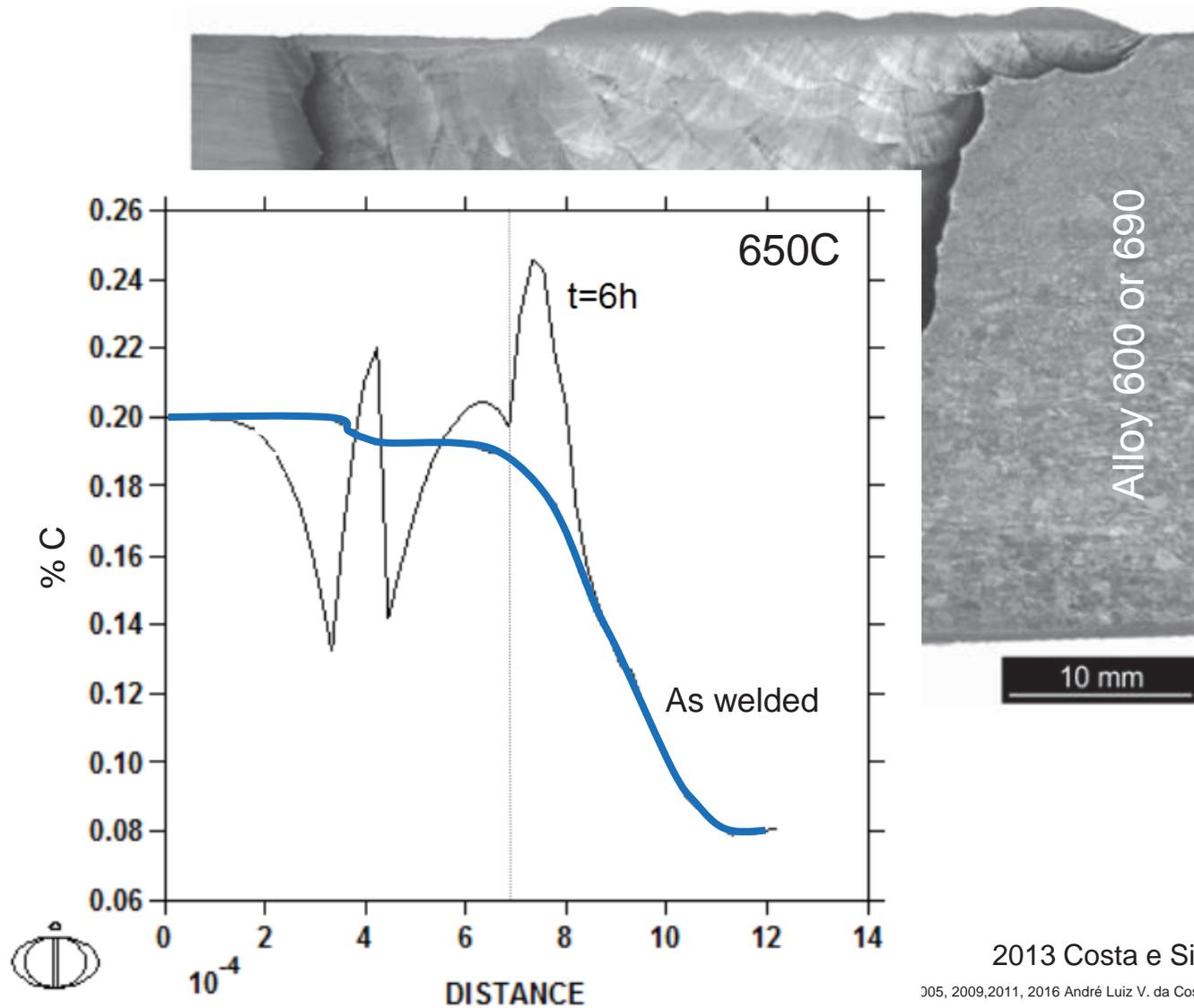


# Ageing of “safe ends” in PWR’s 40, 60, 100 yrs?

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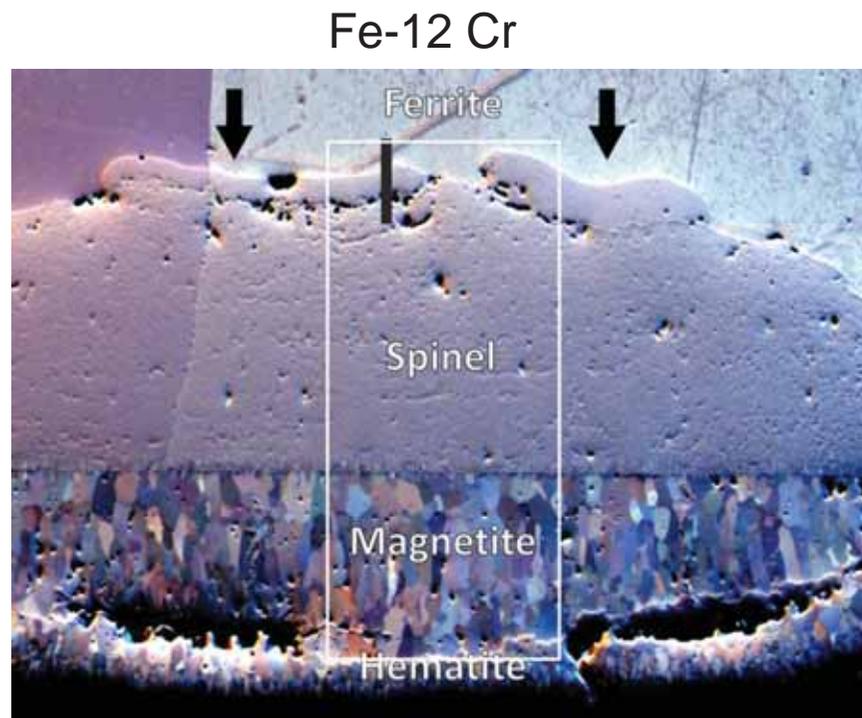
2013 Costa e Silva, Avillez

2005, 2009, 2011, 2016 André Luiz V. da Costa e Silva



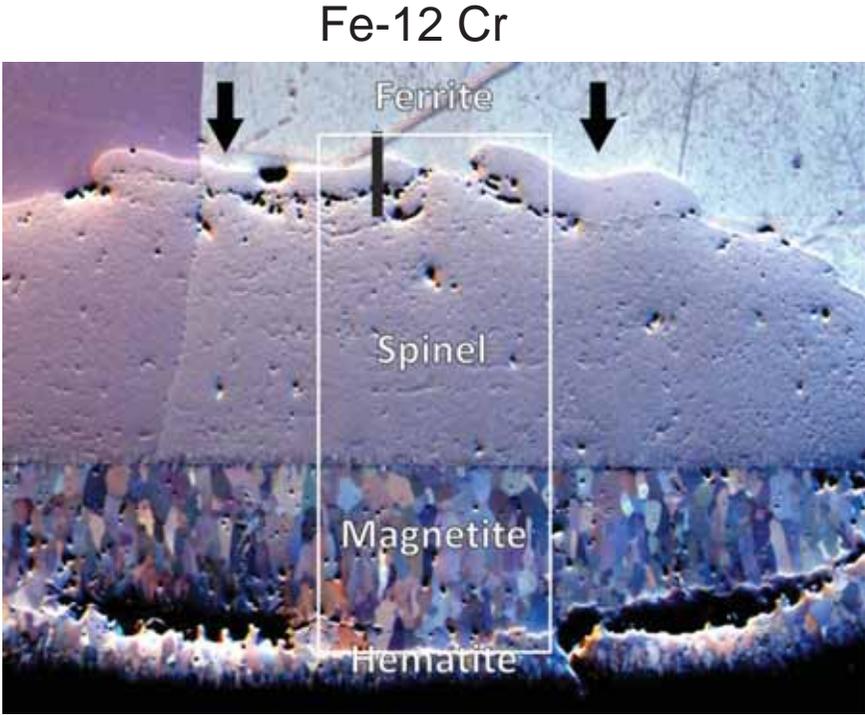
# Understanding internal oxidation for alloy design

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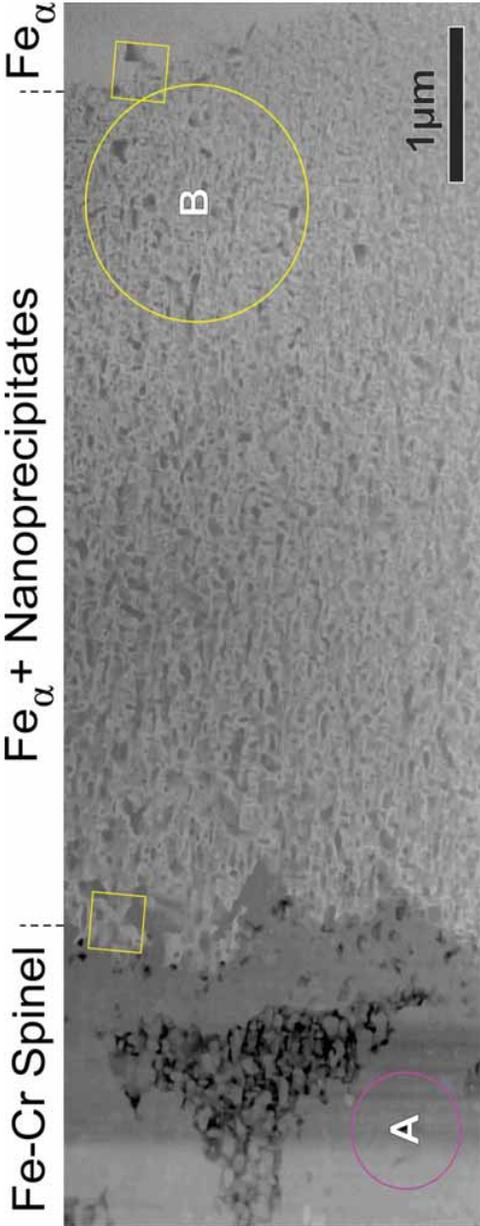


High T water vapor

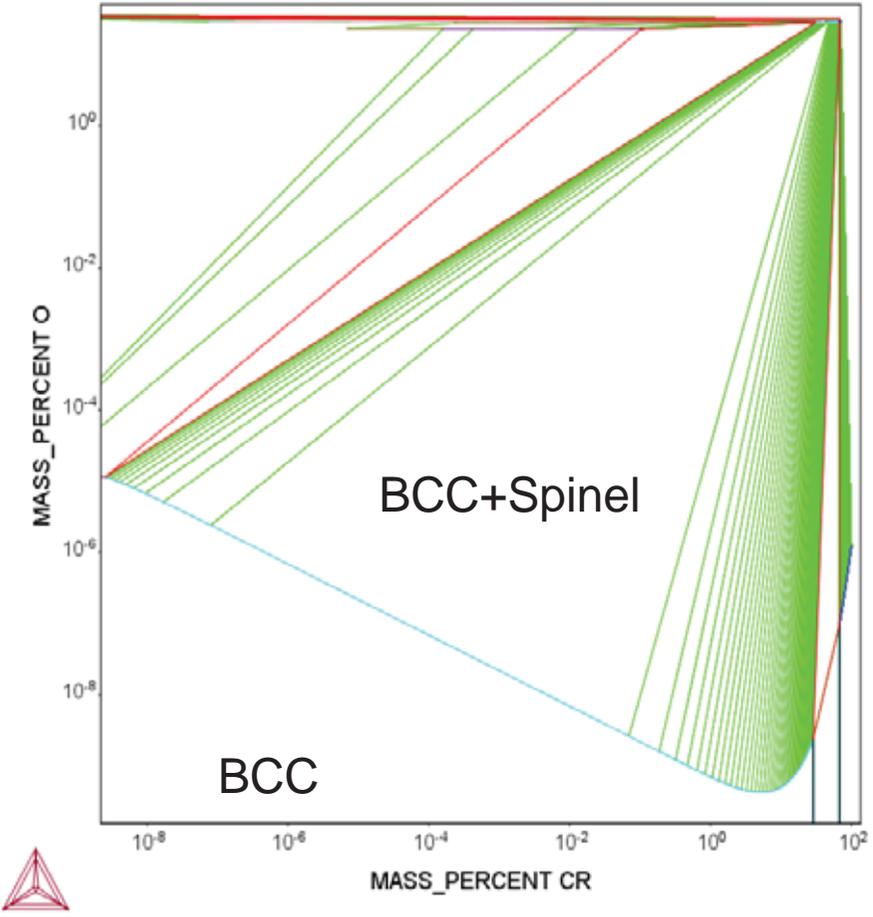
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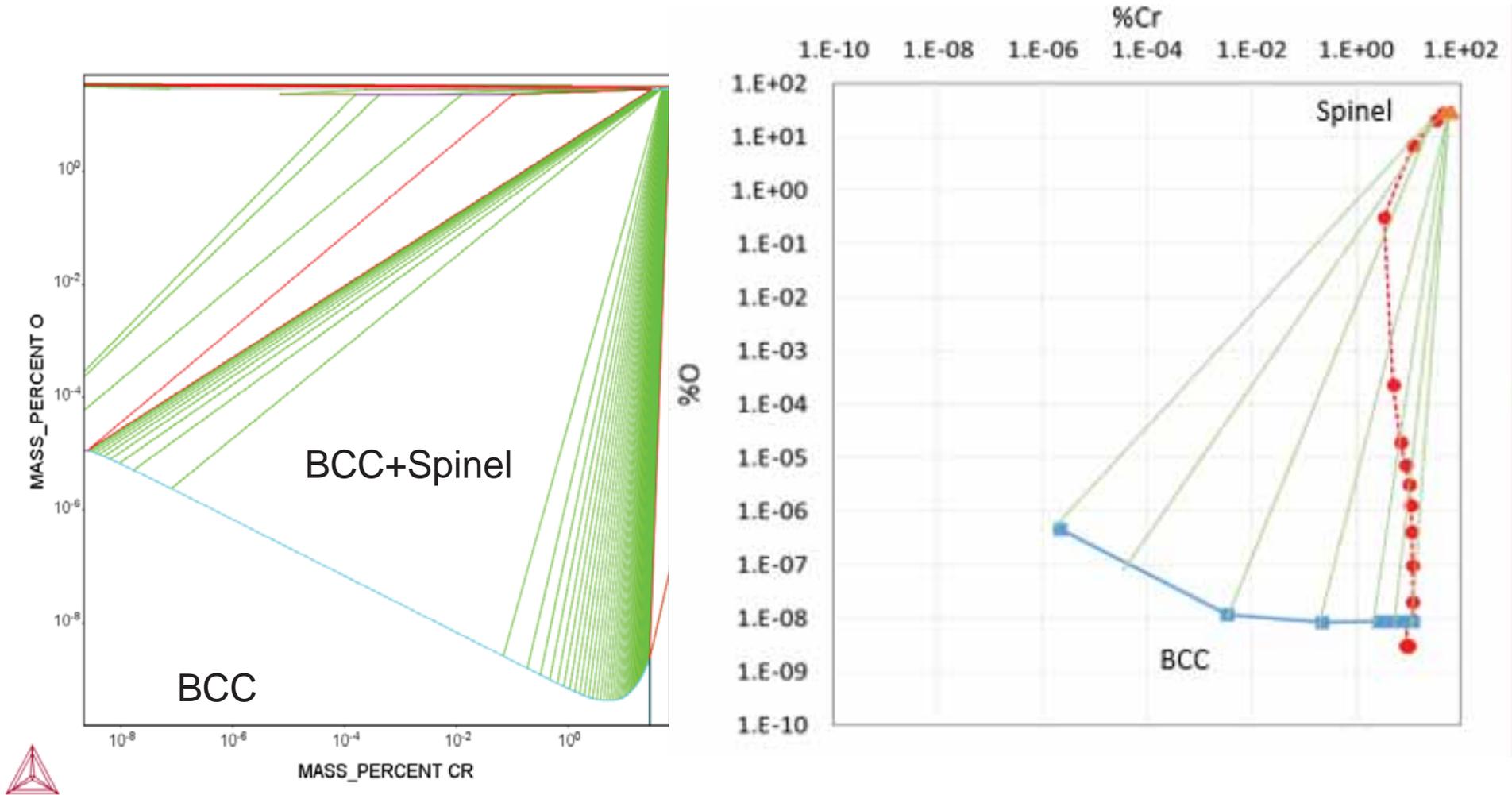
High T water vapor



# Diffusion path considering metastable oxidation



# Diffusion path considering metastable oxidation



Rizzo et al 2016

# Summary

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- **These range from steelmaking to alloy use and include alloy design.**
  
- **Computational thermodynamics can provide important insights that will help understanding the performance of steels during processing and applications.**
  
- **Computational thermodynamics can “perform” some “impossible” experiments that are relevant in alloy design.**



**Thank you!**