

Calculating phase transformations in duplex stainless steels using computational thermodynamics

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Acknowledgment:

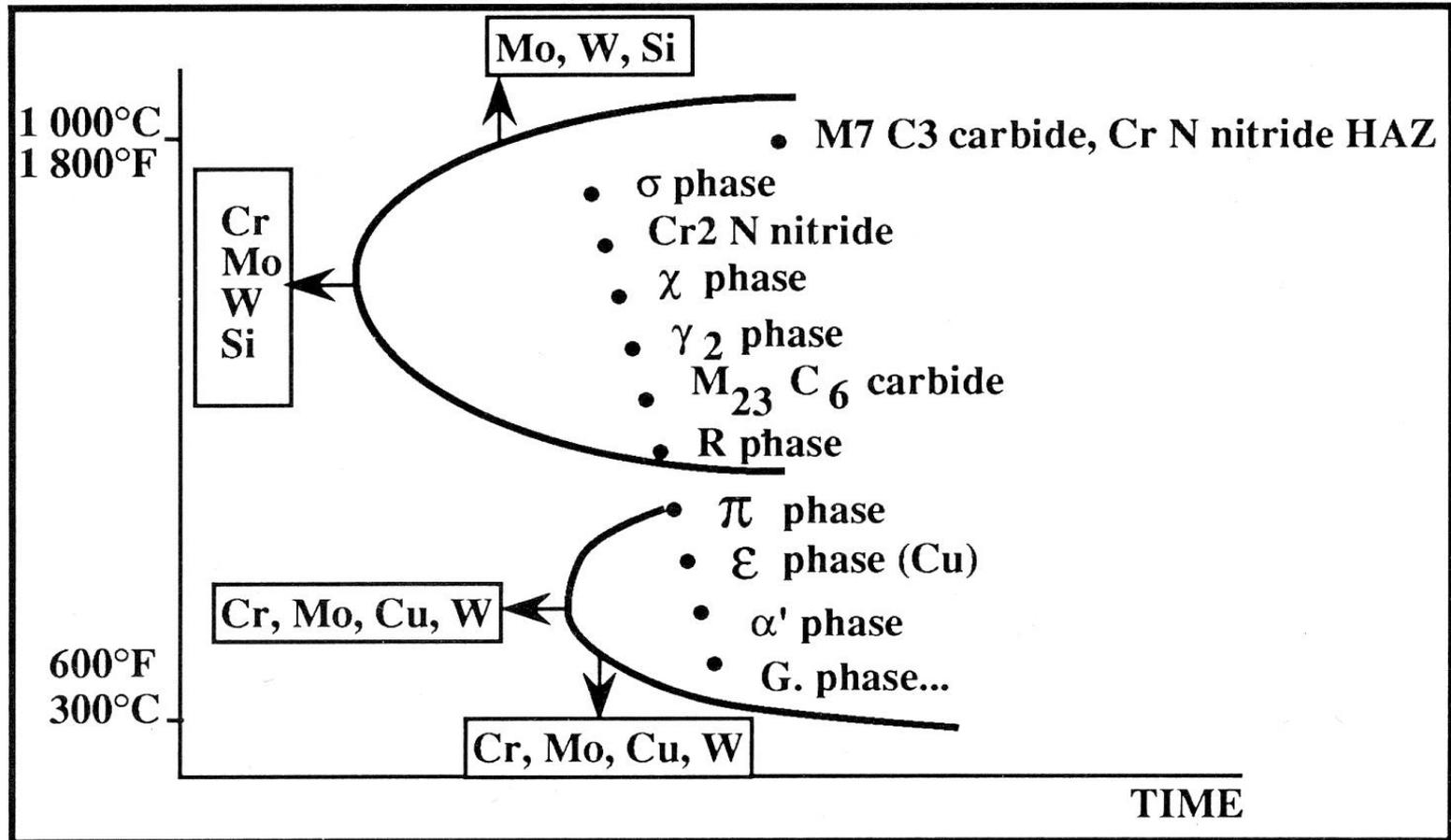
- The member research consortium MRC Stainless at Swerea KIMAB AB
- Outokumpu Stainless AB
- AB Sandvik Materials Technology
- KTH Royal Institute of Technology
- Jernkontoret - The Swedish Steel Producers' Association
- Thermo-Calc Software AB

Duplex stainless steels:

- Consist of about equal amounts of ferrite and austenite

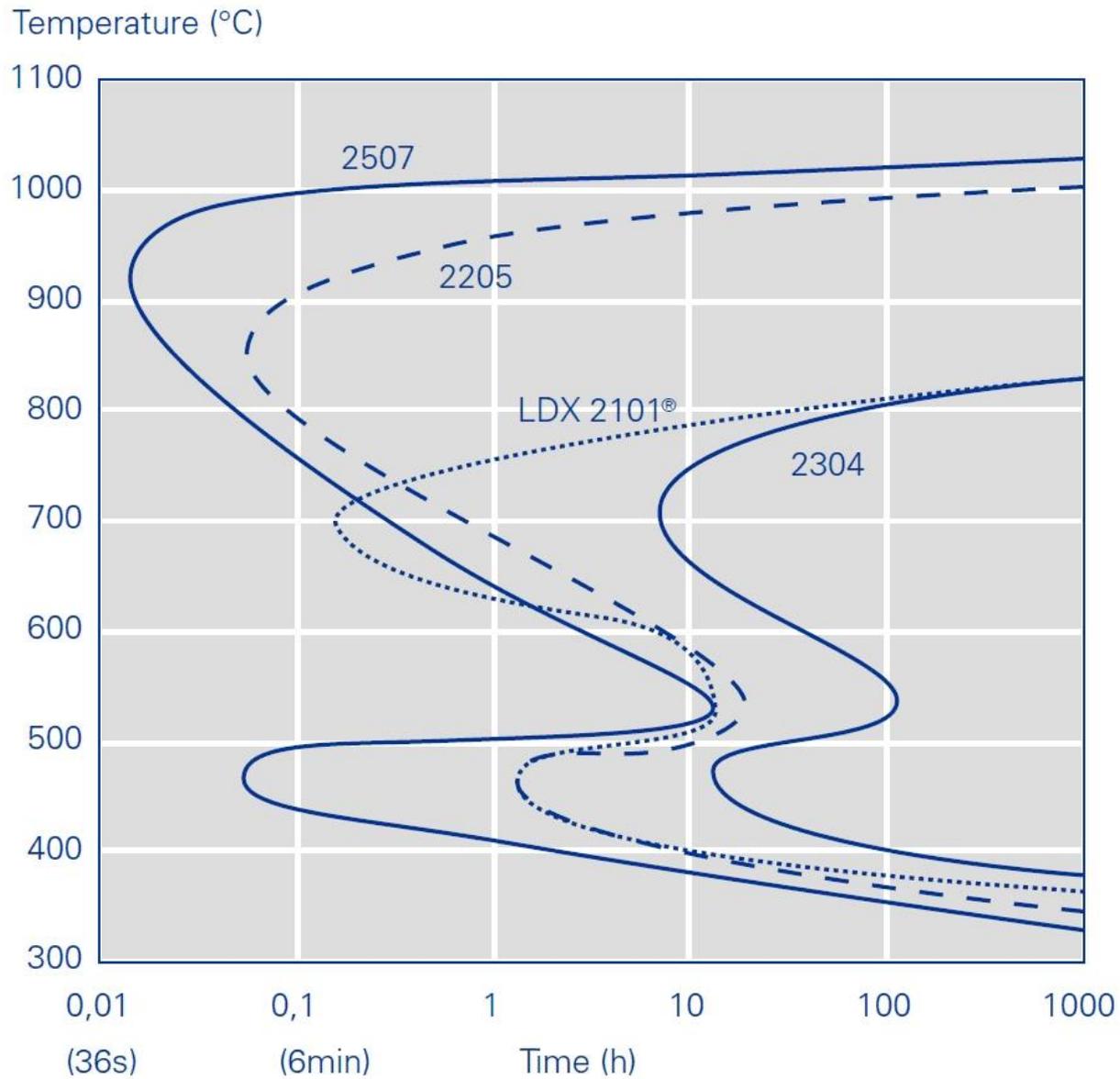
Lean: LDX 2101 [®]	Fe-5Mn- 21.5Cr - 1.5Ni -0.3Mo-0.22N
Standard: 2205	Fe- 22Cr - 5.5Ni -3Mo-0.17N
Super: 2507	Fe- 25Cr - 7Ni -4Mo-0.27N
Hyper: SAF 3207 HD [™]	Fe- 32Cr - 7Ni -3.5Mo-0.50N

- Doubled strength, compared to austenitics
 - Excellent corrosion resistance
 - Good weldability properties
 - Low Ni, compared to austenitics, i.e. favourable cost
 - Lean duplex: Ni partly substituted by Mn+N
- ☺ Cr+Mo = Corrosion resistance
☹ Cr+Mo = Intermetallic phases



J. Charles, Proc. Duplex Stainless Steels '91, Beaune, France, 1 (1991)

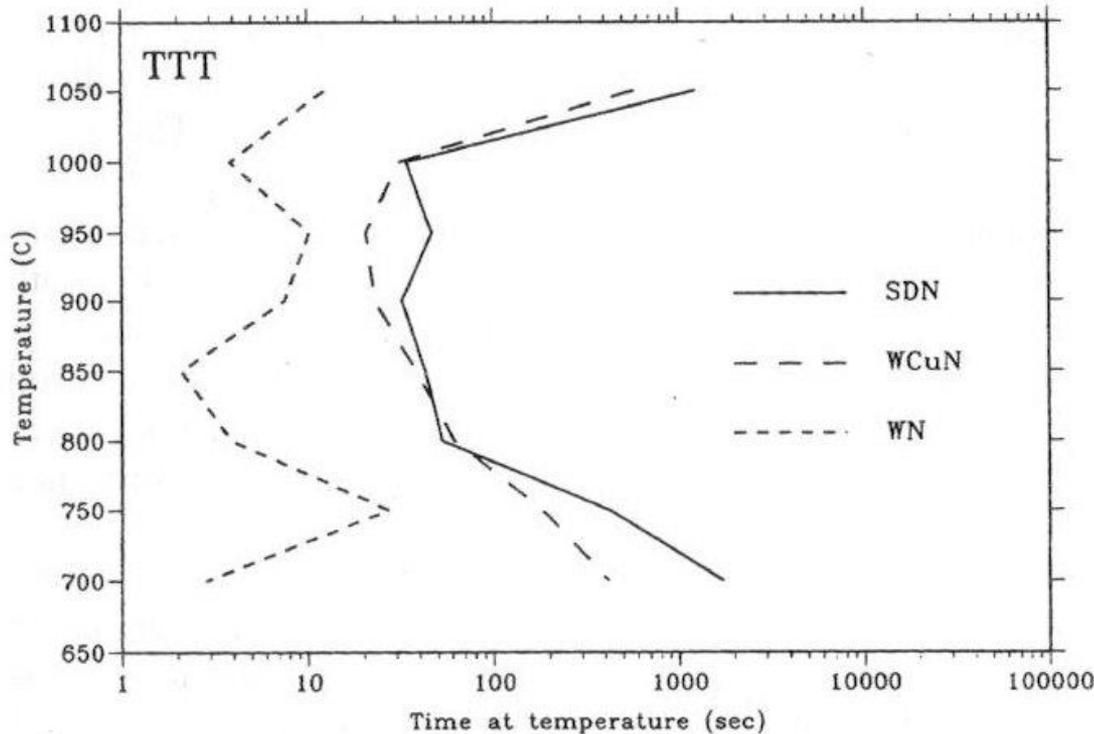
Schematic TTP (time to-precipitation) diagram



From Outokumpu Stainless AB, 50 % toughness loss

Chemical compositions [wt.-%]:

	C	Si	Mn	Cr	Ni	Mo	N	Cu	W
SDN	0.034	0.75	0.72	25.59	9.26	3.79	0.25	0.09	<0.003
WCuN	0.038	0.38	0.93	25.47	9.20	3.60	0.23	0.62	0.92
WN	0.046	0.69	1.34	25.67	9.60	3.09	0.24	0.09	2.34



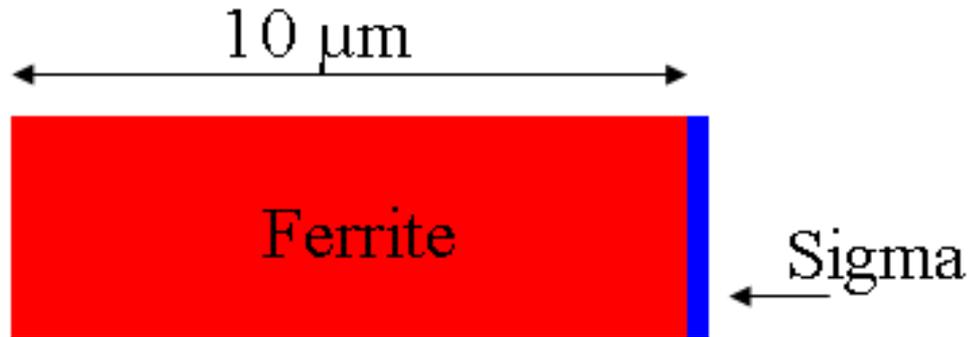
MMA welds:

- Isothermal heat treatments
- Optical microscopy
- SEM
- Point counting

TTT diagram showing C-curves representing 1 % of intermetallic phase.

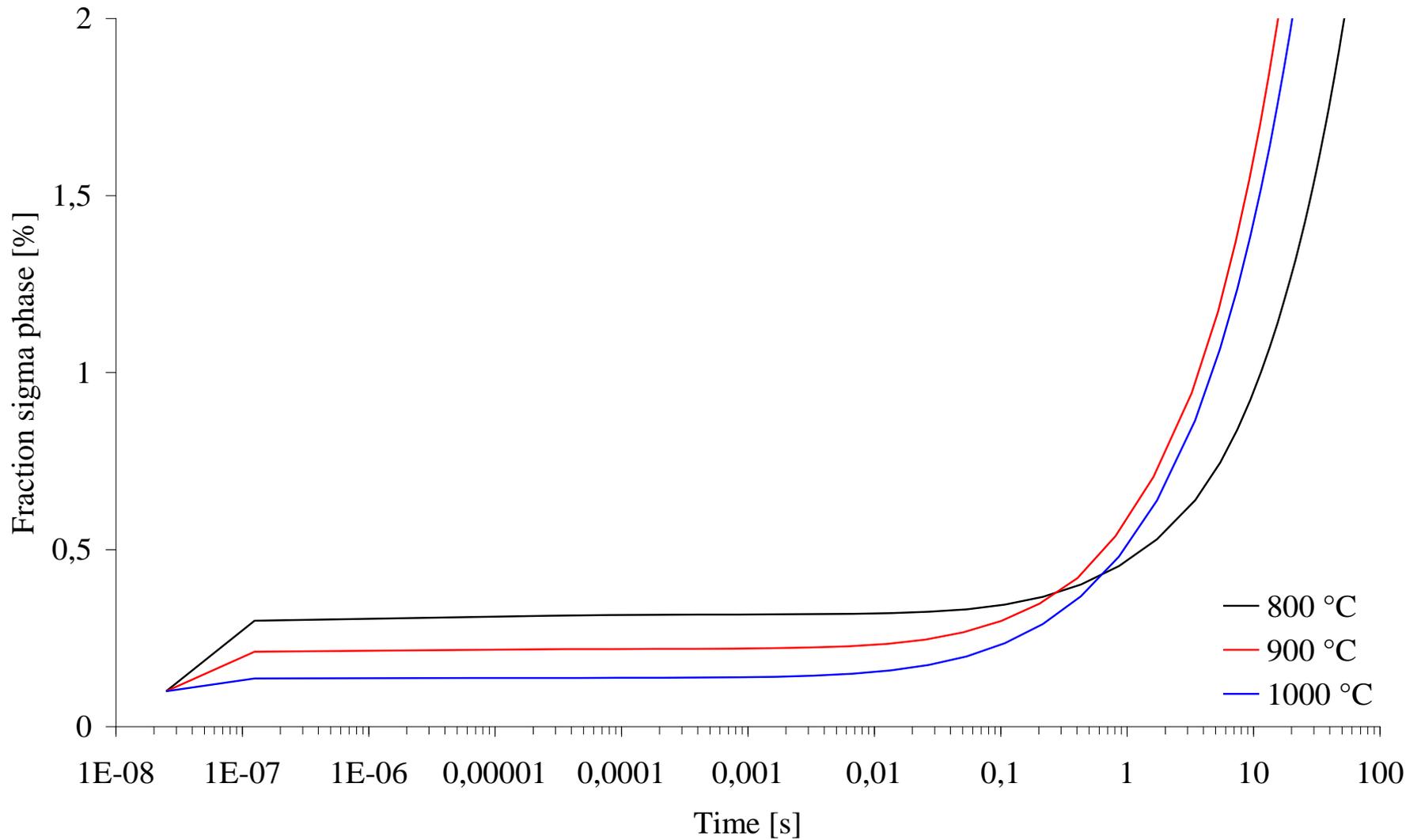
Simplified chemical compositions [wt.-%]:

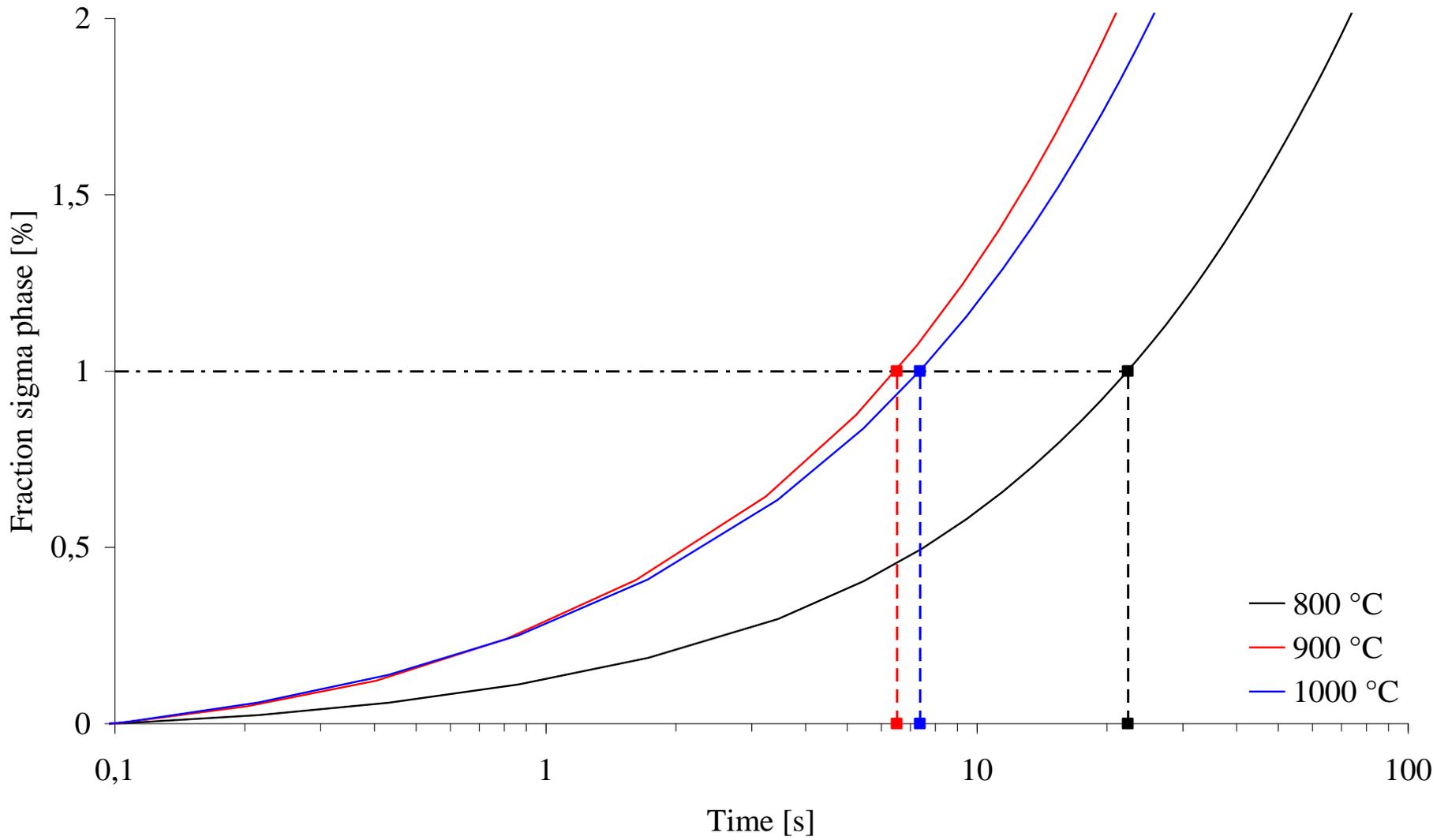
	C	Si	Mn	Cr	Ni	Mo	N	Cu	W
SDN	-	-	-	25.59	9.35	3.79	0.284	-	-
WCuN	-	-	-	25.47	9.82	3.60	0.268	-	0.92
WN	-	-	-	25.67	9.69	3.09	0.286	-	2.34

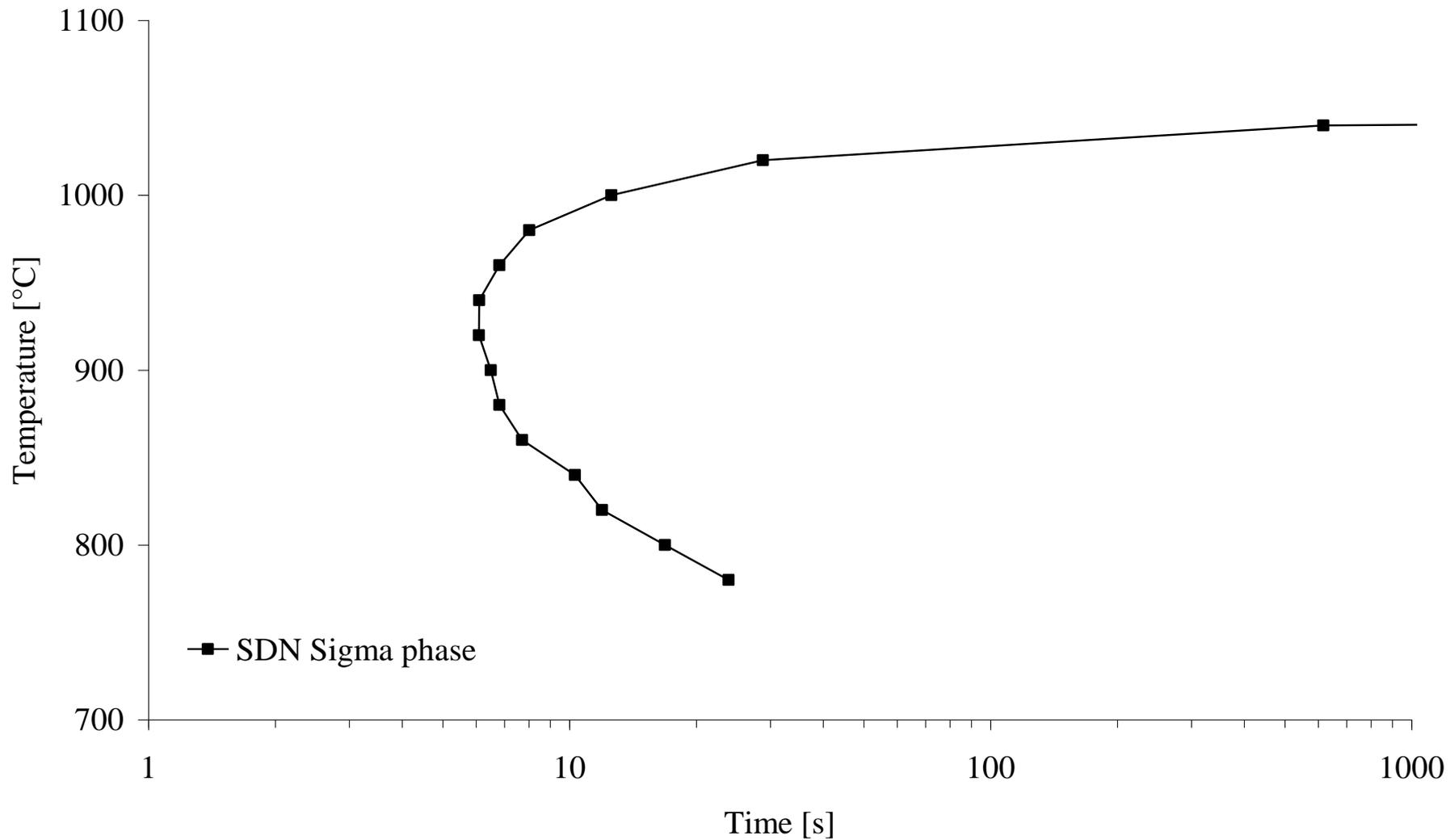


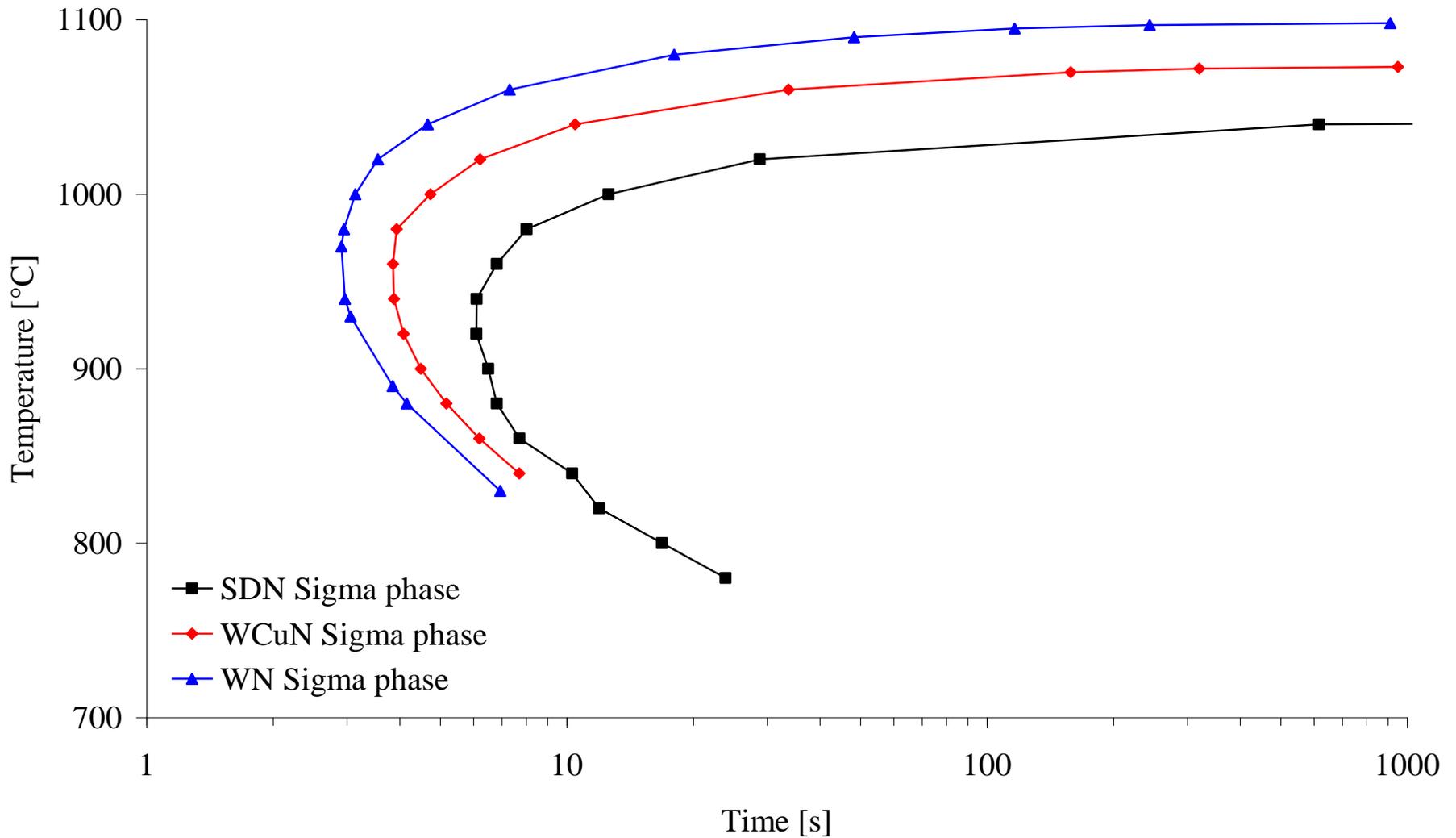
Approach:

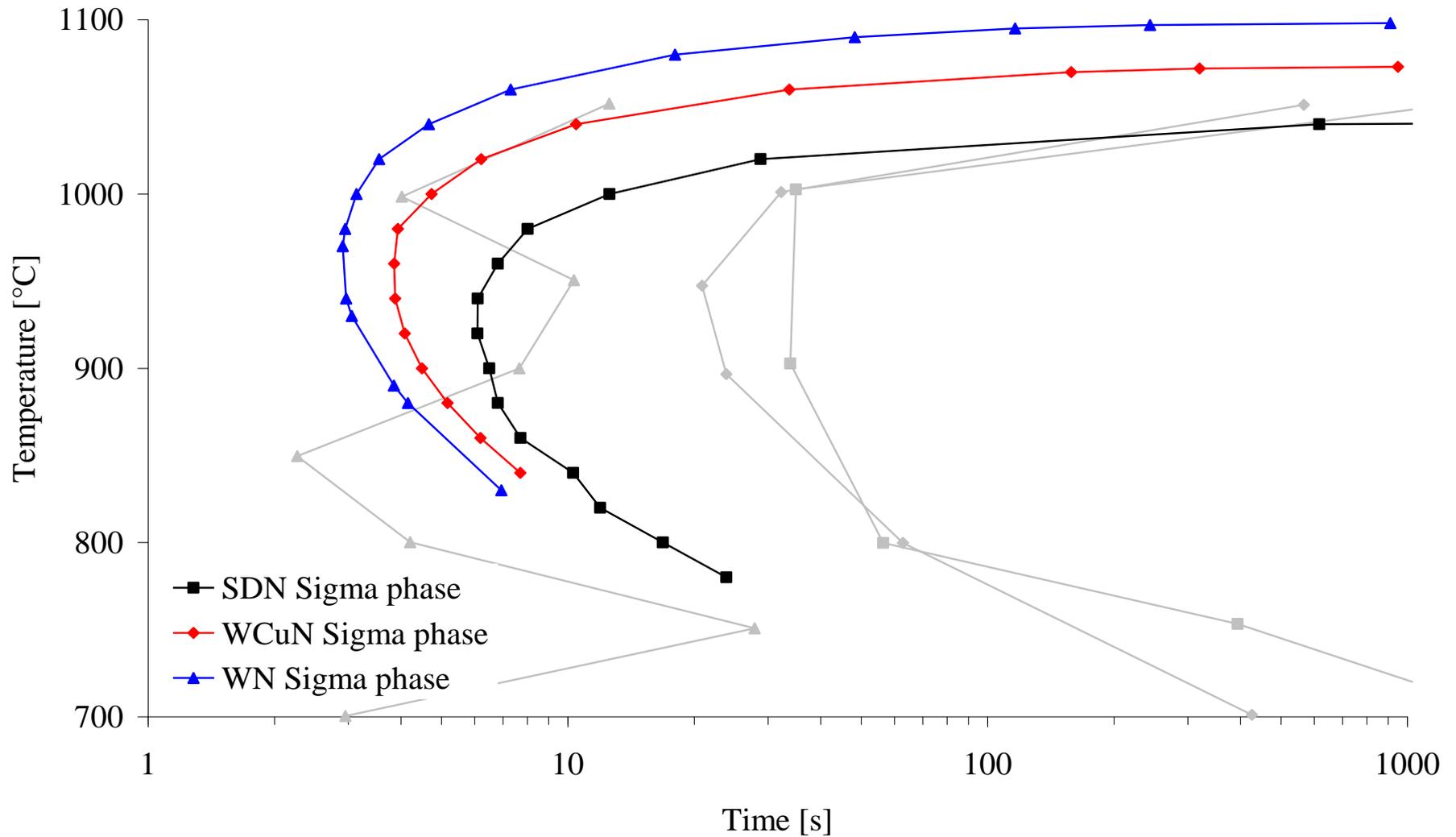
- Omit the austenite, only ferrite and austenite
- Simplifications: Si & Mn omitted; C → N and Cu → Ni
- Thermo-Calc: Calculate the ferrite composition at 1100 °C (1150 °C WN)
- 10 µm ferrite → DICTRA; TCFE6 & MOB1
- **Time to grow 1 % sigma phase**



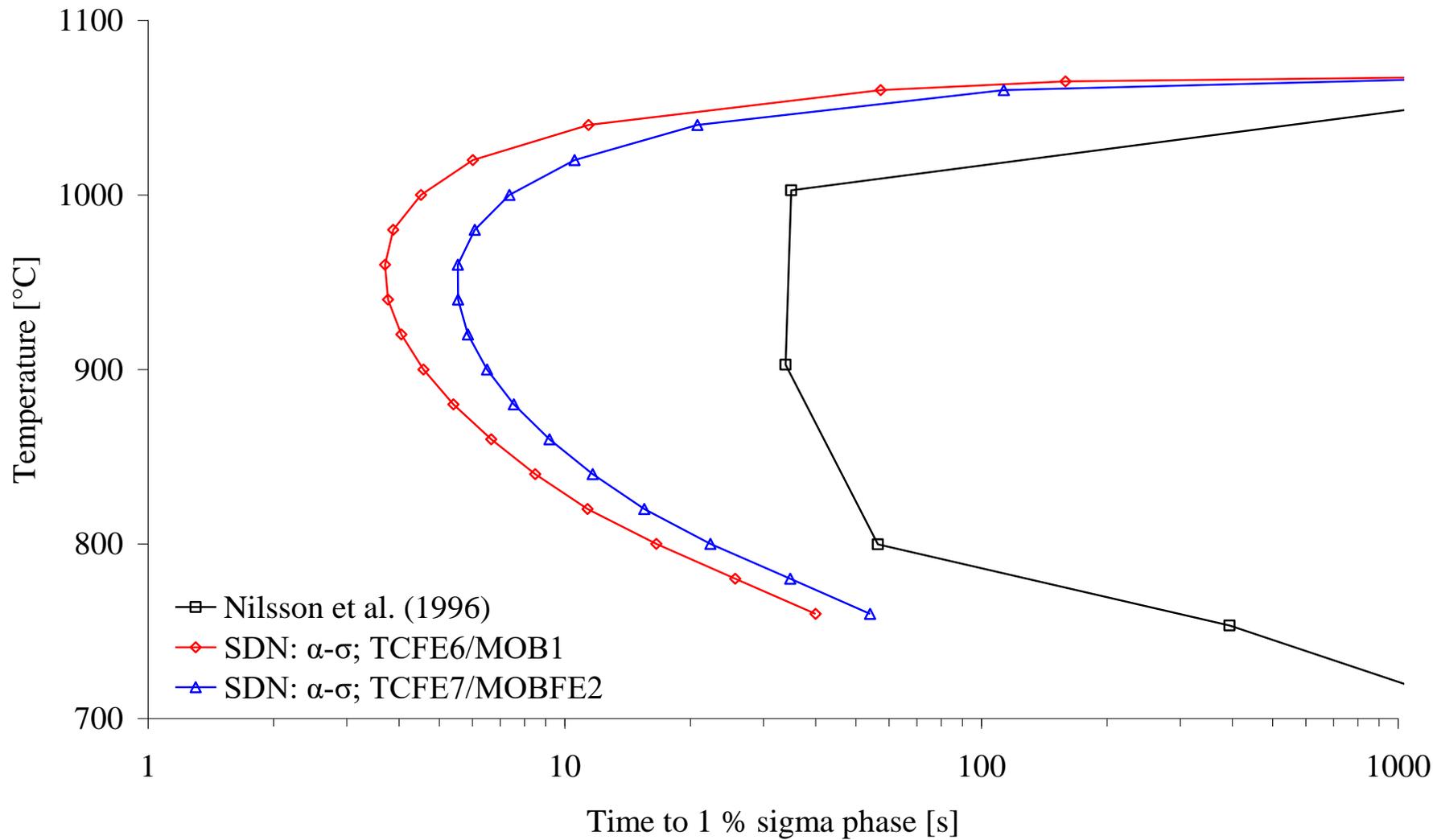


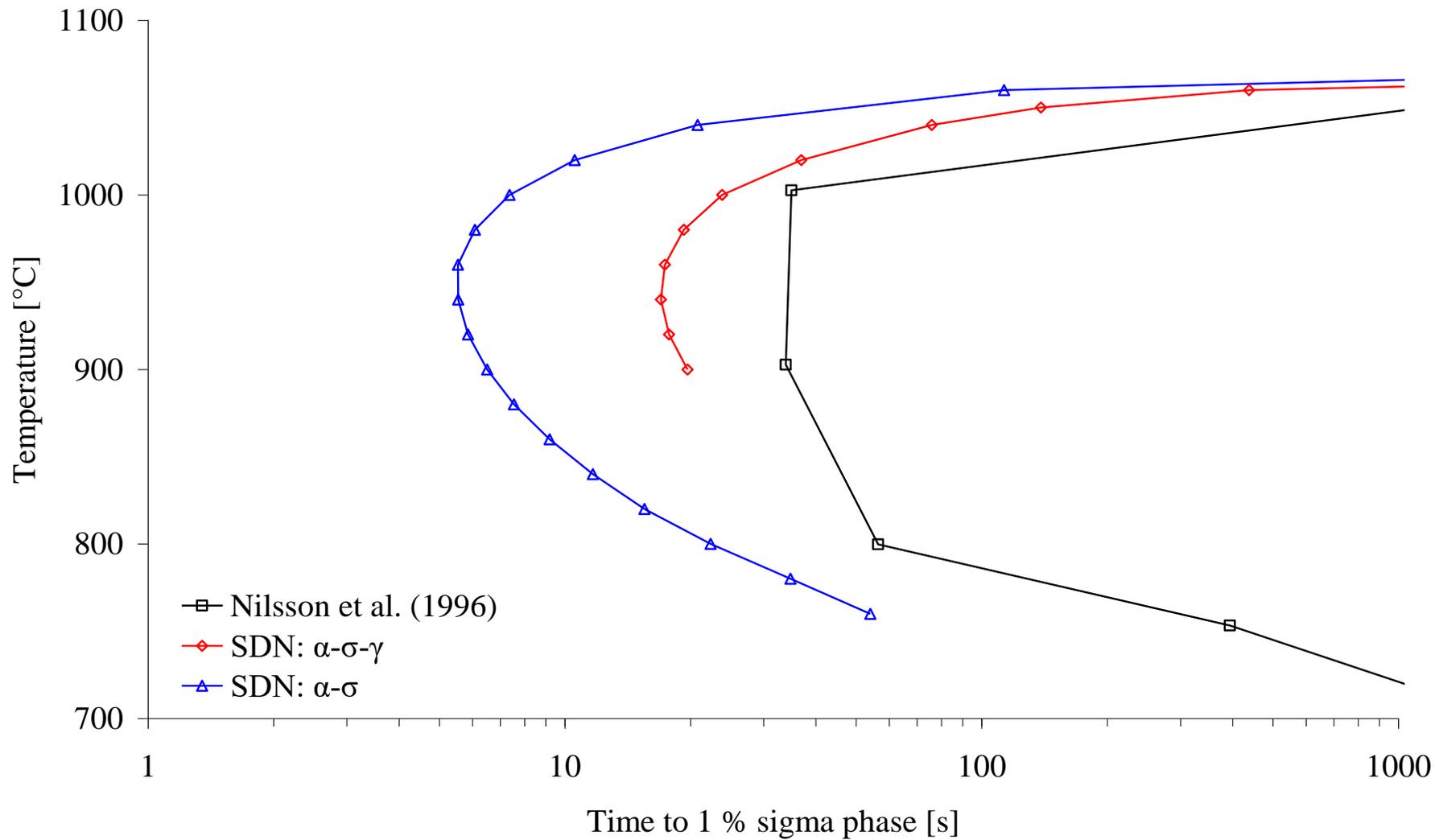




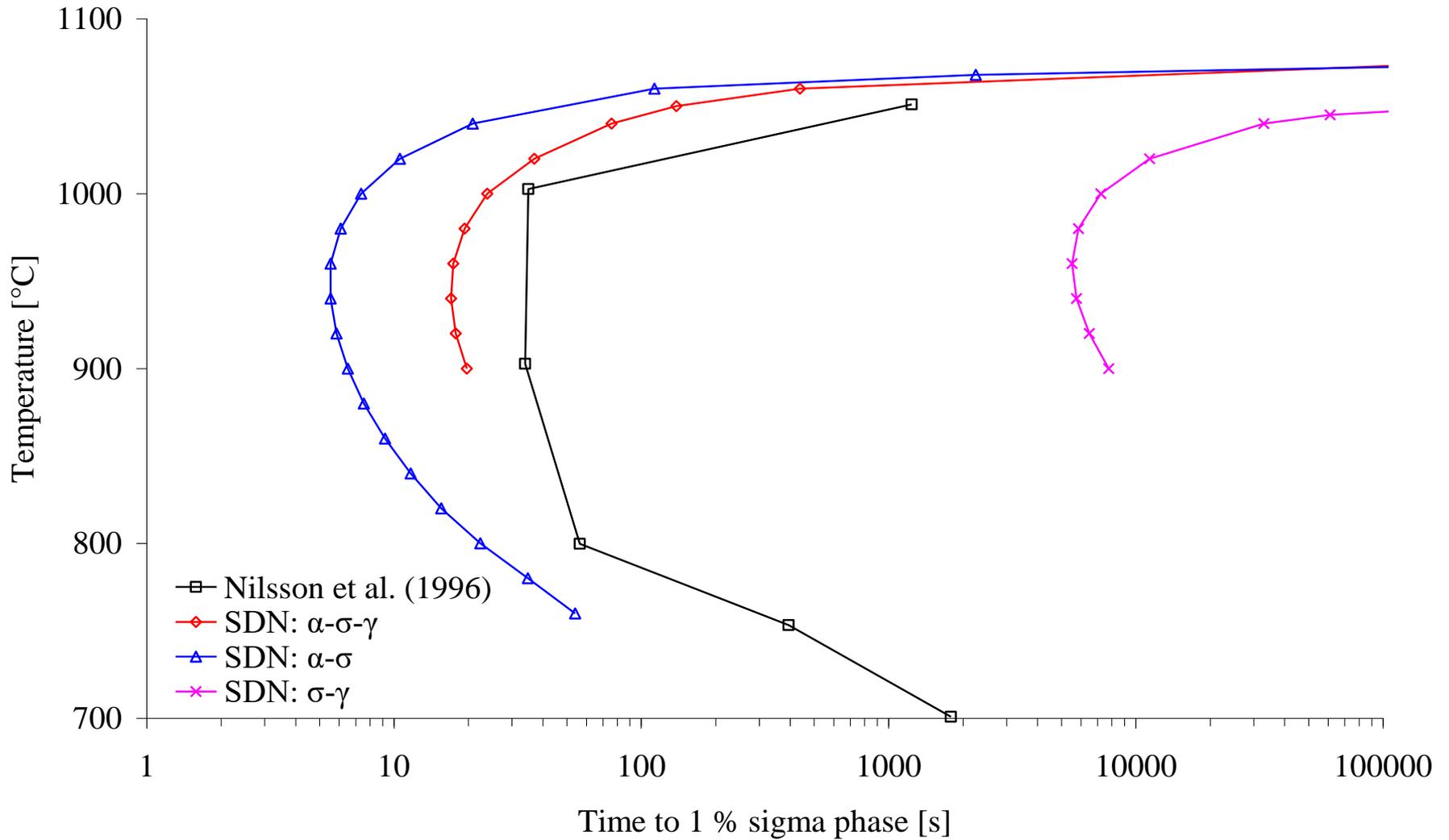


SDN, WCuN & WN: TTT-diagrams

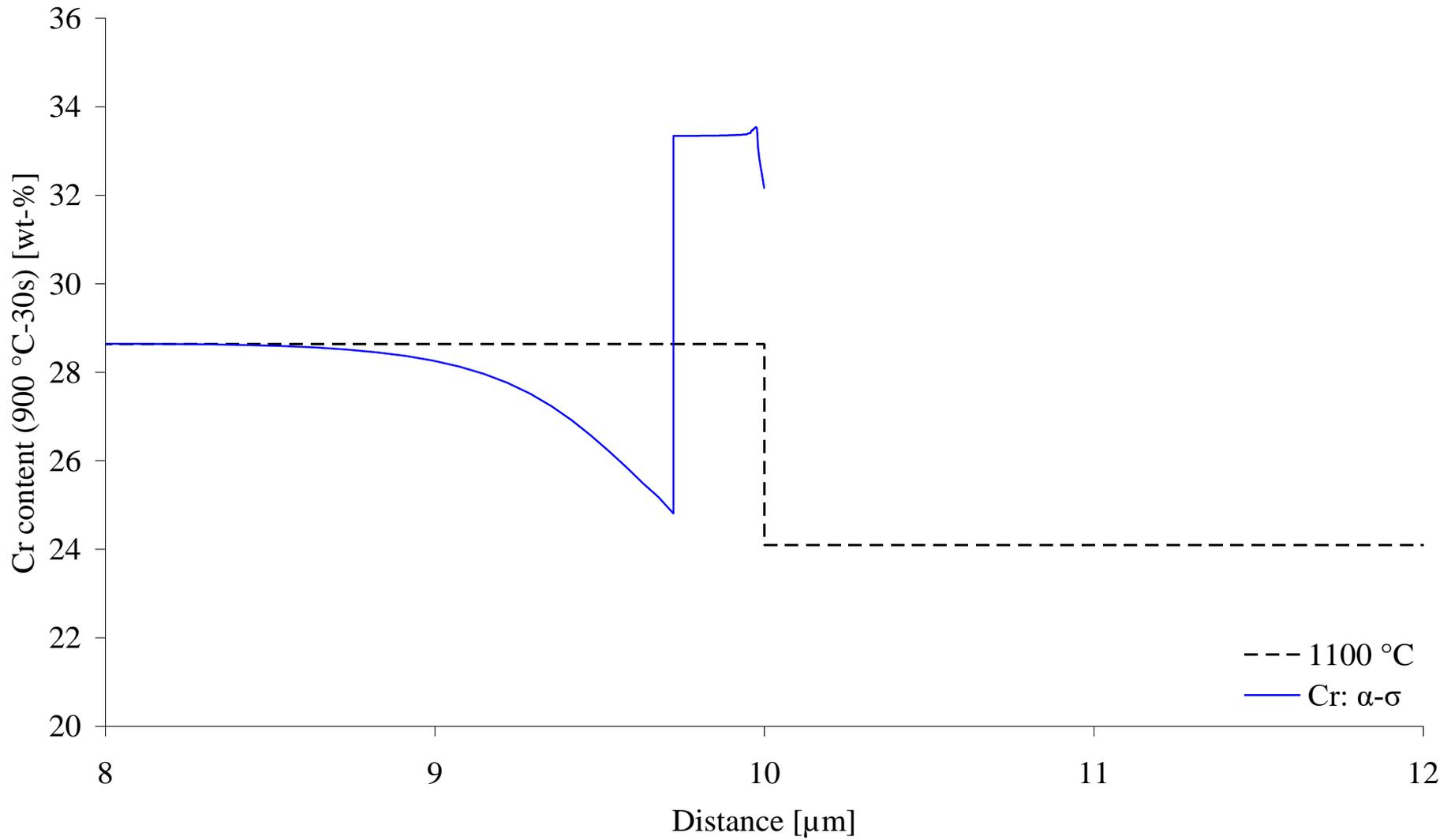




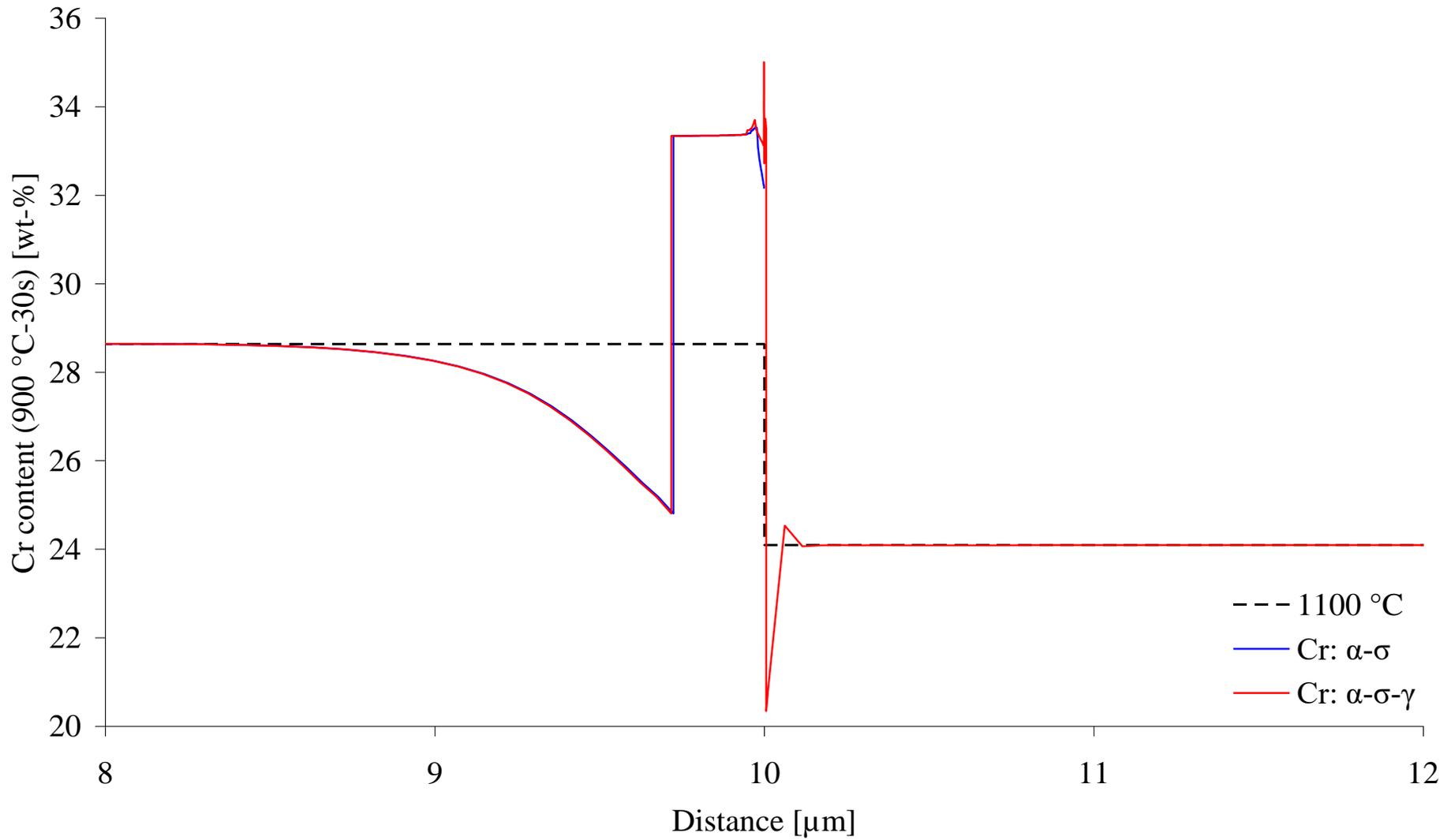
SDN: α - σ - γ versus α - σ



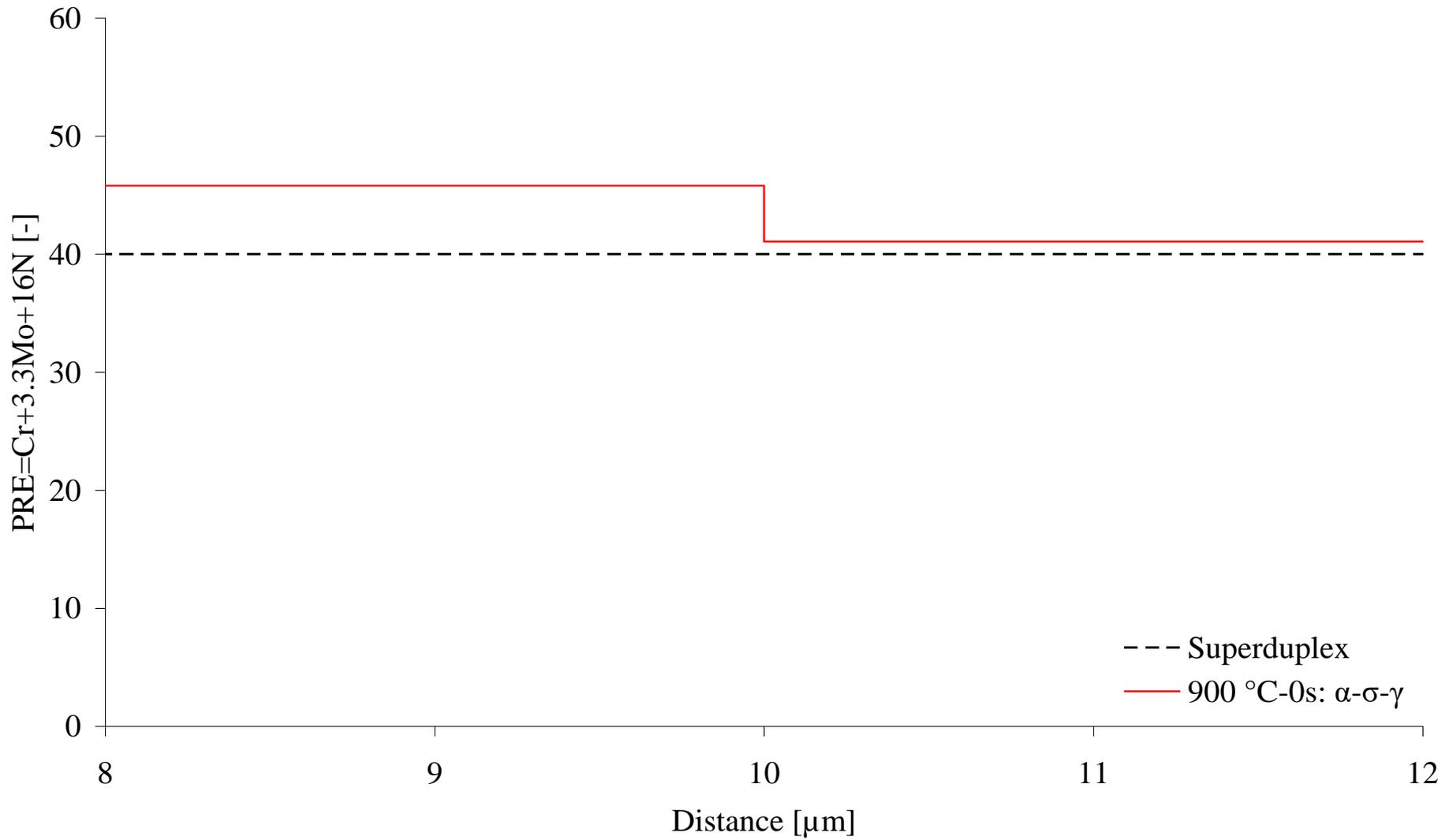
SDN: α - σ - γ & α - σ versus σ - γ



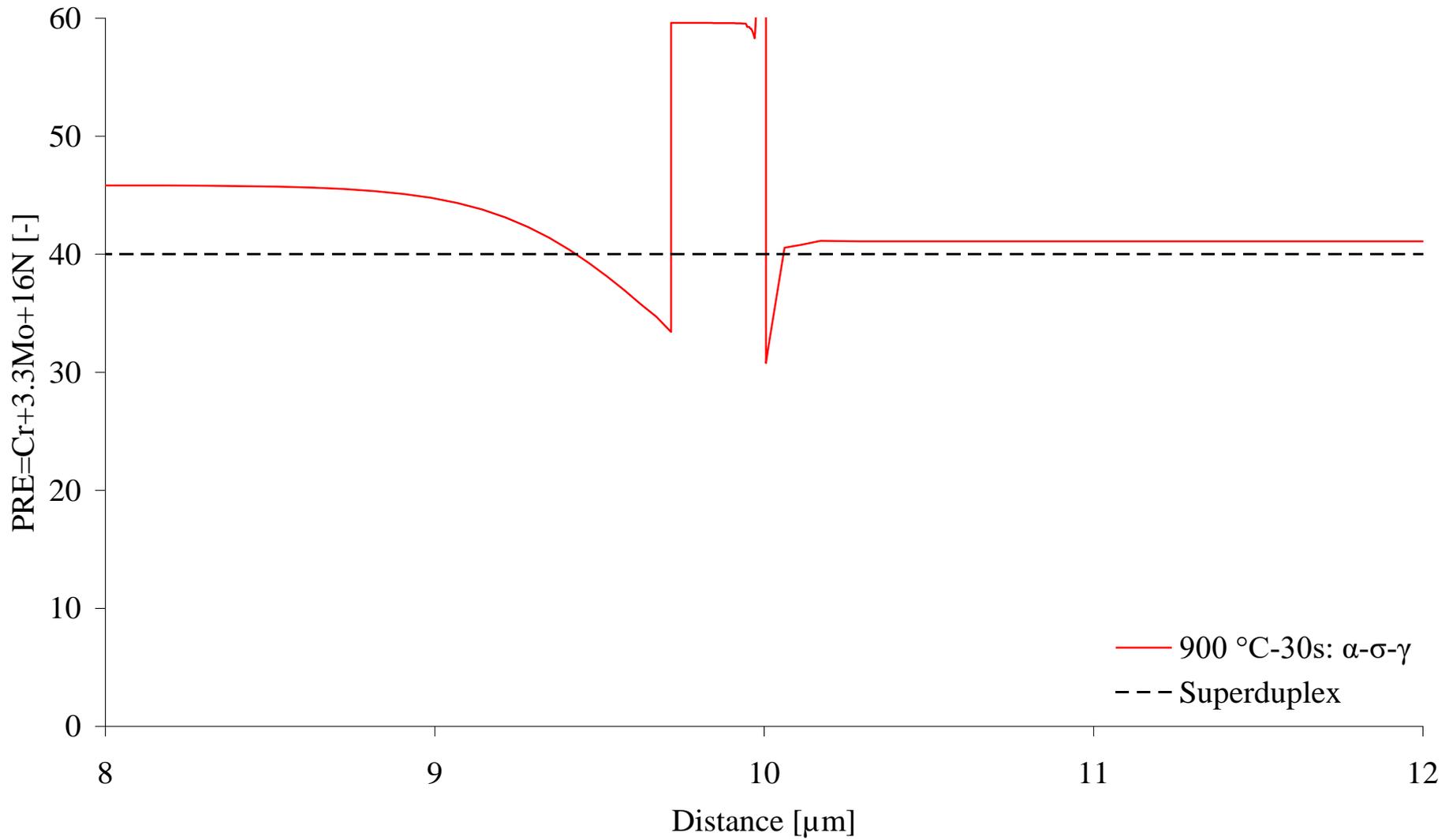
SDN: Depleted zones, 900 °C-30 s



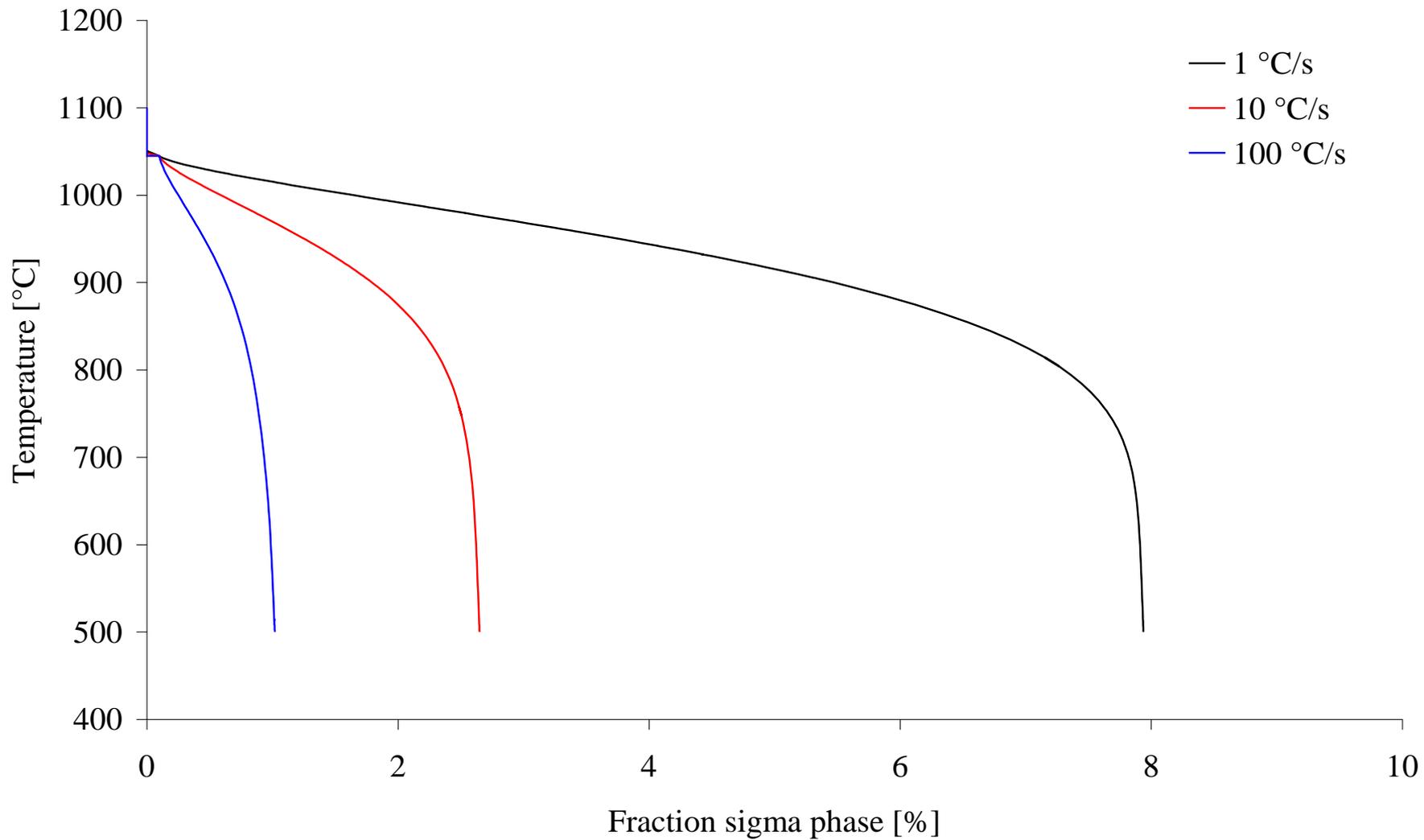
SDN: Depleted zones, 900 °C-30 s

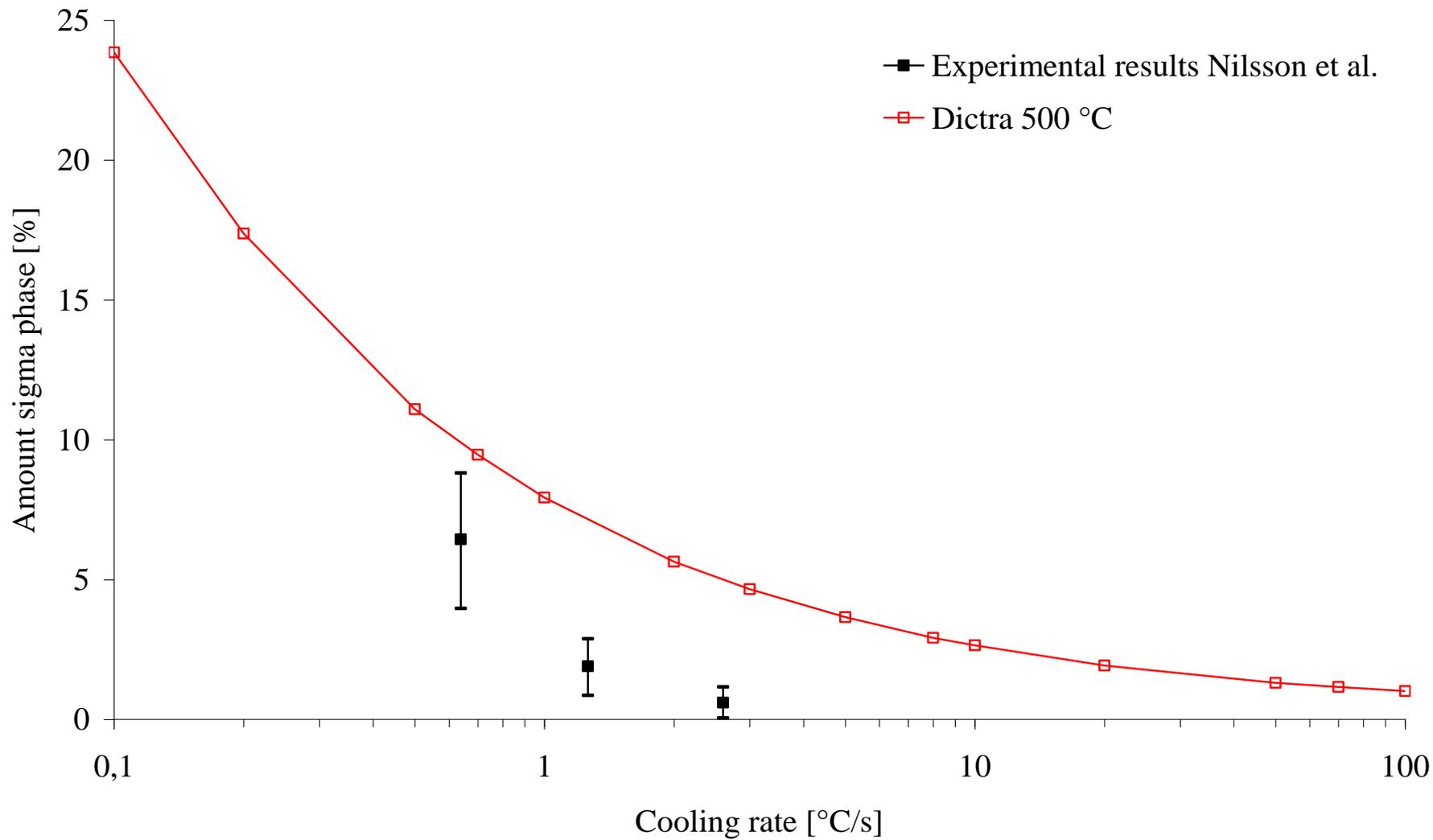


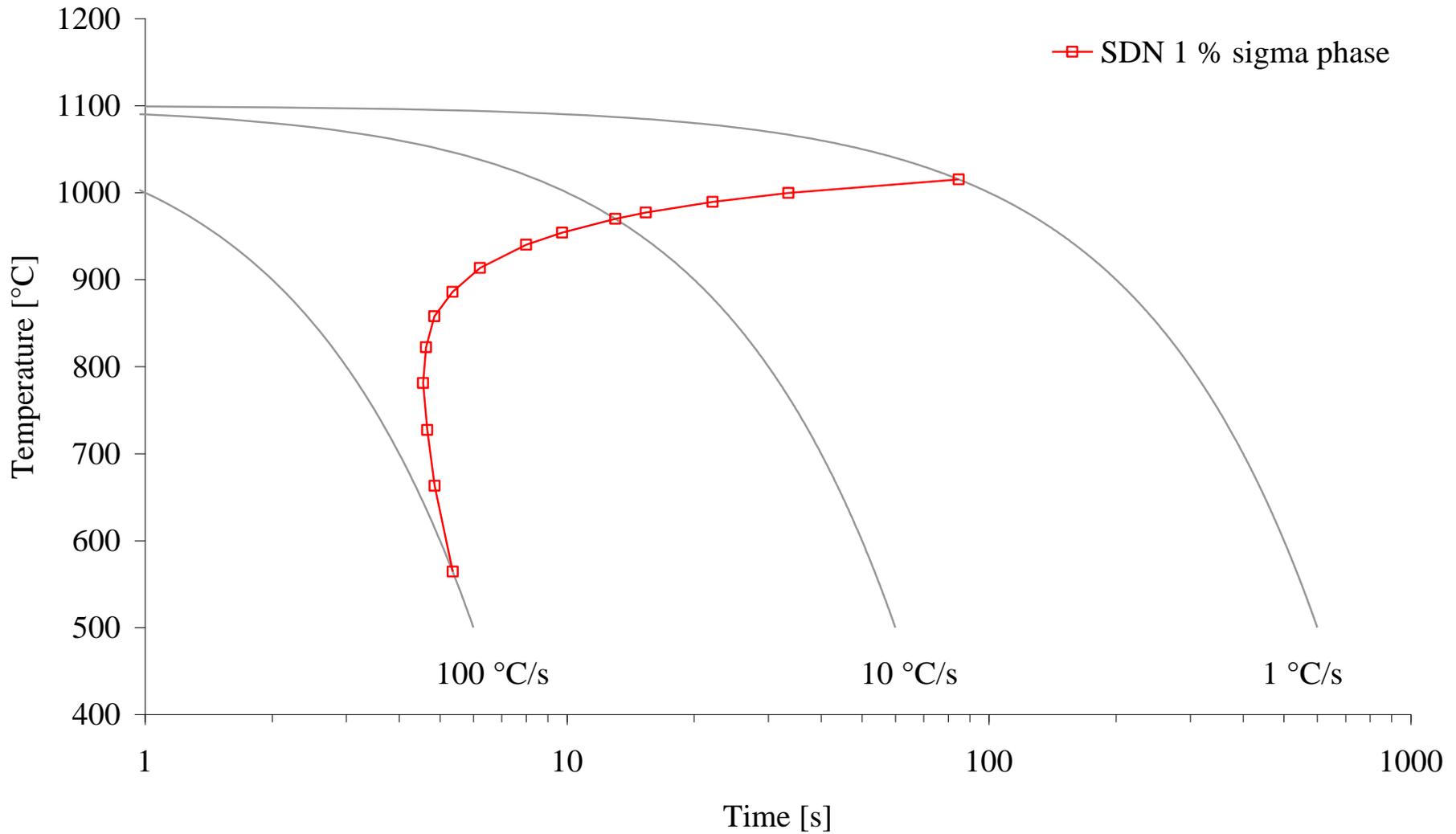
SDN: Pitting resistance equivalent $PRE=Cr+3.3Mo+16N$



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Conclusions:

- DICTRA can be used successfully to reproduce TTT and CCT diagrams.
- Growth of depleted zones adjacent to secondary phases can be studied.
- The influence of alloy composition on precipitation and growth kinetics of intermetallic phases can be derived.
- A good theoretical reproduction of tedious experimental and metallographic work was achieved.
- An extension with nominal compositions gave further insight to the influence of the individual elements W and Cu.

Critical issues :

- Austenite phase was omitted for sake of simplicity.
- Multiple phases was not considered, but the intermetallics one by one.
- Cell size is a **very** critical parameter !
- Nucleation was omitted, by good reason ?

S. Wessman, R. Pettersson: Application of computational thermodynamics to predict growth of intermetallic phases in superduplex stainless steels, Steel Research International 86 (2015).